

AN INVESTIGATION OF THE REACTIONS OF CARBON DIOXIDE,
CARBON MONOXIDE, METHANE, HYDROGEN, AND WATER OVER
IRON, IRON CARBIDES, AND IRON OXIDES

by

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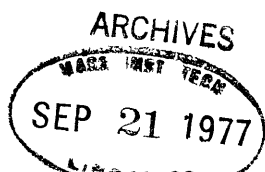
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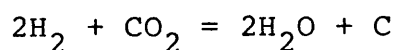
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ABSTRACT

Future long-duration manned space flights will require regenerative life-support systems. The Bosch process is one of several alternative regenerative life-support systems presently being evaluated by the National Aeronautics and Space Administration.



Prototype Bosch units to reduce metabolic CO_2 to carbon have been constructed and operated successfully with large recycle ratios. The water obtained per pass was, however, far less than that predicted from equilibrium calculations.

An investigation was performed which determined this limitation to be due to oxide formation; the catalyst condition being controlled by $P_{\text{H}_2}/P_{\text{H}_2\text{O}}$ and/or $P_{\text{CO}}/P_{\text{CO}_2}$ ratios. Carbide formation was shown to be slow, having little effect on efficient Bosch operation. Conclusions were drawn and the optimal recycle configuration and operating conditions were specified.

In addition, a metallurgical investigation of morphological changes occurring during reaction was performed. Results indicate two distinct morphologies developed depending on temperature during the oxidation-reduction sequence. One is typical of a process controlled by solid state diffusion; the other by diffusion (gas)/interfacial reaction control. The effects of carbon fiber formation were shown along with the examination of individual carbon fibers.

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Oh yes, to He who gave me all those mentioned above; to Him I say, "thank You".

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Professor Irving Kaplan
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Dear Professor Kaplan:

In accordance with the regulations of the faculty, I herewith submit a thesis, entitled "An Investigation of the Reactions of Carbon Dioxide, Carbon Monoxide, Methane, Hydrogen, and Water Over Iron, Iron Carbides, and Iron Oxides", in partial fulfillment of the requirements for the degree of Doctor of Philosophy in Chemical Engineering at the Massachusetts Institute of Technology.

Respectfully submitted,

Albert Sacco, Jr. V

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1. Summary

Future manned space vehicles will require full utilization of all metabolic waste products. In particular, regeneration of oxygen from all oxygen-bearing waste compounds will be essential in maintaining a closed, habitable ecological system. The Bosch process, of interest in this investigation, is one of several alternative regenerative life-support systems presently under consideration.

The Bosch process is the reaction of hydrogen with carbon dioxide to produce water and carbon.



The reaction is catalyzed by transition metals in the temperature range 800 to 1000 K.

Conceptually, water would be removed from the reactor effluent and electrolyzed. The hydrogen product would be recycled back to the reactor. The sum of these two processes would produce carbon and oxygen from metabolic carbon dioxide.



NASA has investigated several prototype Bosch recycle reactors utilizing an iron catalyst. Although moderately successful (if success is measured by carbon deposition), the effluent water concentration has been far below that expected from equilibrium considerations. Also, sizeable quantities of

carbon monoxide and methane have been found in the reactor effluent. This has resulted in recycle penalties (i.e., low energy utilization and high volume and weight requirements).

This investigation was undertaken to determine the limitation on water production; to ascertain the most efficient way to overcome and/or utilize this limitation and to specify optimal Bosch reactor operating conditions.

1.1 Pertinent Investigations

The stoichiometry indicated in reaction A is by no means representative of the reaction mechanism. The mechanism has been determined (Meissner and Reid {1972}, Manning {1976}) to consist of three major reaction systems: the carbon deposition reactions consisting of reaction D and/or reaction E



the reverse water-gas shift reaction , *



and the methane formation reactions



1.1.1 Carbon Formation

Walker et al. (1959) performed an extensive study on carbon deposition from carbon monoxide-hydrogen mixtures over

* Note: Reaction F + D = E

a reduced iron catalyst. Typically, the carbon deposition versus time plots (Figure 1) were sigmoidal in shape showing a weak induction period followed by a period of constant carbon deposition; the carbon deposition rate gradually decreased to zero. This general behavior was similar to that observed by other investigators (Manning, 1976).

1.1.2 Carbide Inhibition

High intensity X-ray diffraction patterns for cementite (Fe_3C) were observed on the spent catalyst in Walker's investigation. Walker et al. felt that carbide formation caused the drop in carbon deposition rate shown in Figure 1.

Tsao (1974), investigating carbon deposition from carbon monoxide over alpha-iron, also found cementite detrimental to carbon formation. Utilizing Mossbauer Spectroscopy, Tsao observed that formation of Fe_3C (reaction I), caused a rapid drop off in carbon deposition from carbon monoxide



Podgurski et al. (1950), in support of the hypothesis of both Walker et al. and Tsao, found carbon monoxide would not adsorb on a carbided surface. Thus, carbides would not be expected to catalyze reactions requiring carbon monoxide adsorption.

1.1.3 Oxide Inhibition

Manning (1976) determined that iron oxide inhibits carbon deposition from binary gas mixtures of carbon monoxide-carbon

dioxide at 823 K. Figure 7 indicates, when the CO/CO_2 ratio was such as to favor magnetite formation, reaction J-A, no carbon deposition occurred even though it was thermodynamically favorable.



If the $P_{\text{CO}}/P_{\text{CO}_2}$ ratio favored α -iron formation (i.e., $P_{\text{CO}}/P_{\text{CO}_2} = 1.76$), rapid weight gain was observed.

Additional evidence suggesting iron oxide may not be catalytic for carbon deposition was provided by Everett (1967). Everett, while investigating the effects of trace quantities of water, hydrogen, carbon monoxide, and carbon dioxide over an iron wire catalyst, observed carbon deposition to cease when the $P_{\text{H}_2}/P_{\text{H}_2\text{O}}$ ratio approached 10. This occurred in the temperature range 548-848 K and was attributed to iron oxide formation.

1.1.4 Reverse Water-Gas Shift Reaction

The reverse water-gas shift reaction is perhaps the most widely studied reaction in the Bosch sequence.

Kusner (1962) performed a detailed investigation of the reverse water-gas shift reaction in a packed bed.



His results indicated reaction F will go to completion, at 922 K, over both an iron and/or iron oxide (Fe_{1-y}O) catalyst.

Barkley et al. (1952), while investigating the shift reaction as a possible means of adjusting the H_2/CO ratio in

synthesis gas, determined an iron oxide-copper catalyst will catalyze reaction F at 811 K. Again, the reaction went to completion in a packed bed reactor.

1.1.5 Methane Formation

Manning (1976) studied methane formation in binary gas mixtures of hydrogen-methane at 823 K. The iron catalyst was preconditioned prior to introduction of the reactant gases. Figure 14 indicates at high hydrogen contents (i.e., 75-100%), carbon rapidly reacts; in 25-40% hydrogen mixtures, both reactions G and H should proceed to the left. However, no weight change was noticed suggesting carbide inhibits carbon deposition from methane.

Virtually no data are available on methane formation in five component gas mixtures. However, the equilibrium investigations of Browning et al. (1950, 1951) indicate reaction G was a problem in measuring the equilibrium for reaction H above 930 K. This implies that reaction G becomes kinetically more favorable at high temperature (i.e., ≥ 930 K).

Although nothing definitive can be said concerning methane formation, the inhibiting effects of iron oxide and iron carbide are clearly suggested in the literature. By simultaneously adjusting the gas phase composition to favor both carbon deposition and the solid phase of interest, the catalytic effects of that phase can be determined.

1.2 Equipment and Procedure

The experimental apparatus consisted of three integrated sections: the feed-gas delivery section, the reactor section, and the analytical equipment section.

In the feed-gas delivery system, chemically pure gases were individually metered, mixed, and fed dry or saturated with water to the reactor section.

In the reactor (Figure 32), the feed gases were preheated and passed into the bottom of a 28 mm vertical quartz tube. A thermocouple well allowed two thermocouples to be positioned under the catalyst bed; one was used with a proportional controller to maintain a preset temperature; the other provided a continuous reading of the reactor temperature. The catalyst assembly was positioned midway up the vertical quartz reactor tube. Clean steel wool was employed as the catalyst and, normally, 500 mg were charged. During a run, the reacting gases were forced to pass through the catalyst. At intervals of 10 to 20 minutes, the feed-gas flow was diverted and the catalyst carrier weighed in situ with an analytical balance. Weight changes within ± 1 mg could be detected.

The inlet and outlet streams were sampled and analyzed using an on-line gas chromatograph. An external standard was used in conjunction with the method of Dal Nogare and Juvet (1962) to obtain all gas compositions except hydrogen; hydrogen being determined from an empirical calibration curve following the method of Purcell and Ettore (1965).

1.3 Results and Conclusions

1.3.1 Effect of Preconditioning

Clean #2 steel wool catalyst was preconditioned by oxidizing with a carbon dioxide-water mixture followed by reduction with hydrogen. Two distinct surface morphologies were observed depending on conditioning temperature.

At 800 K, Figure 54 shows a thin, dense layer characteristic of oxidation-reduction processes controlled by solid state diffusion (Landler and Komarek, 1970).

Preconditioning at 900 K showed a highly porous, sponge-like shell surrounding a dense core. The shell region (Figure 39) is seen to consist of two distinct layers, and thus indicates the possibility of multiple oxide formation.

Spitzer et al. (1966) has shown multiple oxidation morphology occurs when the catalyst behaves as a porous body under diffusion (gas) or mixed diffusion-interfacial reaction control. Multiple oxide formation poses some interesting kinetic problems. Depending on the past history of the oxygen activity, changes in P_{H_2}/P_{H_2O} and/or P_{CO}/P_{CO_2} ratio (i.e., oxygen activity) will elicit a different response from the system. This indicates a system response time is to be expected and this response time may vary.

Also of interest is the fact that the effective catalyst area is increased by 500% after preconditioning at 900 K.

1.3.2 Carbon Inhibition by Fe_3O_4 (Magnetite)

The Bosch reaction systems and the various solid phases which may form during reaction can be conveniently represented on triangular phase diagrams.

Figure 55 illustrates the system behavior for six runs at 800 K for three different O/H ratios over a preconditioned, pre-carboned* steel wool catalyst. At each individual O/H ratio, two different experiments were run; one approached the α -iron/ Fe_3O_4 phase boundary from the reduced side; the other approached the boundary from the oxide side. The phase field of interest was investigated by adjusting the $P_{\text{H}_2}/P_{\text{H}_2\text{O}}$ ratio of the 5-component gas mixture, at a fixed O/H value, to a value thermodynamically favoring formation of that phase. Then, by observing if and when carbon deposition would start or stop, the catalytic activity of that phase for carbon deposition would be determined and the position of the phase boundary of interest established.

Figure 55 indicates the excellent agreement between the experimentally determined phase boundary and the theoretical phase boundary. The effluent concentration in all runs indicated methane remained constant; the weight gain during carbon deposition coming from carbon monoxide conversion. Due to the errors involved in experimentation, data acquisition and reduction; no definitive statements on the carbon

*After preconditioning, a 50% H_2 - 50% CO mixture was used to deposit a carbon bed on the catalyst.

deposition mechanism could be made.

1.3.3 Carbon Inhibition by Fe_{1-y}O (Wustite)

Figure 46 represents several experimental runs at various O/H ratios at 900 K. With the one exception of O/H equal to 0.17, all these data concur with the data obtained at 800 K. That is, iron oxide, in this case Fe_{1-y}O , inhibits carbon deposition. Again, the experimentally determined α -iron/ Fe_{1-y}O phase boundary is seen to be in agreement with that predicted by theory. The runs at O/H ratios equal to 0.17 are to be viewed with caution due to equipment limitations imposed by the high water concentrations necessary to obtain these O/H ratios.

Some scatter was associated with the various run conditions. That is, carbon deposition did not always start or stop where expected. Most of these "errors" were within 10% of the expected equilibrium $P_{\text{H}_2}/P_{\text{H}_2\text{O}}$ ratio. A propagation of error analysis indicates the maximum error in $P_{\text{H}_2}/P_{\text{H}_2\text{O}}$ can be as large as 11%. Undoubtedly part of this scatter comes, therefore, from normal inherent error associated with the experimental procedure, data collection, and data analysis methods.

It was felt, however, that some of the scatter may reflect another process(es) which affect the rate of both carbon deposition and oxide formation. An intriguing possibility is the formation and reduction of multiple oxide phases.

The catalyst from run A-43 is shown in Figure 49. Experiment A-43 was performed at high water concentration (i.e., O/H ratio equal to 0.17) with a reactor set temperature of 900 K. The run was prematurely terminated due to equipment malfunction. Figure 49 is a series of scanning electron micrographs of catalyst morphology found. Shown clearly in Figure 49 is the double layer structure indicative of multiple oxide formation. In this case the local oxygen potential was apparently higher than the bulk composition; which should not have supported multiple oxide formation. This high localized oxygen potential is believed to be caused by product poisoning; product poisoning has been reported by Everett (1967) and Wilson (1971) under similar conditions.

Since the rate of reduction or oxidation in a system capable of multiple oxides varies (Spitzer et al., 1966), the scatter in these data at 900 K was not unexpected.

1.3.4 The Effect of Carbides on Carbon Deposition

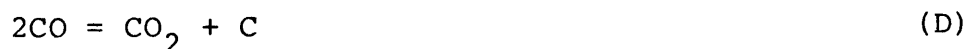
The role of carbides during carbon deposition is still unclear. The problem lies in the fact that carbides are difficult to identify and can form not only during reaction but, as the reaction goes through wide temperature fluctuations such as in hot spot formation or during the reactor cooling-down sequence.

In an attempt to determine the effect of carbides on carbon deposition, runs A-58 and A-59 were performed at fixed C/H values of 0.2 and 0.35, respectively. The catalyst was

preconditioned using the standard procedure and a carbon bed laid down.

Figure 51 indicates that a weight gain was observed in the region where cementite (Fe_3C) would be expected to be the stable solid phase. However, the effluent in both runs A-58 and A-59 indicate no change in methane concentration; the observed weight gain resulted from carbon monoxide conversion.

There are two possible reactions which could account for the weight gain observed, reaction I and/or reaction D.



From stoichiometric considerations it is impossible to determine which reaction accounts for the observed weight gain. The fact that 3.5 times as much weight gain is observed as needed for complete carbiding (35 mg) and, considering the work of Podgurski et al. (1950) and Walker et al. (1959) suggesting carbiding with carbon monoxide is very slow, indicates that the weight gain is probably due to carbon deposition.

Although the data are inconclusive in determining if carbides inhibit or catalyze carbon deposition, from an operational standpoint, it appears that oxide formation is the most immediate problem in efficient Bosch reactor operation.

1.3.5 Structural Changes During Reaction

Varying oxygen potential in the system to control carbon

deposition will cause structural changes depending on initial morphology and temperature. In addition, carbon deposition has been shown to alter the catalyst structure through the formation of carbon fibers (Walker et al., 1959, Ruston et al., 1969). In an attempt to illustrate the morphological changes occurring on a Bosch catalyst, a metallographic examination was performed.

The catalyst from run A-18 was examined using a scanning electron microscope. In run A-18 the P_{H_2}/P_{H_2O} ratio was varied between (∞) (i.e., no water and rapid carbon deposition) to a value of 1 (little or no observable weight gain). The reactor set temperature was 900 K and the total pressure $1.01 \times 10^5 \text{ N/m}^2$.

Micrograph (a) illustrates the shell and core type morphology which results from the oxidation-reduction sequence used in A-18. Carbon-bearing gases diffuse through the porous-iron shell and deposit carbon. This deposition results in a brittle external shell that breaks easily from the more structurally solid core. In micrograph (b), carbon fibers are seen to form in bundles or nodules. Transmission electron micrographs indicate these nodules are electron-dense material, presumably iron or iron compounds.

Micrographs (c), (d), and (e) represent a small section of the exterior shell. Here, the nodule-like fiber bundles are clearly seen; micrograph (e) reveals carbon fibers having both tubular and circular shape.

Figure 43 is a transmission electron micrograph of a typical carbon fiber formed in the external shell region of a catalyst fiber from run A-18.

The "camel" shaped fiber was suspended securely from the main catalyst surface by carbon fibers. This is indicative of the type of shell structure developed during carbon deposition following catalyst pretreatment. The shell region appears to consist of an intricate network of interwoven fibers connected securely by fiber bundles or nodal points. These nodal points consist of iron and/or iron compounds.

The shaft region is seen to be hollow, with some electron dense material along the outside. Most fibers appeared to have an electron dense tip.

These results are in agreement with the fiber structures noticed by Walker et al. (1959) and Ruston et al. (1969).

Fiber formation is fascinating, but little can be said concerning the growth mechanism from this investigation. However, these micrographs clearly indicate an increase in total as well as effective (metallic) surface area. Also, they suggest possible diffusion limitations as reaction proceeds. A detailed knowledge of structural changes as a function of time is necessary, however, before a definitive statement can be made.

1.4 Recommendations

The conclusions suggested from this investigation lead to the following recommendations.

First, if it is desirable to use steel wool as a catalyst, there is an optimum system design. This optimum design utilizes the knowledge that carbide formation is slow and will not influence process operation. However, care must be taken to avoid and/or utilize oxide formation to maximum advantage; second, if other transition metal catalysts are investigated, care must be taken to evaluate the effects and amount of water and/or carbon dioxide necessary for oxide formation. Also, the effects and rate of carbide formation should be evaluated.

1.4.1 Optimal Reaction Conditions

It can be shown that the maximum water concentration occurs at the intersection of the graphite-gas/iron-iron oxide phase boundaries. The optimal O/H ratio (i.e., the intersection point) will increase with temperature. Ideally, one would like to operate at this point; however, if running a Bosch reactor alone or in combination with a reverse water-gas shift prereactor, having fixed the total moles in the system at the inlet H_2/CO_2 ratio of 2.0; the O/H ratio throughout the system is constrained by material balance to be 0.5. The optimal operating conditions would be determined by the intersection of an operating line drawn from the carbon apex to the point where the O/H ratio equals 0.5 (i.e., position of H_2O) with the phase boundary intersection point where the O/H ratio equals 0.5. This has been shown by Manning (1976) to occur at 915 K with a corresponding recycle ratio of 10.0 total moles recycled per mole CO_2

processed.

This limitation can be overcome, however, if one allows for the addition or removal of hydrogen to the system on start-up. A balance can then be made which equates the product of the recycle rate from the Bosch reactor times the water concentration (at the optimum O/H ratio desired) to the rate at which oxygen, hydrogen, and carbon are being fed to the reactor. This type of design is optimal using a shift reactor to initially take out some of the water. A conceptual reactor design is shown in Figure 57. An actual reaction gas flow path is shown in Figure 56 as envisioned on a triangular phase diagram at 875 K.

Initially, 2 moles of hydrogen are mixed with 1 mole of CO_2 and fed to the shift reactor. Point 1 represents the position of the mixture which must fall on the intersection between the O/H operating line for the shift reactor (i.e., 0.5) and a line drawn from the position of carbon dioxide to the position of hydrogen as represented on the phase diagram. Removing the water formed in the shift reactor, the gas mixture moves along the O/H operating line to point 2. The gas mixture at point 2 is mixed with a gas mixture at point 3 to give point 4. The mixture at point 3 was composed of the Bosch reactor effluent at an O/H ratio of 0.204 minus the water formed. At this time, it should be again pointed out hydrogen was added on start-up to initially get this optimum O/H ratio. The gas composition at point 4, having an O/H

ratio of 0.204, proceeds down the Bosch reactor operating line to the intersection point, at point 5.

Evaluation of this process indicates a minimum recycle ratio of 9.2 at a reactor temperature of 875 K for both the shift and the Bosch reactors.

The effect of hydrogen recycle in the shift reactor (i.e., decreased O/H ratio) is to increase the total moles to be recycled per mole of CO_2 processed. Increasing the O/H ratio in the shift reactor (i.e., decreasing the inlet H_2/CO_2 ratio), again, increases the minimum recycle rate. The minimum recycle rate can be decreased by increasing the shift reactor temperature but this decrease is small being only 3% for every 100 degree increase in shift temperature.

In conclusion, therefore, the optimum operating conditions are the shift reactor-Bosch reactor configuration with both reactors operating at 875 K; the minimum recycle rate in the Bosch reactor being set at 9.2 total moles recycled per mole CO_2 processed.

1.4.2 The Use of Nickel and Cobalt as Catalysts

The iron system has been shown to be restrictive due to oxide formation. Other transition metals such as nickel and cobalt are believed to catalyze all the reaction systems involved in the Bosch sequence but, no determination as to the behavior of their oxides is known. Garmirian and Reid (1977) have shown that oxide formation may not be a problem for these systems. That is, the nickel/nickel oxide and cobalt/cobalt

oxide systems have equilibrium water concentrations well above that expected for the Bosch system (i.e., the graphite-gas equilibrium). Thus, these two metals show promise as efficient catalysts for the Bosch process.

In all metallic catalytic systems, the various phases which form during reaction should be carefully evaluated. A tractable way to accomplish this is the phase diagram type of analysis used in this investigation.

2. Critical Literature Review

The Bosch process consists of a complex set of reactions thought to occur in both parallel and series combinations. Some of these reactions are catalyzed by transition metals and/or their oxides and carbides; others are involved in solid gas reactions that change the catalyst structure and phase during reaction. It is believed a clear understanding of these processes is necessary to understand the experimental methods and results obtained in this investigation.

2.1 Gas Phase Reaction Systems of Interest

2.1.1 Carbon Formation

Walker et al. (1959) performed one of the more extensive studies on carbon formation from carbon monoxide-hydrogen mixtures. This investigation was conducted in a reactor consisting of a Vycor tube into which the catalyst was placed. The catalyst itself was in a porcelain combustion boat. The cumulative weight of carbon formed during reaction was determined from the volume of gas measured before and after the reactor, assuming reactions D and B were the only reactions of significance.



Most runs were conducted with a carbon monoxide-rich gas (carbon monoxide-hydrogen ratios were normally between 99.2/0.8 to 80.8/19.2). The temperature range covered was between

723-973 K. The flowrate was varied and a number of reduced iron powders tried. Baker analyzed reagent grade powder, 100 μm , reduced in hydrogen was found to be the most active.

A typical plot of carbon deposition versus time is shown in Figure 1. The curves were generally sigmoidal in shape showing a weak induction period followed by a period of constant carbon deposition. Finally, after a period of decreasing rate, carbon deposition stopped entirely.

Walker et al. studied the effects of hydrogen addition on carbon deposition rates at a variety of different temperatures. Figure 2 was typical of the behavior found.

As the hydrogen content was increased, the temperature at which the maximum rate of carbon deposition occurred generally increased. Also, the amount of carbon deposited per gram of catalyst increased. At temperatures below 801 K, the change in gas composition had little effect on carbon deposition. Above 849 K, Walker observed, the maximum rate of carbon deposition increased with hydrogen content to a point and then decreased.

2.1.1.1 Carbide Inhibition

Walker et al. noted that x-ray diffraction analysis on deactivated* catalyst indicated predominantly cementite (Fe_3C) and carbon peaks. No diffraction patterns were reported for Fe_3O_4 , Fe_2O_3 , or $\alpha\text{-Fe}$. They speculated that cementite was not a catalyst for carbon deposition; i.e., carbon deposition ceased when the fraction of available $\alpha\text{-Fe}$ became negligible.

* After no more carbon would deposit.

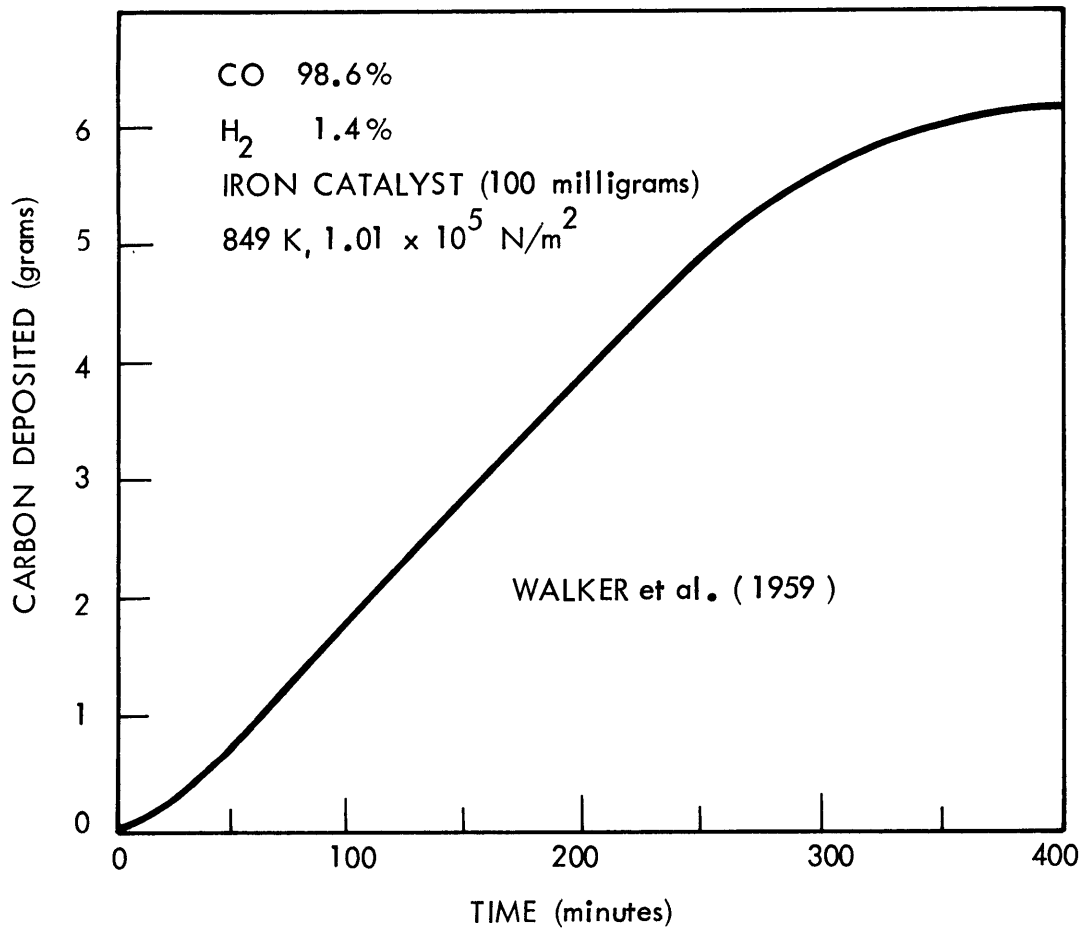


FIGURE 1
CARBON DEPOSITION VERSUS TIME, GENERAL SYSTEM RESPONSE

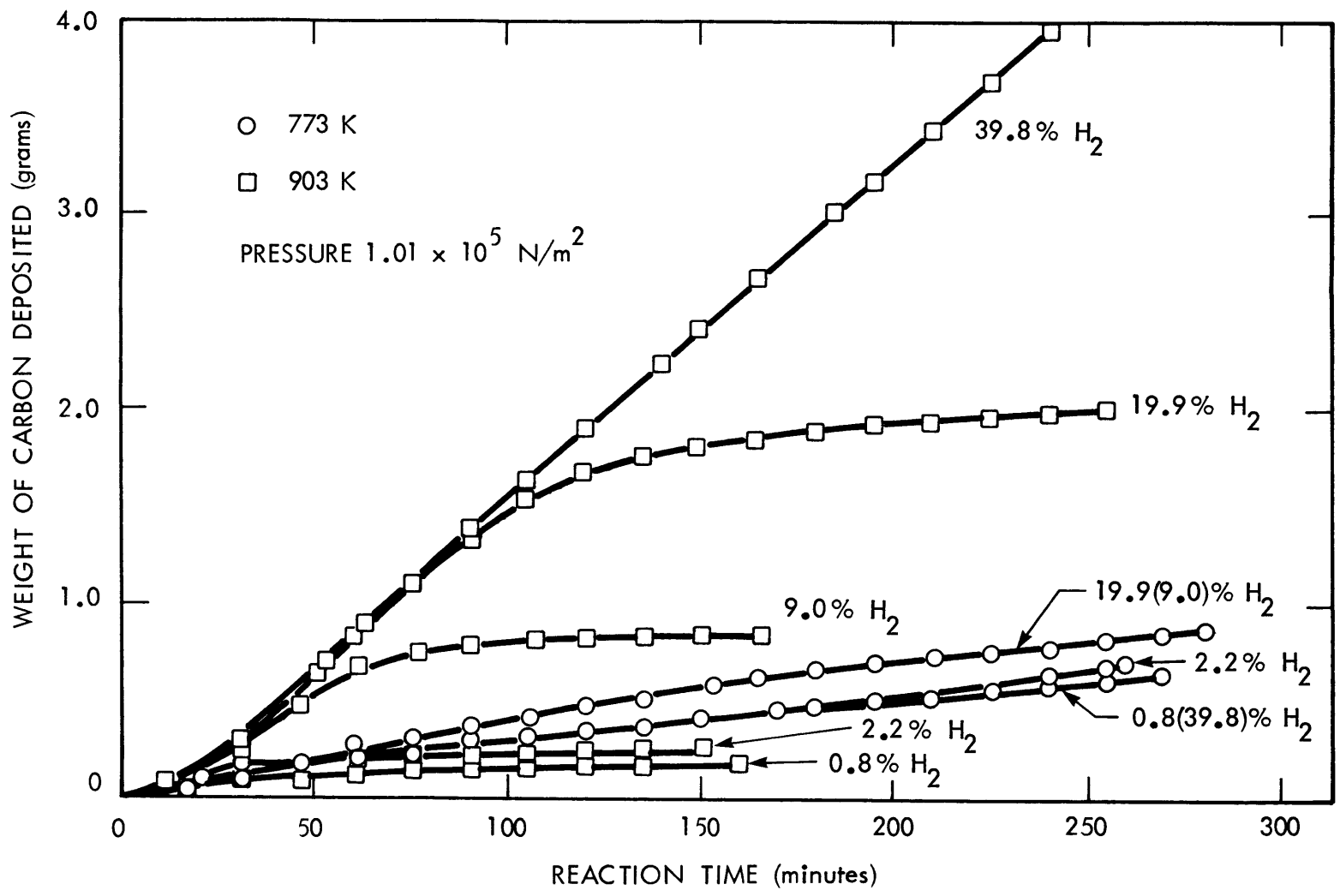


FIGURE 2 EFFECT OF HYDROGEN ADDITION ON CARBON DEPOSITION FROM CARBON MONOXIDE AT 773 K AND 903 K (WALKER et al., 1959)

The rather confusing behavior observed with hydrogen addition was qualitatively explained using this hypothesis.

At low temperatures carbide formation occurred at a very slow rate. The addition of hydrogen to the system would not be expected, therefore, to have a substantial effect on the fraction of α -Fe available for reaction. At high temperatures, however, carbide formation is rapid and the high hydrogen contents are necessary to keep a sufficient fraction of the catalyst in the reduced form. The odd behavior of carbon deposition first increasing then decreasing with increased hydrogen content at high temperatures, could be explained through the influence of reaction G but, no qualitative state-



ment was made. This proposed mechanism of carbide inhibition agrees with that suggested by Tsao (1974).

Tsao studied the dissociation of carbon monoxide over reduced, porous iron disks. The disks were suspended from a Ni-span C spring balance into a vertical alumina reaction tube housed in a resistance furnace. Operating temperatures were between 903-1027 K and a flowrate of 0.3 l/min (STP) was normally used.

Figure 3 illustrates the behavior Tsao observed. Three distinct regions were found: initially, the rate of carbon deposition was constant; then, it suddenly dropped off; this was followed by a slow increase in reaction rate (the one exception being at 903 K).

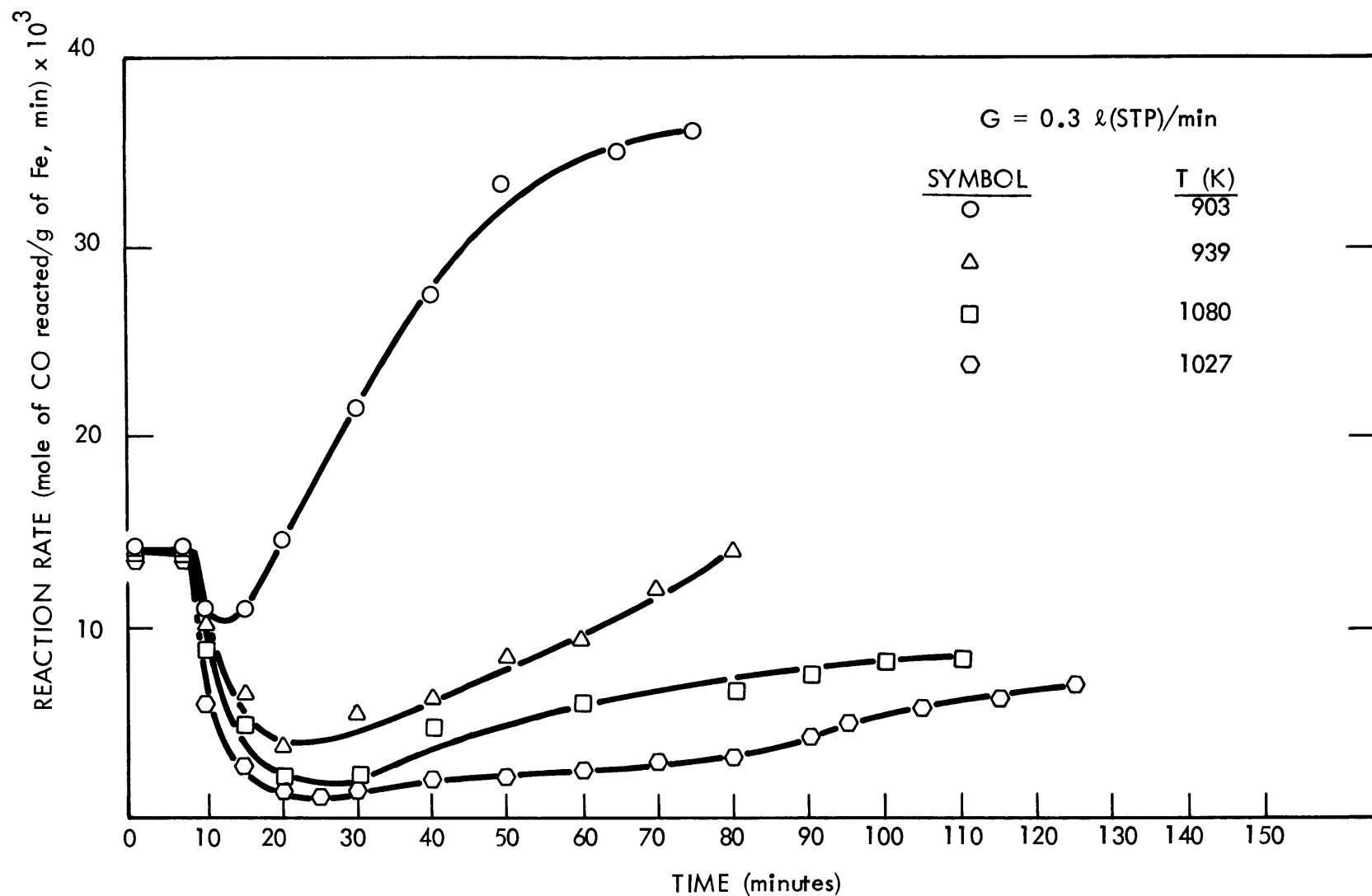


FIGURE 3 REACTION RATE VERSUS TIME (TSAO, 1974)

Tsao explained this behavior based on competition between reactions D and I.



The data for Figure 4 were obtained by Mossbauer analysis of the iron-bearing compounds on the surface of the disks. As shown, cementite appeared to have formed at a faster rate initially than did "free" carbon. Tsao concluded that the observed drop in carbon deposition rate was due to the formation of carbide. The slow increase in deposition rate noted after complete carburizing was attributed to the increase in "free" carbon surface area; which, he measured.

Tsao's proposed mechanism agrees in principle with Walker and co-workers. Unfortunately the results at low temperature (i.e., 903-939 K), in the range Walker studied, could be interpreted as Fe_3C being a more active catalyst than $\alpha\text{-Fe}$, as well as the way it was interpreted.

The question of carbides being promoters or inhibitors is a complex one. The problem lies in the nature of analysis, as well as the age-old question: Does one have on completion of reaction what one had during reaction? Most data indicate that carbides should not be catalysts for carbon deposition from carbon monoxide (Podgurski et al., 1950). However, the question of the catalytic effects of carbides is still open to interpretation.

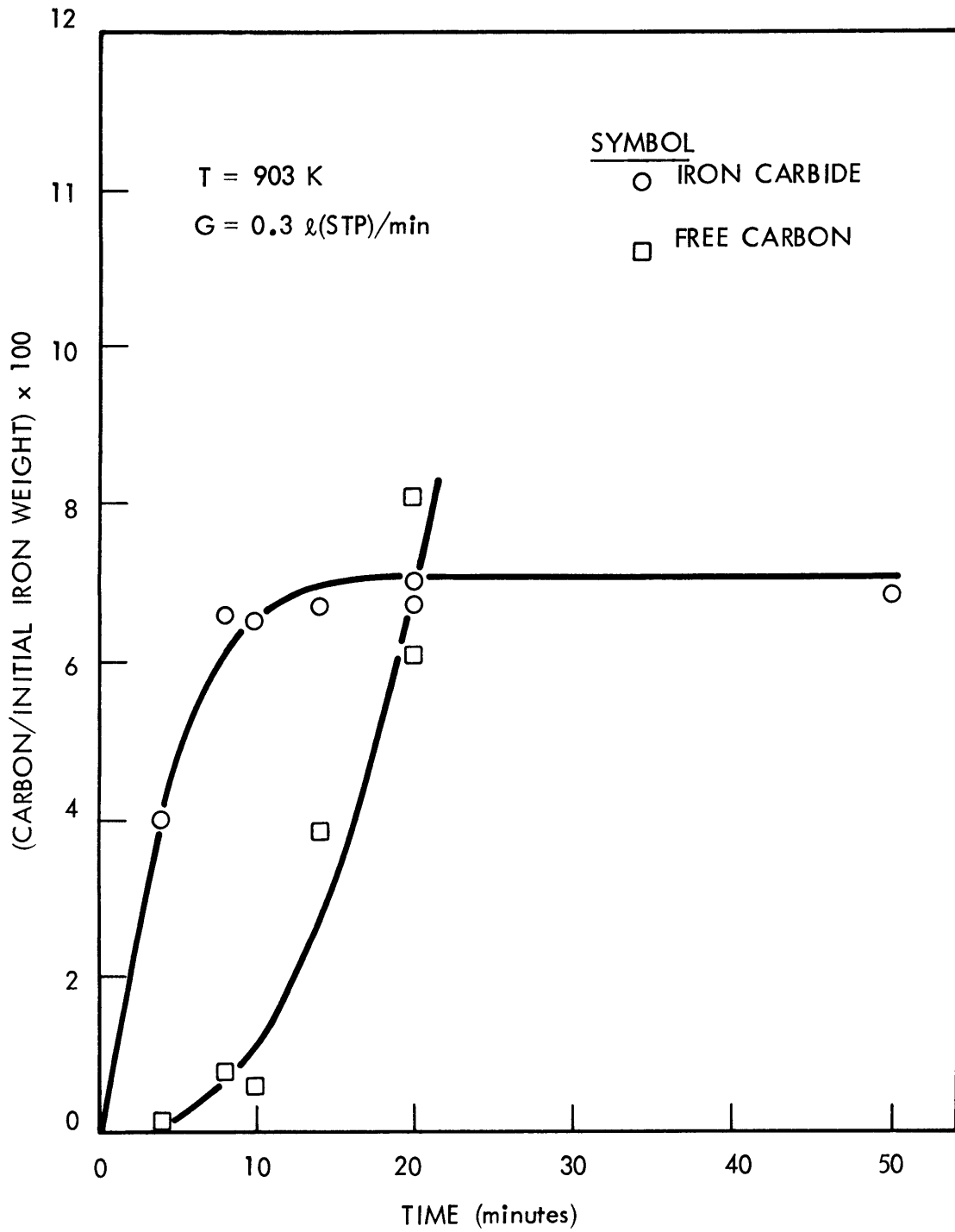


FIGURE 4
CARBON ANALYSIS OF THE DISKS REACTED AT 903 K (TSAO, 1974)

2.1.1.2 Recent Research at MIT

Manning (1976) studied the formation of carbon from various hydrogen-carbon monoxide mixtures at 823 K. The catalyst used was a commercial grade steel wool. A detailed description of the apparatus used is given in the Apparatus and Procedure Section.

Figure 5 is a plot of weight of carbon deposition versus time for a typical run. In this particular experiment, a 1:1 mole ratio of hydrogen to carbon monoxide was passed over 250 mg of steel wool catalyst. As shown, carbon deposition was initially a linear function of time; this period corresponds with the Walker et al. (1959) induction period. However, after a short period (120 minutes) the carbon deposition rate increased. This change occurred in the same time frame as Walker's region of increased rate and is believed similar.

During the initial linear period, Manning found trace amounts of carbon dioxide and water. These corresponded, at the given flowrate {19.8 cm³/s (STP)}, to a conversion of less than 1% for both hydrogen and carbon monoxide.

The reactions by which carbon forms from CO are not clear. There are several proposed routes by which carbon may be produced. One is the carbon monoxide disproportionation reaction (reaction D).



Alternately, carbon may be formed by reaction B coupled with

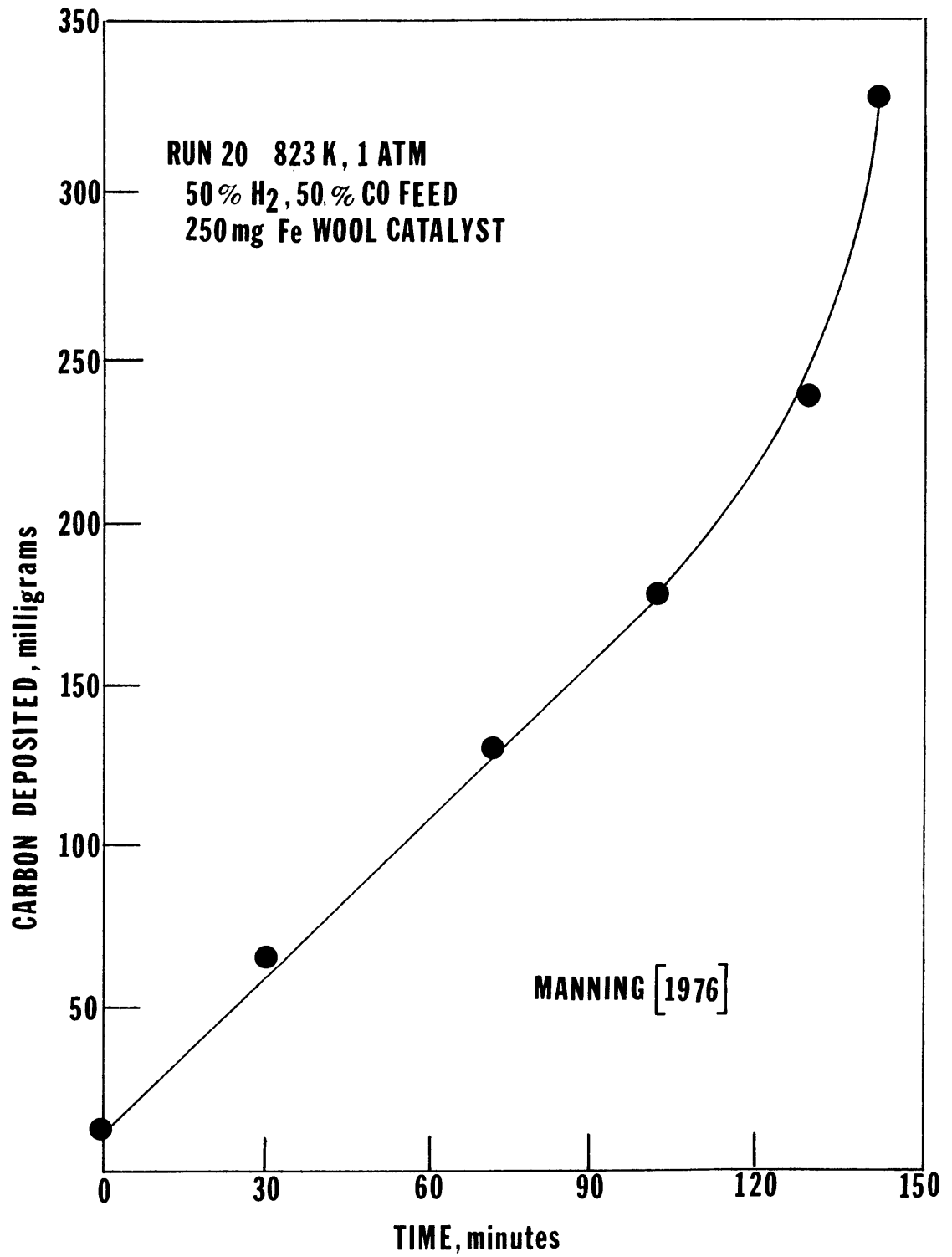


Figure 5 Carbon Deposition During Start-Up With CO-H₂ Feed

reaction F giving the observed effluent.



Also possible, as Walker et al. (1959) suggested, reaction D may occur in parallel with reaction B.

Manning's data, obtained at low conversion, were unable to ascertain the mechanism for carbon deposition. However, several interesting results were noted. Using a least square analysis, Manning obtained the following statistically significant correlation:

$$\text{rate (g mole carbon/cm}^2\text{-s)} = 4.3 \times 10^{-8} (P_{\text{CO}} P_{\text{H}_2})^{0.42 \pm 0.10} \quad (1)$$

The carbon monoxide and hydrogen dependence was similar to that found by Everett (1967) for high content hydrogen-carbon monoxide mixtures.

Also, Manning determined the incipient reaction rate for carbon deposition could be increased significantly by pre-oxidation of the catalyst. Initially, 250 mg of steel wool catalyst were oxidized in CO_2 for several hours. A 75% H_2 —25% CO reducing gas mixture was then fed to the reactor. Figure 6 illustrates the observed effect. It was speculated that reduction of the surface occurred rapidly, providing a highly reactive, high area α -Fe surface.

The effects of an oxidized surface were further investi-

* Note reaction D = reaction B + reaction F

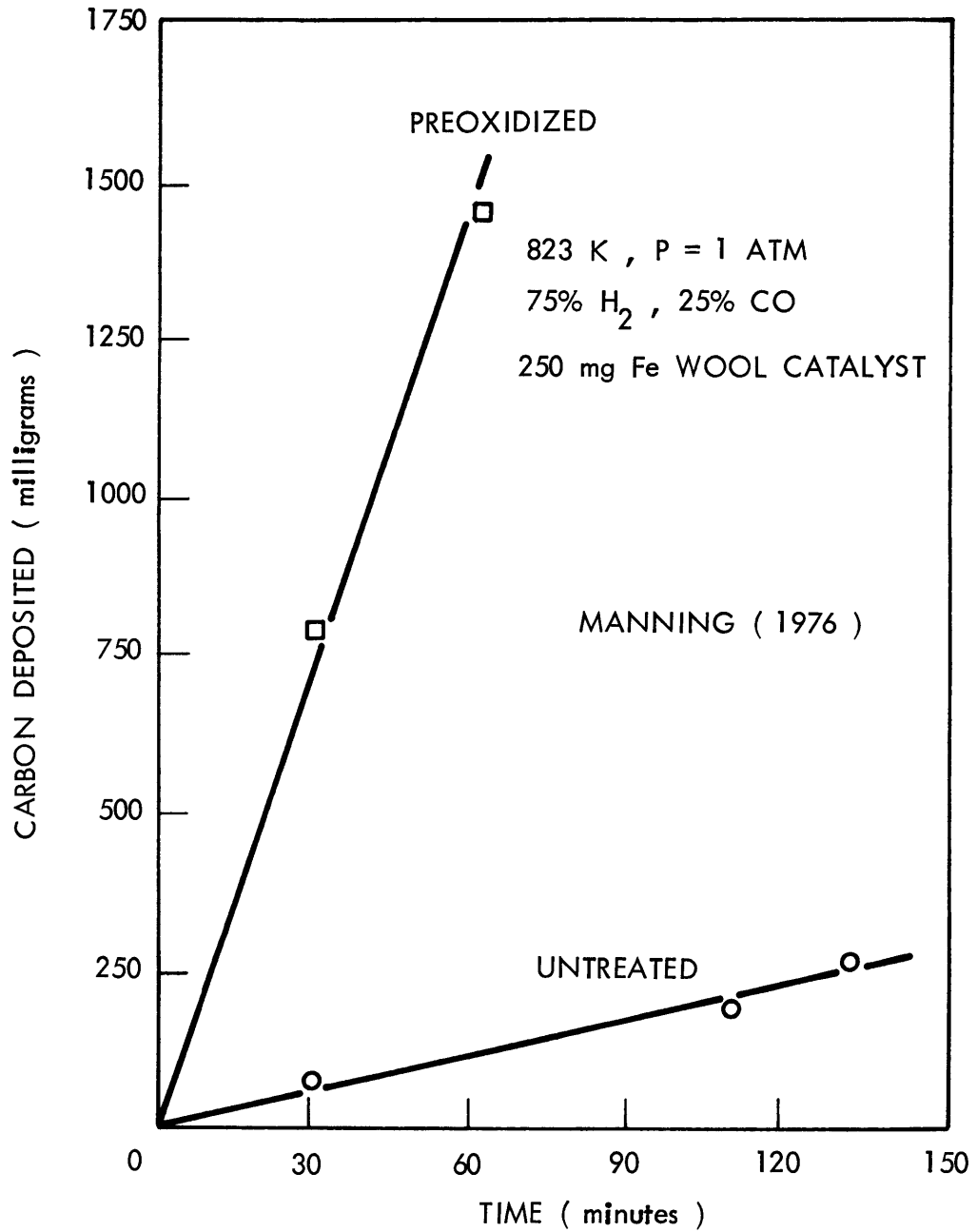
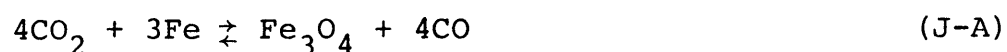


FIGURE 6
COMPARISON OF START-UP BETWEEN AN OXIDIZED AND AN
UNTREATED BOSCH CATALYST

gated by studying the effect of binary gas mixtures of carbon monoxide-carbon dioxide over a pre-carbonized catalyst, Figure 7. When carbon monoxide-carbon dioxide ratios of 1-1.3 were passed over the catalyst, no significant weight loss or gain was noted. Increasing the carbon monoxide-carbon dioxide ratio to 1.5, produced erratic weight loss and weight gain; and, a ratio of 1.94 produced rapid weight gain.

Analysis of Figure 7 indicates the formation of Fe_3O_4 (reaction J-A) inhibits reaction D from depositing carbon.



This conclusion is based on thermodynamic considerations.

All the binary carbon monoxide-carbon dioxide gas mixtures fed exceeded the equilibrium carbon monoxide partial pressure (i.e., 13%) for reaction D. Thus, thermodynamically, reaction D should have proceeded to the right depositing carbon. However, when the carbon monoxide-carbon dioxide ratios were such as to favor the formation of Fe_3O_4 rather than Fe (i.e., $\text{CO}/\text{CO}_2 \leq 1.13$), no carbon deposition occurred. Further, when the carbon monoxide-carbon dioxide ratio favored α -Fe formation (i.e., 66% CO, 34% CO_2), rapid weight gain was observed, the implication being that iron oxide (Fe_3O_4) is not a catalyst for carbon deposition.

2.1.1.3 Oxide Inhibition

The idea of oxides of iron inhibiting carbon deposition from carbon monoxide was developed in an investigation per-

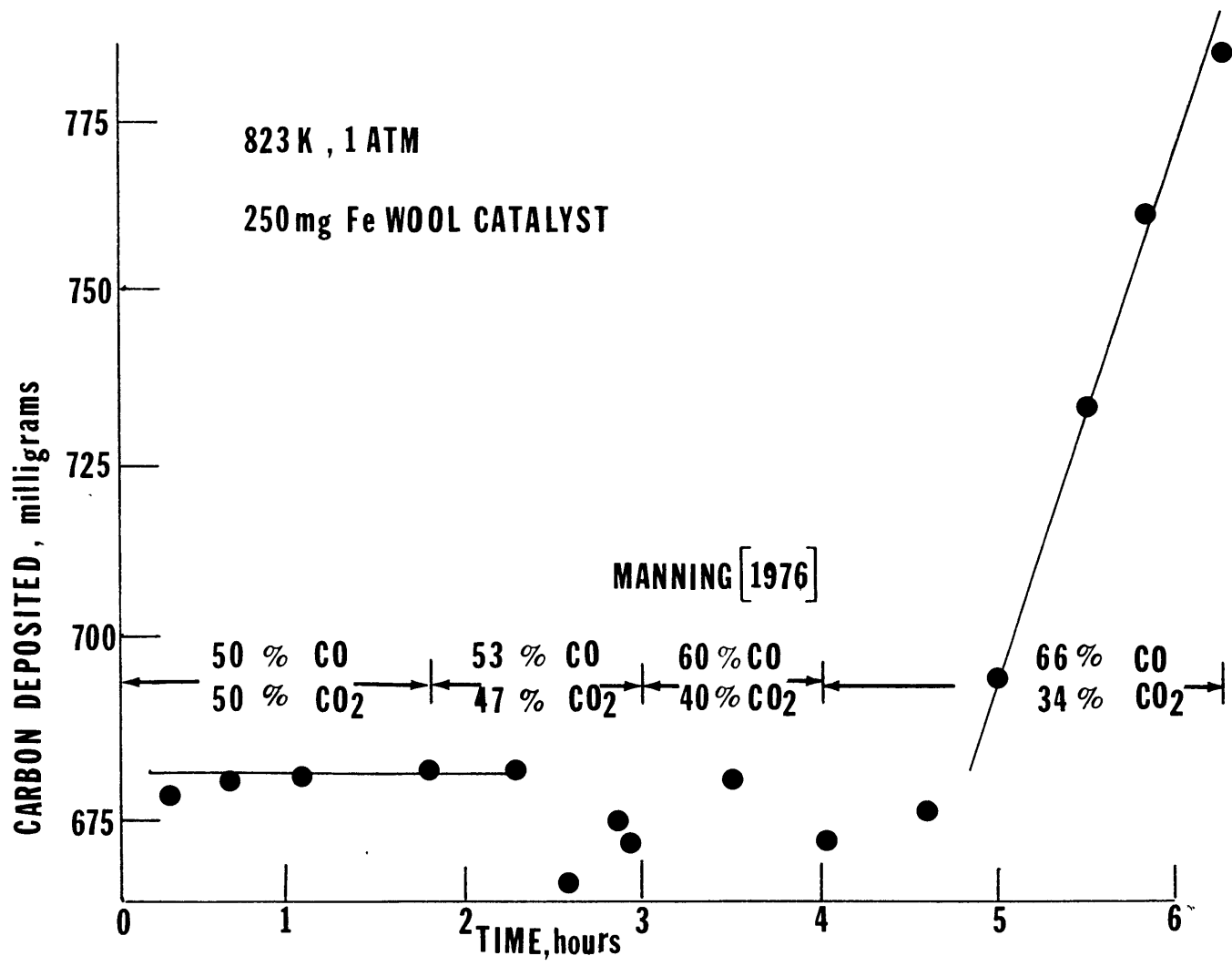


Figure 7 Carbon Deposition From CO-CO₂ Mixtures

formed by Everett (1967).

Everett studied the kinetics of carbon deposition reactions in high temperature gas-cooled nuclear reactors. This investigation was undertaken to ascertain the likelihood of metal fatigue caused by trace quantities of water, carbon dioxide, carbon monoxide and hydrogen.

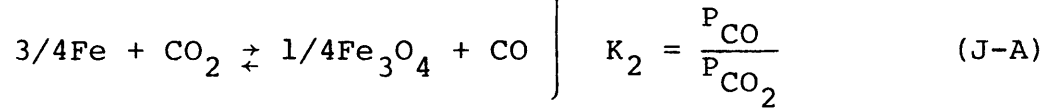
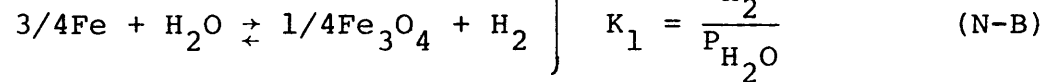
The experimental apparatus consisted of a furnace housing a silica reaction tube and a set of steel samples. The temperature could be varied between 548-848 K. A high temperature graphite furnace for partial reconversion of the water and carbon dioxide products back to hydrogen and carbon monoxide was used to maintain a steady hydrogen-carbon monoxide feed. The entire system was connected in a closed loop; if desired, reaction gases could be recirculated until equilibrium was obtained.

Everett ran helium containing carbon monoxide and hydrogen over identical iron specimens. He observed that the rate of carbon deposition was highest for specimens at the entrance to the reactor. The rate of reaction decreased to zero in the direction of flow. This type of behavior is typical of reactions which are inhibited by their products (in this case, carbon dioxide and water).

Everett noticed if he shut down the graphite reconversion furnace, the hydrogen-to-water ratio decreased to a constant value of 10 and carbon deposition ceased. He proposed that the metal catalyst was in the oxide state and as such no

longer acted as a catalyst for carbon deposition.

In an effort to verify his hypothesis, Everett constructed phase diagrams based on the following reaction systems:



Everett assumed carbide and methane formation could be ignored and constructed several phase diagrams. An example of which is shown in Figure 8. Diagrams using both the $P_{\text{H}_2}/P_{\text{H}_2\text{O}}$ and $P_{\text{CO}}/P_{\text{CO}_2}$ ratios were constructed.

In Figure 8 the line labelled K_1 represents the change in equilibrium $P_{\text{H}_2}/P_{\text{H}_2\text{O}}$ ratio as a function of temperature for reaction N. Above K_1 , metallic iron (Fe) is the stable solid iron phase. Below K_1 , iron oxide (Fe_3O_4 or FeO) is the stable solid iron phase. The sharply sloped solid lines represent the equilibrium for reaction E and F at a fixed CO/ H_2 ratio of 1.0 and at a fixed total pressure. To the right of these solid lines carbon deposition is favored, while

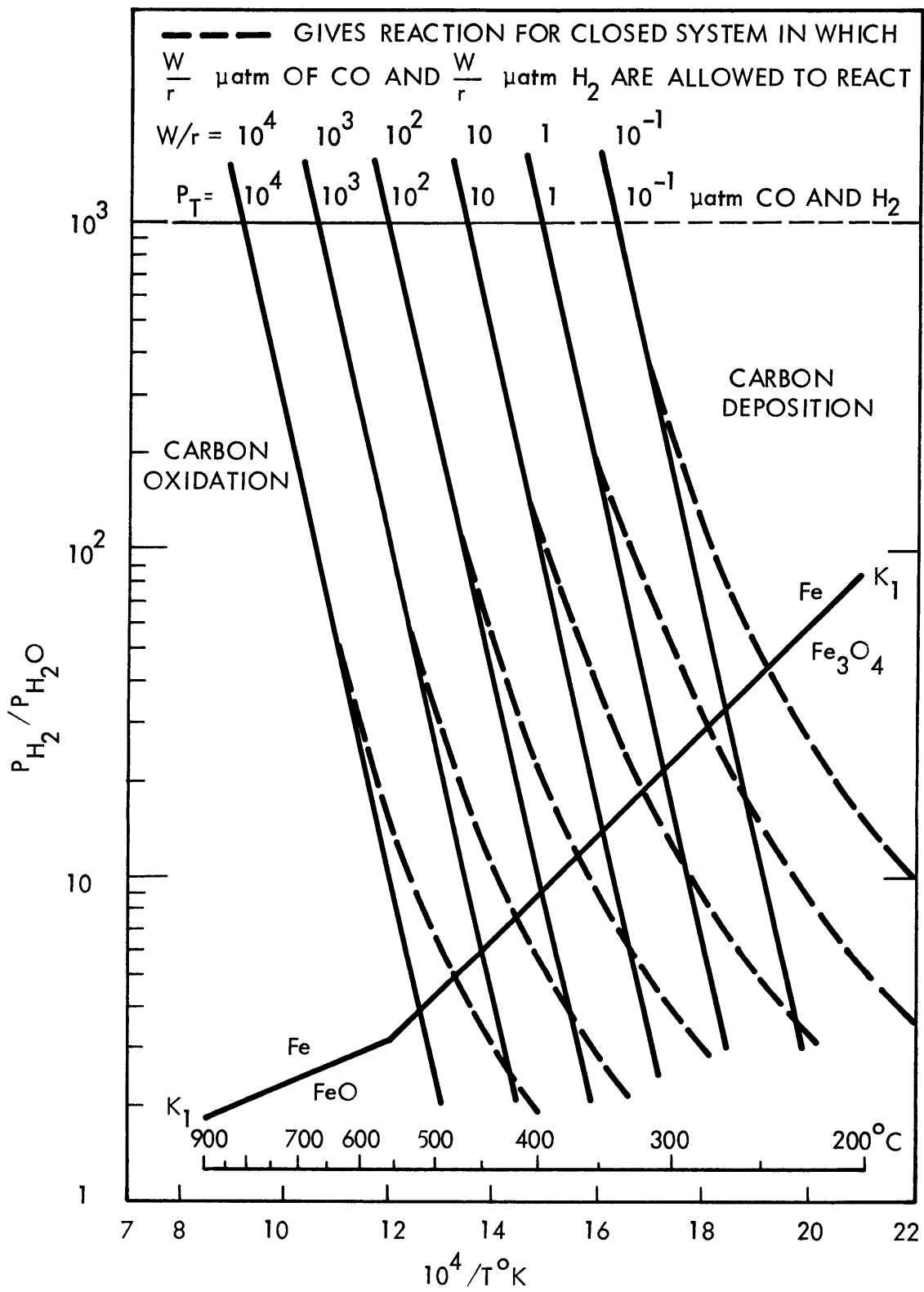


FIGURE 8

$P_{\text{H}_2} / P_{\text{H}_2\text{O}}$ VERSUS $1/T$ K, FOR SYSTEM WHERE H₂/CO RATIO EQUALS 1
 (EVERETT, 1967)

to the left carbon should react (oxidize). If a 1:1 molar mixture of carbon monoxide-hydrogen are initially fed, and the pressure is allowed to vary, then the reaction path for the P_{H_2}/P_{H_2O} ratio is represented by the dashed line.

Everett used these diagrams to analyze his data. He found good agreement with regard to the prediction of limiting hydrogen to water ratios and carbon monoxide to carbon dioxide ratios.

The importance of Everett's research is two-fold: first, it suggested the hydrogen to water or the carbon monoxide to carbon dioxide ratios may be used to control carbon deposition; secondly, he was one of the first to use an equilibrium phase diagram as a means of analyzing a complex reaction system.

Karcher and Glaude (1971) also determined that water was a "strong" carbon deposition inhibitor. In their investigation a steel sample was suspended from an electrobalance into a combustion tube. Normal operating temperature was 823 K. Using argon gas as a carrier median, various amounts of carbon monoxide, hydrogen, and inhibitor were passed over the steel samples, carbon deposition being monitored by changes in the sample mass.

Figure 9 shows some of the data reported. The first plot shows the ratio of carbon deposition rates with and without water versus the parts per million of water by volume in the gas stream. Replotting the data against the ratio of partial pressure of hydrogen-to-water gives the second plot. In Figure 9 the data for two concentrations of hydrogen and

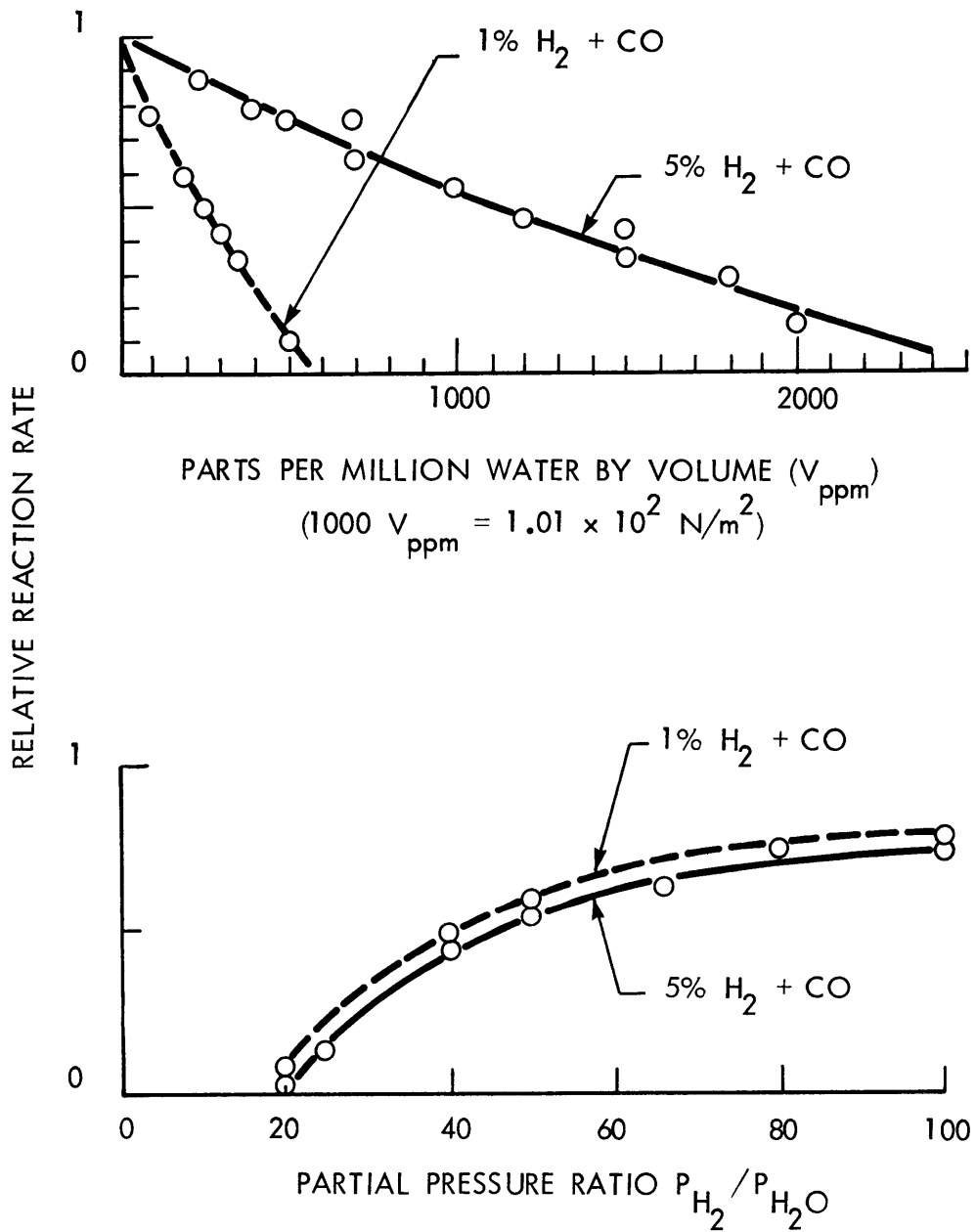


FIGURE 9
EFFECT OF WATER ON CARBON DEPOSITION AT 823 K
(KARCHER and GLAUDE, 1971)

carbon monoxide are reduced to one curve when plotted on these coordinates, thus, it is the ratio P_{H_2}/P_{H_2O} which is important in determining the carbon deposition rate. Also apparent is that at a ratio of P_{H_2}/P_{H_2O} of 20, the rate of carbon formation is zero. The cause of this apparent "equilibrium" is not known. Karcher and Glaude speculated that the phenomenon observed was due to competitive adsorption of the inhibitor on the catalyst sites with respect to the reaction partners carbon monoxide and hydrogen. In view of the fact that under the conditions reported, a P_{H_2}/P_{H_2O} of 20, metallic iron would be the stable equilibrium phase; this seems reasonable.

In a similar manner, Karcher and Glaude found carbon dioxide inhibited carbon formation from gas mixtures of carbon monoxide and hydrogen. However, with carbon dioxide, unlike the results found for water, carbon deposition could never be completely stopped. They speculated that the behavior of carbon dioxide may reflect side reactions such as reaction F. Reaction F would produce water which in turn was the actual inhibitor.

Although possible, the results of Manning (1976) and Everett et al. (1967) indicate that if the carbon dioxide concentration is high enough, carbon deposition will cease.

2.1.2 Reverse Water-Gas Shift Reaction

A detailed study on the reverse water-gas shift reaction was done by Kusner (1962).



The reaction was carried out in a continuous flow vertical reactor, the reactor tube being positioned in an electric furnace which provided heat for the preheating of reaction gases and for the heat of reaction. A reduced iron catalyst was positioned mid-way up the reaction tube on a support grid. Inlet and exit gas compositions were monitored using a gas chromatograph.

Several different iron catalysts were tried and all were found to be active. Normal catalyst preparation involved grinding the iron or iron-oxide powder, pelletizing with a starch binder, followed by drying and firing to burn off the starch binder and to partially sinter the pellets. After final reduction in hydrogen at 1092 K, they were cooled (in hydrogen), crushed, and sieved. The final average particle size was 3 mm, the density 3.07 g/cc, the porosity 61% and the BET surface area 0.11 m²/g. The amount of catalyst used ranged from 100 to 600 g, corresponding to a packed bed height between 2.6 and 15.6 cm. The bed porosity was determined to be 0.46.

Experiments were performed with hydrogen-to-carbon dioxide ratios between 0.5 and 3.0. Normal operating temperature was approximately 922 K. Reynolds numbers, based on particle diameter, varied between 1.7 and 16.0. This corresponded to the laminar flow regime with some excursion into the transition region.

In order to determine the equilibrium composition to be expected from a given inlet hydrogen-to-carbon dioxide ratio, Kusner constructed the phase diagram shown in Figure 10. Fixing the system temperature and pressure, and recognizing that the stoichiometry of the process is such that the amount of carbon monoxide formed equals the amount of water formed, allowed Kusner to construct a process operating line based on the inlet H/C ratio which fixed the system composition. For example, if one fixes the hydrogen to carbon ratio at X, in the given coordinate system, one would follow an operating line equal to 2X (which is fortuitously equal to the inlet H_2/CO_2). When the appropriate equilibrium curve (fixed by temperature, pressure, and C/H) is intercepted, the coordinates for the point give the equilibrium gas composition. The equilibrium solid phase expected was also plotted on this diagram in similar fashion.

Figure 11 is a plot of conversion versus inverse space velocity (time required to process a volume of feed at a given catalyst loading). At a fixed inverse space velocity, conversion was seen to increase with particle Reynolds number. Based on these initial results, Kusner modeled his system as an isothermal, packed bed plug flow reactor under mass transfer control. He assumed, initially, no axial or longitudinal diffusion.

This model, however, did not adequately describe the observed quantitative behavior. Having observed some carbon

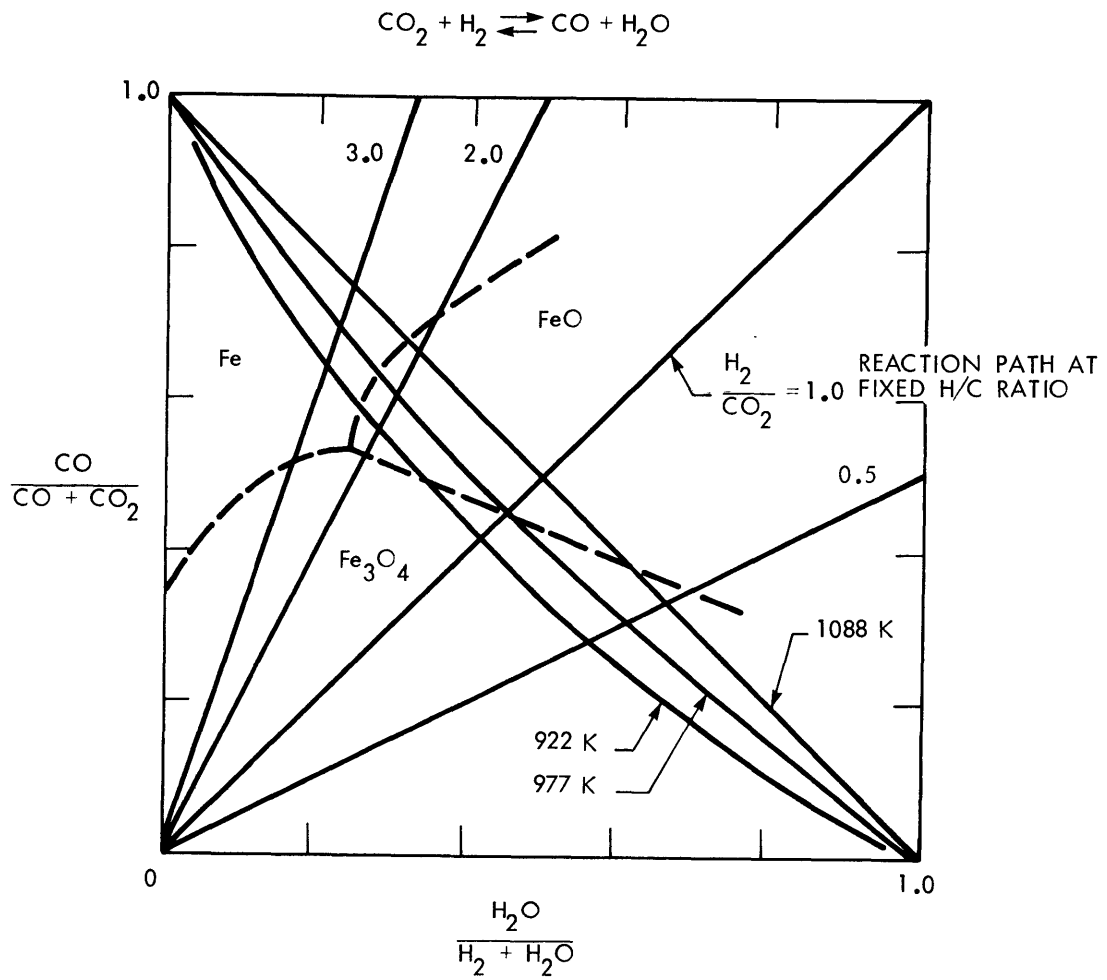


FIGURE 10
 PHASE DIAGRAM FOR THE REVERSE WATER-GAS SHIFT REACTION
 (KUSNER, 1962)

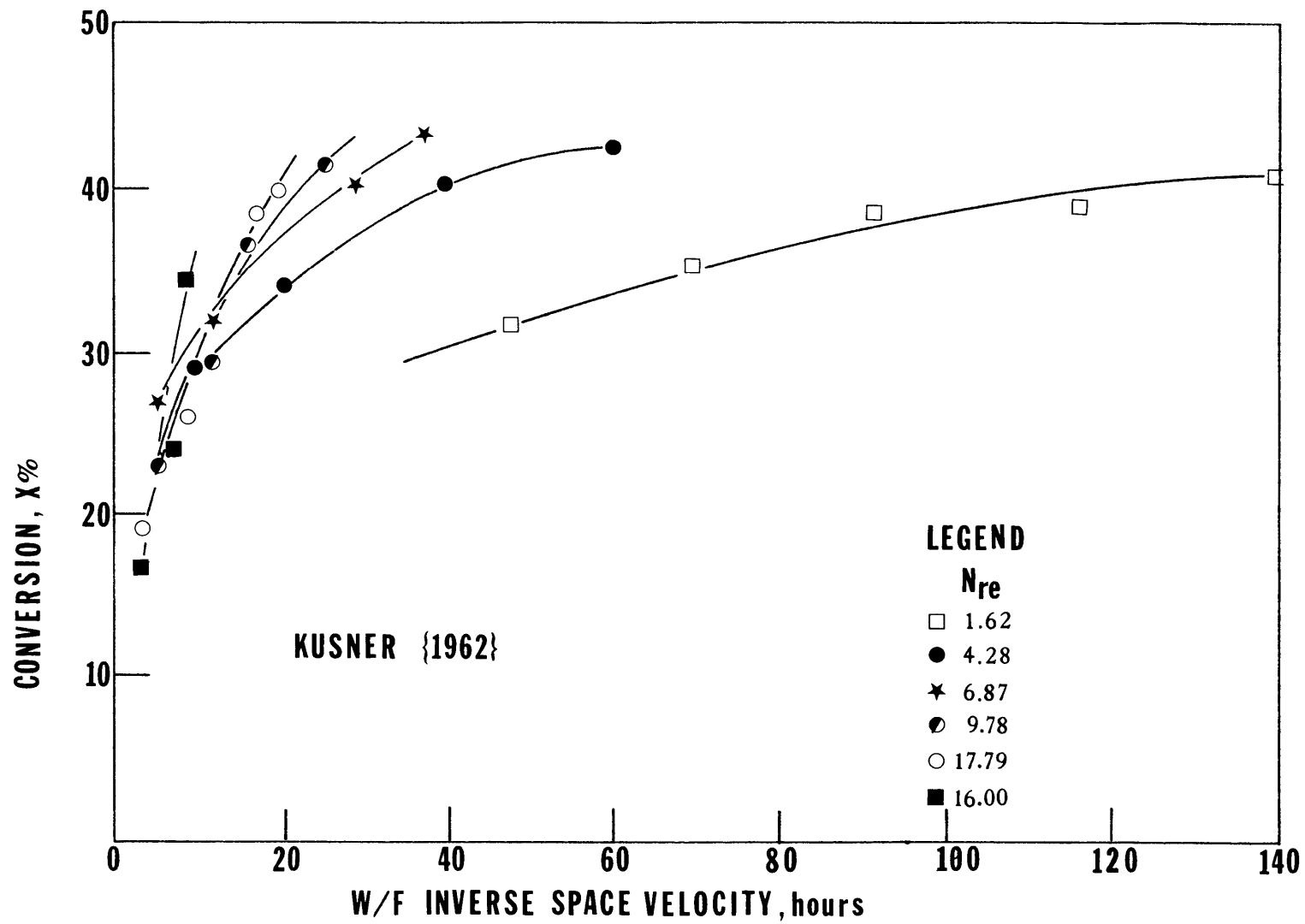


Figure 11 CO_2 Conversion in a Packed Bed For The Reverse Water-Gas Shift Reaction

monoxide and water prior to the catalyst bed, Kusner assumed that axial and longitudinal diffusion may be important.

Kusner next modelled his system as (N) mixed reactors in series and also as a plug flow reactor with axial and longitudinal diffusion. The models both appeared to describe qualitatively the behavior observed but no quantitative comparison could be made.

The importance of Kusner's work lies in the fact that he has conclusively shown that the reverse water-gas shift reaction occurs at a rapid rate in a packed bed at 922 K without carbon formation. Also important is the implication which can be drawn from his phase diagram analysis. That is, both iron and various iron oxides are catalysts for the reaction.

Additional evidence suggesting that iron oxide is a catalyst for the reverse water-gas shift reaction was provided by Barkley et al. (1956).

Barkley studied the reverse water-gas shift reaction over a promoted iron oxide catalyst. The catalyst was in the form of cylindrical pellets 3 mm in diameter and 1.75 mm in length. The bulk density of the catalyst was 300 g/cc. The reactor consisted of a vertical Vycor tube housed in an iron pipe to facilitate temperature distribution. The entire reactor assembly was set in a split type, heavy duty electric combustion furnace with the catalyst being supported on a perforated porcelain dish. The bed height varied between 1.3 and 6.5 cm. Reactor feed rates varied between 0.02 and 0.22 M³/Hr. Hydrogen-to-carbon dioxide ratios varied from 4.0 to

0.25. Samples of both feed and product gases were analyzed for carbon dioxide, carbon monoxide, and hydrogen. The quantity of water vapor formed was obtained by material balance. No other gases were found and no carbon formation was reported.

Figure 12 is a plot from Barkley's data for conversion expressed as lb-moles CO_2 converted per lb-mole CO_2 fed versus inverse space velocity (W/F). Comparing Figure 11 of Kusner's data with Figure 12, one can see that the general shape of the curves are similar. The curve for a hydrogen-to-carbon dioxide ratio of 4.0 in Figure 12 shows an approach to equilibrium of 88%. Similarly, for a hydrogen-to-carbon dioxide ratio of 3.0 in Kusner's study, the approach to equilibrium is approximately 95%. Unlike Kusner, Barkley's apparatus showed no mass transfer limitations. This was determined by varying the amount of catalyst and the feed rate of CO_2 independently while maintaining the W/F (inverse space velocity) constant. No appreciable effect on conversion was observed, thus indicating no mass transfer limitations.

Barkley proposed the following reaction mechanism for the reverse water-gas shift reaction:

- A) A molecule of carbon dioxide is adsorbed on a single active site.
- B) The adsorbed carbon dioxide molecule reacts with hydrogen to form a molecule of adsorbed carbon monoxide and a molecule of water in the gas phase (rate controlling step).
- C) The molecule of carbon monoxide is desorbed.

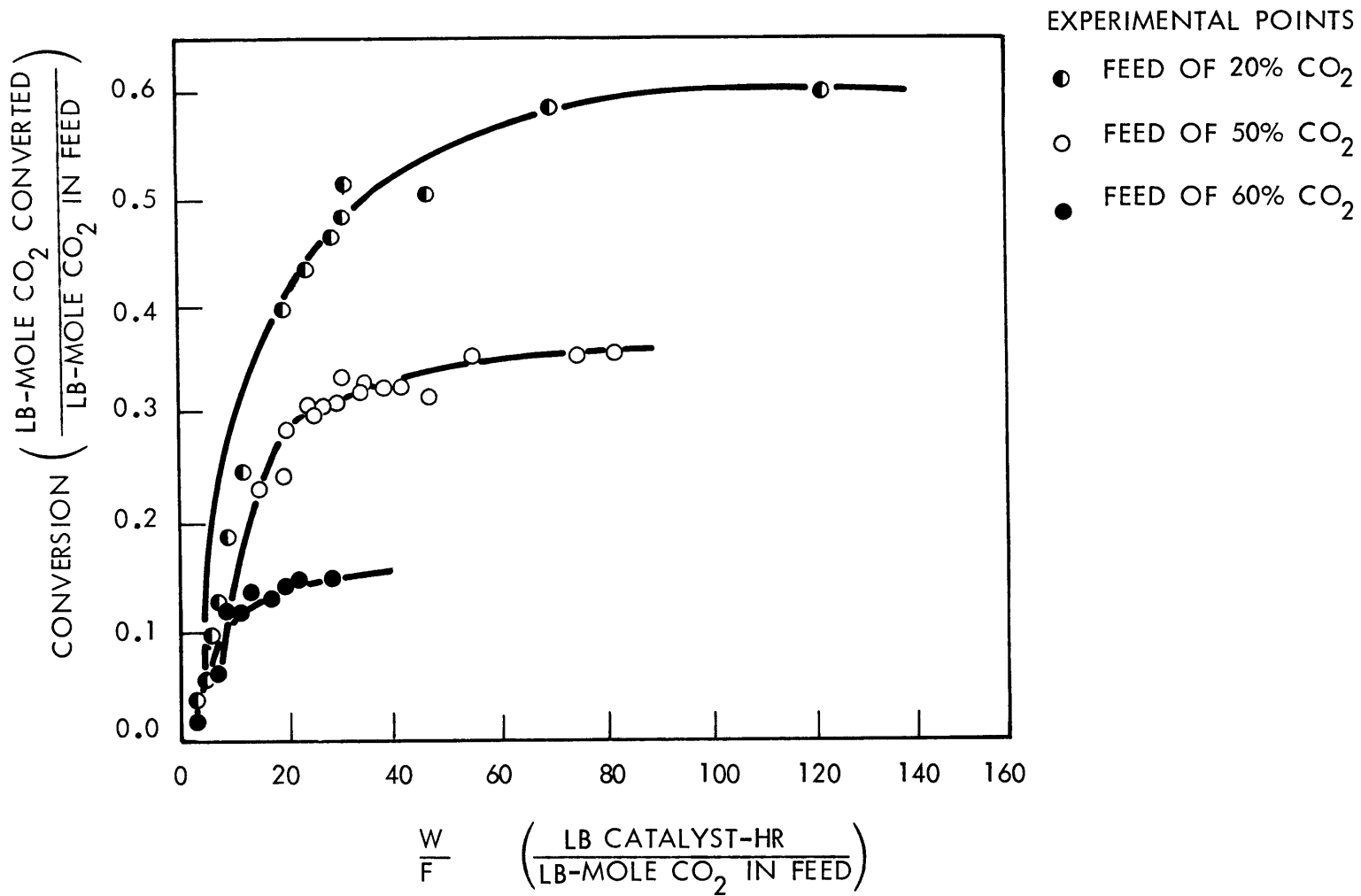


FIGURE 12 CONVERSION VERSUS W/F (BARKLEY et al., 1956)

Using this reaction mechanism and a least square regression analysis, Barkley found the following rate expression fit the obtained data:

$$r = k \left(\frac{P_{\text{CO}_2} P_{\text{H}_2} - P_{\text{CO}} P_{\text{H}_2\text{O}}}{1 + K_A P_{\text{CO}_2} + K_R P_{\text{CO}}} \right) \quad (2)$$

Although this mechanism is intriguing, the fitted constants (i.e., k , K_A , K_R) were not presented, since temperature control in Barkley's reactor was reported poor. Temperatures were reported to fluctuate as much as 50 K.

The importance of this work is that, once again, evidence indicates the reverse water-gas shift reaction occurs readily in a packed bed.

2.1.3 Methane Formation

Methane formation has been found to occur in all Bosch processes. The mechanism by which it forms has never been clearly understood. This lack of understanding is due in no small part to the complexities of the Bosch reaction sequence. The following reactions are suggested as possible methane formers.

2.1.3.1 CH₄-H₂-C System

Browning et al. (1951) studied the carbon-hydrogen-methane system in the presence of an iron catalyst.



The apparatus consisted of a sample container, a trap

for removing water during analysis, a copper oxide trap for conversion of hydrogen to water during analysis, a circulating pump, and a by-pass to allow gases to be either circulated through the catalyst or be by-passed through the analytical train.

Prior to a run, a synthetic ammonia catalyst was reduced in hydrogen at 773 K. The reduced iron catalyst was then carburized to cementite (Fe_3C) in butane at 548 K. Heating to 773 K for 72 hours decomposed the cementite to carbon and iron. The temperature was then lowered to 548 K and hydrogen was again introduced to convert any remaining cementite to iron. During this last step, the carbon formed during cementite decomposition remained relatively unaffected.

Figure 13 represents the data obtained by Browning (solid line), as well as that reported by Rossini (1947) (dotted line). The apparent equilibrium values reported by Browning lay below those reported by Rossini. Comparing the Gibbs energies they calculated with those reported by Rossini, Browning determined the Gibbs energy of formation of the carbon in their system was approximately 300 calories/mole less than the β -graphite used in Rossini's work. Browning approached the "equilibrium" from both the hydrogen and methane-rich sides, obtaining good agreement. However, the conversion of methane when approaching from the methane-rich side was small. Thus, further verification from the methane-rich side would be desirable.

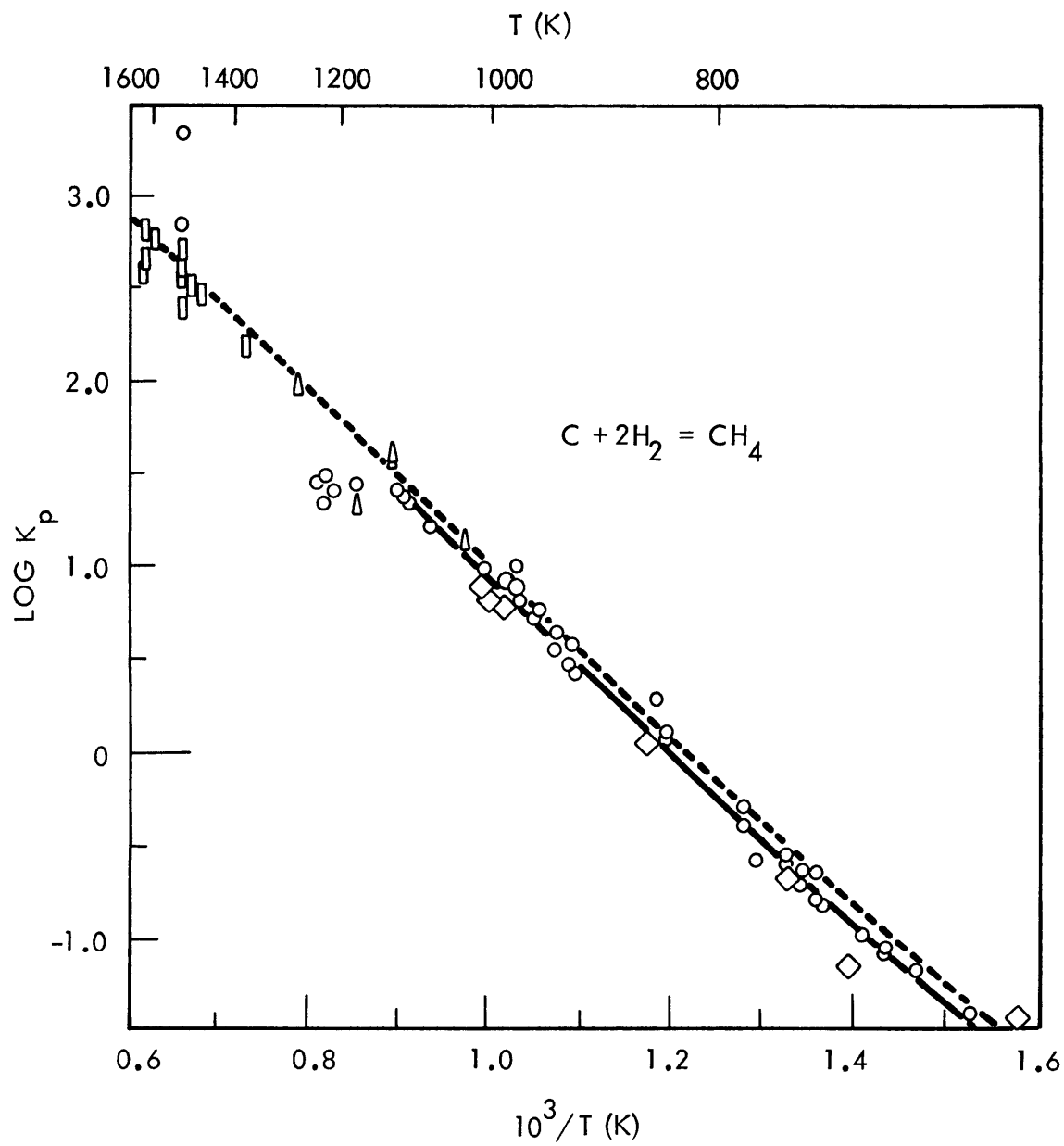


FIGURE 13
EQUILIBRIUM DATA FOR REACTION (G)
(BROWNING and EMMETT, 1951)

Although Figure 13 indicates good agreement with Rossini and others, there remains a problem in the equilibrium study as described.

From a phase rule analysis on Browning's system, n , the number of components, equals 3 (i.e., H_2 , C, CH_4), the temperature was fixed but not pressure. Given two phases exist (i.e., carbon and gas) and the number of independent reactions, R , equals 1. The number of intensive variables left to fix the system is 1 ($f = 3 + 2 - 2 - 2 = 1$).

There are two possible explanations for this, both speculative; one, the experiment as reported is inaccurately described; second, some Fe_3C remains after the final hydrogen reduction. The implication of some Fe_3C remaining is as follows: in this temperature range, reaction G must be faster than reaction H-A in both the forward and reverse direction.



Again, this is speculative and requires data for substantiation.

2.1.3.1.1 Recent MIT Work

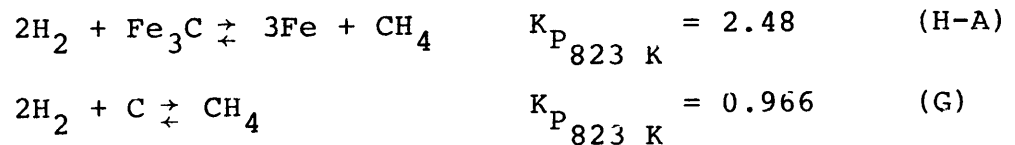
Manning (1975) reacted methane and hydrogen over 250 mg of steel wool catalyst which had had approximately 375 mg of carbon deposited on it. The reaction was carried out at a temperature of 823 K under a total pressure of 1.01×10^5 N/m². The carbon was deposited using a 1:1 molar ratio of carbon monoxide-to-hydrogen at 823 K.

Pure hydrogen was fed to the reactor for one hour; the reactor effluent indicated a hydrogen conversion of 0.24 -

.54%. This corresponded to an outlet concentration of only 0.17 to 0.27% methane. During this same time period, a linear carbon weight loss was observed (Figure 14). As indicated by Figure 14, the carbon weight loss was in excess of 75 mg.

Manning increased the methane content to 75% for two hours. Figure 14 indicates no weight loss or gain was noted. There was no variation in the effluent and this also indicates that no reaction had occurred. Dropping the methane content to 60% for 1.5 hours again produced no apparent reaction. A mixture of 25% methane-75% hydrogen was next run through the reactor for 2.5 hours. Figure 14 indicates 35 mg of carbon were lost. The slope of the 75% hydrogen mixture is less than that for the 100% hydrogen mixture.

Manning suggested the following reactions as the probable sources of methane:



In an attempt to determine the mechanism for carbon formation, pure hydrogen was passed over 450 mg of activated charcoal (BET area $850 \text{ m}^2/\text{g}$) in the absence of iron. No methane was detected in the reactor off-gas. Manning concluded that reaction H-A was therefore responsible for methane formation. He explained the 75 mg carbon loss by assuming that cementite production is occurring faster than its reduction (only 17 mg of carbon would have completely carbided the catalyst.)

This hypothesis is suspect because other established

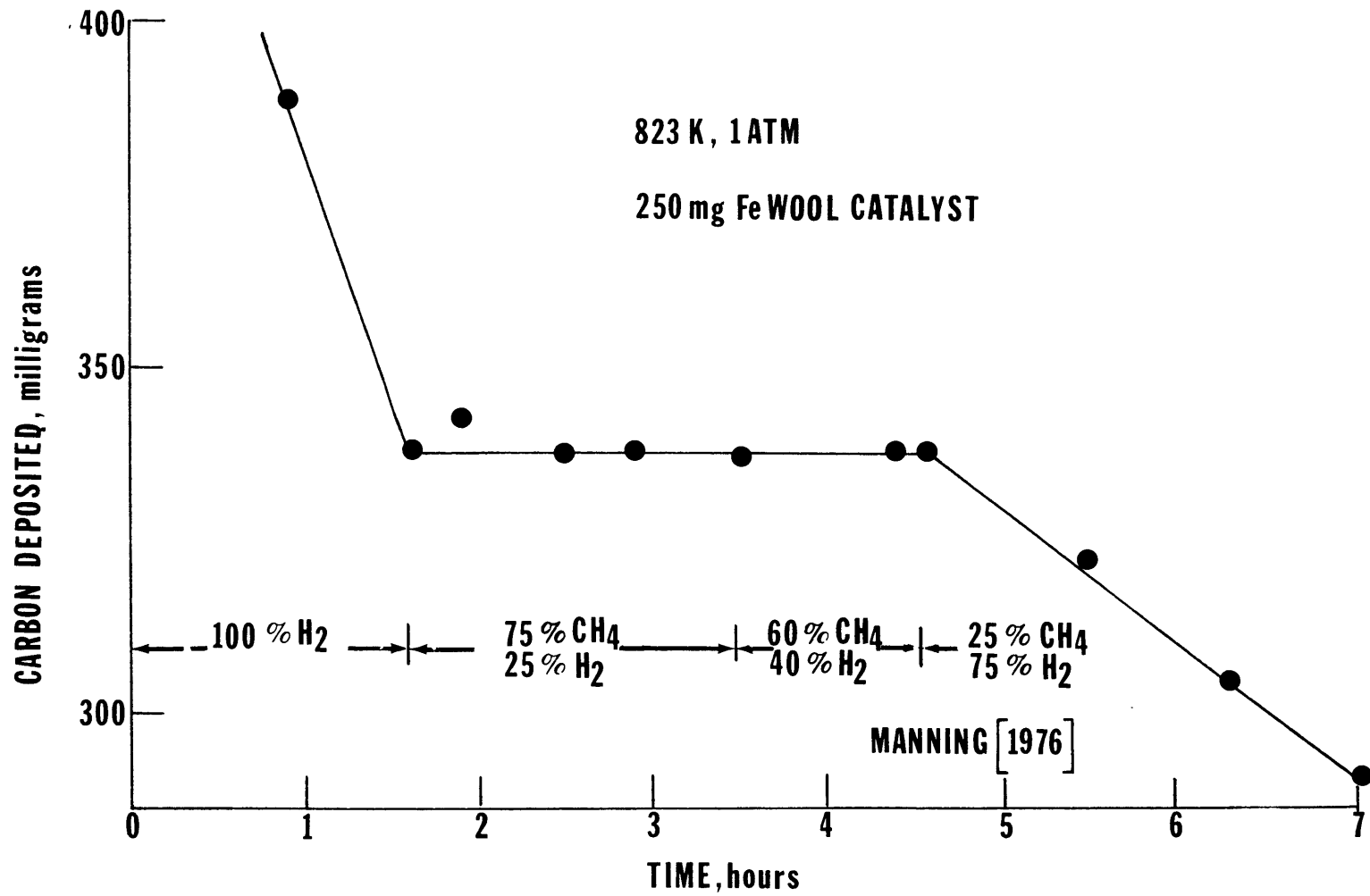


Figure 14 Carbon Deposition From CH₄-H₂ Mixtures

methane catalysts do not form carbides in the temperature range where they are known to catalyze methane formation. Thus, a carbide species as an intermediate in methane formation is dubious.

From a thermodynamic point of view reaction H-A and G would proceed to the right for the runs where the hydrogen concentrations were 100% and 75%. For the runs of 25% and 40% hydrogen, reaction G would be expected to deposit carbon while reaction H-A would be expected to form cementite (Fe_3C).

The implication is that methane decomposition may be inhibited by cementite formation. This, however, is speculative and more data are needed before a definitive statement can be made.

2.1.3.2 $\text{CH}_4\text{-H}_2\text{-Fe}_2\text{C-Fe}$, $\text{CH}_4\text{-H}_2\text{-Fe}_3\text{C-Fe}$ Systems

Browning et al. (1950) also studied the equilibrium represented by reactions H-B and H-A.



Hagg carbide (Fe_2C) was prepared by reducing an iron synthetic ammonia catalyst in hydrogen at 773 K prior to carbiding. The carbiding gas was either carbon monoxide, butane, or methane. Carbiding was done at 473-573 K; the amount and type(s) of carbide formed was determined by x-ray diffraction patterns. The surface area of the catalyst was determined by standard BET methods and was found to be $17 \text{ m}^2/\text{g}$.

Cementite (Fe_3C) was prepared by heating Hagg carbide to 748-773 K for three hours. Browning indicated this treatment caused complete disappearance of Fe_2C lines and the appearance of Fe_3C lines in an x-ray diffraction pattern. Figure 15 is a plot of Browning's data, where $\log K_p$ is defined as

$$K_p = \frac{(X_{\text{H}_2})^2}{(X_{\text{CH}_4})} \quad (3)$$

Curves A, B, and C are plots of the best values of the "equilibrium" data for the systems $\text{C}_\beta\text{-CH}_4\text{-H}_2$ (as given by Rossini {1947}), $\text{Fe}_3\text{C-Fe-H}_2\text{-CH}_4$, and $\text{Fe}_2\text{C-Fe-H}_2\text{-CH}_4$ (Browning et al.), respectively. Figure 15 shows that, below approximately 670 K, curves B and C follow linear behavior. However, above 670 K the data appears to lie halfway between curve A and curve B. The implication of these data is that below 670 K reaction H-A and H-B are kinetically more favorable than reaction G. Above 670 K, however, the rate of reaction G becomes more significant.

A phase rule analysis on the $\text{H}_2\text{-CH}_4\text{-Fe}_2\text{C-Fe}$ and $\text{H}_2\text{-CH}_4\text{-Fe}_3\text{C-Fe}$ systems indicates only one intensive variable need be fixed to completely specify the equilibrium state. From Figure 15 and from the fact that Browning occasionally reported carbon diffraction lines when preparing carbides, one can speculate that some carbon is present in all the "equilibrium" systems measured. This explanation will satisfy the phase rule analysis.

One note of caution is necessary in evaluating Browning

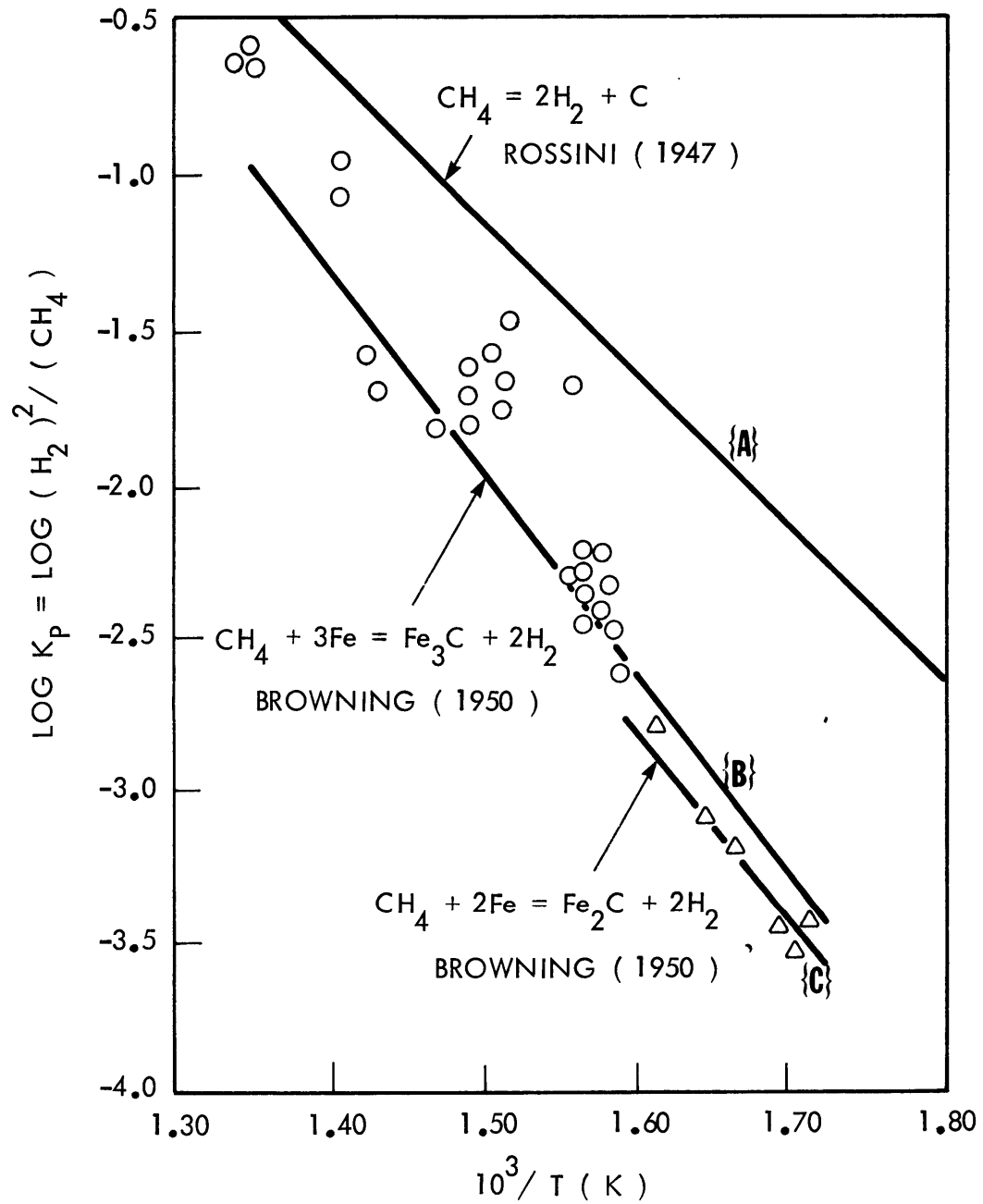


FIGURE 15
 EQUILIBRIUM STUDIES OF $\text{CH}_4\text{-H}_2\text{-Fe-Fe}_2\text{C}$ AND
 $\text{CH}_4\text{-H}_2\text{-Fe-Fe}_3\text{C}$ SYSTEMS

et al.'s data: the methane content in the systems was obtained by difference, that is, no true material balance was performed on the system; the hydrogen content was determined after it was converted to water.

2.1.3.3 H₂-CO-CH₄-H₂O System

The reaction of carbon monoxide and hydrogen to give methane and water is termed methanation, reaction K.



Reaction K is normally run over a nickel catalyst at 523-723 K (Vanice, 1976). However, nickel is not the only active catalyst for methanation. The following metals were described by Vanice (1976) as good methanation catalysts. In decreasing order of activity: Ru, Ir, Rh, Ni, Co, Os, Pt, Fe, and Pd. In general, methanation catalysts deactivate due to sulfur compounds, sintering, and carbide formation.

2.2 Metallurgical Considerations

The gas phase reactions which comprise the Bosch process are seen to be catalyzed by transition metals. The literature indicates that the solid phase and/or phases of the catalyst may change during reaction. The catalytic effect of these new phases are not clear, however, indications are that iron oxides and/or iron carbides may not be catalysts for carbon deposition, carbon deposition being an integral part of the Bosch sequence. Thus, a thorough knowledge of the thermodynamics and kinetics of oxide and carbide formation is necessary.

2.2.1 Thermodynamics of the Iron-Iron Oxide System

Figure 16 is a phase diagram representing the iron-iron oxide system. The solid phase composition is fixed, at a specific temperature, by adjusting the oxygen activity* to the desired value. The oxygen activity in the system can be maintained at a desired value by fixing the hydrogen-to-water ratio according to reaction B or, alternatively, by the carbon monoxide-to-carbon dioxide ratio, reaction L.



If carbon monoxide, carbon dioxide, hydrogen, water, and oxygen are all present in the equilibrium gas mixture, then reaction F can be



used in conjunction with reaction B or reaction L to fix the solid phase composition.

It is of interest to note the wustite phase field is of variable composition. That is, at a set temperature, the activity of oxygen varies across the phase field. Also, the gentle slope of the wustite/iron, wustite/magnetic phase boundaries must be accounted for in determining oxygen partial pressure above a desired equilibrium phase. A further compli-

* Note: $a_{\text{O}_2} = \frac{\hat{f}_{\text{O}_2}}{f_{\text{O}_2}^\circ}$ $f_{\text{O}_2}^\circ \rightarrow 1$ and $f_{\text{O}_2} \cong P_{\text{O}_2}$

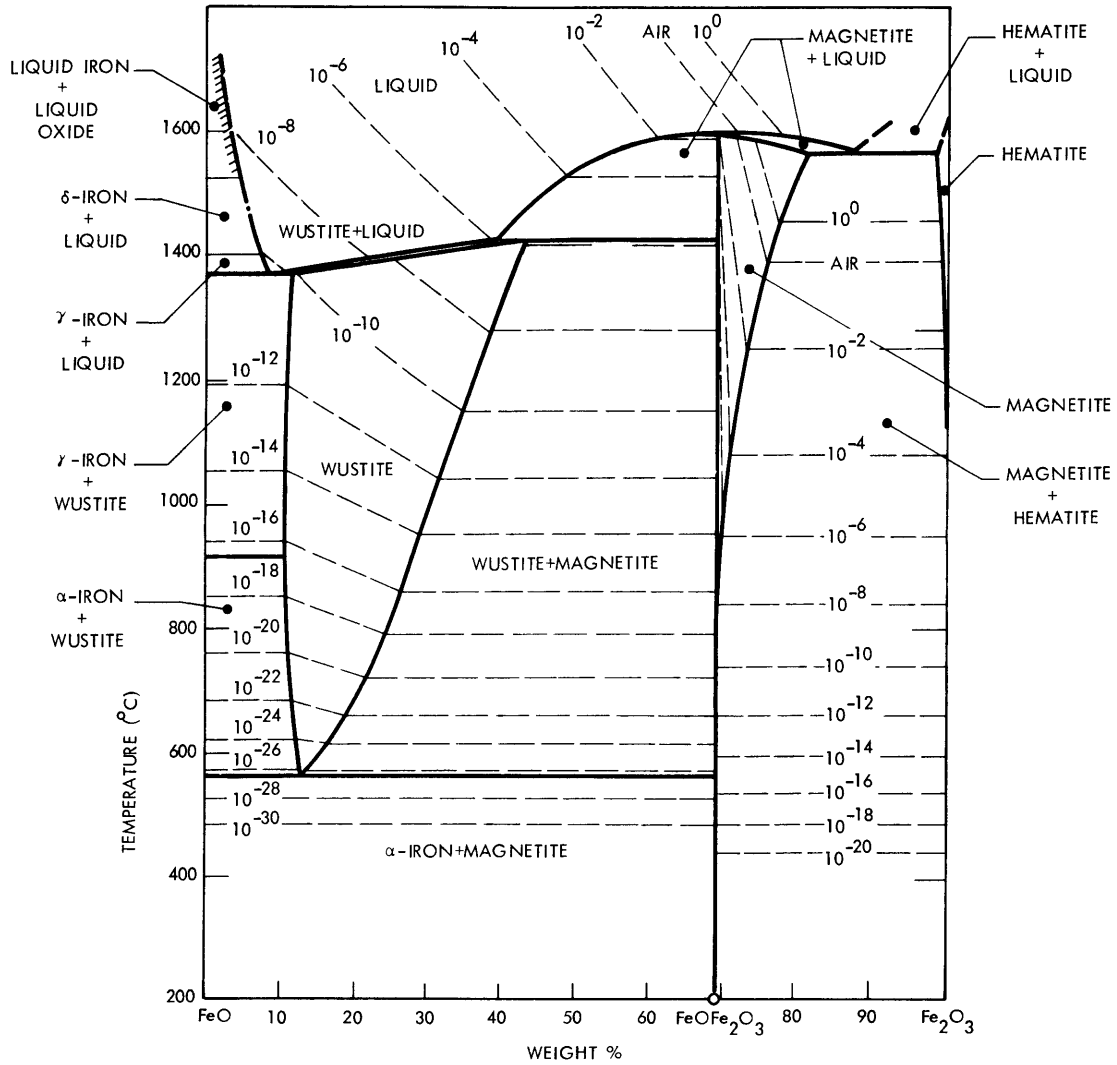


FIGURE 16
 IRON-IRON OXIDE PHASE DIAGRAM
 (MUAN and OSBORN, 1965)

cation, illustrated by Figure 16, is that the stable oxide phase in equilibrium with alpha-iron is a function of temperature. Below 833-843 K, magnetite (Fe_3O_4) is the stable oxide phase; while above 833-843 K, wustite is the expected oxide phase.

2.2.2 Solid-Gas Reactions - Oxidation and Reduction of Iron and Iron Oxides

The complexity of the iron-iron oxide phase diagram is reflected in the voluminous and often confusing literature concerning iron oxidation and/or iron oxide(s) reduction. Additional complications arise from the fact that structural changes occur during reaction and these affect the kinetics. There are, however, three major mechanisms which have been successful in explaining observed data both qualitatively and quantitatively. Each describes the kinetics depending upon initial structure and temperature. The three proposed mechanisms are as follows:

- A) Gas-solid reactions combined with solid-state diffusion
- B) A shrinking core model which assumes a nonporous unreacted core and a porous product layer
- C) A shrinking core model for a porous particle assuming diffusion and/or mixed diffusion-interfacial control

A brief description of each mechanism is felt necessary to insure a better understanding of the process or processes

that will occur on the Bosch catalyst.

2.2.2.1 Gas-Solid Reaction and Solid State Diffusion

When the solid product layer is nonporous, forming a dense layer around the reactant solid, the transport of matter across the product layer occurs by solid state diffusion. Solid state diffusion has been shown to control metal oxide reduction and metal oxidation under certain conditions.

Ladler and Komack (1966) studied the partial reduction of wustite with hydrogen.



They interpreted their results based upon a model which assumed the overall rate was controlled by chemical reaction at the solid-gas interface and solid state diffusion of iron within the oxide. Figure 17 illustrates the proposed mechanism.

Assuming the solid to be a slab of thickness, (L), they derived the transport equation for the unidirectional diffusion of iron. Applying the appropriate boundary conditions they obtained the iron concentration profile as a function of thickness (y) and time (t). This is shown in Figure 17. As additional proof of the validity of the model, the iron concentration profile was used to derive an expression for the overall conversion (\underline{X}) as a function of reduced time ($4D_s/tL^2$).

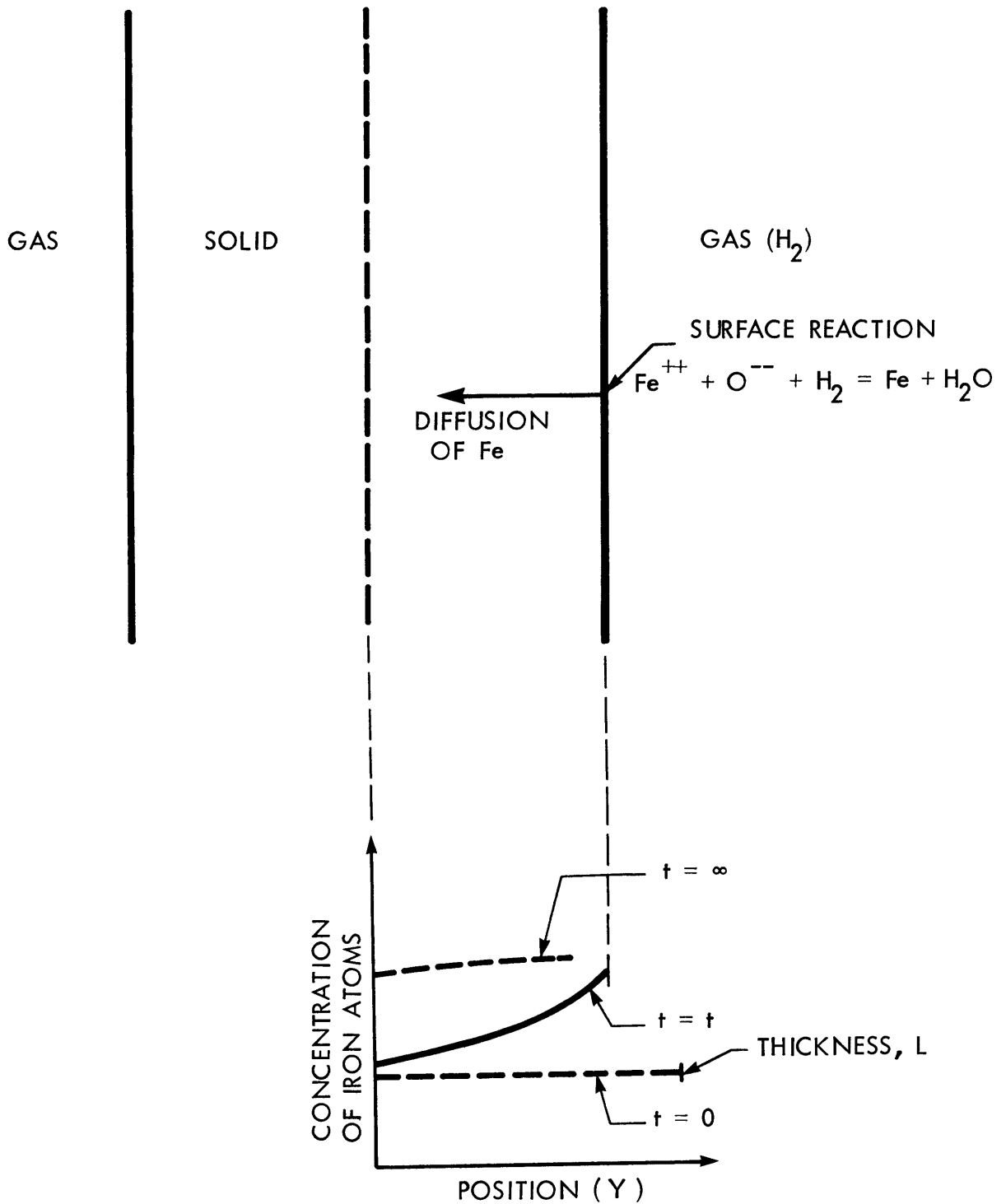


FIGURE 17
 COMBINED SURFACE REACTION WITH SOLID STATE DIFFUSION
 (SZEKELY et al., 1976)

$$\underline{x} = 1 \sum_{n=1}^{\infty} 2 \exp \left(-4 \frac{D_s t}{L^2} \beta_n^2 \right) \left(\beta_n^2 + \frac{2D_s \beta_n^2}{k L} + \frac{2D_s \beta_n^2}{k L} \right)^{-1} \quad (4)$$

Landler and Komack then used this to determine the diffusivity of iron and found it to be reasonable and consistent.

This proposed model is slow relative to the other two mechanisms and would require a high activation energy. Edstrom (1955) has demonstrated that the rate of iron ion mobility in wustite varies over one hundred fold between 973-1273 K. Assuming a rate expression of the form

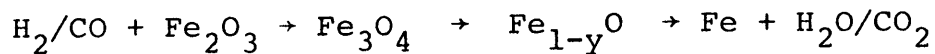
$$R = k e^{-E/RT} \quad (5)$$

one can calculate an $E/R = 1.90 \times 10^4$ K. Extrapolating to 900 K, the relative rate is $0.007 R_{1273 \text{ K}}$. With an iron mobility this low, iron oxidation or reduction would be very slow in the temperature range of interest in the Bosch process (≤ 900 K).

If solid state diffusion is controlling oxidation or reduction, the weight gain or loss during reaction is initially more rapid. As the outer surface is reduced (oxidized), a nonporous shell is formed, the thickness of which increases slowly with time. Frequently, on reaching a critical thickness, reaction essentially stops (i.e., diffusion distance is large). Thus, if seen in a metallograph, one would expect a single, usually thin nonporous oxide or iron layer.

2.2.2.2 Nonporous Shrinking Core Model With Porous
Product Layer

Spitzer et al. (1966) developed a shrinking core model for the reduction of dense hematite (Fe_2O_3) spheres. It was assumed that iron oxide reduction went through the following steps:



All oxygen removed was assumed to occur at the $\text{Fe}/\text{Fe}_{1-y}\text{O}$ interface, while reduction of the intermediate oxides occurred by solid state diffusion. The oxygen density at the core was assumed constant.

They proposed the following steps to occur in series:

- I. Transport of gaseous reactants from the bulk to the outer surface of the particle
- II. Diffusion of reactant through the porous iron layer to the surface of the unreacted core (the $\text{Fe}/\text{Fe}_{1-y}\text{O}$ interface)
- III. Chemical reaction of the gaseous reactant with solid oxide to form gaseous product
- IV. Outward diffusion of the gaseous product through the product shell (iron layer)
- V. Transfer of the product species from the outer surface to the bulk gas stream

An illustration of the proposed model is shown in Figure 18 along with the electrical circuit analogy used in the

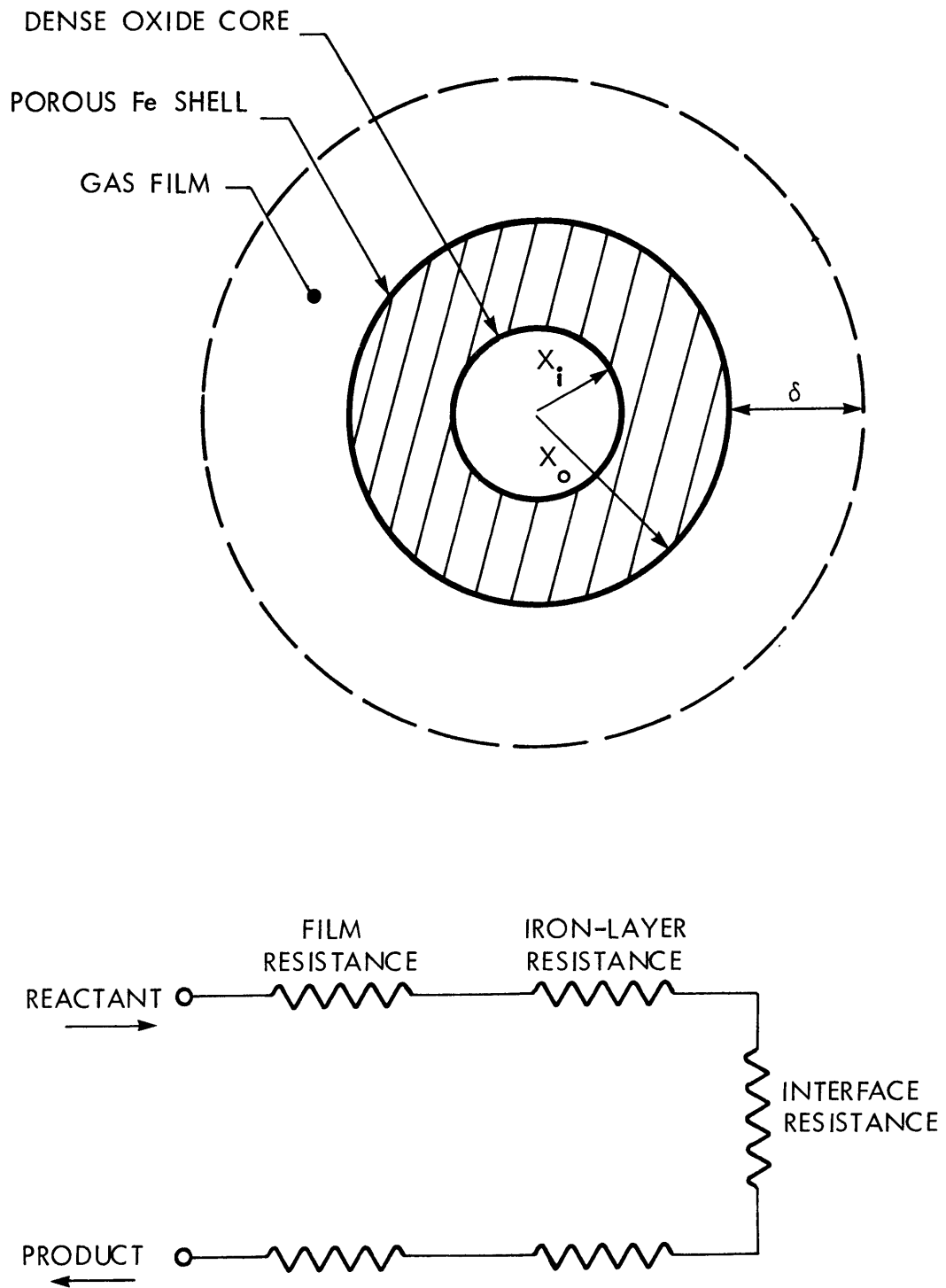


FIGURE 18
 NON-POROUS SHRINKING CORE MODEL WITH POROUS PRODUCT LAYER
 (SPITZER et al., 1966)

mathematical derivation.

The gas film, shell layer and interface resistances* were combined in Ohm's law fashion, and an equation describing the rate of change of core radius was generated.

$$\frac{dX_i}{d\theta} = \frac{-k_{ov}}{R_g TCO} \left[P_A^{(b)} - \frac{P_B^{(b)}}{K_e} \right] \quad (6)$$

where:

$$k_{ov} = \frac{1}{\frac{X_i^2}{X_{O\alpha}^2} + \frac{X_i(X_O - X_i)}{X_{O\beta}} + \frac{1}{\bar{k}_r}} \quad (7)$$

The terms in the denominator were identified as, respectively, the gas film resistance, the shell layer resistance, and the interface resistance.

This model predicted well the linear rate of advance observed by McKewan (1962). Adopting a Langmuir-Hinshelwood rate expression, the model was also able to predict the observed behavior of rate as a function of hydrogen pressure. Since the Langmuir-Hinshelwood formulation requires that strongly adsorbed species be present in the denominator, Equation 8 implies that water is more strongly adsorbed than hydrogen on iron and iron oxide.

* See Appendix 7-1.

$$r_o = \frac{\frac{k_r}{R_g T} \left(P_{H_2}^{(b)} - \frac{P_{H_2O}^{(b)}}{K_e} \right)}{\left(1 + \frac{K_a^{H_2O}}{R_g T} P_{H_2O}^{(i)} \right)} \quad (8)$$

If the solid-gas reactions on the Bosch catalyst followed this mechanism, a metallograph would show a single linearly advancing interface of porous material.

2.2.2.3 Shrinking Core Model for a Porous Particle

The equilibrium diagram in Figure 16 indicates that, depending on the kinetics of individual phase changes, as many as three interfaces could conceivably exist. In recognition of this problem, Spitzer et al. (1966) extended their shrinking core model to the reduction of porous hematite spheres. The situation is illustrated in Figure 19. The electrical circuit diagram was again used as the model for the mathematical formulation. The formulation, as before, followed an Ohm's law analysis.

Each interface was proposed to move at a rate determined by the specific rate constant for the surface reaction and by the gas composition present at the interface. For reversible, first-order kinetics, there were three equations generated of the form:

$$\frac{dx_i(t)}{d\theta} = \frac{-k_r(t)}{R_g T C_o(s) \theta(t)} \left(P_A(t) - \frac{P_B(t)}{K_e(t)} \right) \left| \begin{array}{c} s \rightarrow t \\ w \rightarrow Fe \\ m \rightarrow w \\ h \rightarrow m \end{array} \right| \quad (9)$$

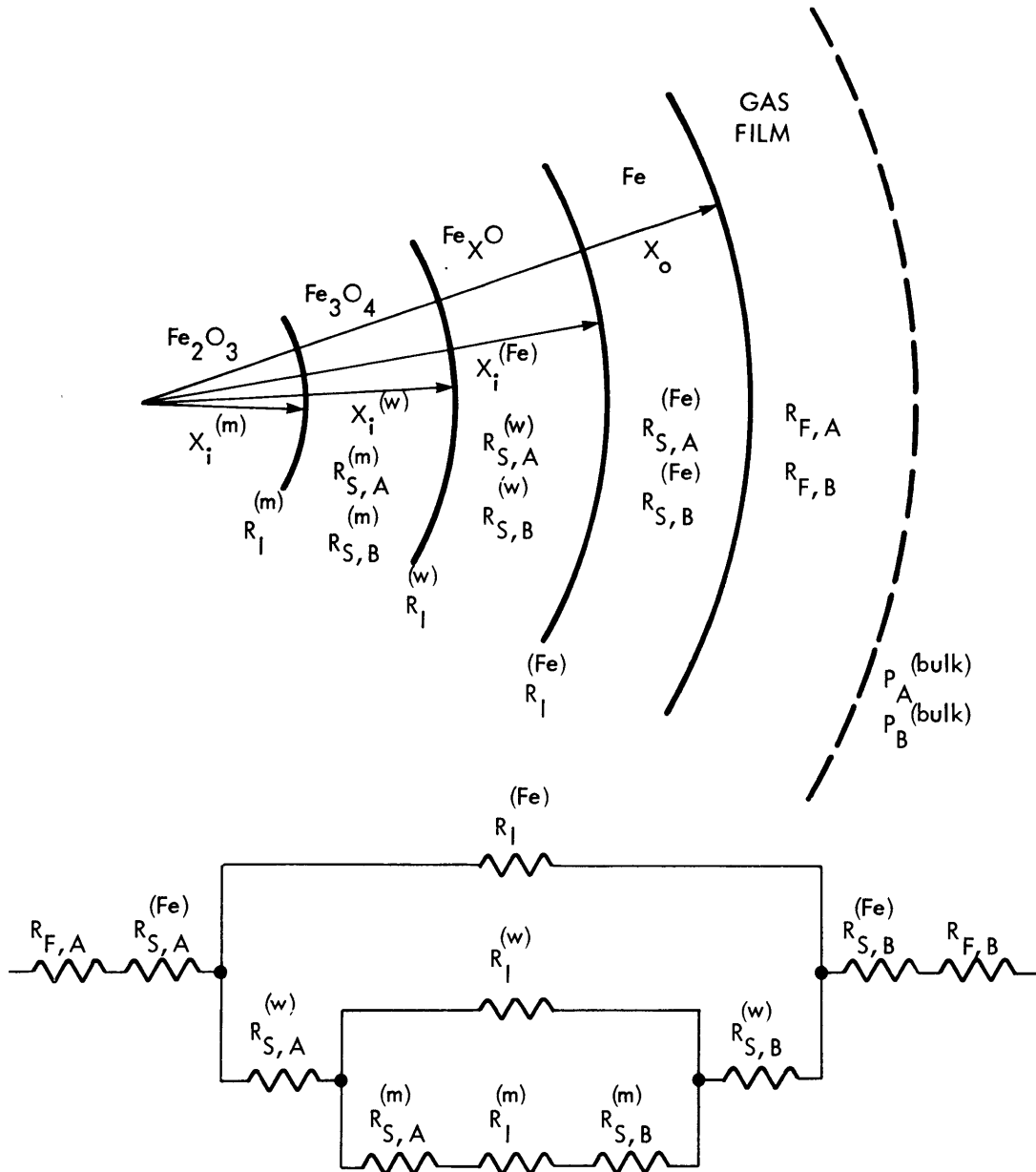


FIGURE 19
 SHRINKING CORE MODEL FOR POROUS PARTICLE
 (SPITZER et al., 1966)

where s is a mnemonic index for the reactant oxide, t is a mnemonic index for the product, and the $s \rightarrow t$ box signifies that equation 9 applies to all three interfaces.

Specific rate constants for each interface were determined by fitting the model to the data of McKewan (1964). An interesting parametric study was then made using this model.

Figure 20 is a computer generated plot of core radii (cm) versus reaction time, θ (min) for hematite reduction in pure hydrogen at 1173 K and $1.01 \times 10^5 \text{ N/m}^2$. Similar to the dense pellet model, a linear advance of the reaction interface was predicted. However, according to this model, thin layers of magnetite and wustite are also predicted.

The interesting thing about this model is that it predicts a substantial increase in wustite layer thickness with addition of water, while the magnetite layer thickness does not increase greatly. If the $P_{\text{H}_2}/P_{\text{H}_2\text{O}}$ is adjusted to a value only slightly above the equilibrium $P_{\text{H}_2}/P_{\text{H}_2\text{O}}$ for wustite, Figure 21 is generated. Figure 21 indicates that for the first ten minutes no iron will be formed. Again, a thick wustite layer was predicted.

This type of system response results because the $P_{\text{H}_2}/P_{\text{H}_2\text{O}}$ ratio is much smaller for the $\alpha\text{-Fe}/\text{Fe}_{1-y}\text{O}$ equilibrium than for either the $\text{Fe}_3\text{O}_4/\text{Fe}_{1-y}\text{O}$ or the $\text{Fe}_2\text{O}_3/\text{Fe}_3\text{O}_4$ equilibria. Therefore, at a given $P_{\text{H}_2}/P_{\text{H}_2\text{O}}$ the driving force for reaction (i.e., distance from equilibrium) is greater for reduction of the Fe_2O_3 , and Fe_3O_4 phases than for Fe_{1-y}O .

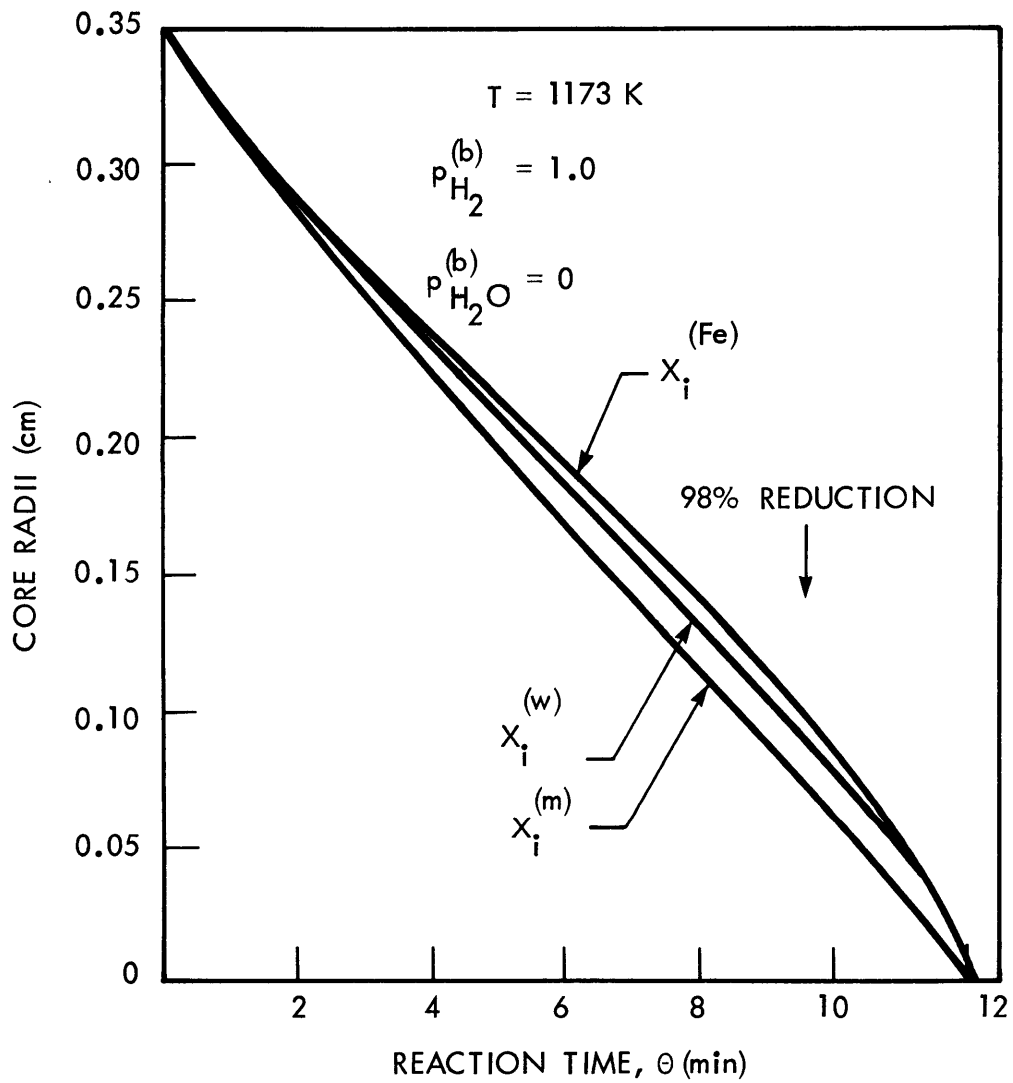


FIGURE 20
 CORE RADII MOVEMENTS COMPUTED FOR POROUS PELLET MODEL
 (SPITZER et al., 1966)

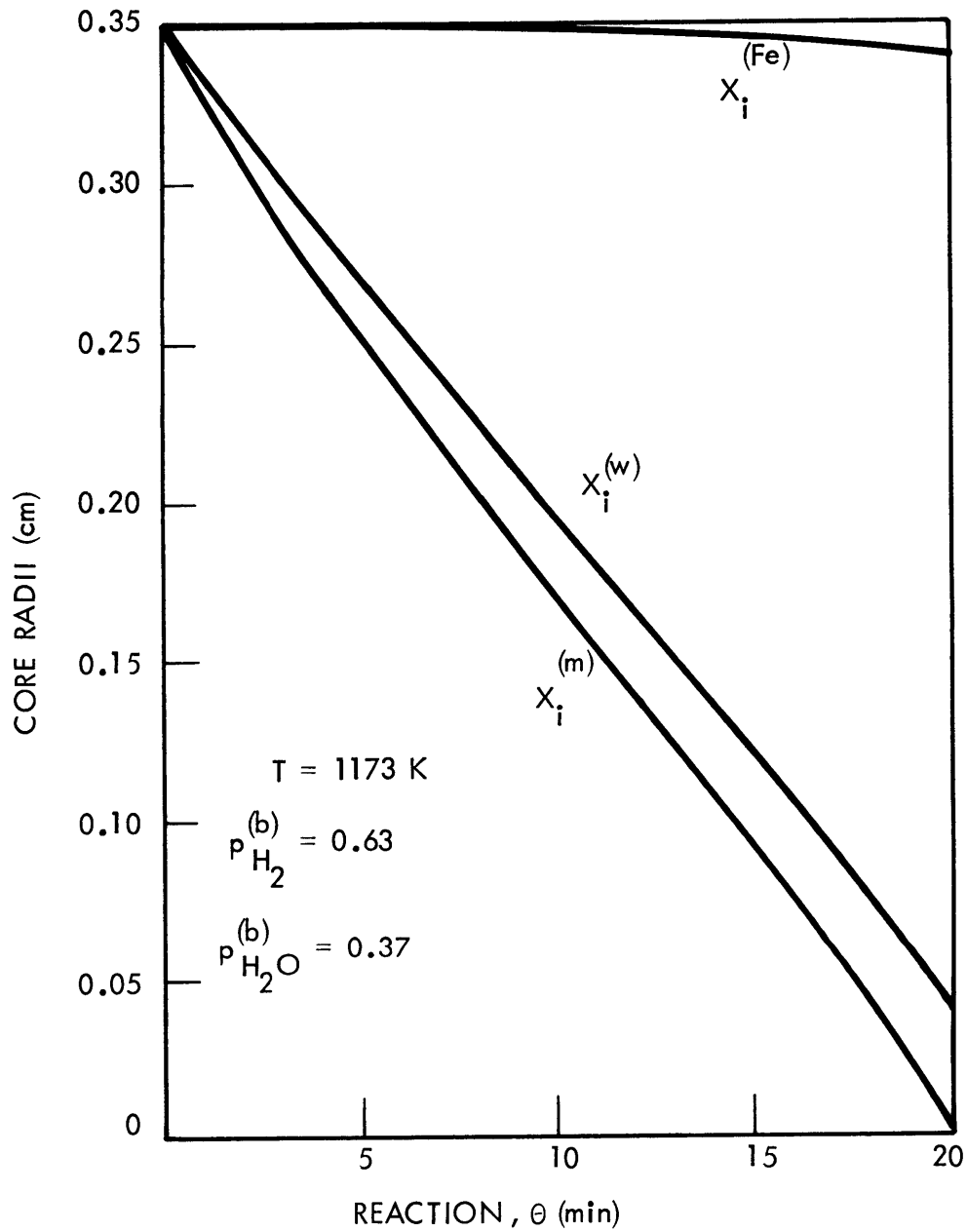


FIGURE 21
 DELAYED ADVANCE OF IRON/WUSTITE INTERFACE POROUS PELLET MODEL
 (SPITZER et al., 1966)

Also, since the various oxide phases are considered porous, these "faster" reactions produce water which maintains the local water concentration higher at the $\text{Fe}_{1-y}\text{O}/\alpha\text{-Fe}$ interface than in the bulk gas phase. This in turn suppresses the reaction rate further for the conversion of Fe_{1-y}O to $\alpha\text{-Fe}$.

A comparison between morphologies predicted from the dense pellet model to that for the porous pellet model (Figure 22) shows that complete reduction is 1.5 faster in the porous pellet than in the dense pellet. Also, the dense pellet model predicts, at the same $P_{\text{H}_2}/P_{\text{H}_2\text{O}}$, a dense wustite layer. The porous pellet model predicts a substantially thicker porous layer.

From the kinetic analysis presented above, the following conclusions can be drawn. Depending on whether the steel wool catalyst behaves as a dense or porous solid and, depending on the mobility of iron ions at the temperature of interest, different morphologies can be predicted. If solid diffusion is important, the rate of oxidation and reduction will be substantially slower. Under certain conditions metastable multiple oxide phases are possible. The number and thickness depend on the temperature, past history (i.e., porous, non-porous), magnitude of the oxygen chemical potential, time at a fixed oxygen activity, and structural changes which may occur during reaction or during oxidation.

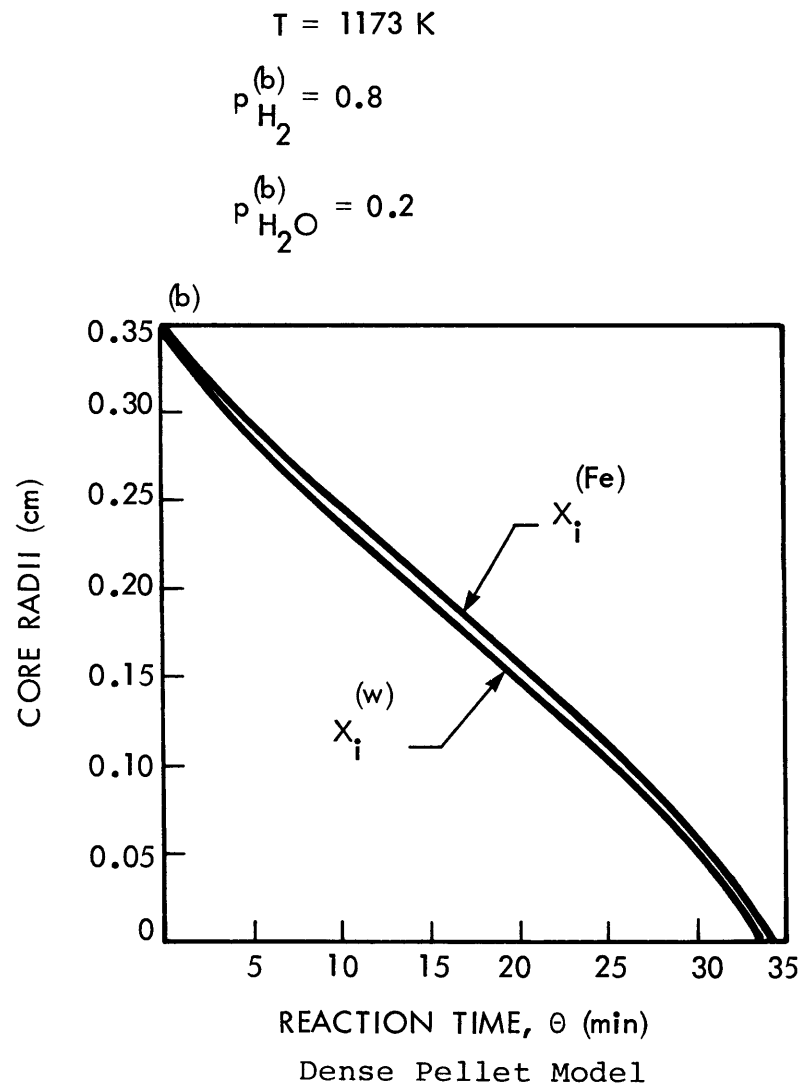
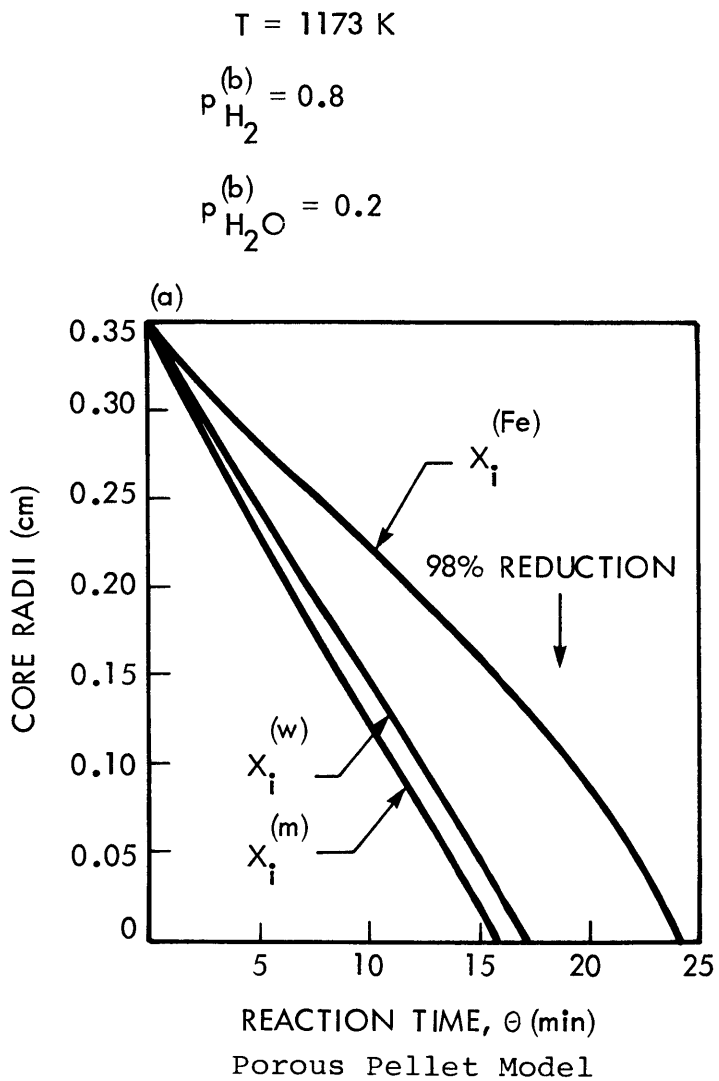


FIGURE 22 COMPARISON OF WUSTITE LAYER THICKNESS BETWEEN POROUS PELLET MODEL AND DENSE PELLET MODEL (SPITZER et al., 1966)

2.2.3 Iron-Carbon System

2.2.3.1 Thermodynamics of the Iron Carbon-System

Figure 23 (1958) is the standard phase diagram for the Fe-C_β system in the form of a double line diagram. The curves for the metastable Fe-Fe₃C system are drawn as solid lines. Those for the stable Fe-C_β are represented by dashed lines. Figure 23 is not a true equilibrium phase diagram because Fe₃C is not an equilibrium phase. Graphite is the stable carbon phase and cementite will eventually decompose to graphite. In ordinary steel, however, graphite precipitation is virtually never observed. Iron super saturated with carbon will precipitate cementite, not graphite. This is due to the fact that nucleation of cementite in iron occurs much more readily. Thus, when carbon is precipitated from solid solution of alpha (bcc) or gamma (fcc) iron, the resulting phase is almost always cementite (Reed-Hill, 1973).

Other iron carbides can form when iron solutions are supersaturated with carbon, but these are less stable than cementite (Cahn, 1965) and form only under special conditions. The most frequently observed of these less stable carbides is Hagg carbide. Hagg originally reported its measurement in 1932. Later, Jack (1946) and Jack and Wild (1966) reported its actual structure to be Fe₅C₂. Cohn et al. (1949) reported Hagg carbide would decompose to cementite above 773 K.

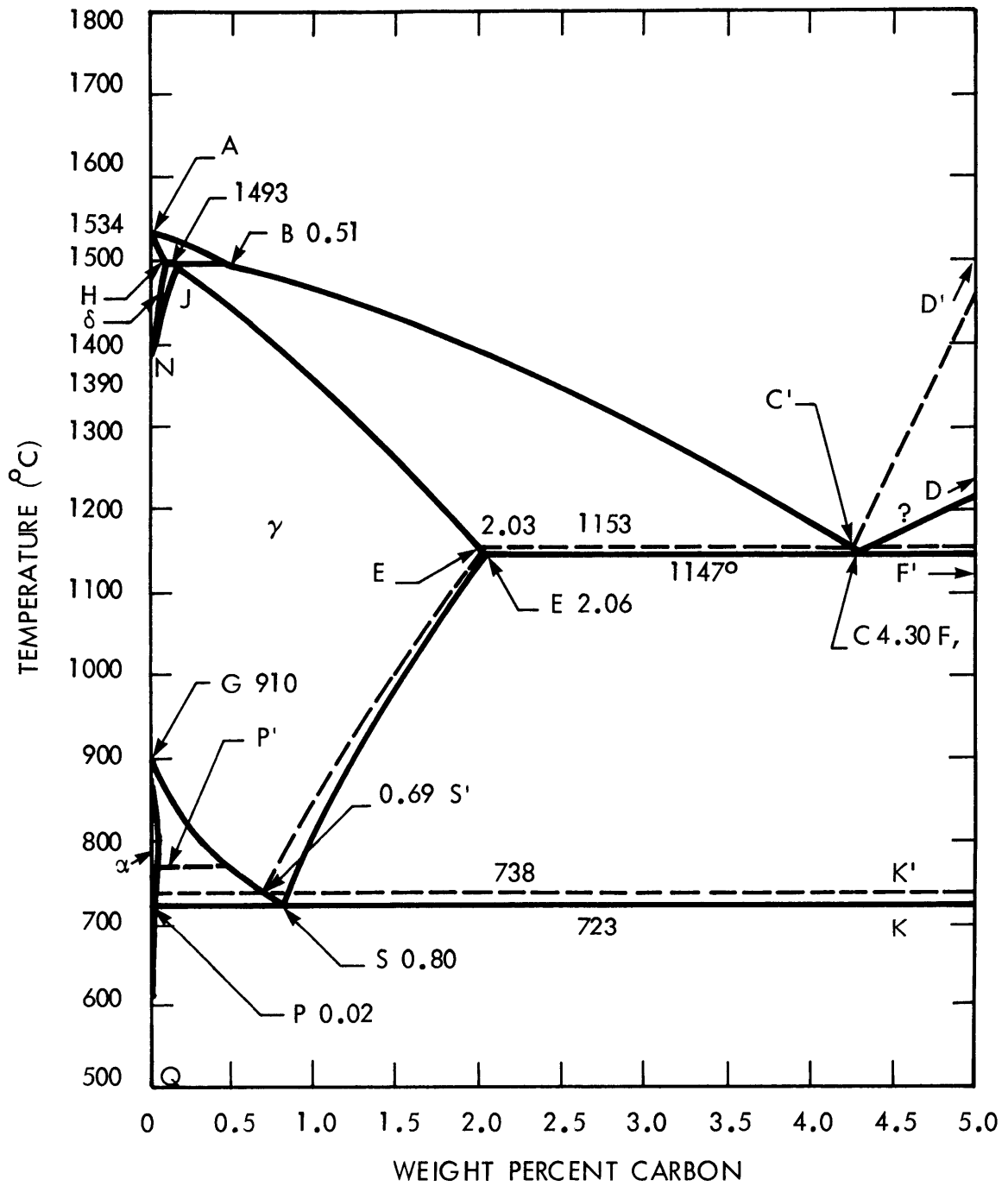


FIGURE 23
 CARBON-IRON PHASE DIAGRAM (WEIGHT PERCENT)

2.2.3.2 Structural Change and Cementite Formation

Podgurski et al. (1950) determined that Hagg carbide could be produced from a variety of different gases. Using hydrocarbons such as butane, propane, and pentane, complete carbiding was reported in just a few hours in the temperature range 548-598 K. Methane was also used; however, the rate of carbiding was very slow. Carbon monoxide was tried, but free carbon was found to form and for that reason the carbiding temperature was lowered to 473-498 K. The Fe_2C and Fe_3C formed from carbon monoxide were found to be more stable than those from other carbiding gases. Podgurski suggested that iron oxide may help stabilize carbides; some iron oxide was produced when carbon monoxide disassociated.

Cementite (Fe_3C) was formed from Hagg carbide by reaction M,



The normal temperature used was 773 K and reaction took three to four hours.

The Fe-C system is a complex one. Metallurgists have for a long time recognized the time-temperature history of an iron-carbon alloy has a profound effect on its structure. In order to be able to adjust physical properties of these alloys an understanding of the time-temperature effects was necessary.

The time-temperature history of a metal catalyst, such as the Bosch catalyst, is also important. Hot spots developing in the bed and/or the cooling rate for the reactor can have

an enormous effect on the final structure and phases observed after completion of an experiment.

A convenient way to represent the effect of temperature excursions on final structure is the so-called Time-Temperature-Transformation diagram (T-T-T diagram). Figure 24 is a typical isothermal T-T-T diagram for eutectoid steel. Depending on the path chosen to bring the steel to room temperature, different structures and phases will be observed.

For example, path 1 shown in Figure 24 shows eutectoid steel quench-cooled to 600°C from above the eutectoid temperature. The steel is held at 600°C for approximately 20 seconds; this converts all the austenite to pearlite, pearlite being a specific cementite- α -Fe structure. When quench-cooled to ambient temperature no structural change will occur since all the austenitic steel has already been converted to pearlite.

On the other hand, if one follows path 2, only 50% of the austenite is converted to bainite before quench-cooling. Thus, the remaining austenite will convert to martensite on quench-cooling to ambient temperature. Bainite and martensite are specific structures common in steel processing (Reed-Hill, 1973). An additional complication arises from the fact that for a gradual cooling rate, the envelope for austenite to pearlite transformation will shift to the right. Thus, a mixture of several different structures is possible.

This may account for some of the dispute in the literature on what is the actual catalyst for carbon deposition. The structure and phases observed on completion of the reaction

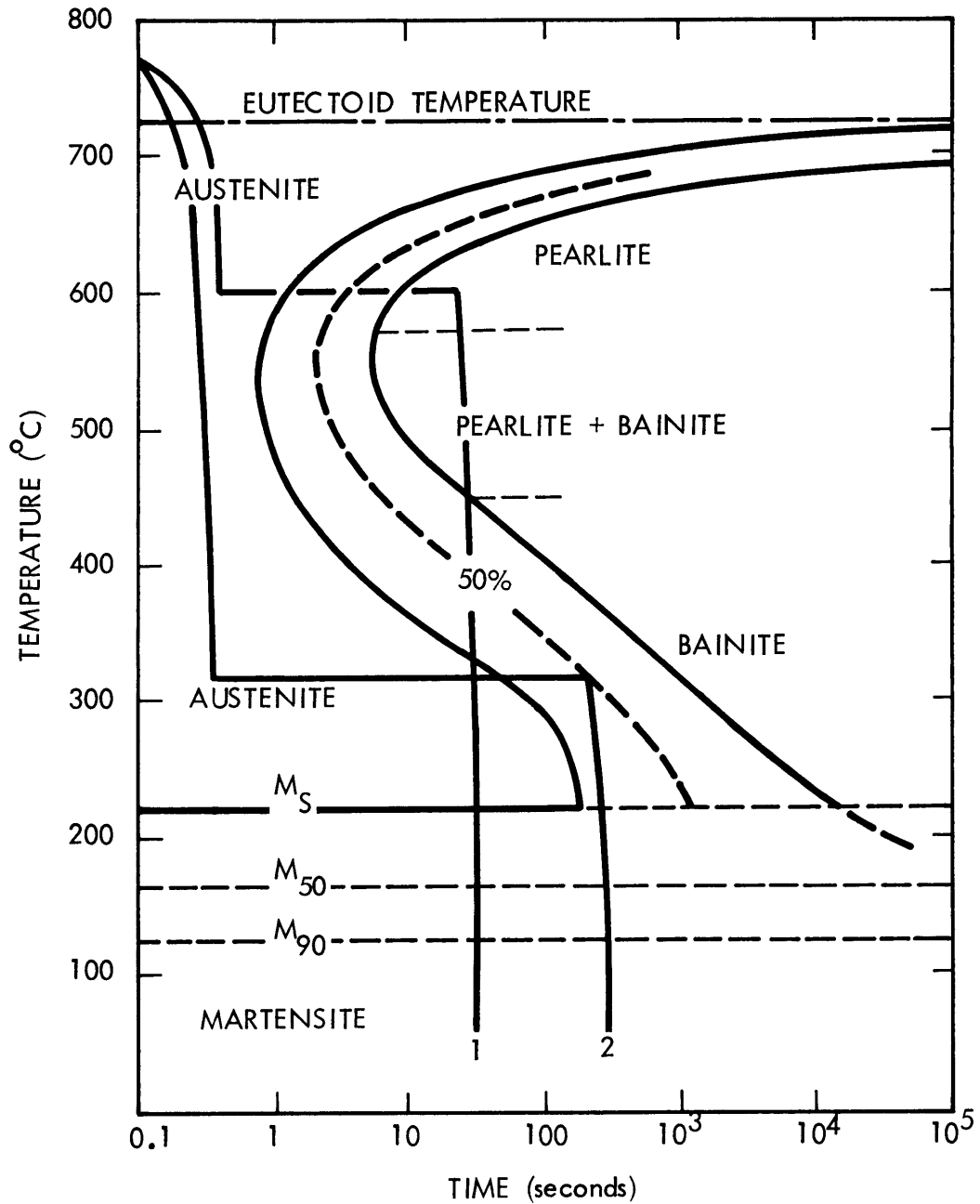


FIGURE 24

COMPLETE ISOTHERMAL TRANSFORMATION DIAGRAM FOR EUTECTOID STEEL
(REED-HILL, 1973)

may not necessarily be that available during reaction.

2.2.3.3 Carbon Fiber Formation

Fiber formation is a peculiar by-product of almost all carbon deposition experiments. Fibers form on many transition metals, notably nickel and iron. The mechanism for fiber growth has been speculated upon by many, however, no complete mechanism has been proposed thus far which can explain the often conflicting data. The fibers themselves are extremely uniform in diameter, usually between 500 - 1000 Å in thickness (MacIver et al., 1955), come in the shape of flat ribbons, solid or hollow tubes, and some are even twisted. Crystals are often found at the end of these filaments and in some cases iron fragments are found throughout their length (MacIver et al., 1955). The filaments have lengths over 1 μ and BET areas up to a range of over 100 m²/g. Fiber C/Fe ratios were sometimes well in excess of 100 and the C/H atomic ratios in the filaments varied between 10 and 30 and, were found to increase with temperature (Walker et al., 1959).

Ruston et al. (1969) performed the most detailed metallurgical investigations on the structural changes which occur on the surface and within the bulk of an iron catalyst during carbon deposition. Utilization was made of optical and electron microscopy, metallographic etching, and x-ray and electron diffraction. Observations were made to determine what occurred on the surface, in the bulk and what solid phases

formed during the reaction. All experiments were carried out with crystalline, metallographically polished iron with pure carbon monoxide at a pressure between 0.01 to 1.01×10^5 N/m^2 and at 823 K.

Two types of carbon were generally found on the surface of their samples after exposure to carbon monoxide. The first was a lamellar form of carbon located close to the metal surface. Ruston speculated that this probably came from decomposition of Fe_3C (cementite) crystals on the surface. The second was a filamentous form which they proposed originated from small crystals of iron carbide epitaxially grown on the surface of metal grains.

The following mechanism was proposed for the formation of the lamellar form of carbon. Carbon monoxide first adsorbs on the reduced iron surface. Depending upon the temperature and carbon monoxide concentration, some carbon monoxide decomposed to carbon and oxygen atoms; the oxygen atoms reacted with adsorbed carbon monoxide to form product carbon dioxide; the carbon atom, being relatively mobile, diffused into the α -iron along intergranular paths. Super-saturated carbon solutions precipitated cementite (Fe_3C) at intergranular sites creating stress in the metal. The stress was released by creep within the surface region; in the absence of compressive stress on the surface, the Fe_3C decomposes to iron and lamellar carbon.

Metallographic examination of the catalyst cross section indicated the phenomenon of carbon deposition was always

accompanied by the formation of cementite within the bulk of the metal. Again, the cementite appears to form almost exclusively at intergranular sites.

To explain the observed fiber growth, Ruston proposed the following reaction sequence. The crystallite formed from cementite decomposition and reacted with carbon monoxide to form Fe_7C_3 . The shape and distribution of these small crystals depend on the orientation of the original iron crystals. The small crystals of Fe_7C_3 were lifted off the metal substrate by carbon formed by catalytic decomposition of carbon monoxide on their surface. This carbon diffused around the back forcing the crystallite off the surface. The small crystals continued to function as a growth center and were carried upward by the growing filaments. They also noted that this process did not continue indefinitely as the Fe_7C_3 crystallite disintegrated, leaving behind iron-rich fragments as the carbon filament grew. The crystallite was no longer active when the carbon-iron filament dropped to less than about 3 wt% Fe (C/Fe atom ratio = 150).

Ruston's findings are valuable for their insight into the overall process occurring on the surface of polycrystalline iron. However, several objections have been raised to his conclusion that the small crystallites grown on single crystal iron surfaces are Fe_7C_3 . Ruston formed this conclusion primarily on the basis of x-ray diffraction data yet both Renshaw et al. (1970) and Ratcliff (1968) contend that his reported x-ray diffraction data are open to other interpretations.

Also, Ruston does not elucidate the mechanism to explain how the original crystallite is lifted off the surface, this issue being perhaps the most important and least understood part of fiber formation.

Robertson (1970) studied carbon formation from methane over iron, nickel, and cobalt surfaces at 923 K and 1023 K. Using transmission electron microscopy, he established two distinct types of carbon were formed. One he designated "flake" carbon, and the other "polycrystalline".

From electron and x-ray diffraction properties, Robertson discovered that the "flake" carbon was highly crystalline and graphitic in nature. This layered form of carbon was very similar to the "lamellar" form reported by Ruston. Surface perfection and reactivity studies also revealed the equivalence of "flake" carbon to natural or synthetic graphite. Only trace amounts of metal substrate were found in this form of carbon. This high degree of crystalline graphite perfection in "flake" carbon is very unusual below formation temperatures in excess of 2000 K.

The second form of carbon was deemed "polycrystalline". This fibrous form of carbon was similar to that reported by Ruston. These fibers grew from the main body of the deposit and ranged in length between 2400 to 14,000 Å and, in width, between 600 to 1350 Å. In a similar manner to Ruston et al. (1969), Robertson (1970) reported: "Dispersed throughout this type of carbon are discreet electron-dense "kernels"; these are surrounded by much more electron transparent bands which

display diffraction contrast effects along their lengths."

Boehm (1973) studied the nature of carbon fibers deposited from hydrogen-carbon monoxide mixtures over nickel deposited from nickel carbonyl, iron deposited from iron carbonyl, Raney nickel, and powdered iron. Runs were made in the temperature range 753-973 K; no pressure was indicated. By comparing the resultant carbon, several conclusions were drawn: first, nickel and iron form different types of carbon fibers. The fibers formed on nickel were hollow, thin-skinned tubes, while those formed on iron were finer, denser fibers apparently containing bands and kernels of iron. Boehm concluded that the formation of carbon fibers on nickel and iron followed different mechanisms. The tubular carbon fibers on nickel were explained by assuming that nickel in the form of globular particles and chains of fused particles (analogous to carbon blacks) had formed during pyrolysis of the carbonyl. Subsequent deposition of carbon on the nickel surface and leaching of the nickel during hydrochloric acid washing of the carbon served to leave only the outside carbon shell.

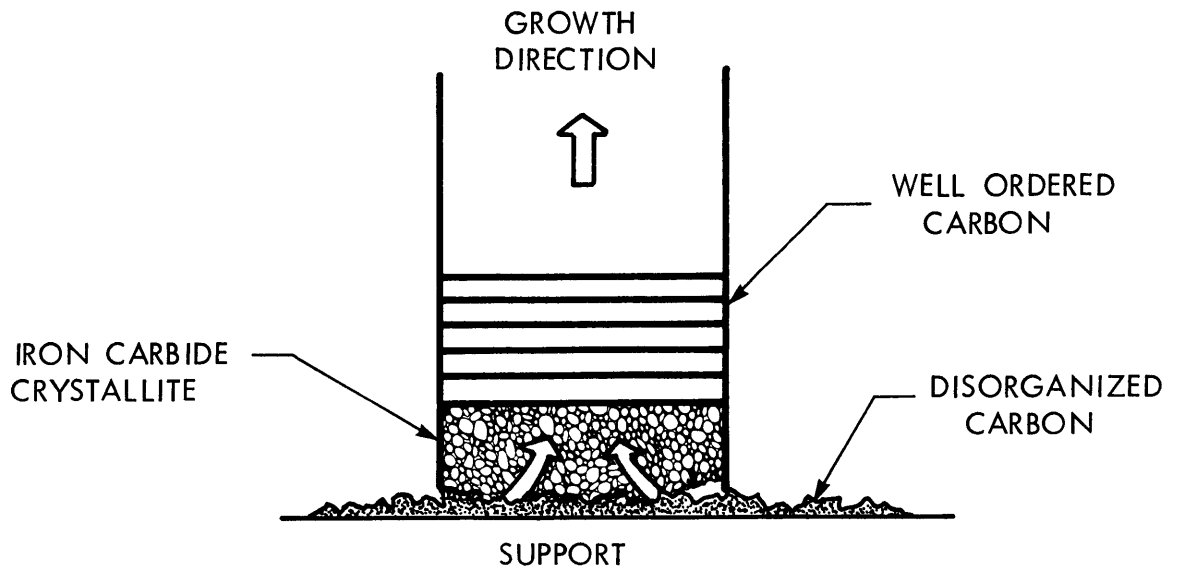
The carbon fibers formed from either iron carbonyl or reduced iron oxide appeared in a variety of shapes. Helically twisted filaments, tubes, and straight strands were all found. Indeed, some of the straight strands were noted to show low contrast indicating ribbon-like structures rather than a rod-like structure of circular or rectangular cross-section. Boehm also reported that extensive fibrous growth was only noted for carbon monoxide-hydrogen mixtures. In a

pure carbon monoxide stream, only "empty skins" and "densely agglomerated material" were reported.

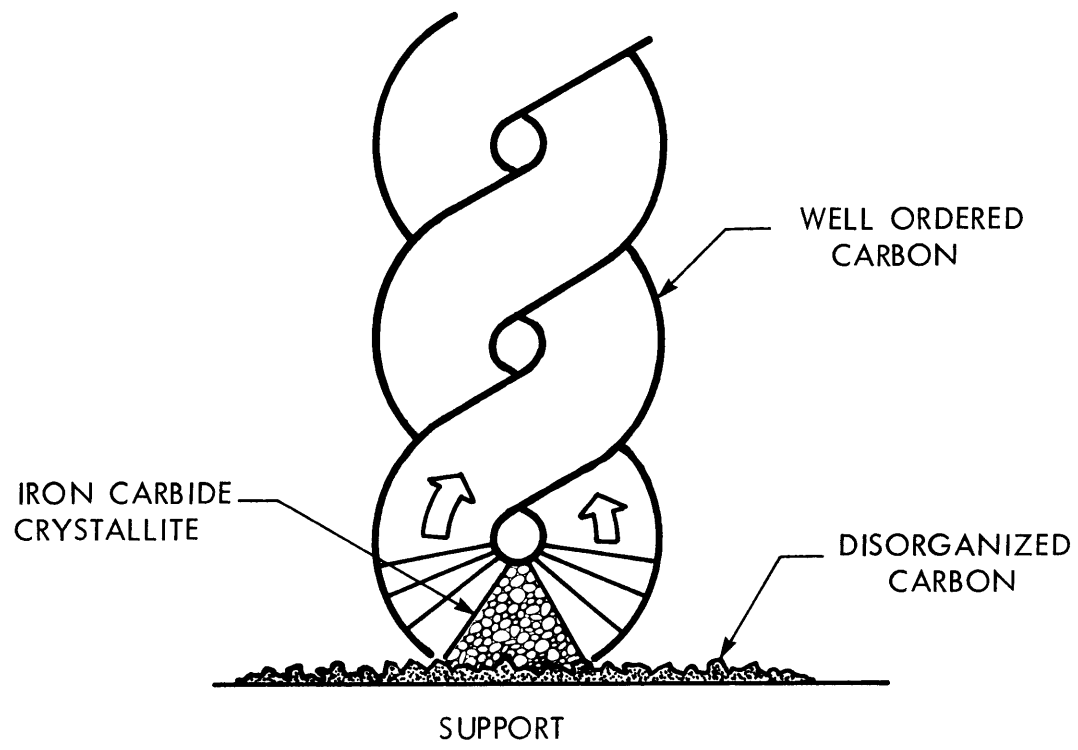
Boehm proposed a mechanism to explain the ribbon-like carbon fibers as well as the twisted carbon fibers he noted in his study. He suggested that carbon filaments grow only from certain crystal faces of a carbided catalyst, the growth rate being controlled by the diffusion of carbon atoms to these specific crystal faces. Disorganized or poorly crystalline carbon in contact with other crystal faces of the carbide phase would be transported by diffusion to the thermodynamically more favorable, well-organized carbon phase. An illustration of this mechanism is shown in Figure 25.

There are several inherent problems in this proposed mechanism of Boehm's. One is that disorganized carbon is assumed always available for diffusion into the carbide crystallite. The formation and transport mechanism for this disorganized carbon to reach the crystallite is not discussed. Also, the direction of the specialized nucleation points must always be oriented in such a way as to give vertical fiber growth, as is almost always found. The reason why the helix structure is developed is unclear. Fourthly, and most importantly, no mechanism is proposed to explain the crystal heads noted in almost all fibers. These points cast some doubt on the proposed mechanism.

Baker et al. (1972) advanced a hypothesis which attempted to explain both the "hollow core" reported by many authors and the "kernel" of metal or carbide at the fiber tip. The



GROWTH OF RIBBON-LIKE FIBERS



GROWTH OF TWISTED FIBERS

FIGURE 25
FIBER GROWTH MECHANISMS (BOEHM, 1973)

proposed model is shown in Figure 26. In step (a), a crystallite has been liberated from the support (for iron this could be by the mechanism proposed by Ruston et al., 1969). Gas phase components such as methane, acetylene, or carbon monoxide are then free to react and deposit carbon on the crystallite as shown in (b). Carbon from the decomposed gases can be taken into solution in the metal, diffuse through the crystallite to be deposited predominantly in the protected regions to produce the situation shown in step (c). The precipitation of carbon at the rear of the particle builds up a deposit of carbon which forces the particle away from the support as shown in (d). If diffusion through the particle is slow enough to limit the rate of the above process, then eventually the surface of the entire crystallite will be covered with carbon and catalytic activity will cease as shown in (e). The shape of the catalytic crystallite and differing diffusion paths lead to the hollow core shown in (d) and (e). Baker alleged that the diffusion of carbon through the crystallite was caused by a thermal gradient. They proposed an exothermic reaction occurring at the exposed surface caused temperature gradients across the particle. In support of their hypothesis, they pointed out that the activation energy for the observed reaction was the same as the reported activation energy for the diffusion of carbon in nickel. Manning (1976) correctly pointed out, however, that the solubility of carbon increases with temperature to 996 K then decreases. Thus, a thermal gradient as the driving force for carbon solution and

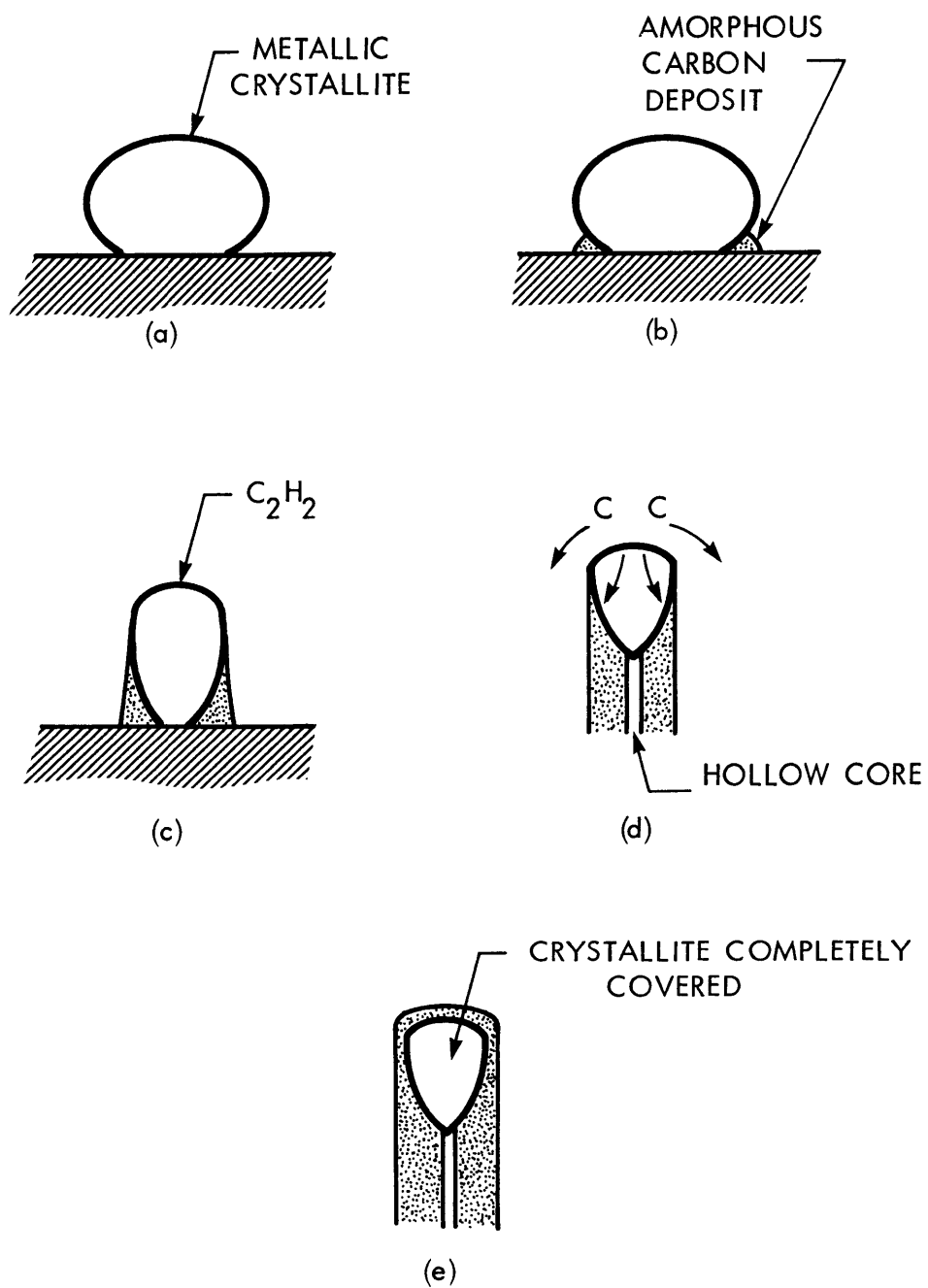


FIGURE 26

FIBER GROWTH MECHANISM (BAKER et al., 1972)

then dissolution appears inconsistent with this fact.

Baker et al. (1972), using controlled atmosphere electron microscopy, also studied the deposition characteristics of various gas mixtures containing carbon dioxide, carbon monoxide, and methane. Single crystal graphite substrates were used onto which iron films could be evaporated. The gas mixtures used were pure carbon dioxide; pure methane; and a 97.5% carbon dioxide, 1.5% carbon monoxide, 1.0% methane mixture. Reaction temperatures varied between 725 K and 1265 K, while pressures were in the order of 8 kN/m^2 . Reaction times were thirty to sixty minutes.

Baker observed two types of carbon formation which he deemed "Type I" and "Type II". "Type I" carbon formation was a flocculent amorphous deposit which only formed in the presence of iron particles, if methane was a component in the gas mixture and only when under an electron beam. In addition, this flocculent amorphous deposit was only found above a temperature of 900 K.

"Type II" carbon was identified as crystalline platelets of graphite (these are very similar to Robertson's "flake" carbon (1970) and Ruston's "lamellar" carbon (1969)). This type of formation was only observed at temperatures above 1200 K and only occurred around iron particles greater than 60 nm in diameter. During this investigation they noticed a high degree of iron crystal mobility. Iron particles 10 nm in diameter diffused along the graphite substrate and coalesced to form spheres 80 nm or more in diameter. Smaller

iron particles were observed to remain mobile and aggregated until they were large enough for platelet growth to begin. Type II carbon was found to occur even if only pure argon (i.e., inert atmosphere) was fed as the gas phase. However, Type II formation only occurred in an iron-graphite system; never if either of these (iron or graphite) were used alone. This lends support to the hypothesis of Baker et al. and Ruston et al. that suggested that lamellar (platelet) carbon is found by Fe_3C disintegration.

Assuming first that some carbon (i.e., graphite) dissolves in these iron particles at the reaction temperature (i.e., 1225 K), Baker postulates the following as the mechanism for Type II formation. At 1225 K the stable Fe-C phase is austenite and in order to maintain the concentration of dissolved carbon at the saturation level for 1225 K, cementite is precipitated. At temperatures above 975 K, the rate of decomposition of cementite to form iron and graphite increases rapidly with increasing temperature. Under the present slow cooling conditions, it is probable that decarburization of austenite will occur, the carbon precipitating as cementite, which in turn will undergo decomposition to form iron and graphite, the latter crystallizing on the edges of the metal particles. If iron particles lose mobility when carbide forms, then carbon transpiration could then take place within the particle causing graphite to crystallize and grow from the gas-cooled upper surfaces. Since carbon dissolves in iron faster from graphite than from other carbonaceous

materials, a high concentration of carbon in iron would be expected to build up readily in the present system. Baker did not observe any fibrous carbon formation in this continuous electron microscopy study. From the work of Ruston et al. (1969), one can say that the probable reason for this was that the partial pressures of the carbonizing gases were very low and also the residence times very short. Ruston et al. (1969) concluded that fiber formation was a function of both residence time and carbon monoxide partial pressure.

Thomas, Thrower, and Walker (1973) studied the growth of filamentary carbon on metallic surfaces during the pyrolysis of methane and acetone. This was done with the use of transmission electron microscopy. They found that neither nickel nor iron produced fibers when heated in ultra-pure methane at temperatures below 1173 K. Above 1173 K both nickel and iron catalyzed fiber growth, nickel being the most active. At this temperature, Thomas also reported platelet formation was observed. This plate formation was of the same type as that found by Baker (1972), Robertson (1970), and Ruston (1969). Thomas next ran commercial grade methane over identical catalyst at 1173 K and found that filament growth did occur. In fact, using commercial grade methane, Thomas was able to get significant fiber formation at temperatures as low as 773 K. This would then seem to indicate that trace impurities present in methane play a major role in the production of filaments at low temperatures.

Acetone, a frequent impurity in methane, was then fed

over the iron catalyst. Iron promoted deposit formation at 673 K, but they found no observable effect on nickel. The deposit found was lamellar in nature. Pyrolysis of acetone at 773 K, however, gave rise to filament formation, each filament associated with a dense particle at its dip. The appearance of these filaments was similar to those found by Robertson (1970). If one checks back into Robertson's work (1970) on methane pyrolysis, one finds that Robertson (1968) in a later work comments on the possibility that carbon deposits could arise from breakdown of trace impurities in the methane employed in his investigations. The fact that impurities may have caused the carbon fiber formation not the methane levels, lends credence to the mechanism proposed by Baker et al. (1972). That is, only carbon-bearing gases which deposit carbon exothermically produce carbon fibers. However, the inconsistency suggested by Manning (1976) is still left unanswered.

3. Apparatus Design Rationale

In order to achieve the previously stated thesis objective, the following design criterion had to be met by the experimental apparatus:

1. Gases, both pure and of specified composition, were to be metered and preheated before entering the reactor. Provisions were necessary for on-line inlet and exit gas analysis. Steady flow conditions, as well as the capability for rapid changes from one composition to the next, were necessary.

2. The reactor had to be capable of operating isothermally. A provision was also necessary to monitor changes in mass of the catalyst as oxidation or carbon deposition proceeds. The capability to monitor continuously catalyst bed temperature was also desirable.

3. Provision must be made to facilitate the determination of experimental run conditions. This is, to solve numerically the appropriate equilibrium relationships and establish the necessary flow condition to achieve a given gas composition over the desired solid phase. This procedure must allow variations in the P_{H_2}/P_{H_2O} ratio at fixed C/H or O/H values.

4. Data acquisition had to be fast and accurate. A continuous record of all pertinent temperatures, as well as a record of the analysis for exit and inlet gas samples were

necessary. Finally, to facilitate data interpretation, it would be necessary to formulate a computer program. This program should have the ability to indicate the equilibrium solid phases present.

The above design criteria were met by the experimental apparatus and procedures described below. As such, they provided an efficient and reliable research tool to explore the multi-faceted Bosch reaction sequence.

3.1 Experimental Apparatus

The experimental apparatus can be broadly divided into three subsections: the feed-gas delivery system in which the inlet gases are individually metered, mixed, and delivered dried or saturated with water; a reactor section consisting of a quartz preheater, quartz reactor tube, and catalyst assembly mechanism; and a data acquisition sub-system.

3.1.1 Feed-Gas Delivery System

Figure 27 is a schematic of the feed-gas delivery subsection. The gases used during the investigation and a typical analysis of each is given in Table 1.

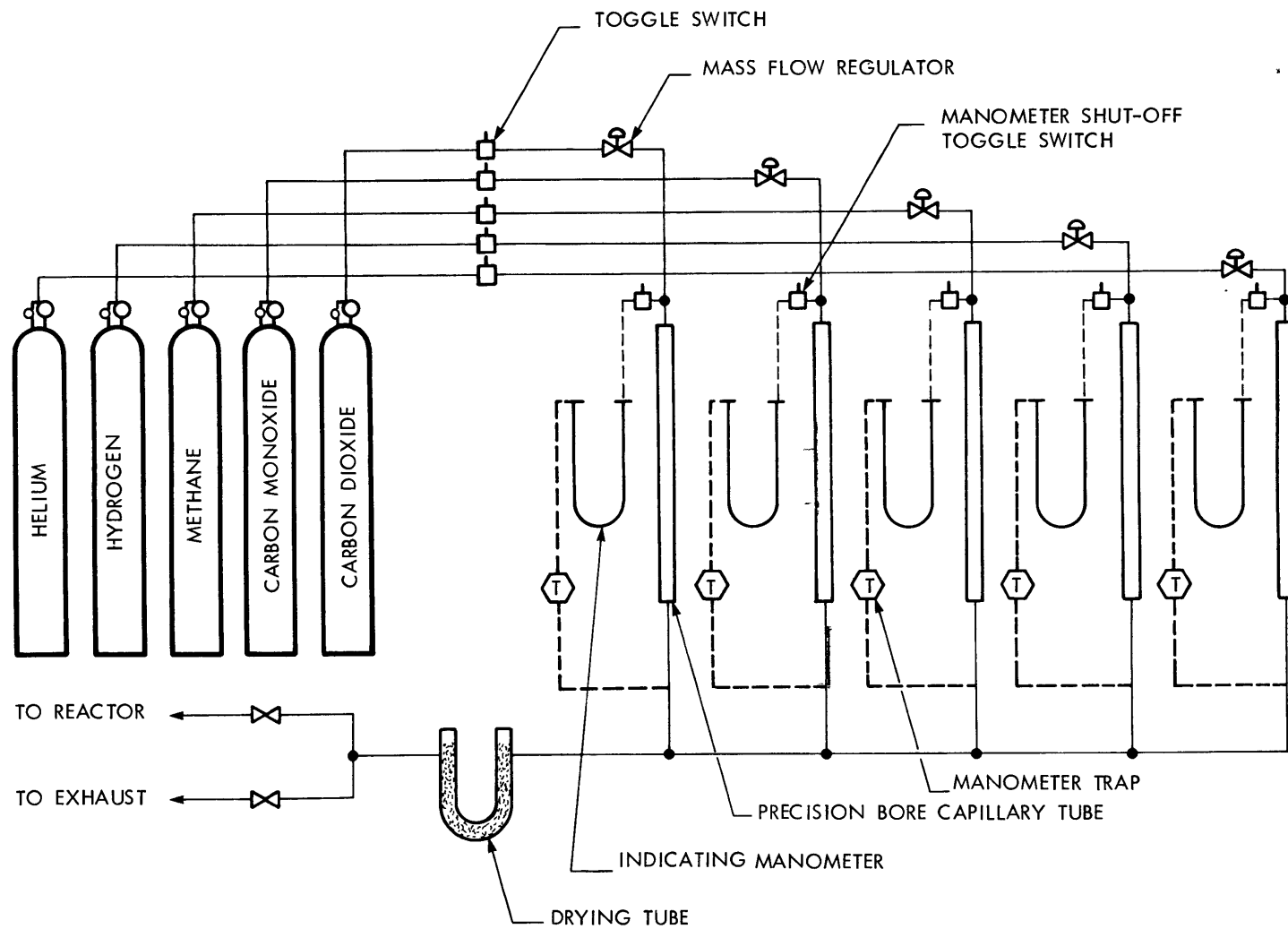


FIGURE 27 FEED GAS DELIVERY SYSTEM

Table 1 Feed-Gas Analysis

	CO ₂	O ₂	N ₂	Dew Point
Hydrogen Prepurified Grade		<20 ppm		213.6 K
Methane (Cp Grade)	0.2 mole %	0.005 mole %	0.6 mole %	
Carbon Dioxide (Bone Dry Grade)	99.95 mole %	0.05 mole %		238.6 K
Carbon Monoxide (Cp Grade)	50 ppm	600 ppm	1500 ppm	
Helium (High Purity Grade)	1 ppm	1 ppm	14 ppm	

The gases were individually fed to Brooks Model 8944 mass flow regulators equipped with digital valve stems for reproducibility and inlet line filters capable of entraining particulates $> 2 \mu\text{m}$. Inlet pressures of $3.45 \times 10^5 \text{ N/m}^2$ (50 psig) maintained choke velocities across the individually sized flow regulators. Precision bore capillary tubes (I.D. $\pm 0.007 \text{ mm}$ of specified value and 91.5 cm in length) in combination with manometers containing Meriam high vacuum manometer fluid (specific gravity equals 1.04) were used to obtain calibration curves of flowrate in cm^3/s versus pressure drop in cm of Meriam fluid for each reactant gas. These calibration curves, corrected to standard temperature and pressure, were used to establish the desired gas phase composition. The gases were mixed and dried in a 15.24 cm Kimax U-tube filled with

indicating dessicant (Grade 42, Silica gel 6-16 mesh, Fisher Scientific Company). The mixed gases were then either fed to the reactor or by-passed to the atmosphere.

When the gases were sent to the reactor, they could be fed either dry or saturated with water. Figure 28 illustrates the components and flowpaths for the saturator system. The feed gases flowed through a preheater constructed from a 3 m length of 0.64 cm I.D. refrigeration tubing. Heat was supplied by a Briskeat-silicone-rubber-imbedded flexible heating tape, the power output being controlled by a Superior Electric Company Powerstat. Asbestos insulating tape was used to reduce heat losses. The gases from the preheater (attaining temperatures as high as 333 K) entered three 1000 ml Pyrex three-necked flasks. These flasks were connected in series and submerged in a 55.32 liter Precision Scientific Company constant temperature bath. A Chemical Rubber Company contact heater maintained the bath temperature to ± 0.1 K. The bath circulation was provided by a motor-impeller assembly. During normal operations, each bubbler contained 900 cm^3 of water. The gas stream was fed to each bubbler through Pyrex-fritted gas-dispersion tubes; the gas stream passing from bubbler to bubbler through 0.95 cm Vycor tubing which was also kept submerged. After the third bubbler, the gases passed into a 20.5 cm deep vertical bed packed with 0.38 cm O.D. glass-beads, and 5 cm of quartz wool. This assembly was used to eliminate any entrained water droplets in the saturated gas stream. The feed gas stream then entered the reactor through

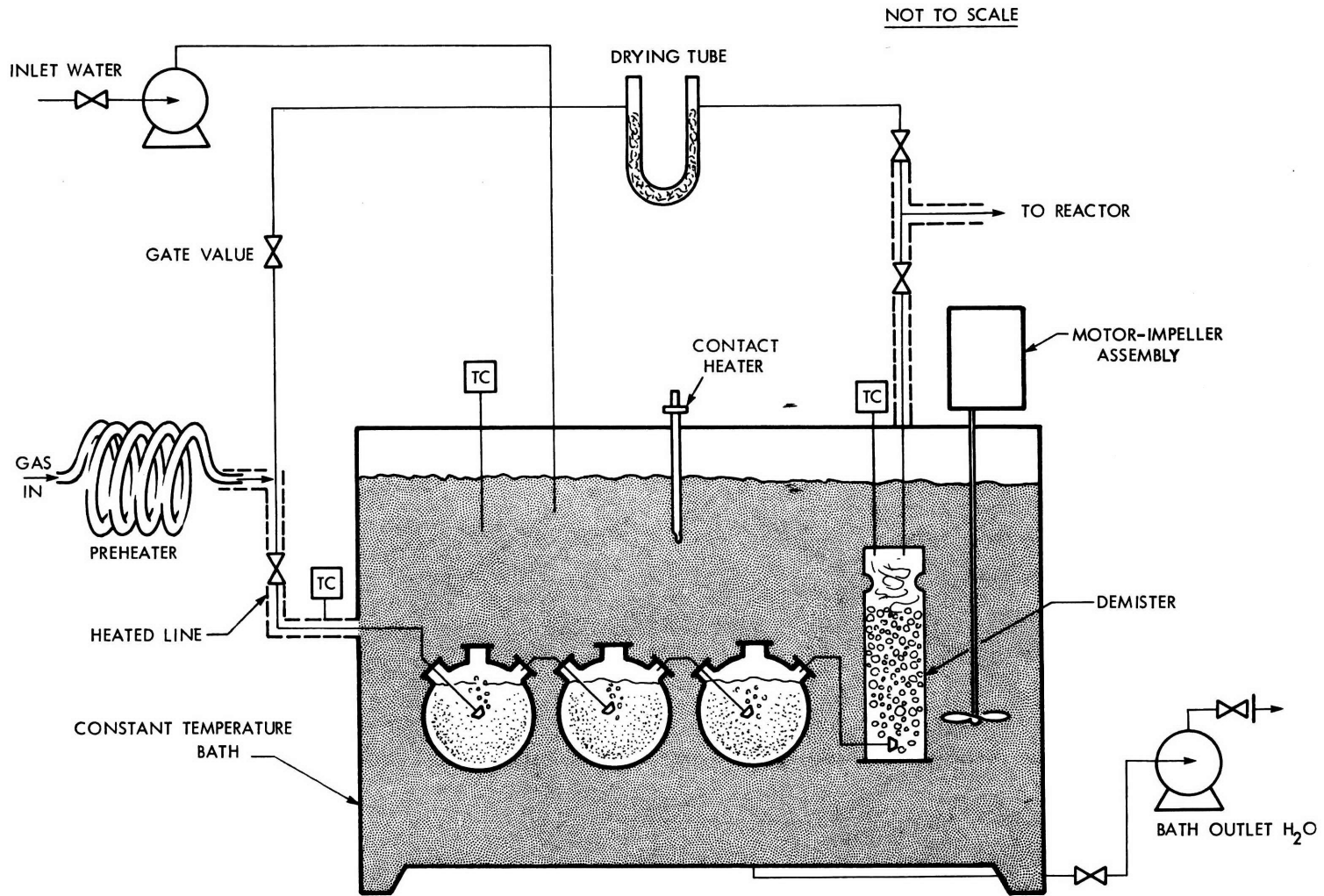


FIGURE 28 SATURATOR SUB-SECTION

heated feed lines. A combination of March Company centrifugal pumps (maximum capacity 6.0 liters/min) and a 1.27 cm NUPRU regulating valve allowed readjustment of the bath and bubbler temperatures in 3 to 5 minutes. The highest water partial pressure attained was $3.13 \times 10^4 \text{ N/m}^2$ (30%). As indicated in Figure 28, 3 chromel alumel thermocouples were used to monitor the gas temperature just prior to entering the first bubbler, the exit gas temperature and the bath water temperature.

If the gases were to be fed dry, after passing through the copper preheater, they were sent through another dessicant-drying tube and on to the reactor.

3.1.2 Reactor Section

The reactor consisted of three basic parts: the pre-heater and lower reactor support tube; the top section of the reactor support tube; and the catalyst assembly. Figure 29 illustrates the top and lower reactor sections including the preheater.

3.1.2.1 Preheater and Lower Reactor Support Tube

The preheater was constructed from a 6 m quartz tube with a 7 mm I.D. formed into a helical coil. The preheater coil extends from approximately halfway up the top reactor section down over the entire length of the lower reactor support tube and connects at the base. Mounted concentrically around the reactor support tube, it has an inside diameter of 60 mm. The lower reactor support section was constructed from a

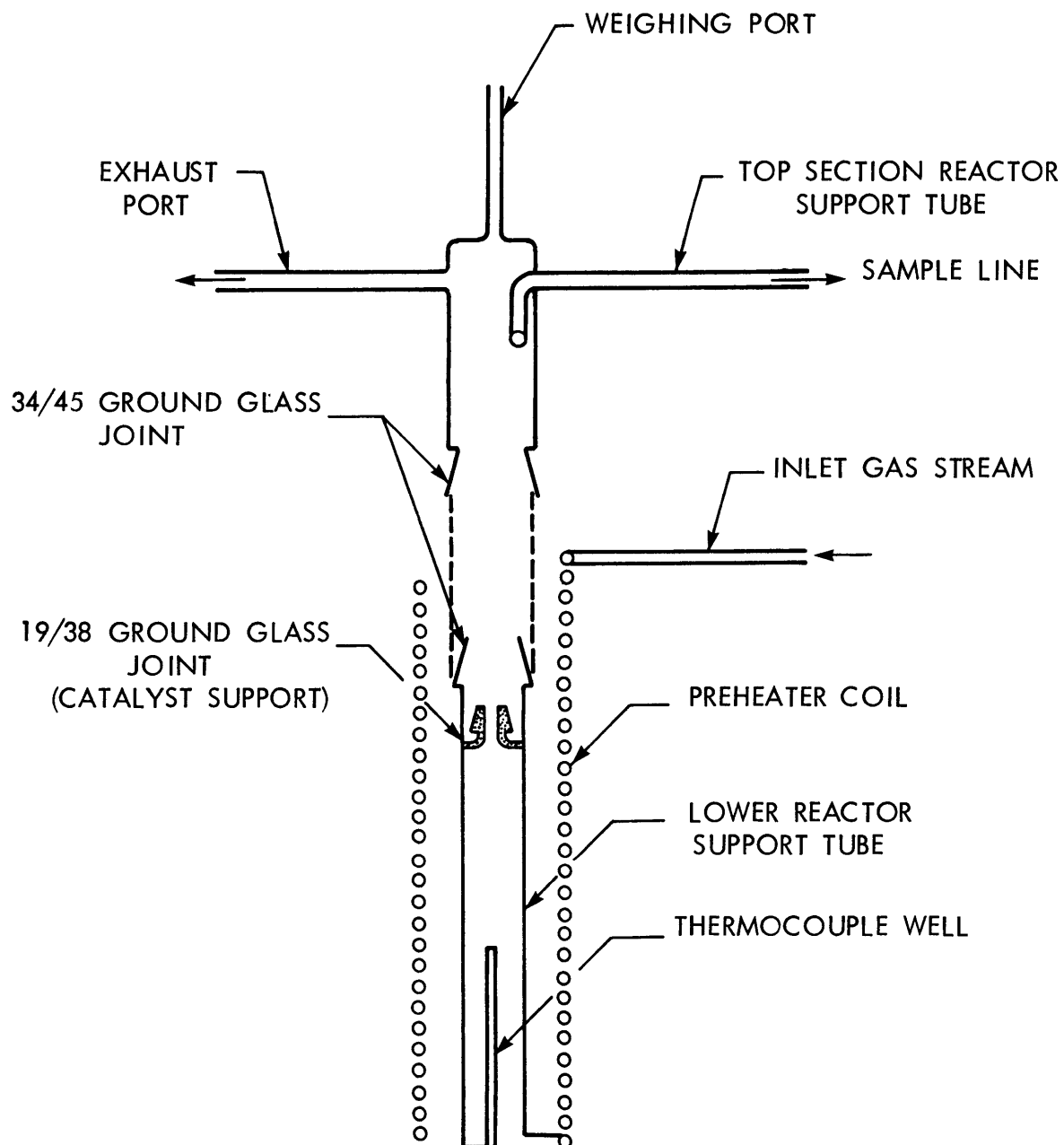


FIGURE 29

TOP AND LOWER REACTOR SECTION INCLUDING PRE-HEATER

28 mm I.D. quartz tube 28 cm in height. A 34/45 tapered male ground glass joint is attached to one end, the other is connected to the preheater. A 110 mm quartz thermocouple well is provided at its base to support a two-hole Alundum thermocouple sheath used to position two 0.051 cm type K chromel-alumel thermocouples. One thermocouple activates a Thermolyne proportional controller used to maintain reactor temperature, the other provides a continuous indication of the reactor temperature. Located 20 mm from the base and on the inside wall of the reactor support tube is a 19/38 male ground glass joint used to support the catalyst assembly. Figure 30 illustrates the temperature profile obtained over the catalyst mechanism at typical run condition (i.e., reactor set temperature 823 K, pressure = 1.01×10^5 N/m², flowrate He = 20 cm³/s {STP}).

3.1.2.2 Top Section Reactor Support Tube

The top section consisted of a 28 mm I.D. quartz tube 15.24 cm in height. The lower end was fitted with a female 34/45 ground glass joint, the upper end had an 8 mm I.D. quartz tube concentrically mounted. This concentric 8 mm tube was used as a weighing port.

Mounted at 90° angles to the support tube center-line were two additional 8 mm I.D. quartz tubes. One of these extensions was used as an exhaust line, the other had a 2.6 cm elbow extending into the effluent stream and was used as a sample line.

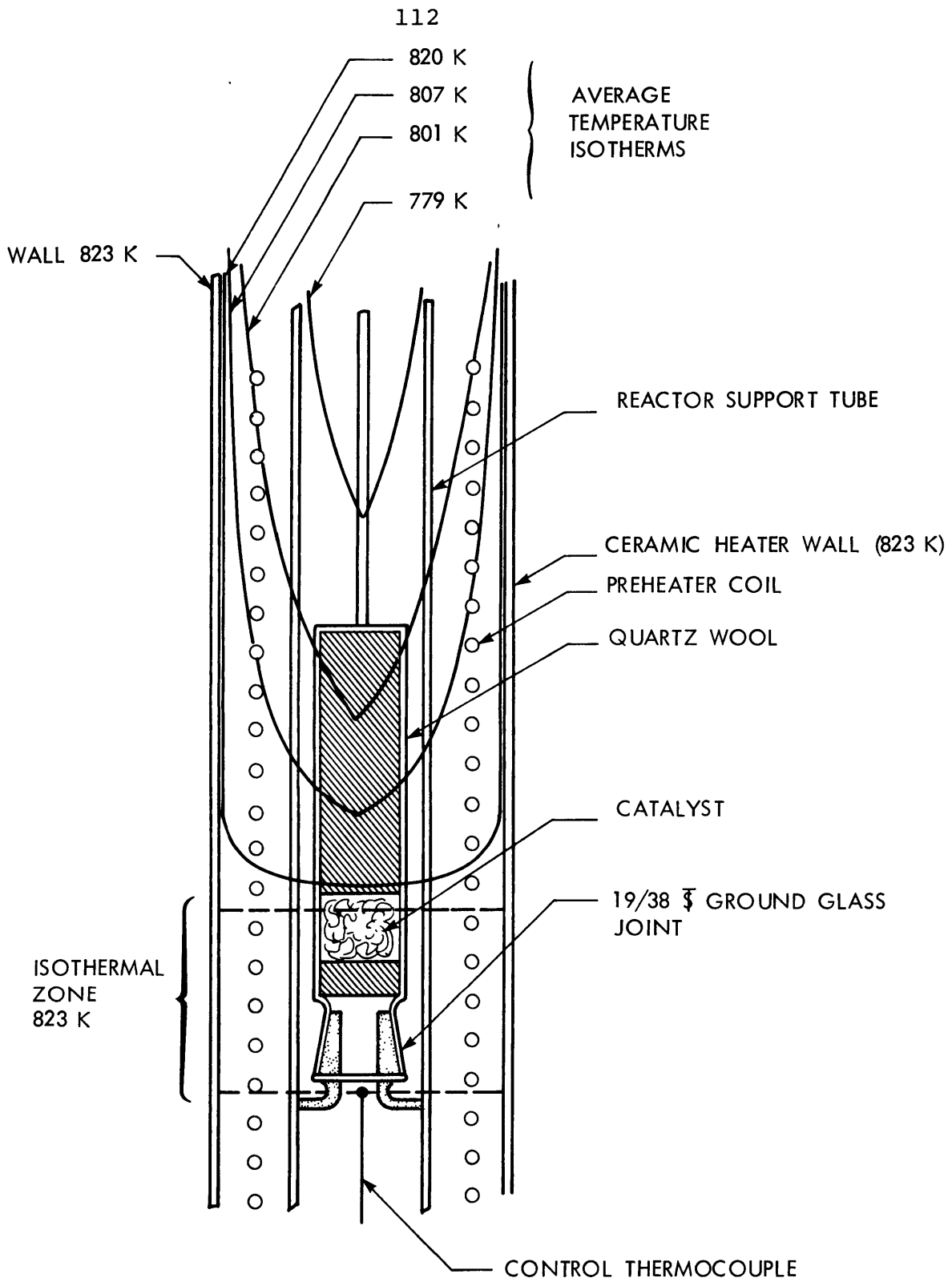


FIGURE 30

REACTOR TEMPERATURE PROFILE

3.1.2.3 Catalyst Assembly

Figure 31 is a representation of the catalyst assembly which consisted of a 120 mm, 20 mm I.D. quartz tube with one end fitted with a 19/38 female tapered ground glass joint. Attached to the catalyst carrier was a concentrically positioned quartz tube 23.62 cm in height with an inside diameter of 2 mm. A 1.5 mm I.D. Alundum sheath provided support for a chromel-alumel thermocouple which was positioned in the catalyst bed. A standard Omega Engineering Company thermocouple connector attached to the quartz tube suspension bar allowed continuous recording of the catalyst bed temperature. A matched thermocouple connector suspended from a Sartorius electrobalance allowed weight measurements to be made periodically during an experiment.

The entire three-piece reactor assembly was housed in a heating furnace (Figure 32). Two Thermcraft Model RH 254 semicylindrical ceramic heating elements 75 mm in diameter and 30.5 cm in length encompass the preheater-reactor heated cavity. These heating elements were wired in parallel to a 220 volt electrical line and were capable of delivering a maximum power output of 2300 watts. The temperature in the cavity was maintained by a Thermolyne Dubuque III solid state proportional controller activated by a chromel-alumel thermocouple. The Dubuque III controller maintained the set value to ± 10 K and, in combination with the heating elements, had a range of 1450 K. The heating elements are enclosed by 10.92

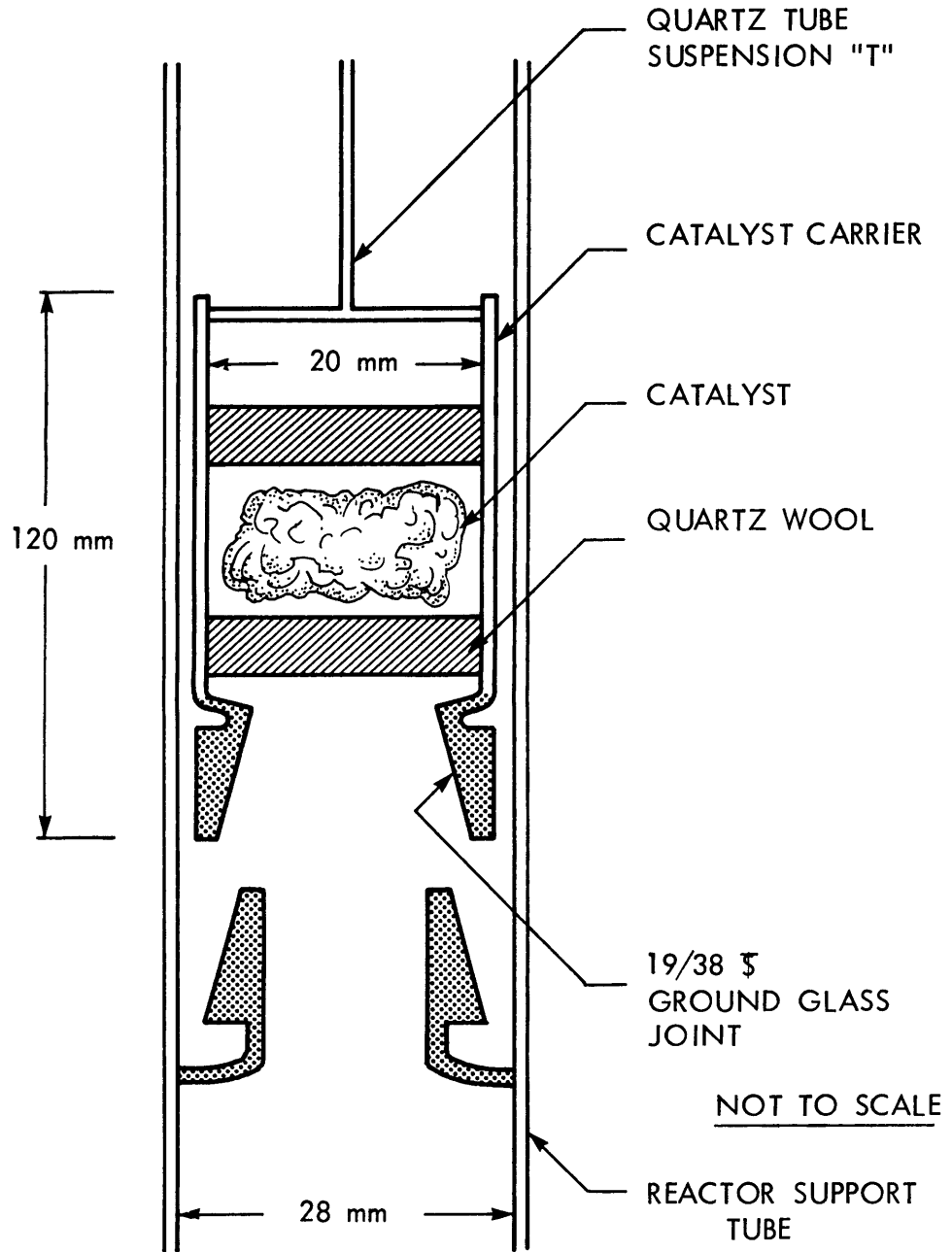


FIGURE 31
CATALYST CARRIER IN SUPPORT TUBE

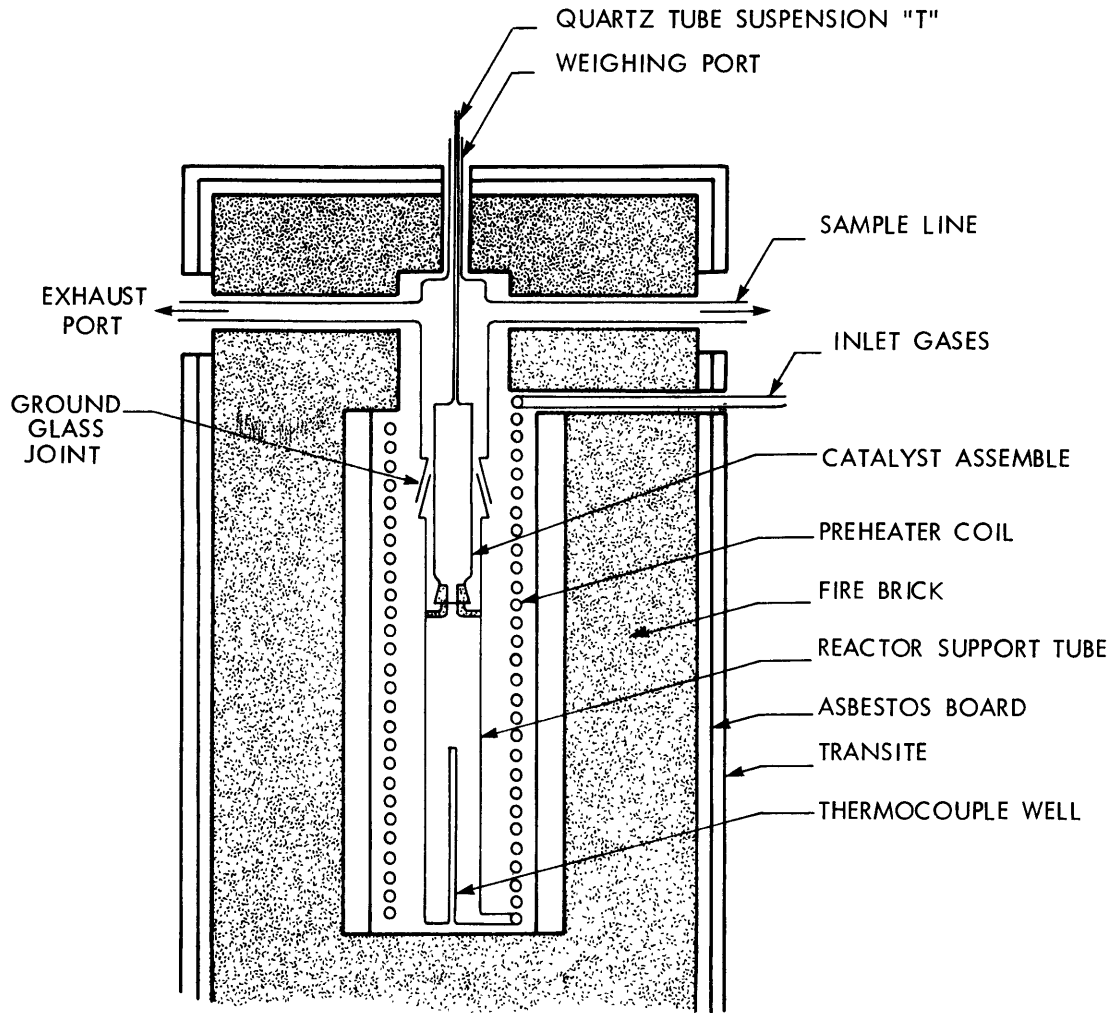


FIGURE 32 REACTOR ASSEMBLY IN FURNACE

cm of Babcock and Wilcox type K-30 insulating firebrick which in turn was enclosed by 6.5 mm of asbestos board. The entire furnace was structurally supported by an outer layer of 6.5 mm Transite, an asbestos-concrete composite.

3.1.3 Data Acquisition Section

Inlet and exit gas samples were analyzed using an on-line series 700 Hewlett-Packard gas chromatograph with a thermal conductivity detector.

Table 2 is a summary of critical operating parameters.

Table 2 Critical Gas Chromatograph Parameters

<u>Filaments</u>	<u>GOW-MAC Rhenium Tungsten Code 13.002 (wx)</u>
Sample Size	0.25 cm ³
Carrier Gas Flowrate	30 cm ³
Column Operating Temperature	348 K
Detector Operating Temperature	473 K
Detector Filament Current	200 m A

Porapak Q and Porapak QS were used interchangeably as packing in the analytical columns. These columns were hand-packed in 3.15 mm X 3 m Teflon tubes which were subsequently made into coils in order to fit into the chromatograph oven. A special carrier gas mixture supplied by the Matheson Gas Company of Massachusetts (19.5% He, 8.5% H₂) allowed direct measurement of hydrogen by the method recommended by Purcell and Ettore (1965). Following the method described by Dal Nogare

and Juvet (1962), an external standard was used to determine response factors of the thermal conductivity detector. These in turn were used to calculate component compositions (except hydrogen).

An Autolab 6300 digital integrator in conjunction with a Honeywell chromatograph recorder was used to give a quantitative as well as visual record of all samples analyzed. The Autolab 6300 was found to have a precision of 0.8% and an accuracy of 1.0% of the indicated values.

An on-line sample valve system was used to obtain reproducible samples for injecting into the gas chromatograph. A Hewlett-Packard Model 19020 sample valve-sample loop assembly was housed in an insulated aluminum box. Cartridge heaters imbedded in an aluminum block provided a constant temperature heat sink. The temperature of the sample valve system was maintained to ± 0.1 K by a model 220 Hewlett-Packard temperature controller activated by an iron-constantan thermocouple. A series of inter-connected toggle switches allowed the selection of samples of inlet, outlet, and standard gases. Another toggle switch allowed the entire sample valve system to be evacuated. During normal operation a vacuum would be drawn on the sample valve system. With the desired toggle switch open, gas entered the sample valve system until the pressure in the sample loop equaled the barometric pressure. The sample loop pressure was measured using a U-tube mercury manometer, one leg of which was open to the atmosphere. The sample size used in the course of the investigation was 0.25

cm³ although the capability was available for larger or smaller sized samples, if desired.

Temperatures were continuously monitored by a Honeywell 27 channel multipoint recorder with an accuracy of ± 3 K. An Omega Engineering Company series 200 digital pyrometer allowed instantaneous analysis of the catalyst bed temperature, reactor center line temperature, and the water saturator temperatures. Accuracy was specified to be $\pm 0.25\%$ of the indicated reading.

The raw data were reduced using a modified data reduction computer program originally developed by Manning (1975). A copy of this analytical program is given in Appendix 7.4.

3.2 Experimental Procedure

The experimental procedure varied depending on the specific test objectives; but, as a general rule, the following procedures were used.

3.2.1 Determination of Run Conditions

The experimental procedure in this thesis centered on determining the catalytic effects of iron oxides and iron carbides on carbon deposition. The method chosen for use during the investigation was to vary the P_{H_2}/P_{H_2O} ratio at a fixed O/H or C/H value and through the use of triangular phase diagrams (Appendix 7.2), determine the iron-iron oxide, iron-iron carbide phase boundaries; the initial assumption being that carbon deposition is not catalyzed by either iron oxides or iron carbides.

To achieve this end, it was necessary to develop a computer program which, for a 5 component gas mixture at fixed temperature, pressure, C/H or O/H, and desired P_{H_2}/P_{H_2O} ratio, gave an overall gas composition lying in the desired solid phase envelope.

A convenient mathematical "trick" was used to achieve a rigorous solution. This was to assume all solid phases which could be present were catalysts for all gas phase reactions.

From the Gibbs phase rule: the number of components, $N = 5$ (CH_4 , H_2 , CO , CO_2 , H_2O), temperature and pressure were fixed along with either the C/H or O/H ratio, and the desired P_{H_2}/P_{H_2O} ratio chosen. The number of degrees of freedom then equals, $f = 5 + 2 - 1 - 4 - R = 2 - R$.

The number of independent reactions must be two to fix the gas phase composition. Reactions B and K were chosen.

Therefore, by specifying the temperature, pressure, C/H or O/H ratio, and the P_{H_2}/P_{H_2O} ratio, the equilibrium gas phase composition could be numerically solved for and specified along with the associated flow settings. Figure 33 and 34 are examples of typical output from the program used to set up experimental conditions. The phase diagram shown in Figure 34 is generated simultaneously with the numerical solution to give a visual check on the position of the projected reactant gas composition.

Figure 33 Example of Experimental Set Conditions

RUN NUMBER A- 51
 TEMPERATURE 800.K , PRESSURE 1.ATM.

CONDITION 1: PH2O/PH2 0.42, C/H 0.24, O/H 0.17
 GAS COMPOSITIONS

PCO2	PCO	PCH4	PH2O	PH2	PTOTAL
0.117493	0.065903	0.416459	0.118352	0.281790	0.999999
PERCENTAGE H 70.75		PERCENTAGE O 12.02		PERCENTAGE C 17.21	
FLOW SETTING					
6.462116	4.718698	35.399093	49.425949	9.299097	

CONDITION 2: PH2O/PH2 0.39, C/H 0.25, O/H 0.17
 GAS COMPOSITIONS

PCO2	PCO	PCH4	PH2O	PH2	PTOTAL
0.122386	0.073508	0.436562	0.103546	0.263994	0.999998
PERCENTAGE H 70.18		PERCENTAGE O 11.93		PERCENTAGE C 17.88	
FLOW SETTING					
6.731242	5.263215	37.107841	46.774284	8.711818	

CONDITION 3: PH2O/PH2 0.33, C/H 0.27, O/H 0.17
 GAS COMPOSITIONS

PCO2	PCO	PCH4	PH2O	PH2	PTOTAL
0.129538	0.091551	0.475464	0.075860	0.227580	0.999995
PERCENTAGE H 69.07		PERCENTAGE O 11.74		PERCENTAGE C 19.17	
FLOW SETTING					
7.124616	6.555103	40.414459	40.793052	7.510158	

CONDITION 4: PH2O/PH2 0.25, C/H 0.30, O/H 0.17
 GAS COMPOSITIONS

PCO2	PCO	PCH4	PH2O	PH2	PTOTAL
------	-----	------	------	-----	--------

0.131625 0.120905 0.519347 0.046562 0.181550 0.999992
PERCENTAGE H 67.81 PERCENTAGE O 11.92 PERCENTAGE C 20.65

FLOW SETTING
7.239427 8.656816 44.144554 31.917293 5.991176

CONDITION 5. PH2O/PH2 0.20, C/H 0.32, O/H 0.17
GAS COMPOSITIONS

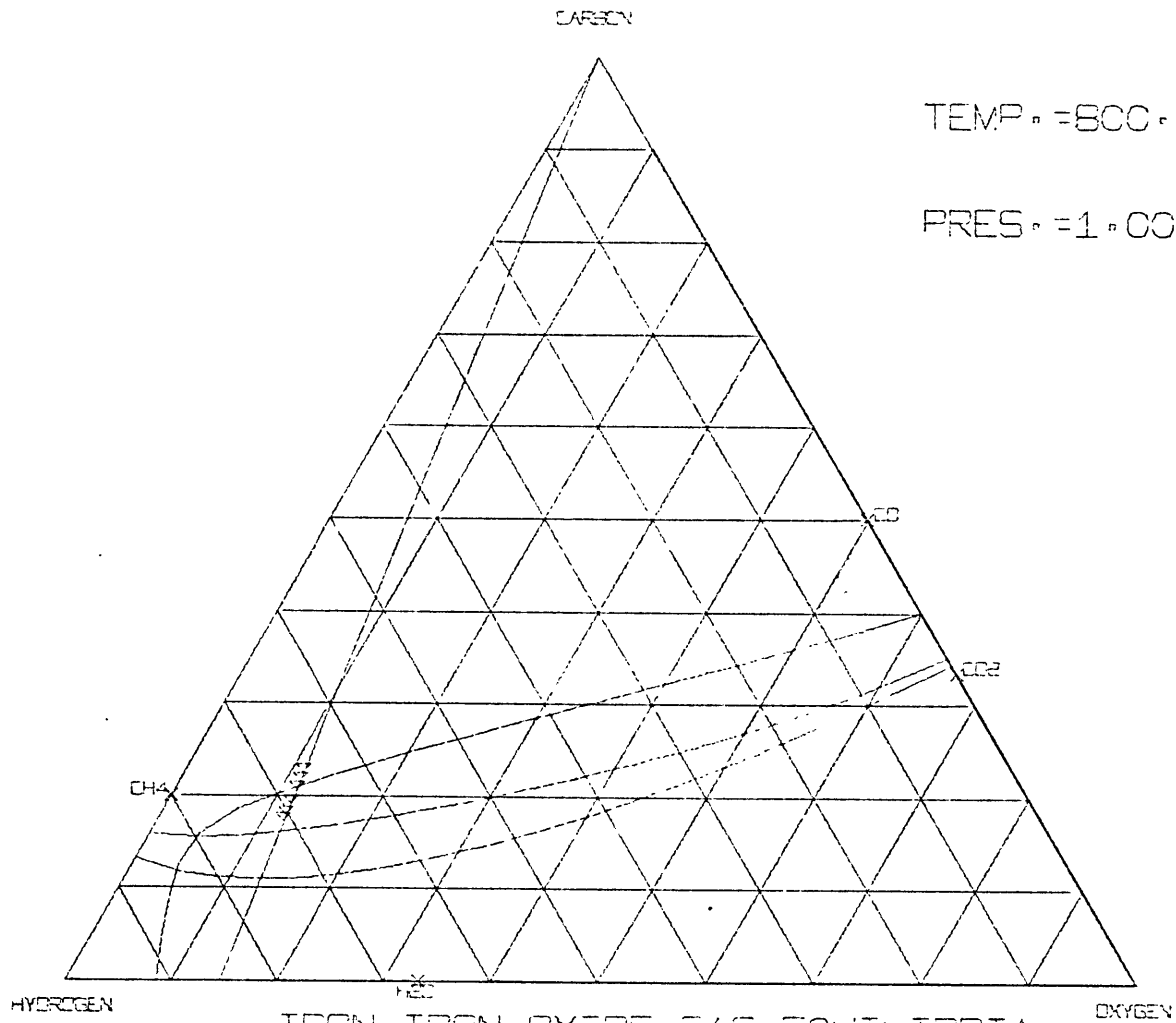
PCO2 PCO PCH4 PH2O PH2 PTOTAL
0.126634 0.149165 0.546503 0.029612 0.148063 0.999979
PERCENTAGE H 66.95 PERCENTAGE O 11.38 PERCENTAGE C 21.66

FLOW SETTING
6.964885 10.680233 46.452781 24.195034 4.886093

CONDITION 6. PH2O/PH2 0.15, C/H 0.34, O/H 0.17
GAS COMPOSITIONS

PCO2 PCO PCH4 PH2O PH2 PTOTAL
0.115832 0.181921 0.566275 0.017728 0.118191 0.999949
PERCENTAGE H 66.19 PERCENTAGE O 11.25 PERCENTAGE C 22.54

FLOW SETTING
6.370766 13.025583 48.133377 15.978784 3.900330



UPPER CURVE IRON-IRON OXIDE-GAS EQUILIBRIA
 MIDDLE CURVE IRON-IRON CARBIDE-GAS EQUILIBRIA
 LOWER CURVE GRAPHITE-GAS EQUILIBRIA

Figure 34 Visual Check on Experimental Set Conditions

3.2.2 Equipment Start-Up

Initially, each piece of electronic equipment was turned on for at least thirty minutes prior to run time. This allowed sufficient time for steady-state behavior to be established in all electrical components. During this thirty minute warm-up period, the catalyst assembly was prepared for the experiment.

The quartz catalyst assembly was first weighed on the Sartorius pan balance to an accuracy of ± 1 mg. An appropriate length of quartz wool was then inserted into the catalyst assembly; the assembly was again weighed and the weight recorded. The next step was to weigh out the amount of catalyst desired and insert this into the catalyst carrier. Normally 450 mg of number 2 steel wool were loaded and positioned in the assembly so as to be completely encompassed by the isothermal region (Figure 30). In Table 3 both the composition of the steel wool catalyst and the surface area are shown.

Table 3 Catalyst Composition

<u>ELEMENT</u>	<u>WT %</u>
Fe	\approx 98.882
Mn	0.700
C	0.360
P	0.040
S	0.018
N	0.400 ppm

BET Area by Krypton Adsorption: $389 \text{ cm}^2/\text{g}$

At this point, a check was performed to ascertain if the combined weight of the quartz catalyst assembly, quartz wool, and steel wool catalyst was equivalent to that calculated from the individual components. A tolerance of ± 2 mg was allowed; if a difference of more than 2 mg was observed, the catalyst loading procedure would be repeated. A final piece of quartz wool was inserted below the catalyst to hold the steel wool in place as well as to distribute evenly the gases as they enter the catalyst bed.

The catalyst assembly was then placed in the lower reactor support tube; the top section of the support tube placed over it and the upper portion of the furnace, in turn, over that. The furnace was next centered under the Sartorius electrobalance and all associated lines connected. When in the furnace, the catalyst assembly was weighed by attaching a monofilament Nylon line to the quartz tube suspension bar. The pre-furnace weight measurements were used as a criterion to determine if the catalyst assembly was freely suspended. Adjustments were made by positioning of the overhead Sartorius balance.

The next step in the start-up procedure was the heating of feed and sample lines, the sample valve system, the saturator subsystem, and the reactor furnace.

The feed and sample lines were heated using powerstats which controlled Briskeat heating tapes. Line temperatures were maintained at approximately 423 K.

The sample valve subsystem was heated by two 65-W

cartridge heaters. A 220 Hewlett-Packard proportional controller maintained the sampling system at 423 K.

The saturation preheater powerstat was next set to the desired levels. This was closely followed by setting the bath contact control heater to the appropriate temperature. The saturator bath temperature often took several hours to reach steady state; and, thus, the bath contact heater was often energized several hours prior to the anticipated run time. If water was not to be a component in the reaction mixture, the saturator system would be by-passed and neither the preheater nor bath heater were turned on.

Prior to activating the furnace heaters, a helium feed rate of 20 cc/s (STP) was fed through the appropriate feed lines to the reactor. The Dubuque III reactor controller was then set to 673 K and activated.

While the system components were being heated, a vacuum was drawn on the sample valve system. A vacuum of at least 400 N/m² was used as a criterion of whether the sample valve was vacuum tight. If a vacuum of 400 N/m² could not be drawn on the sample valve system, corrective action was taken.

After all initial set temperatures were obtained, an external standard was flushed three times through the sample valve system; and, four 0.25 cm³ samples were sequentially fed to the chromatograph analytical columns.*

* All critical gas chromatograph conditions (Table 2) were set 24 hours prior to start-up in order to establish steady state.

Their peak shapes were noted and integrated areas recorded. These established the response factors for the thermal conductivity cell necessary for the data reduction program (Appendix 7.4.1).

If the water saturator was operational, the helium feed gas would be by-passed through the bubbler; this eliminated the water vapor initially accumulated in the closed saturator loop. The helium was then shut off and $10 \text{ cm}^3/\text{s}$ (STP) of CO_2 would be passed through the bubbler. Two calibration samples were then taken; hydrogen at $20 \text{ cm}^3/\text{s}$ (STP) was turned on; the saturator was by-passed (i.e., hydrogen was fed dry); and, the furnace temperature set to the experimental run conditions. These calibration samples were necessary to establish the thermal conductivity cell response factor for water, which was not present in the external standard. Carbon dioxide was used as the carrier gas because its response gave the best precision (<0.8% error) of all gases measured.

Steel wool which has been oxidized by carbon dioxide or water and then reduced in hydrogen, had been shown by Manning (1975) to have a high initial activity. This was desirable since this increased activity allowed a larger variety of conditions to be tested in a given time period (10 to 18 hours on stream).

The reactor was allowed to come up to temperature in hydrogen; the hydrogen was then by-passed out of the reactor and the catalyst assembly was attached to the Sartorius electrobalance. The weight of the assembly was taken, recorded, and used as an initial starting weight in the data reduction program.

Two samples of pure hydrogen were also taken to measure impurities as well as to establish the hydrogen calibration curve following the method of Purcell and Ettore (1965).

3.2.3 Run Procedures

The desired gas compositions were obtained by adjusting the Brooks mass flow regulators in combination with the appropriate manometers. The desired settings were obtained from the predetermined calibration curves. Approximately one minute was necessary to obtain steady-state behavior.

Throughout the duration of the run, inlet and exit gas samples were taken every six minutes. The following procedure was normally followed: first, a vacuum of 400 N/m^2 was continuously drawn on the sample valve system; second, prior to the injecting of any sample, the sample loop was flushed three times with the desired gas mixture; third, the appropriate sample was drawn into the sample valve loop, the temperature and pressure recorded and the sample injected into the analytical columns. Two inlet gas samples, a weight reading, then a sequence of effluent samples was the normal sequence followed.

Weight measurements were taken periodically (usually every 12 minutes) by exhausting the reaction gas to the atmosphere and attaching the catalyst assembly to the electrobalance. During weight measurements, the furnace would be automatically shut off while the gases were by-passed to the atmosphere. This was done as a safety precaution. The reactor

center-line temperature would drop as much as 5 K during these weight measurements. Upon completion of the weighing procedure, the furnace would automatically come back on and, the center-line set temperature was quickly obtained.

A continuous record of all pertinent temperatures was kept with a 27 channel Honeywell recorder. An instantaneous reading of catalyst bed temperature, reactor center-line temperature, and saturator temperatures were obtained with an Omega 200 series digital pyrometer.

3.2.4 Shut-Down Procedures

At the conclusion of a given run, all reactant gases were shut off and $20 \text{ cm}^3/\text{s}$ (STP) of helium was fed to the reactor. All powerstats were turned down and the Dubuque III furnace controller shut off. Helium flow was continued until the center-line temperature was at ambient temperature.

The feed and sample lines were next disconnected from the furnace and the furnace was, in turn, taken out from under the Sartorius balance. The catalyst assembly was removed from the furnace and visually examined; any pertinent observations were recorded. The catalyst and deposited carbon were then put in specimen bottles, sealed, and labelled.

4. Results

4.1 Preliminary Experiments

In the first experiments binary gas mixtures were normally used. In all cases, the catalyst charge was 250-500 mg of steel wool and the reactor temperature was approximately 825 K. The total reactor pressure was 1 atmosphere.

4.1.1 Hydrogen-Carbon Monoxide Mixtures

The initial experiments were made to determine the maximum amount of carbon that could be deposited per unit weight of catalyst as well as to elucidate the carbon deposition mechanism. Figure 35 illustrates the typical behavior observed. In this experiment 450 mg of steel wool catalyst were exposed to a gas mixture of 50% hydrogen-50% carbon monoxide.

Carbon deposition was periodically measured by weighing the catalyst assembly and determining the differential weight change. Correcting for the time off stream, these measurements were used to determine change in rate as a function of time. As shown, the rate of carbon deposition is seen to increase rapidly with time reaching a maximum of 38 mg/min after approximately 204 minutes. The rate was then observed to fall. The minimum P_{H_2}/P_{H_2O} was found to coincide with the point of maximum carbon deposition rate; the minimum P_{H_2}/P_{H_2O} value being equal to 11.0. The reason for the rapid drop-off in rate is unclear. The bulk P_{CO}/P_{CO_2} and P_{H_2}/P_{H_2O}

RUN A-4.0

TEMPERATURE 823 K

PRESSURE $1.01 \times 10^5 \text{ N/m}^2$

CATALYST CHARGE 450 mg

CO/H₂ FEED RATIO = 1.0

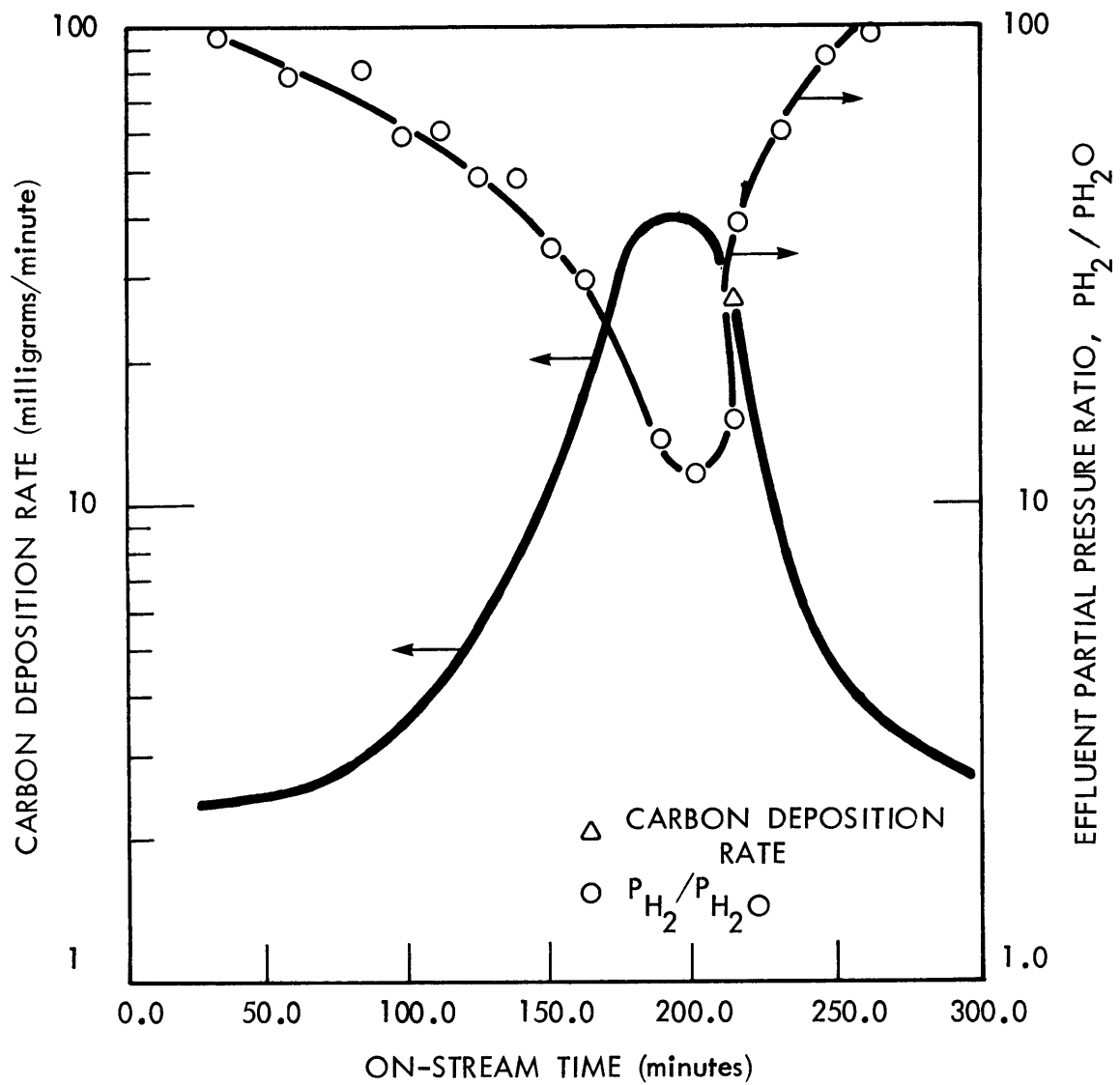


FIGURE 35

CARBON DEPOSITION RATE AND P_{H_2} / P_{H_2O} RATIO VERSUS ON-STREAM TIME

ratios were not in the range where oxide would be expected to form. The catalyst assembly was securely fastened down, eliminating any possible gas by passing the catalyst bed. Slight pressure fluctuations were noted in the reactor back pressure; this suggested plugging with concomitant channeling in the catalyst bed itself. Visual examination of the catalyst bed did indicate a dense packed carbon slug. However, no preferential area of carbon deposition, indicating channeling, was observed. It is interesting that 30% of the total carbon deposited occurred after the fluctuations were recorded in the reactor. This indicated that a substantial portion of the catalyst was seeing reactant gases during the entire experiment. Also, the minimum P_{H_2}/P_{H_2O} at which carbon deposition stopped was found to correspond to that observed by Everett (1967). Their value was reported to be ten.

During the entire experiment, the conversion of hydrogen was virtually constant at less than 1.5%. The carbon monoxide conversion was observed to increase to a maximum of 15%. When conversion of carbon monoxide was $\geq 9\%$, based on the Boudouard reaction, the amount of carbon predicted was within 70% of that found.

Interestingly, no methane was observed until the rate of carbon deposition was almost at a maximum; and then only trace amounts were observed. However, once begun the methane concentration remained virtually fixed until run completion. The final catalyst weight recorded was 3.048 grams, corresponding to a C/Fe atom ratio of 32. Walker et al. (1959) reported

C/Fe ratios as high as 100, before catalyst deactivation.

In an effort to explain this rather odd behavior, an experiment was run at identical conditions with a chromel-alumel thermocouple imbedded in the catalyst bed. The bed temperature was recorded periodically using a potentiometer. Figure 36 illustrates a temperature rise of 100°K occurred in the catalyst bed during the period of rapid carbon deposition. A plot of volume percent CO₂, H₂O, and CH₄ in the effluent indicates the time of maximum temperature rise corresponds to the time when CO₂, H₂O, and CH₄ reach their maximum concentrations (see Figure 36). Although CO₂ and H₂O concentrations were seen to fall off after the peak temperature was achieved, methane concentration appeared to stay fixed.

The following conclusions may be drawn from Figure 36. The increase in carbon deposition rate with time noted by Manning is, at least in part, due to a rapid temperature rise in the catalyst bed. Methane formation appeared to be significant only after the hot spot developed in the bed. This is in agreement with the observation made by Browning et al. (1950) who indicated reaction G interfered with his equilibrium



measurements at temperatures above 930 K (which is approximately equal to the hot spot temperature).

At high carbon monoxide conversion (i.e., $\geq 9\%$), carbon

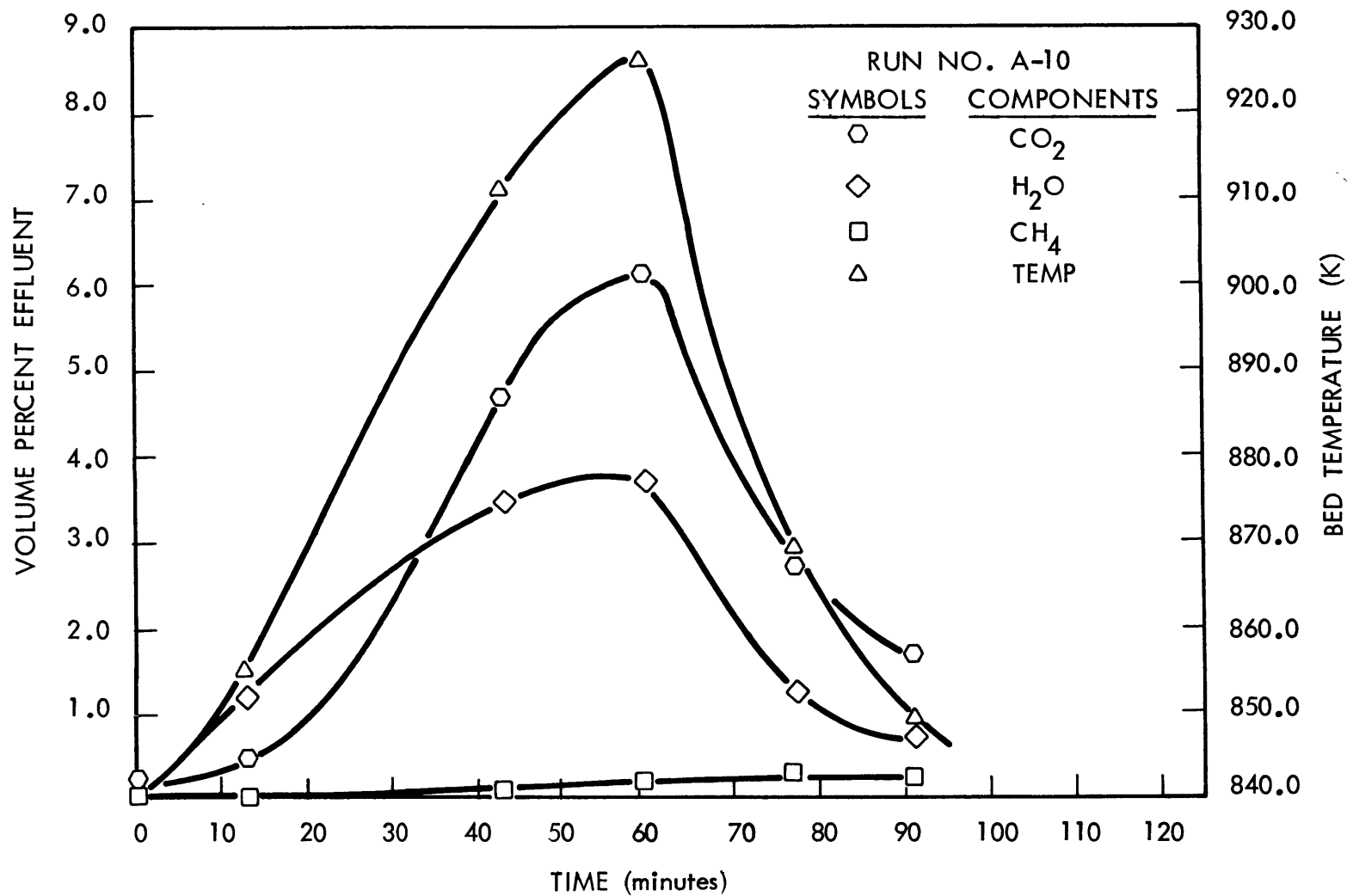


FIGURE 36 VOLUME PERCENT EFFLUENT VERSUS TIME AND BED TEMPERATURE VERSUS TIME

deposition appears to come primarily from the Boudouard reaction.



It is believed that the drop in carbon deposition rate and temperature were probably due to channeling in the bed; and thus, no conclusive statements concerning the observed temperature or concentration peaks could be made.

4.1.2 Surface Oxidation Study

Next, experiments were run to verify the proposed inhibiting effect of iron oxide. Figure 37 is representative of the behavior observed when binary gas mixtures of carbon monoxide and carbon dioxide were run over a pre-carboned steel wool catalyst at 825 K and $1.01 \times 10^5 \text{ N/m}^2$.

A 50% H_2 - 50% CO mixture was first used to deposit 650 mg of carbon on 251 mg of iron catalyst. A stream of carbon dioxide was next fed to the reactor. Figure 37 shows carbon deposition immediately stopped. The small increase in weight is presumably due to oxide formation (reaction J-A). It was thought that reaction D would proceed in the reverse direction, but these data suggest that oxide formation is fast and also, that iron oxide (Fe_3O_4) apparently is not catalytic for reaction of carbon with carbon dioxide.

Following oxidation with carbon dioxide, binary gas mixtures having $P_{\text{CO}}/P_{\text{CO}_2}$ ratios of 1.13 and 4.00 were alternately fed to the reactor. At a value of 1.13, no carbon deposition was observed. The equilibrium concentration of

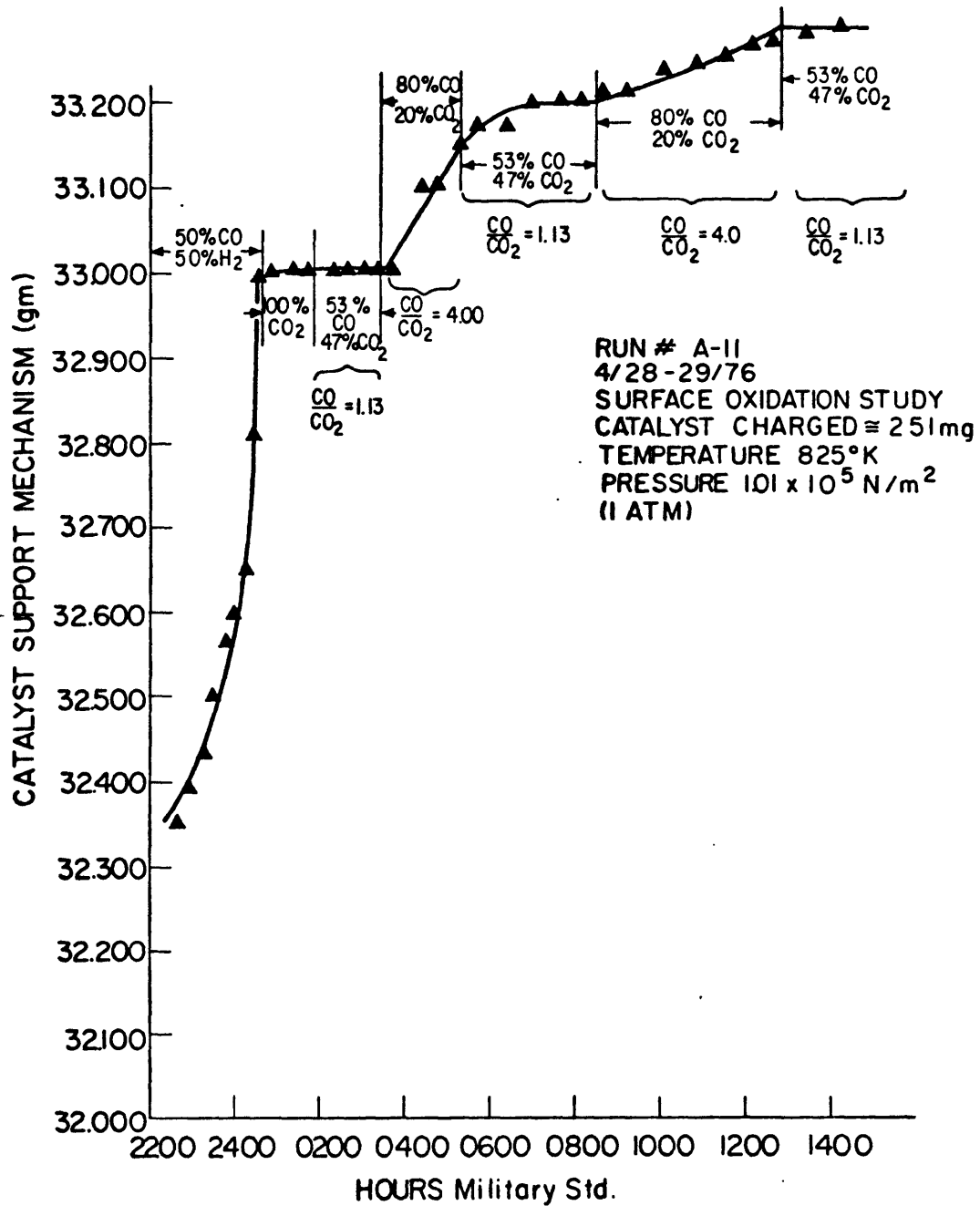


FIGURE 37 EFFECTS OF OXIDE INHIBITION IN BINARY GAS MIXTURES OF CO-CO₂

carbon monoxide for reaction D is 13%; thus, carbon deposition would be expected. However, at a $P_{\text{CO}}/P_{\text{CO}_2}$ ratio of 1.13, α -Fe is at equilibrium with Fe_3O_4 (reaction J-A). It is believed that the "catalyst" was in the oxidized state and as such was not a catalyst for carbon deposition.

In support of this hypothesis, when the $P_{\text{CO}}/P_{\text{CO}_2}$ ratio was increased to 4.0 (i.e., strongly reducing atmosphere), rapid carbon deposition occurred. This reducing atmosphere favored α -Fe formation which in turn catalyzed reaction D.

The step-like system response shown in Figure 37 was typical. The system was seen to respond more sluggishly after the initial oxidation and reduction. This behavior is due to the increased carbon deposition on the catalyst bed which acts to dilute the effective area.

Quantitatively, the amount of carbon formed was usually within 10% of that predicted by carbon monoxide conversion. However, variations as high as 50% were noted. Considering the degree of carbon monoxide conversion (i.e., < 1%), this discrepancy was considered reasonable.

Figure 37 implies that reaction D is inhibited by the formation of iron oxide in both the forward and reverse direction. The quantitative results show that carbon deposition can be accurately represented by the Boudouard reaction.

4.2 Oxide Inhibition in 5 Component Gas Mixtures

The results from the preliminary experiments verified that iron oxide (Fe_3O_4) will inhibit carbon deposition from

binary gas mixtures of carbon monoxide and carbon dioxide. The next step in the investigation was to determine the effect of oxide formation on carbon deposition under normal Bosch operating conditions, i.e., those in which there is a gas phase consisting of methane, hydrogen, carbon monoxide, carbon dioxide, and water. The iron catalyst would have carbon present along with small amounts of iron oxide and/or iron carbide.

In order to study and evaluate the inhibiting behavior of iron oxide(s), a series of experiments were performed to determine: the effect(s) of oxidation and reduction on catalyst structure prior to and during reaction, to ascertain whether the system response was more sensitive to variations in P_{H_2}/P_{H_2O} or the P_{CO}/P_{CO_2} ratio, and, finally, to determine the precise location of the phase boundary for the α -iron/wustite and α -iron/magnetite systems. The phase boundary location was determined by the point where carbon deposition was observed to start and stop.

4.2.1 The Effects of Preconditioning at 900 K

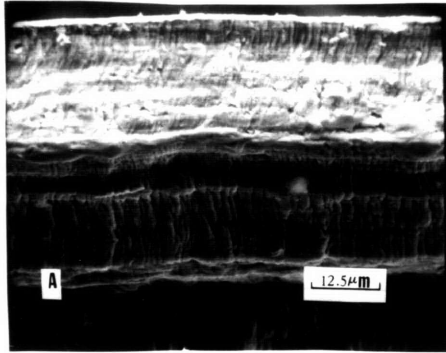
Manning (1976) reported the initial rate of carbon deposition increased substantially on preoxidation of the steel wool catalyst. In order to be able to utilize this effect, a clear understanding of the oxidation-reduction process and/or processes occurring on the catalyst surface was necessary.

A series of micrographs of an unconditioned #2 steel

wool fiber is shown in Figure 38. Micrograph (a) is an overview of the entire fiber. Micrographs (b), (c), and (d) are high-magnification photographs illustrating three distinctive surface structures. The top region of the fiber (micrograph {b}) is seen to be a relatively smooth area, showing some stress cracks and holes. Micrograph (c) illustrates the central region of the fiber. Here the structure appears to be layered. The lower section of the catalyst fiber seems to be a combination of the top and central regions showing both regular layering and periodic cracks and holes. This type of surface structure was found on all unconditioned #2 steel wool fibers and is apparently the result of the continuous shaving process used in steel wool manufacture. Surface area measurements using BET methods with krypton adsorption indicated an area of $389 \text{ cm}^2/\text{g}$ for the unconditioned catalyst.

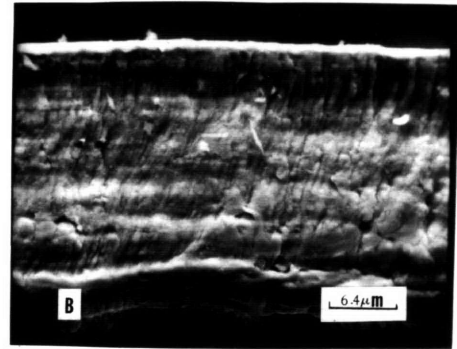
The steel wool catalyst prior to introduction of the reactant gases was typically preconditioned according to a standard procedure. This procedure consisted of oxidizing in a carbon dioxide-water atmosphere for several hours at 900 K and $1.01 \times 10^5 \text{ N/m}^2$. This step was followed by reduction in a flowing stream of hydrogen.

Scanning electron micrographs of the preconditioned catalyst are shown in Figure 39. Micrographs (a) and (b) are different areas of the same fiber. Note the sponge-like appearance which has led to this type of structure being named "sponge-iron". Electron micrograph (c) clearly shows the sintered iron grains which make up this porous structure.

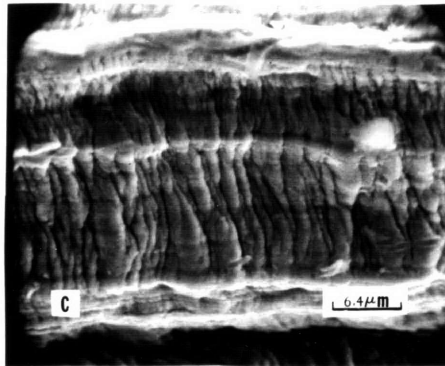


OVERVIEW, STEEL WOOL CATALYST

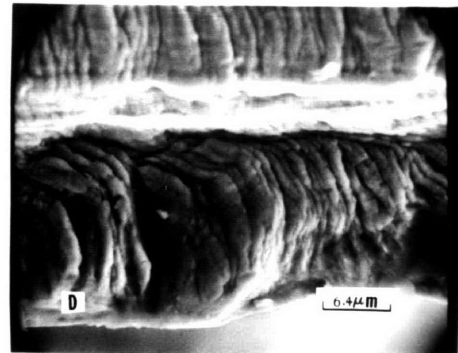
{unconditioned}



TOP REGION, STEEL WOOL CATALYST

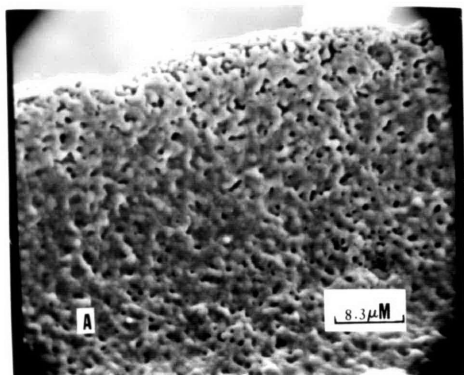


CENTRAL REGION,
STEEL WOOL CATALYST

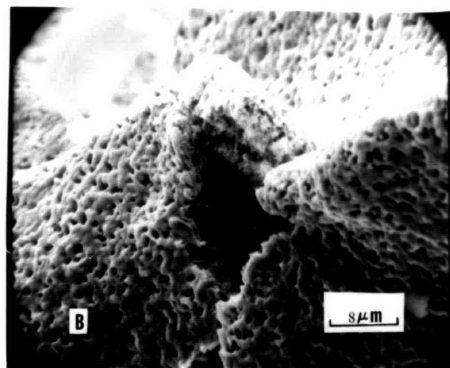


BOTTOM REGION
STEEL WOOL CATALYST

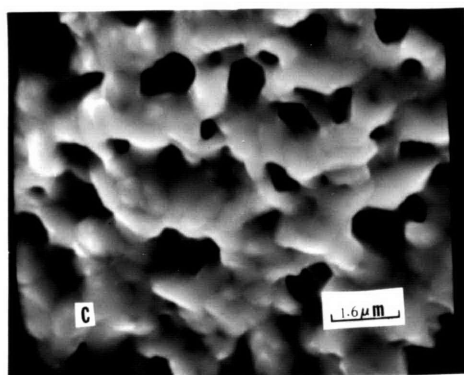
FIGURE 38 UNCONDITIONED #2 STEEL WOOL CATALYST



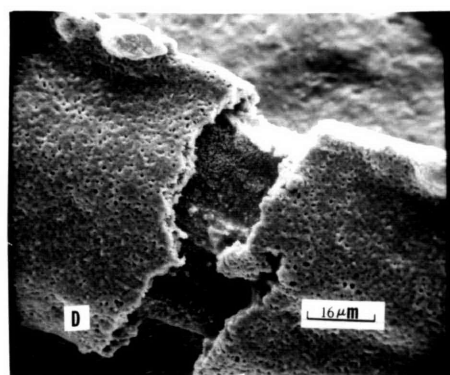
A
PRECONDITIONED CATALYST
AT 900 K



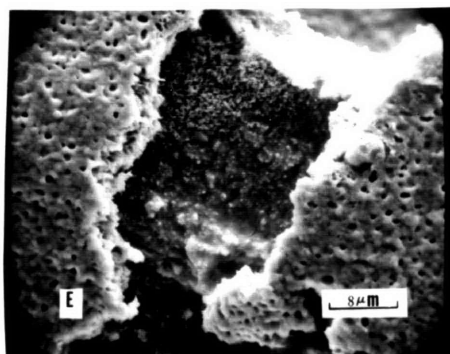
B
SPONGE-IRON STRUCTURE



C
SPONGE-IRON MORPHOLOGY
note: sintered iron grains



D
FISSURE SHOWING SHELL AND CORE
MORPHOLOGY



E
STRUCTURE INDICATING MULTIPLE
OXIDE FORMATION

FIGURE 39 PRECONDITIONED CATALYST AT 900 K

Also shown in Figure 39 are two micrographs of a fissure observed on another catalyst strand.

The shrinking-core type behavior described by Spitzer et al. (1966) is clearly shown in micrographs (d) and (e). The porous shell is seen to be composed of two distinct layers, indicating multiple oxide formation. As discussed by Spitzer (section 2.2.2.3), multiple layers will be observed if the catalyst behaves as a porous particle under diffusion control. BET surface area measurements indicated an area of $1912 \text{ cm}^2/\text{g}$. This corresponds to an increase in effective surface area of 500% relative to the unconditioned catalyst.

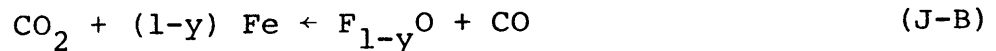
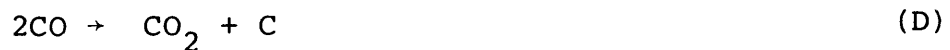
The electron micrographs shown in Figures 38 and 39 indicate the reason for the increase in initial reaction rate noted by Manning, the increase being the result of the increased effective surface area. Figures 38 and 39, however, point out a much more important structural effect. This is the possibility of multiple oxide formation with its concomitant kinetic problem. Spitzer et al. (1966) clearly showed that if multiple oxide formation occurs, depending on the past history of the oxygen activity and the time at a given oxygen activity, the response of the system to changes in oxygen activity will vary. That is, changing $P_{\text{H}_2}/P_{\text{H}_2\text{O}}$ and/or $P_{\text{CO}}/P_{\text{CO}_2}$ ratios will cause a different response. This would indicate a system response time is to be expected and that this response time may vary.

4.2.2 Carbon Deposition-Control by P_{H_2}/P_{H_2O}

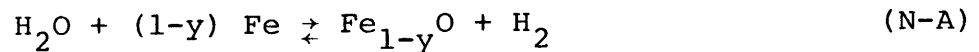
The question arises as to whether the P_{H_2}/P_{H_2O} or P_{CO}/P_{CO_2} ratio controls oxidation and reduction on iron catalysts. This question as well as the question of the inhibition of carbon deposition in a five component gas mixture was next examined.

In run A-18, 415 milligrams of steel wool catalyst were oxidized in a flowing stream of 77.2% carbon dioxide-22.8% water at 900 K and 1.01×10^5 N/m².

The P_{H_2}/P_{H_2O} ratio was varied between a value of ∞ (i.e., no water) and a value of 1.0. The remaining gas compositions were set so that reaction D proceeds to the right while; at this P_{CO}/P_{CO_2} ratio, reaction J-B should proceed in the direction to reduce iron oxide (i.e., to the left).



If the surface condition is controlled by reaction N-A, alternating the P_{H_2}/P_{H_2O} ratio between a value of ∞ and 1.0 should effect carbon deposition.



The results from A-18 are shown in Figure 40. As shown, carbon deposition was controlled by adjusting the P_{H_2}/P_{H_2O} ratio which in turn controlled the surface condition according to reaction N-A. Also, as had been expected,

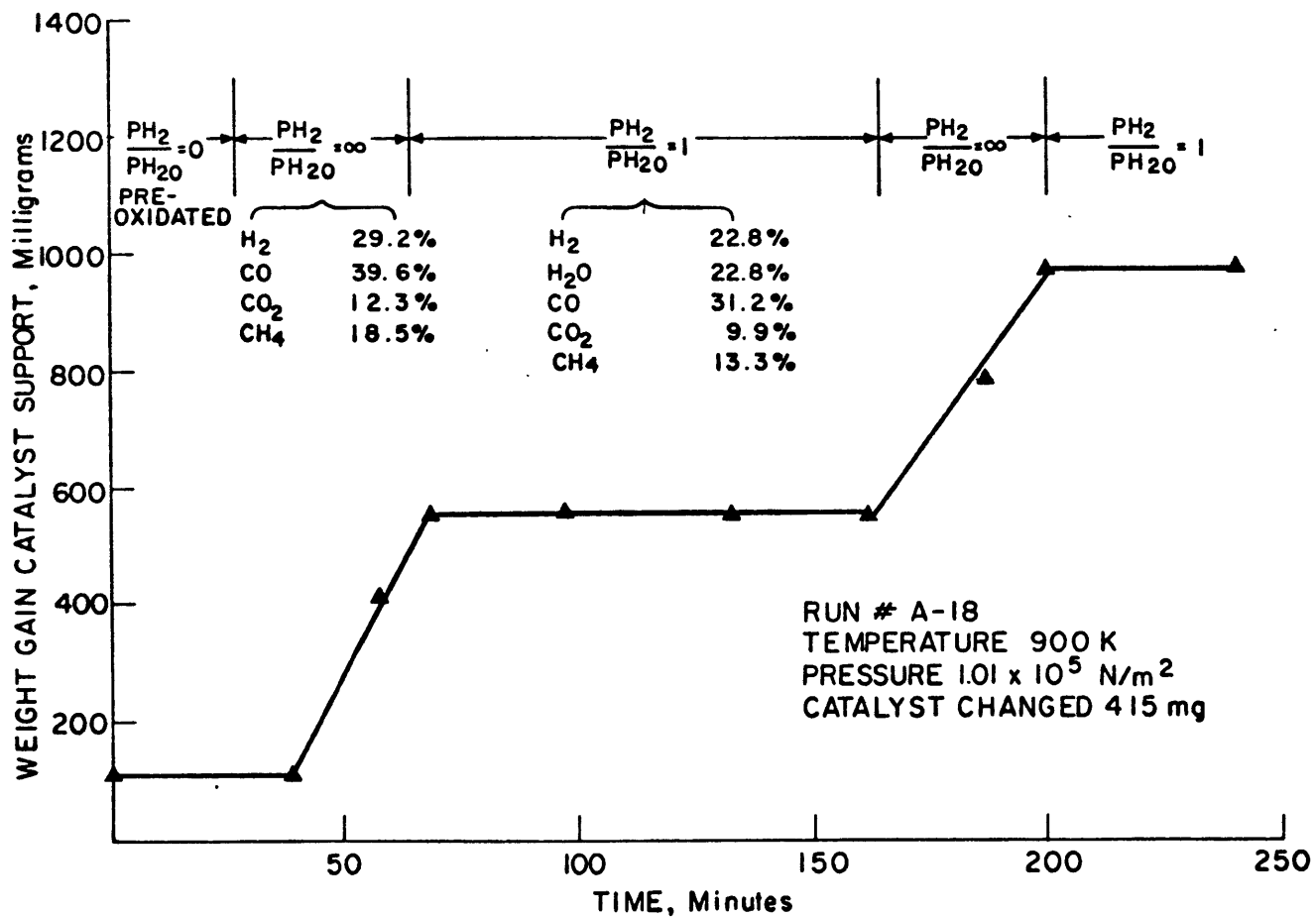


FIGURE 40 CONTROL OF CARBON DEPOSITION THROUGH VARIATION IN PH₂/PH₂O RATIO

iron oxide appeared not to be a carbon deposition catalyst. Unfortunately, due to the inherent error in effluent concentration measurements, at low conversions, nothing quantitative could be said about the reaction mechanism.

The effect of deposited carbon on the response of the metallic iron catalyst was next explored. This was necessary because a Bosch catalyst will routinely have carbon deposited on it. Run A-19 is representative of the behavior found. Initially, a P_{H_2}/P_{H_2O} ratio of (∞) was used to deposit approximately 350 mg of carbon on 415 mg of steel wool catalyst. The reactor set temperature was 900 K and the total pressure $1.01 \times 10^5 \text{ N/m}^2$. As shown in Figure 41, no appreciable weight gain was noticed with P_{H_2}/P_{H_2O} ratios between 1.0 and 2.26. Upon changing the P_{H_2}/P_{H_2O} to 4.58, rapid carbon deposition was observed. This behavior is in complete agreement with the results shown from A-18 and indicates that reaction N-A controls the surface condition, the P_{H_2}/P_{H_2O} ratio at equilibrium for reaction N-A being 2.75.

Although only runs which approach the iron-iron oxide equilibrium from the oxide side of reaction N-A were shown, the same type of behavior was observed coming in from the reduced side.

The runs approaching the phase boundary from the reduced side were not shown because carbon deposit was rapid and it could only be shut off once before the catalyst assembly exceeded its design capacity. That is, the pressure drop across the bed became so great it equaled its weight. This,

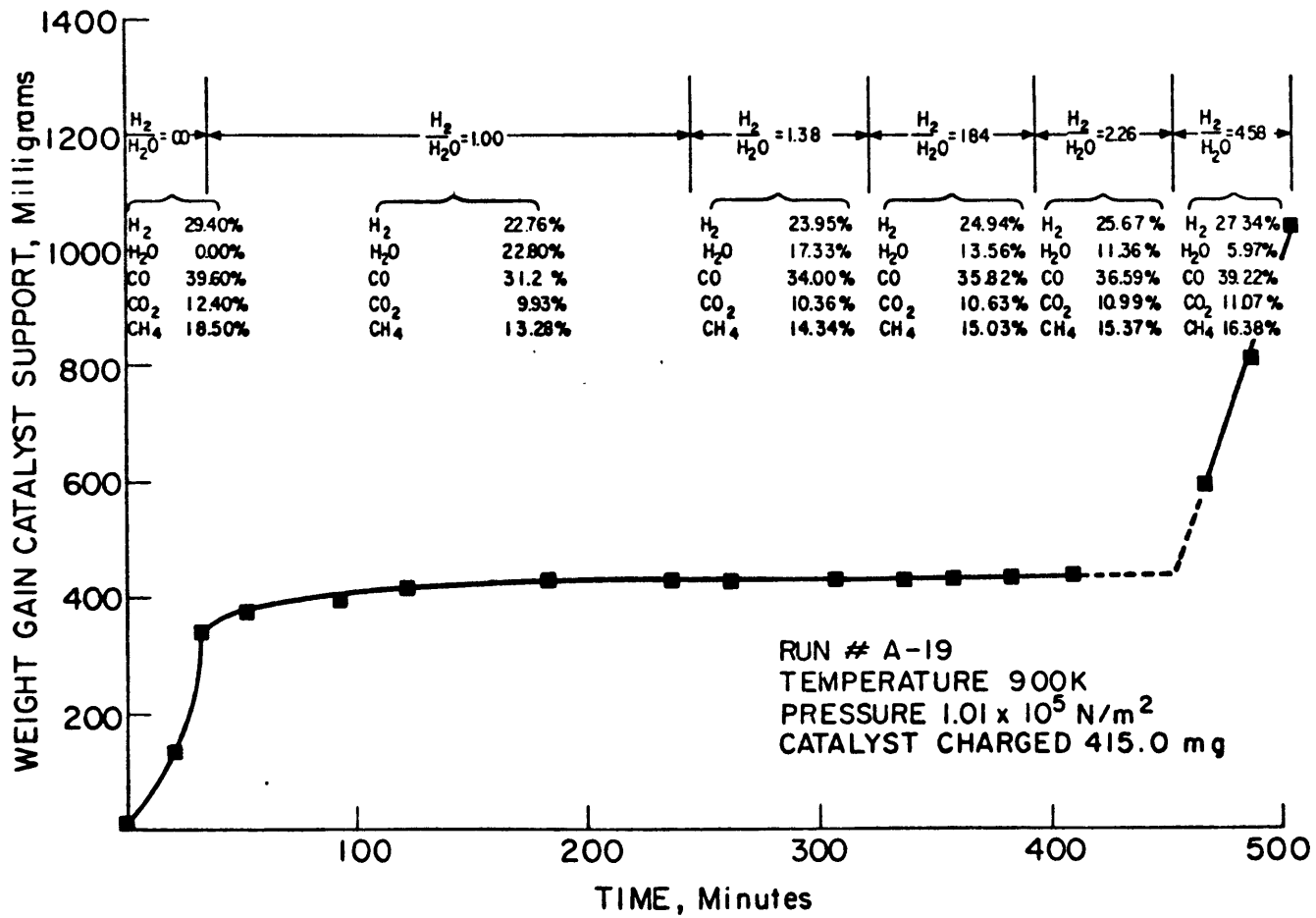


FIGURE 4I CARBON INHIBITION AS A FUNCTION OF PH₂/PH₂O RATIO

in turn, caused the assembly to float. In all cases, carbon deposition ceased when the water content was above that for equilibrium according to reaction N-A.

The question may arise as to the effect of oxide inhibition on reaction

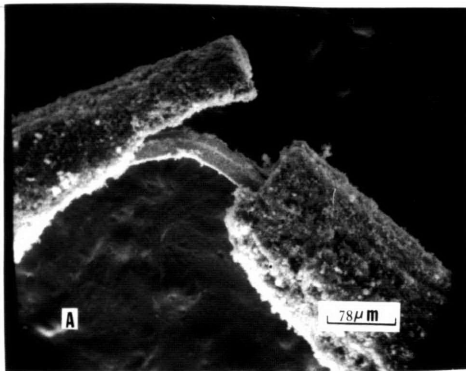


From the data of Figure 40 and 41 it is evident that reaction D is inhibited by oxide formation, but what of reaction E? Run A-19 indicates this reaction is also inhibited by oxide formation. Thus it can be conclusively stated that oxide formation inhibits deposition from both reaction D and reaction E at 900 K. Also, once again, the surface condition is controlled by reaction N-A through adjustments in the $P_{\text{H}_2}/P_{\text{H}_2\text{O}}$ ratio.

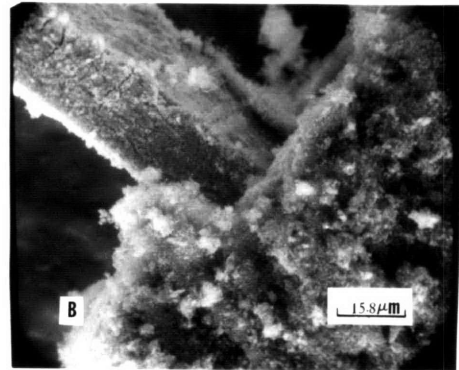
4.2.3 Structural Changes During Reaction

In varying the $P_{\text{H}_2}/P_{\text{H}_2\text{O}}$ ratio to control carbon deposition, the discussion in Section 4.2.1 indicated the catalyst structure may change. In addition, carbon deposition has been shown to alter the catalyst structure (Section 2.2.3.3). In order to better understand these simultaneous structural changes, a metallurgical examination was performed.

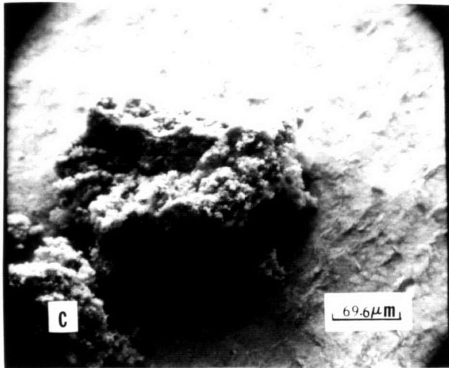
Figure 42 shows a series of electron micrographs of the catalyst from run A-18. Micrograph (a) reveals the shell and core type structure formed during reaction. This type of morphology results from the oxidation-reduction sequence used in run A-18. Carbon-bearing gases diffused through the porous iron shell depositing carbon along its length. This



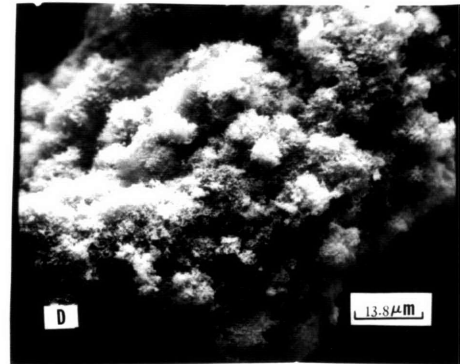
CATALYST FIBER RUN A-18
OXIDATION-REDUCTION MORPHOLOGY



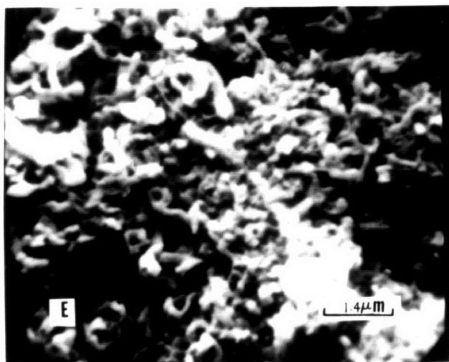
SHELL AND CORE STRUCTURE



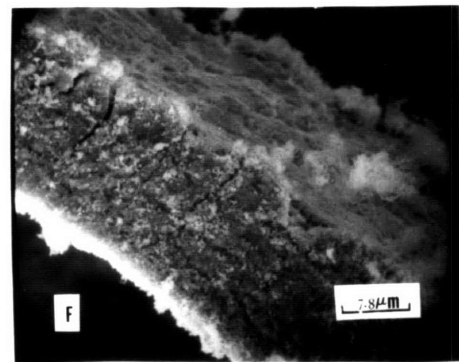
PORTION OF EXTERIOR SHELL
{gold coated}



NODULE-STRUCTURE OF EXTERIOR
SHELL



TUBULAR AND CIRCULAR CARBON
FIBERS



CORE REGION, RUN A-18

FIGURE 42 CATALYST STRUCTURE AFTER REACTION, RUN A-18

carbon deposition appears to make the exterior shell brittle, breaking away easily from the more structurally solid core. A high magnification photograph of the shell surface is shown in micrograph (b). Carbon fibers are seen to form in bundles or nodules. Transmission electron micrographs indicate these nodules are large chunks of electron-dense material, presumably iron or iron compounds. A small portion of the exterior shell was removed for examination and is shown in micrographs (c), (d), and (e). Again the nodule-like fiber bundles are clearly shown. In micrograph (e) carbon fibers can be seen having both tubular and circular shape.

Interestingly, the core region shown in micrograph (f) is relatively smooth showing none of the surface features noticed on the untreated steel wool. Also, very little carbon was deposited on the core fiber indicating both its low effective surface area as well as probable diffusion limitations through the shell for carbon-bearing gases.

4.2.4 Structure of Carbon Fibers

Manning (1976), Walker et al. (1959), and Baker et al. (1972) and many other investigators have observed carbon fiber formation occurring during carbon deposition experiments. Manning reported flat ribbon-shaped fibers, while Walker and Ruston observed both tubular and coil-shaped fibers which were hollow and often, although not always, associated with an electron dense tip. Transmission electron microscopy (TEM) was used to observe the type or types of fiber formed in this

investigation.

Figure 43 is a TEM micrograph of a typical carbon fiber found in the exterior shell region of a catalyst fiber from run A-18. Three points are worth mentioning: first, the "camel" shaped fiber is suspended securely from the main catalyst surface by carbon fibers. This is indicative of the type of shell structure developed during carbon deposition. The shell region appears to consist of an intricate network of interwoven fibers connected securely by fiber bundles or nodal points. These nodal points consist of iron and/or iron compounds. Second, in agreement with Baker and Walker, the fiber shaft appears to be hollow. Third, most fibers were observed to have an electron dense tip.

Electron diffraction patterns were taken of the head and shaft regions of the fiber. The results are shown in Table 4.

The electron dense tip was seen to have some Fe_3O_4 and Fe_2O_3 , along with some graphite (well ordered). A small amount of $\alpha\text{-Fe}$ was observed along with what appears to be Fe_{20}C_9 .^{*} The shaft region had intense patterns for graphite and indicated trace amounts of Fe_2O_3 and $\alpha\text{-Fe}$. Very little Fe_3O_4 and no iron carbide was found on the shaft. These results agree for the most part with those of Walker and

* The diffraction patterns for various carbides are very similar. Consequently, identification is often a matter of experience and judgment.

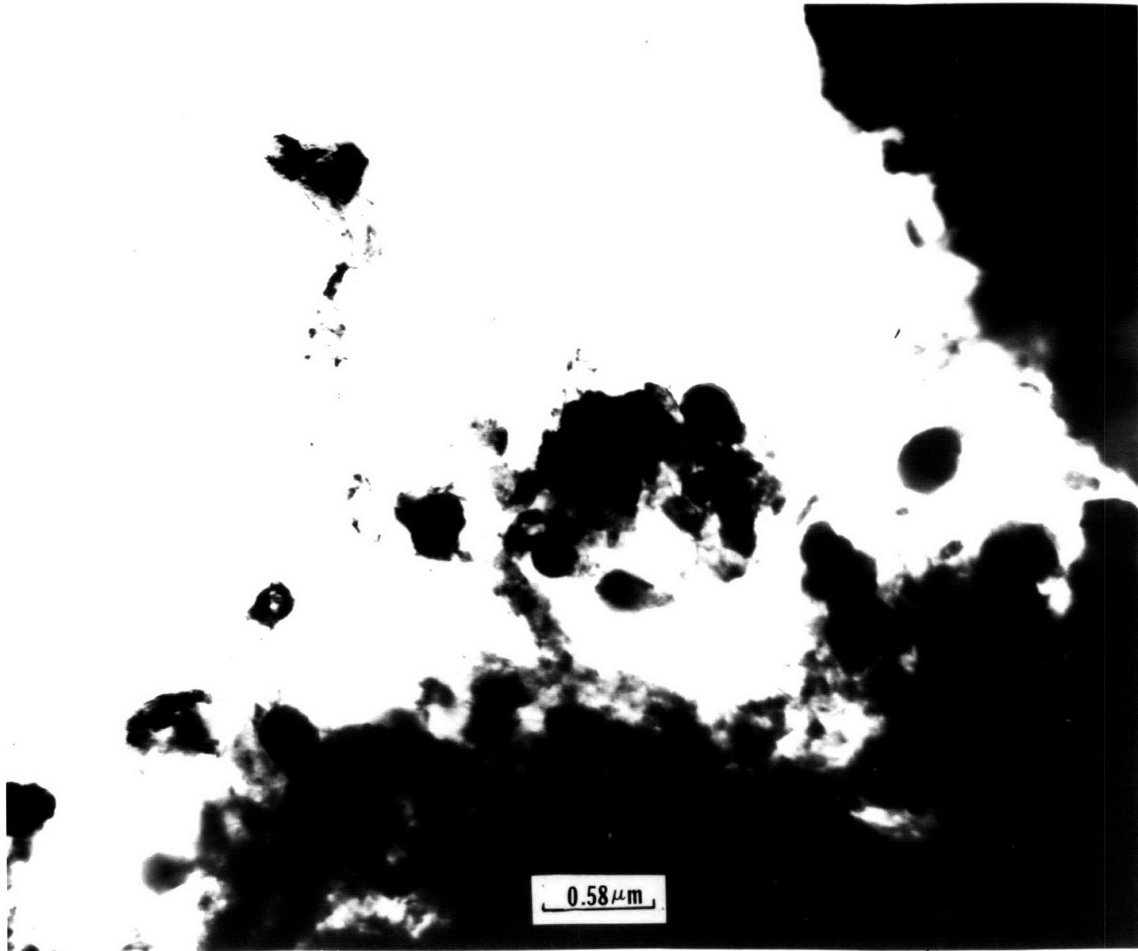


FIGURE 43 CARBON FIBER FROM EXTERNAL SHELL REGION , A-18

Diffraction Pattern	Experimental Data d I/IO (A) Visual Est.		Graphite ASTM Card 13-148	Graphite ASTM Card 12-212	α -Fe ASTM CARD 6-0696	"Fe-C" 6-0686	ϵ -Fe ₂ C 6-0670	Fe ₂₀ C ₉ 8-370 9-369	Fe ₃ C 6-0688	α -Fe ₂₀ O ₃ 13-534	Fe ₃ O ₄ 11-614	Fe _x O 6-0615
	Electron Dense Tip	3.49	100	3.35 100	3.37 100						3.66 25	
2.07		60	2.04 50	2.04 3	2.03 100		2.08 100	2.08 100	2.06 70	2.07 2	2.09 70	
1.73		50				1.72 60		1.73 50	1.76 15		1.71 60	
1.27		20	1.23 90	1.23 6			1.24 60			1.26 8	1.26 10	1.24 15
1.17		20	1.15 90	1.16 6	1.17 30	1.17 30	1.16 60			1.19 10		
Shaft Region	3.49	100	3.35 100	3.37 100						3.66 25		
	2.05	80	2.04 80	2.04 3	2.03 100		2.08 100			2.07 2	2.09 70	
	1.27	10	1.23 90	1.23 6			1.24 60			1.26 8	1.26 10	1.24 15
	1.18	20	1.15 90	1.16 6	1.17 30	1.17 30	1.16 60			1.19 10		

TABLE 4 Electron Diffraction Pattern Head and Shaft of Carbon Fiber

Ruston. The high intensity hematite (Fe_2O_3) and magnetite (Fe_3O_4) diffraction patterns are presumably due to the oxidation-reduction sequence used in run A-18.

A variety of tubular shaped fibers were formed. Figures 44 and 45 show two additional types. Figure 44 is a micrograph of a fiber taken into the direction of growth. Again, the hollow shaft is seen along with the electron dense tip. Interestingly, in all micrographs taken in the growth direction, a slit or hole is noticeable in the tip.

Figure 45 is a circular-shaped fiber. The hollow shaft is again seen, but this fiber has no metallic tip. The graphitic nature of the skin is shown in Figure 45 along with indication of disintegration fragments of iron and/or iron compounds.

Electron diffraction patterns on all fibers formed gave similar results to those presented in Table 4. No ribbon-shaped fibers were formed nor were any coiled fibers in evidence.

4.2.5 $\text{Fe}_{1-y}\text{O}/\alpha\text{-Fe}$ Phase Boundary

Having determined that carbon deposition can be controlled by adjusting the $P_{\text{H}_2}/P_{\text{H}_2\text{O}}$ ratio, the next step in the investigation was to determine precisely the location of the wustite/ α -iron phase boundary.

The iron-iron oxide, graphite-gas and iron-iron carbide phase boundaries can be conveniently plotted on a triangular phase diagram. Figure 46 shows such a representation with

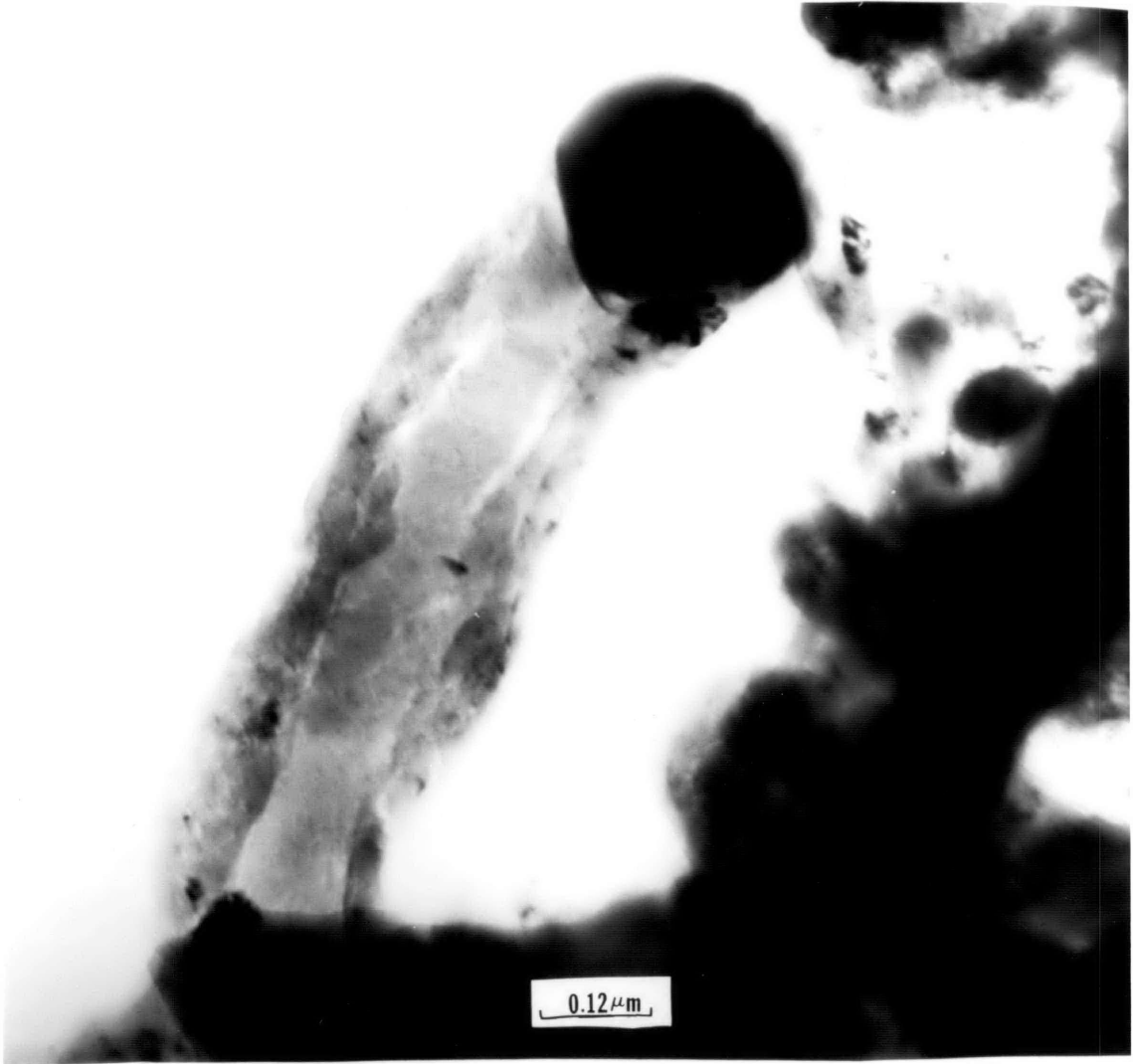


FIGURE 44 CARBON FIBER-MICROGRAPH TAKEN IN GROWTH DIRECTION



FIGURE 45 CIRCULAR-SHAPED CARBON FIBER

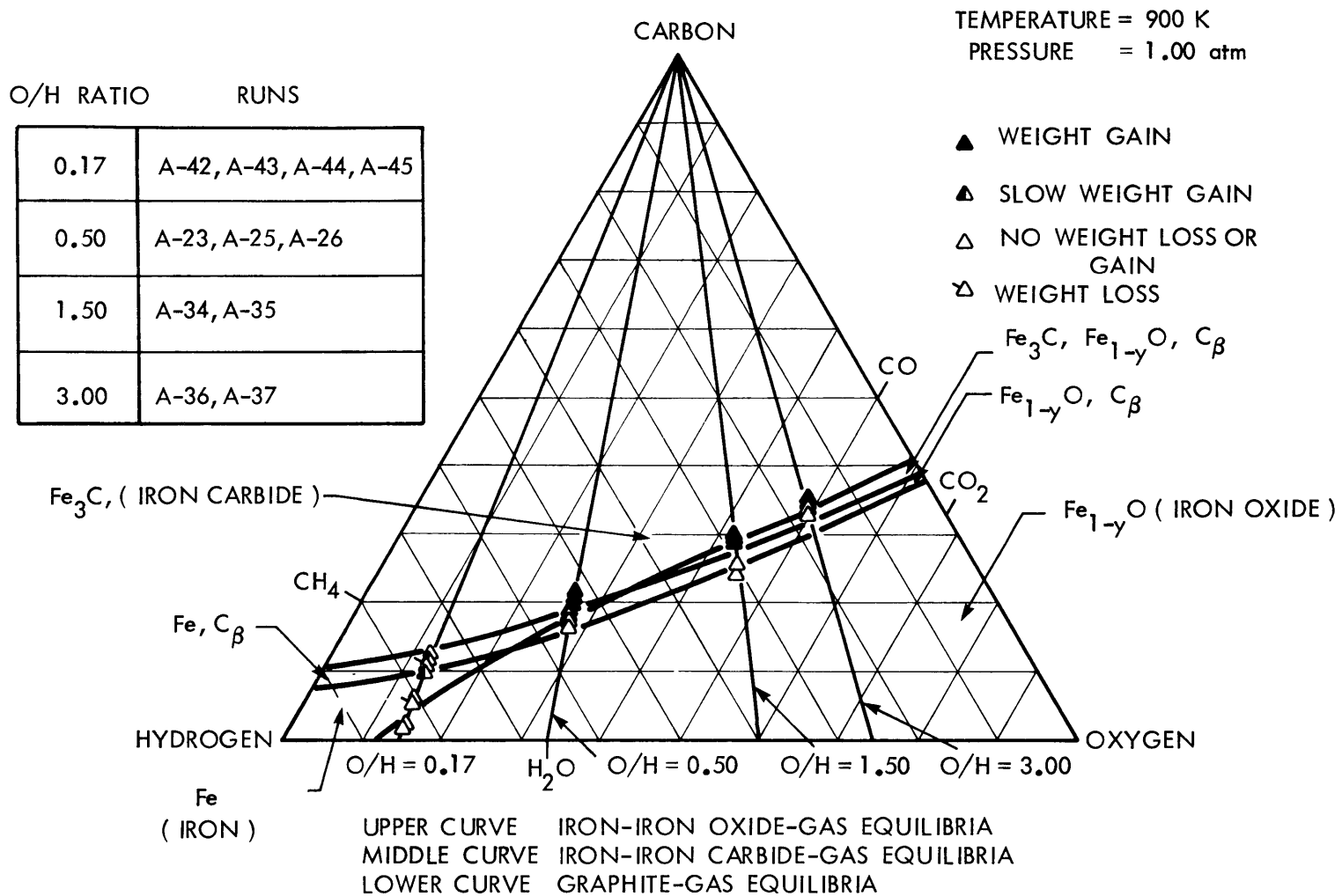


FIGURE 46 DETERMINATION OF THE IRON/WUSTITE PHASE BOUNDARY IN A FIVE COMPONENT GAS MIXTURE AT VARIOUS O/H RATIOS

several runs plotted at various O/H ratios. With the one exception of O/H = 0.17, all the data shown represents well the conclusion established earlier; that is, that iron oxide inhibits carbon deposition. Also, the phase boundary itself appears to be located at the predicted region from theoretical calculations. Due to the proximity of the various phase boundaries at 900 K, these data with some additional data are best represented as shown in Figure 47.

Figure 47 is a plot of the percentage of the theoretical P_{H_2}/P_{H_2O} for the surface controlling reaction N-A, versus the experimental P_{H_2}/P_{H_2O} ratio. Each data point is seen to have associated with it error bars as determined by a propagation of error analysis (Appendix 7.4.2).

Figure 47 indicates some scatter was associated with different run conditions. That is, occasionally, carbon deposition did not stop or begin when expected. Most of this scatter is within 10% of the equilibrium P_{H_2}/P_{H_2O} ratio. Again a propagation of error analysis indicates the maximum error in P_{H_2}/P_{H_2O} ratio could be as high as 11%. Undoubtedly, part of this scatter comes from normal inherent error associated with the experimental procedure, data collection, and data analysis methods.

Since a propagation of error analysis gives the maximum error possible, it is felt that some of the scatter shown possibly reflects another process which affects the rate of either carbon deposition or oxide formation. An intriguing possibility is the formation and reduction of multiple oxide

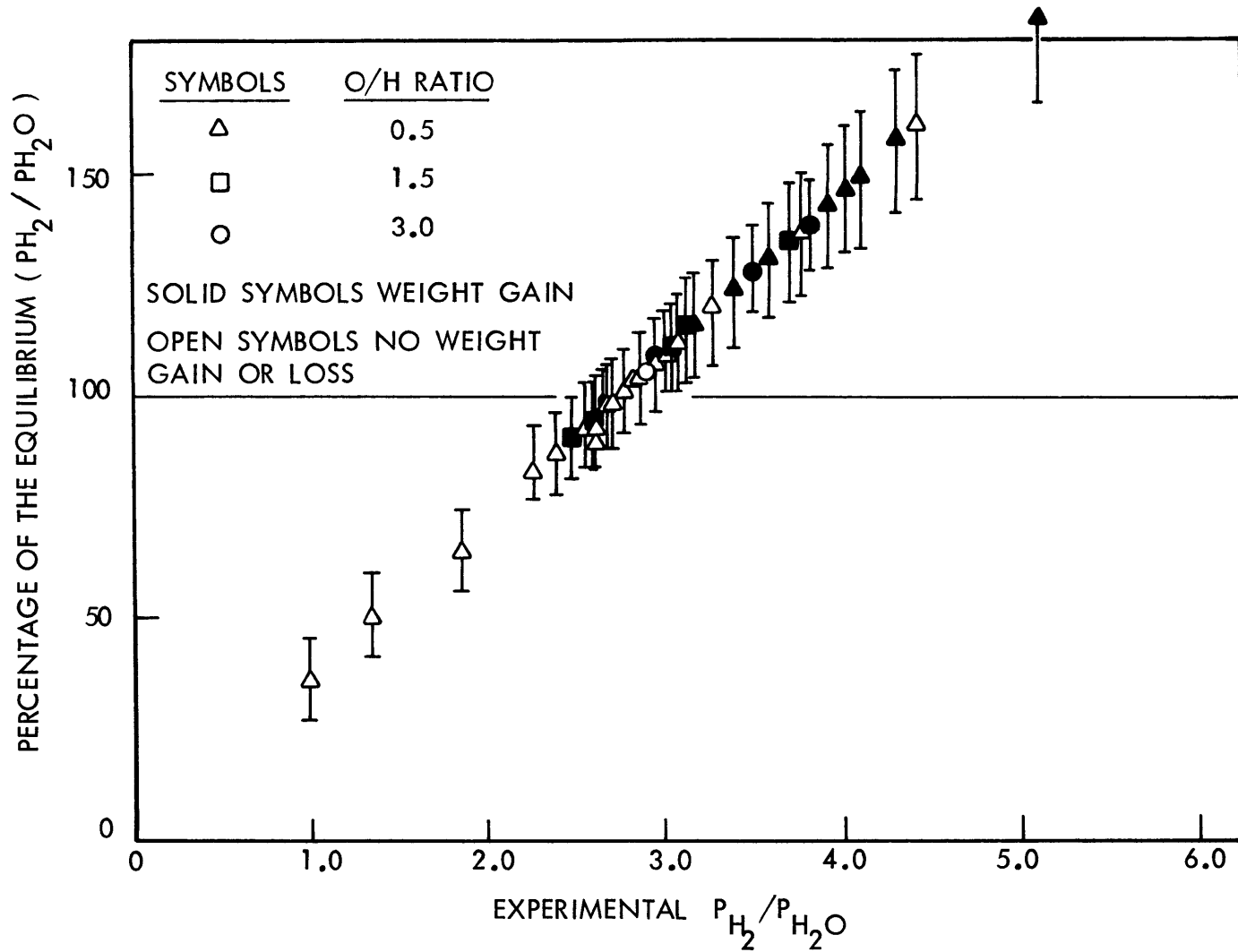
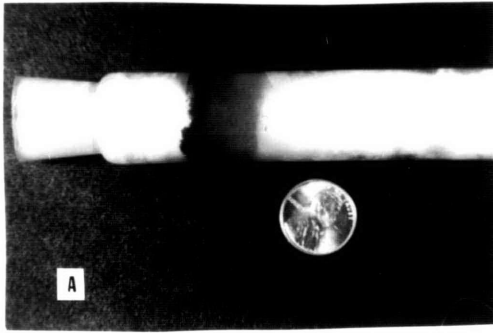


FIGURE 47 PERCENTAGE OF THEORETICAL P_{H_2} / P_{H_2O} VERSUS EXPERIMENTAL $P_{H_2} / P_{H_2O} |_{eq}$ FOR VARIOUS O/H VALUES

phases. It was shown in Section 4.2.1 that multiple oxide formation does in fact occur on the steel wool catalyst. However, multiple oxide formation can only occur if the oxygen activity in the system at some time exceeds that necessary to support the particular multiple oxide phase of interest. For example, if a layer of magnetite (Fe_3O_4) is to form at 900 K, the oxygen partial pressure (activity) has to exceed that for the equilibrium between wustite and magnetite. While this was clearly the case during normal catalyst pretreatment, it was not generally the case during most of the experimental runs. This fact would normally tend to rule out this effect. However, the results from run A-43 as well as several others suggest that multiple oxides may indeed form.

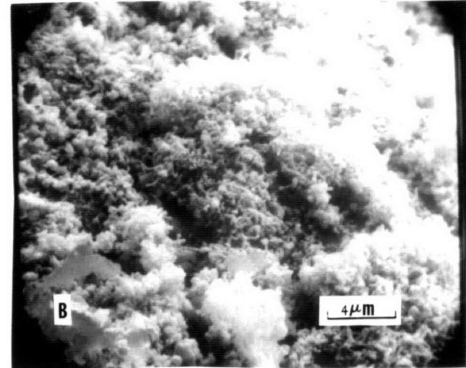
Run A-43 was carried out to establish the position of the iron-iron oxide phase boundary at an O/H value equal to 0.17. The normal catalyst pretreatment was carried out, i.e., 402 mg of carbon were deposited and an initial set of run conditions fed. This initial set of run conditions created an oxidizing atmosphere with a $P_{\text{H}_2}/P_{\text{H}_2\text{O}}$ equal to 2.38. On conclusion of the first set of run conditions a leak in the inlet line developed and the reactor was immediately shut down.

Figure 48 shows the catalyst carrier after removal from the reactor. Carbon was shown to be preferentially deposited at the bed inlet. It is important to point out that this carbon deposition occurred prior to the time when the inlet leak occurred. Above the carbon bed was a region where very little carbon was observed to deposit, followed by a reddish-



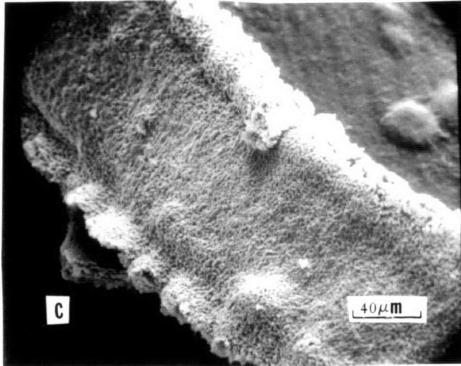
CATALYST CARRIER RUN A-43

note. three distinct regions



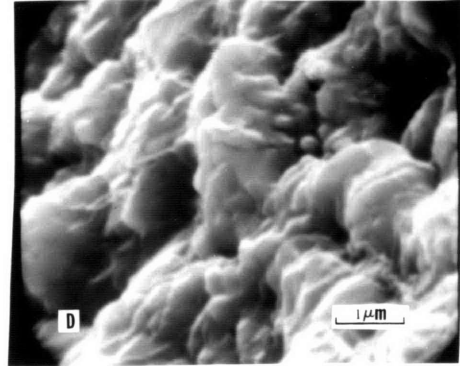
CARBON FIBER FORMATION

{lower region}



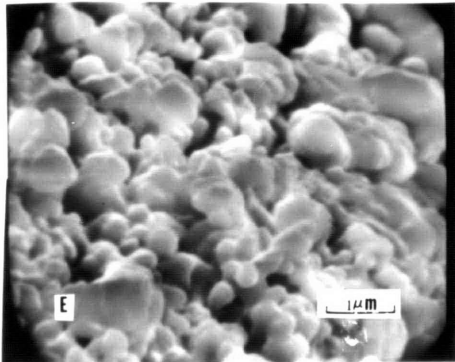
OVERVIEW CATALYST STRAND

{central region}



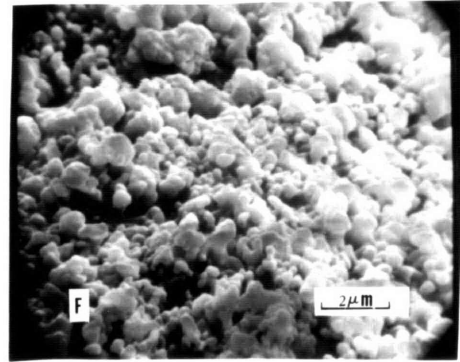
SURFACE STRUCTURE

{central region}



SURFACE MORPHOLOGY

{upper central region}



HEMATITE, Fe_2O_3 , STRUCTURE

{top region}

FIGURE 48 CATALYST SHOWING STRUCTURAL CHANGE, RUN A-43

orange region where no carbon deposition occurred. This reddish-orange region is Fe_2O_3 (hematite). At this $P_{\text{H}_2}/P_{\text{H}_2\text{O}}$ ratio no Fe_2O_3 should have been able to form. Since similar results were noticed in other runs and, also, since this type of behavior has been reported by Everett et al. (1967) and Wilson (1971) under similar conditions it is felt that this oxide formation is not a function purely of the inlet leak reported. This type of behavior reflects product poisoning; in this case water and carbon dioxide. If product poisoning is occurring, it means that multiple layer formation could take place. If so, this could easily explain the scatter noted in Figure 47 (see Section 4.2.5). Figure 48 is important from another point also: it visually shows that carbon deposition does not occur on oxidized iron.

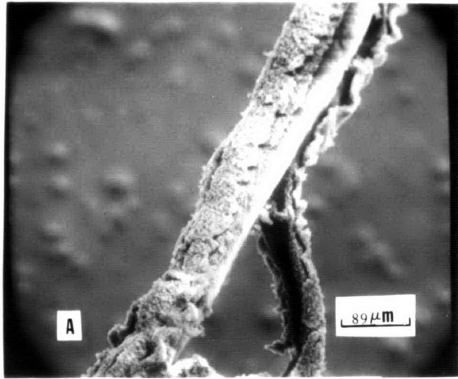
The three regions observed in Figure 48 provide a unique opportunity for metallurgical examination of what could possibly be the structural change sequence which occurs during reaction.

Micrograph (b) shows the familiar carbon fiber formation which was similar to those seen from micrographs of the shell region from run A-18. A strand taken from the central region is shown in micrograph (c). At first glance the structure looks very similar to the sponge-like structure which develops during preconditioning. However, close examination of the surface shows rough irregular features with significantly less porosity than that observed on a preconditioned catalyst

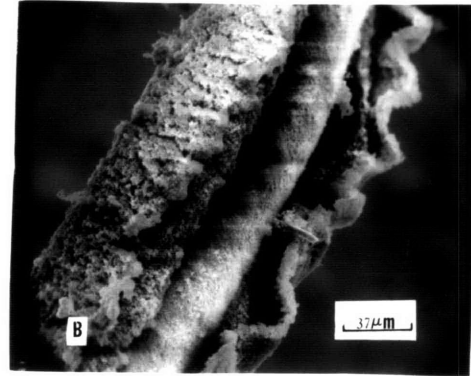
(micrograph {d}). Micrograph (e) is a high magnification photograph of the surface features just prior to entering the reddish-orange hematite (Fe_2O_3) region. Notice there is more porosity here suggesting less structural sintering. The final micrograph shown in Figure 48 is the Fe_2O_3 surface oxide; note the particle-like nature of the hematite structure and the associated high porosity. It should be remembered in viewing the micrographs in Figure 48 that the original structure was that of sponge-iron developed during standard catalyst preconditioning.

Figure 49 is a final series of micrographs showing the multi-layer structure formed in run A-43. Micrograph (a) is a typical catalyst fiber showing the core and double layer regions. Micrographs (b) and (c) are high magnification micrographs of the double layer structure observed in micrograph (a). These micrographs clearly show two distinct layers, neither of which appear to have any carbon deposited on them. A close examination of the core region micrograph (c) shows what appears to be some kind of projections. A close up of this core region shown in micrograph (d) show iron whiskers. These were identified as pure iron by Energy Dispersion Analysis. It is interesting to compare the core region shown in micrograph (d) with the core region from micrograph (f) in Figure 42. In both photographs very little carbon if any is seen. However, no iron whiskers are noticed in micrograph (f) of Figure 42.

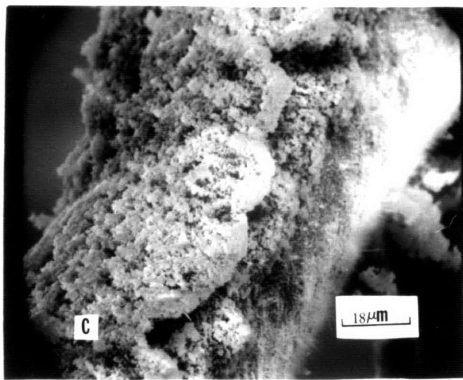
Whiskers are curious structures which have not been fully



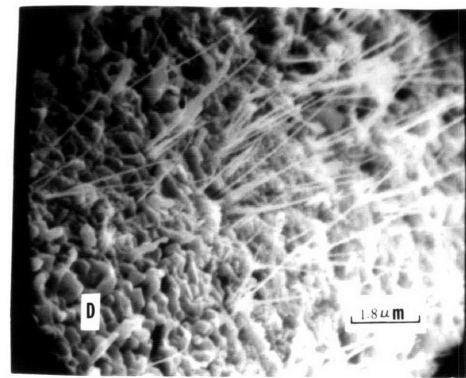
RUN A-43 CATALYST FIBER
note. shell double layer



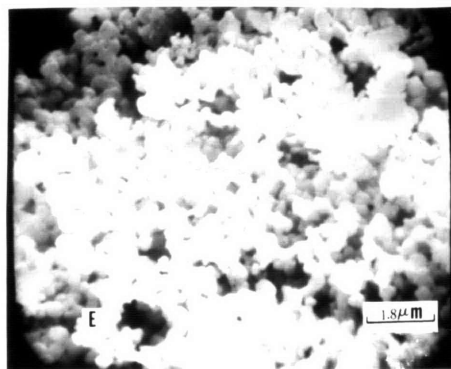
DOUBLE LAYER MORPHOLOGY



SHELL AND CORE STRUCTURE
note. iron-whiskers on core



IRON-WHISKER FORMATION



Fe₂O₃ OUTER SHELL

FIGURE 49 CATALYST EXHIBITING MULTIPLE OXIDE FORMATION, RUN A-43

explained. Several theories have been put forth, but none have been experimentally verified. However, whisker formation has been found to frequently occur in structures reduced in hydrogen with hematite (Fe_2O_3) present (Cahn, 1973).

It was mentioned earlier that the data in general agreed with the conclusion that iron oxide is not a catalyst for carbon deposition. However, one of the major problem areas was at low O/H values (i.e., < 0.2). Here the data were inconsistent. The reason for this was that the water concentration was so high that it exceeded system design. Condensation was observed to occur in both the inlet and exit lines. For completeness the runs at O/H values of 0.17 were shown in Figure 46, but the results presented must be viewed with caution.

In conclusion, the $\alpha\text{-Fe/Fe}_{1-y}\text{O}$ phase boundary is seen to be in the position predicted by theory. The scatter noticed can be justified by consideration of normal error involved in experimentation and data analysis. Also, evidence has been shown which suggest multiple oxide formation can be important in explaining the observed system behavior.

4.2.6 Iron-Iron Carbide Equilibrium

The question of the role of carbides during carbon deposition is a difficult one. The difficulty lies in the fact that carbides are difficult to identify and can form not only during reaction but as the reactor goes through wide temperature fluctuations, such as in a hot spot or during the cooling

down sequence.

Several experiments were run in an attempt to determine the effect of carbides formed during reaction on carbon deposition.

Run A-57 was the initial experiment performed. In run A-57, a 514 mg steel wool catalyst with 577 mg of carbon was exposed to a variety of different methane-hydrogen mixtures at 823 K and 1.01×10^5 N/m². At all run conditions, the methane content was maintained at a value above that predicted by equilibrium for reaction G, 38%. Thus, reaction G would be expected to proceed to the left.



The methane content was adjusted to favor either Fe₃C or α-Fe formation according to reaction H-A.

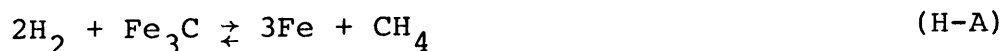


Figure 50 represents the results obtained. Condition 1 was run with a methane content of 58%; this corresponded to a composition slightly above the 52% required for equilibrium of reaction H-A.

From equilibrium considerations alone, carbide formation as well as carbon deposition would be expected to occur. As shown a slow weight gain was recorded, however, due to low conversion, it was impossible to determine whether or not carbide formation accounted for the weight gain. The methane content was then reduced to 49% so as to bring the gas composition below the iron-iron carbide phase boundary.

RUN NO. A-57

Condition %Hydrogen %Methane Comment

Condition	%Hydrogen	%Methane	Comment
1	41.23	58.51	Slow Weight Gain
2	50.50	49.29	Very, Very Slow Weight Gain
3	60.57	39.25	No Weight Gain or Loss
4	25.45	74.21	No Weight Gain
5	98.86	00.13	Rapid Weight Loss

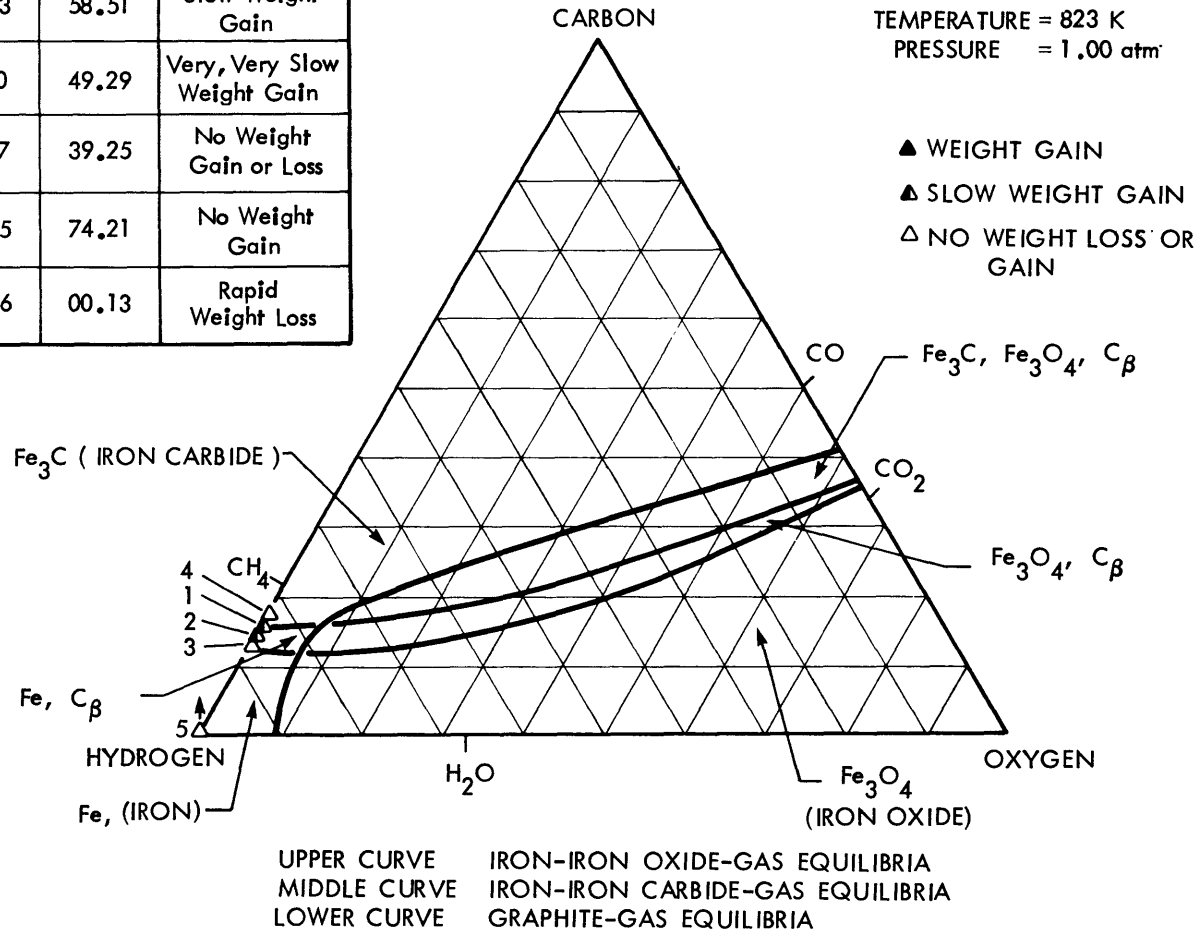


FIGURE 50 EFFECT OF Fe₃C ON CARBON DEPOSITION FROM METHANE-HYDROGEN MIXTURES

According to reaction G carbon should have formed but a methane content this low would require H-A to proceed to the right, thus reducing any carbide present. A slow weight gain was observed but effluent gas composition did not appear to change substantially. For the next two runs at methane contents of 39 and 74%, respectively, there was neither weight loss nor gain. The run at 74% methane should have formed carbide and carbon, however, no weight change was observed nor did the effluent composition change. When the methane content was dropped to less than 1%, rapid weight loss was observed, with methane appearing in the product gas stream.

Manning (1976) suggests that methane formation may be formed through a carbide intermediate, but here is proof that not enough, if any, carbide was formed to account for 115 mg of weight loss which occurred at run condition 5. Condition 1 indicated (assuming the weight gain noted was entirely from Fe_3C formation) that carbide formation is much too slow, from methane, to account for the rate of carbide formation necessary to give the weight loss observed. These observations agree with those observed by Podgurski (1950) who stated that carbide formation from methane was extremely slow. Carbon formation from reaction G at this temperature can also be inferred to be slow by Browning et al.'s (1950) data on α -iron/ Fe_3C equilibrium and thus little carbon weight gain would be expected.

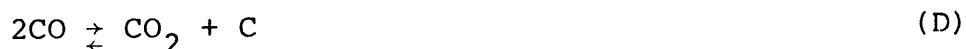
In order to confirm these results and to try and establish if Fe_3C is a catalyst for carbon deposition, a series of

runs were performed at 900 K with a five component gas mixture. At this temperature carbon formation from reaction G should become more significant and the rate of carbide formation should also increase.

Runs A-58 and A-59 were performed at fixed C/H values of 0.2 and 0.35, respectively. The catalyst was preconditioned using the standard procedure and a carbon bed laid down. Figure 51 illustrates the observed behavior.

During both experimental runs, weight gain was observed in the region where cementite (Fe_3C) would be expected to be the stable solid phase. However, the effluent in both runs A-58 and A-59 indicated little if any change in methane concentration, and the observed weight gain resulted primarily from carbon monoxide conversion.

There are two reactions which could account for the weight gain observed, reaction I and/or reaction D.



From stoichiometric considerations it is impossible to determine if reaction I or reaction D accounts for the weight gain reported. However, the results of the work by Podgurski et al. (1950) and Walker et al. (1959) indicated that carbon deposition resulted when carbiding with carbon monoxide, even at temperatures as low as 598 K. Taking this into account, along with the fact that the total weight gain was 3.5 times that required for complete carbiding (35 mg), suggests that

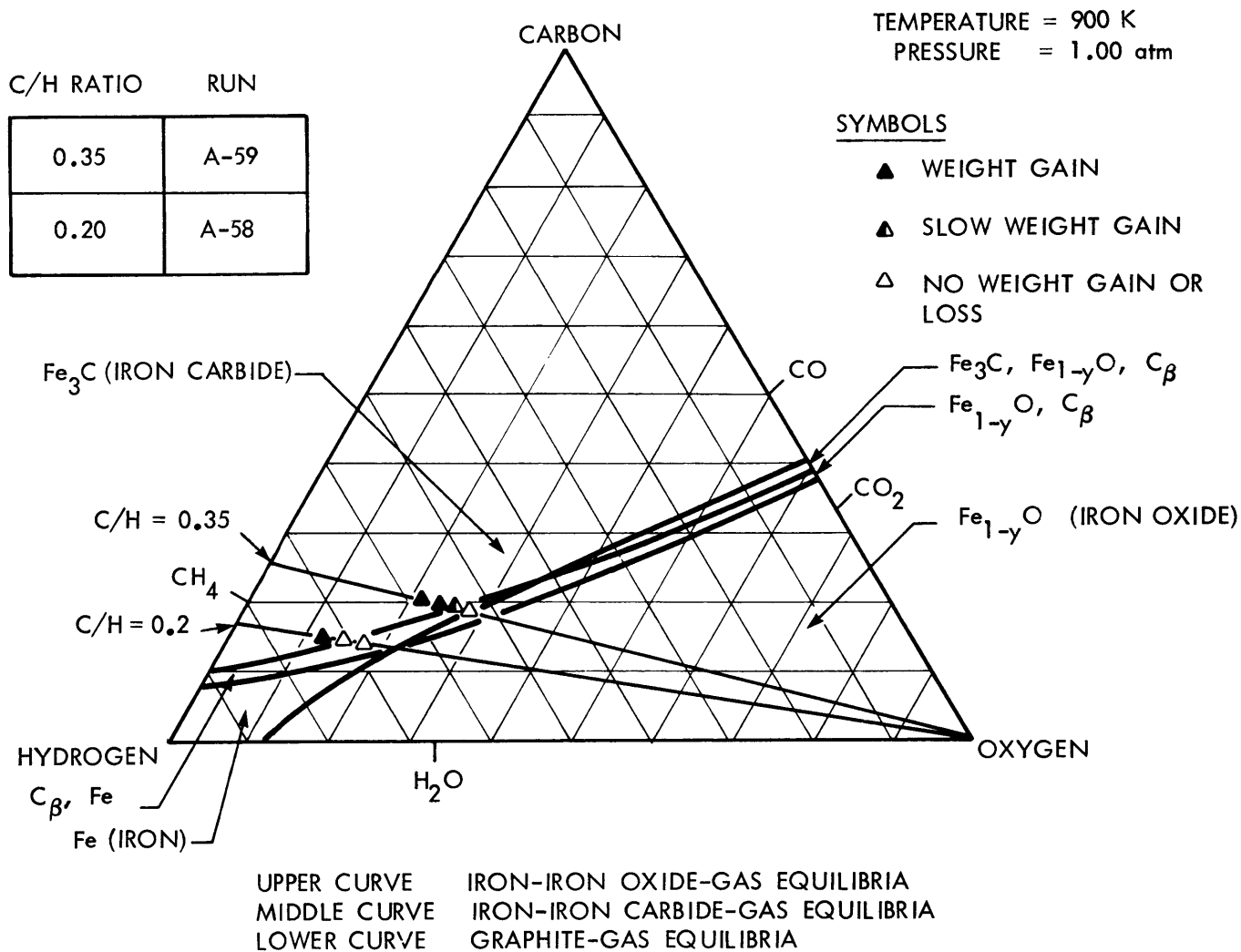


FIGURE 51 DETERMINATION OF IRON / IRON CARBIDE PHASE BOUNDARY IN A FIVE COMPONENT GAS MIXTURE AT VARIOUS C/H RATIOS

the weight gain observed was due to carbon deposition not carbide formation.

These data indicate that at 900 K, carbide formation from methane does not occur readily. The data were inconclusive in determining if weight gain was due to carbon formation entirely or a carbide-carbon combination. For this reason, these data do not conclusively show whether carbide formation inhibits or catalyzes carbon deposition. However, from an operational standpoint, it appears that oxide formation is the more important problem in efficient Bosch reactor operation.

If one looks closely at the data plotted in Figure 51, an interesting question arises: why did not carbon deposition occur in the region between the iron-iron carbide and graphite-gas phase boundaries, i.e., in the region where the stable solid phase should be α -iron and carbon?

The reason appears to be related to kinetics. For example, assuming the rate of reaction E can be represented by a power law relationship:

$$r_{(E)} = \frac{k_1}{R_g T} \{P_{H_2}\} \{P_{CO}\} - \frac{k_2}{R_g T} \{P_{H_2O}\}$$

assuming a one step mechanism

$$K_e = \frac{k_1}{k_2}$$

$$\therefore r_{(E)} = \frac{k_1}{R_g T} \left[(P_{H_2}) (P_{CO}) - \frac{(P_{H_2O})}{K_e} \right]$$

or

$$r_{(E)} = \frac{k_1}{R_g T} \left(1 - \frac{\frac{P_{H_2O}}{P_{H_2} P_{CO}}}{\frac{P_{H_2O}}{P_{H_2} P_{CO}} \Big|_{eq}} \right) \quad (10)$$

Equation 10 expresses the fact that rate can be related to a driving force or distance from equilibrium (i.e.,

$$\left\{ 1 - \frac{\frac{P_{H_2O}}{P_{H_2} P_{CO}}}{\frac{P_{H_2O}}{P_{H_2} P_{CO}} \Big|_{eq}} \right\} .$$

A visual representation of this "driving force" is given in Figure 52. In Figure 52 a plot of the percentage of the theoretical equilibrium constant versus the actual experimental ratio for reaction E and reaction D is presented.



Again, solid symbols represent runs where weight gain was observed; open symbols were runs where no weight change was observed. The data for runs A-58 (triangles) and A-59 (diamonds) clearly show a critical driving force for reaction (distance from equilibrium) must exist before weight gain will be observed. This appears to be true even though α -iron is present to catalyze the reaction.

In order to verify if this critical driving force could account for some of the deviation observed when determining the iron-iron oxide phase boundary, several runs were plotted

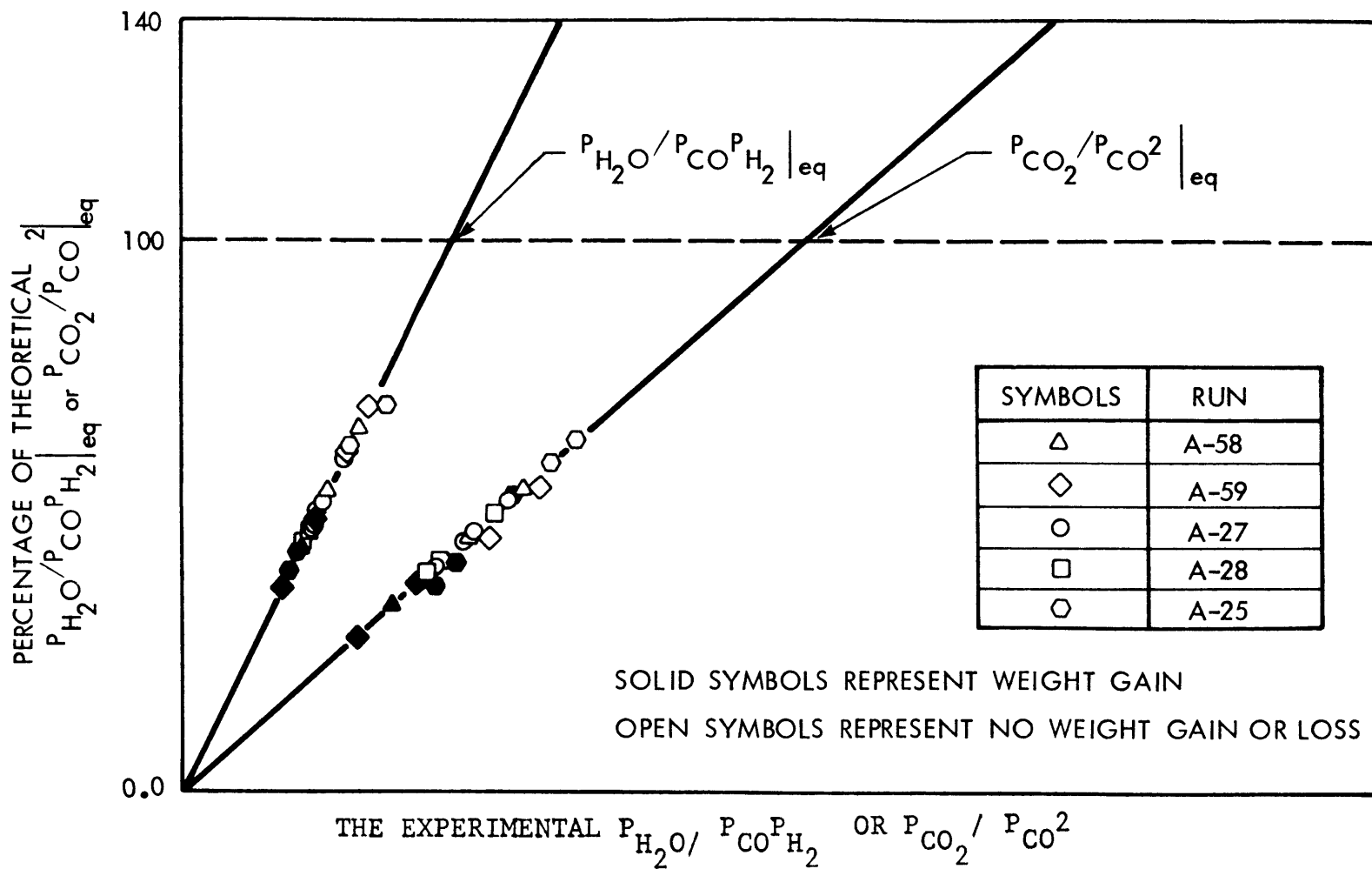


FIGURE 52 PERCENTAGE OF THEORETICAL $P_{H_2O}/P_{CO}P_{H_2}$ |_{eq} AND P_{CO_2}/P_{CO^2} |_{eq} VERSUS
THE EXPERIMENTAL $P_{H_2O}/P_{CO}P_{H_2}$ OR P_{CO_2}/P_{CO^2}

from that study. Runs A-27 (circles) and A-28 (squares) were two runs at an O/H ratio equal to 0.5. No appreciable weight gains were noticed in either runs A-27 or A-28, even though the P_{H_2}/P_{H_2O} ratio was in the region where α -iron would be the expected iron phase. Figure 52 indicates that in both these runs, within experimental error, the critical driving force was not achieved.

Run A-25 is also plotted in Figure 52 and is represented by hexagon-shaped symbols. This run was found to predict accurately the iron-iron oxide phase boundary at an O/H ratio equal to 0.5. Within experimental error, one again sees that carbon deposition will not occur until the driving force for carbon deposition is in the critical region.

It should be pointed out that this critical driving force or distance from equilibrium will not cause carbon deposition to occur if the surface is oxidized (run A-19).

This analysis suggests that two factors are important in determining if carbon deposition will occur: first, the catalyst surface must be in the reduced state; second, the $P_{H_2O}/P_{H_2} P_{CO}$ and/or P_{CO_2}/P_{CO}^2 ratio must be sufficiently far away from their equilibrium values to insure a large driving force for reaction.

4.2.7 Effect of Iron-Iron Oxide ($Fe_{3-4}O_x$) at 800 K

Having determined $Fe_{(1-y)}O$ (wustite) is not a catalyst for carbon deposition, the next step was to determine the effect of Fe_3O_4 on carbon deposition, Fe_3O_4 (magnetite) being

the stable oxide in equilibrium with alpha-iron at 800 K.

4.2.8 Catalyst Preconditioning at 800 K

The standard method for preconditioning was used at 800 K. Again, this consisted of running a predetermined carbon dioxide-water mixture over the catalyst at reaction temperature, followed by reduction with hydrogen. Figure 53 is a comparison between the rate of oxidation at 800 and 900 K. As shown the rate of oxidation at 900 K was substantially greater than at 800 K. The behavior shown in Figure 53 at 800 K is characteristic of a process controlled by solid state diffusion. As indicated in Section 2.2.2, if solid state diffusion is important, electron micrographs of the reduced surface should indicate a dense thin iron layer.

In Figure 54 are several electron micrographs of a catalyst strand after standard preconditioning. Micrograph (a) shows a boundary which delineates between the smooth core and the rough, irregular and dense outer surface. The morphology shown in micrograph (a) indicates, in agreement with Figure 53, that solid state diffusion is important in oxidation and reduction at 800 K. Micrograph (b) is a high-magnification photograph of the boundary shown in micrograph (a). Again, notice the rather irregular dense outer shell compared to the featureless inner core. A final micrograph of the outer shell is shown in micrograph (c). In micrograph (c) a crack showing the thin dimensions of the reduced iron skin is shown.

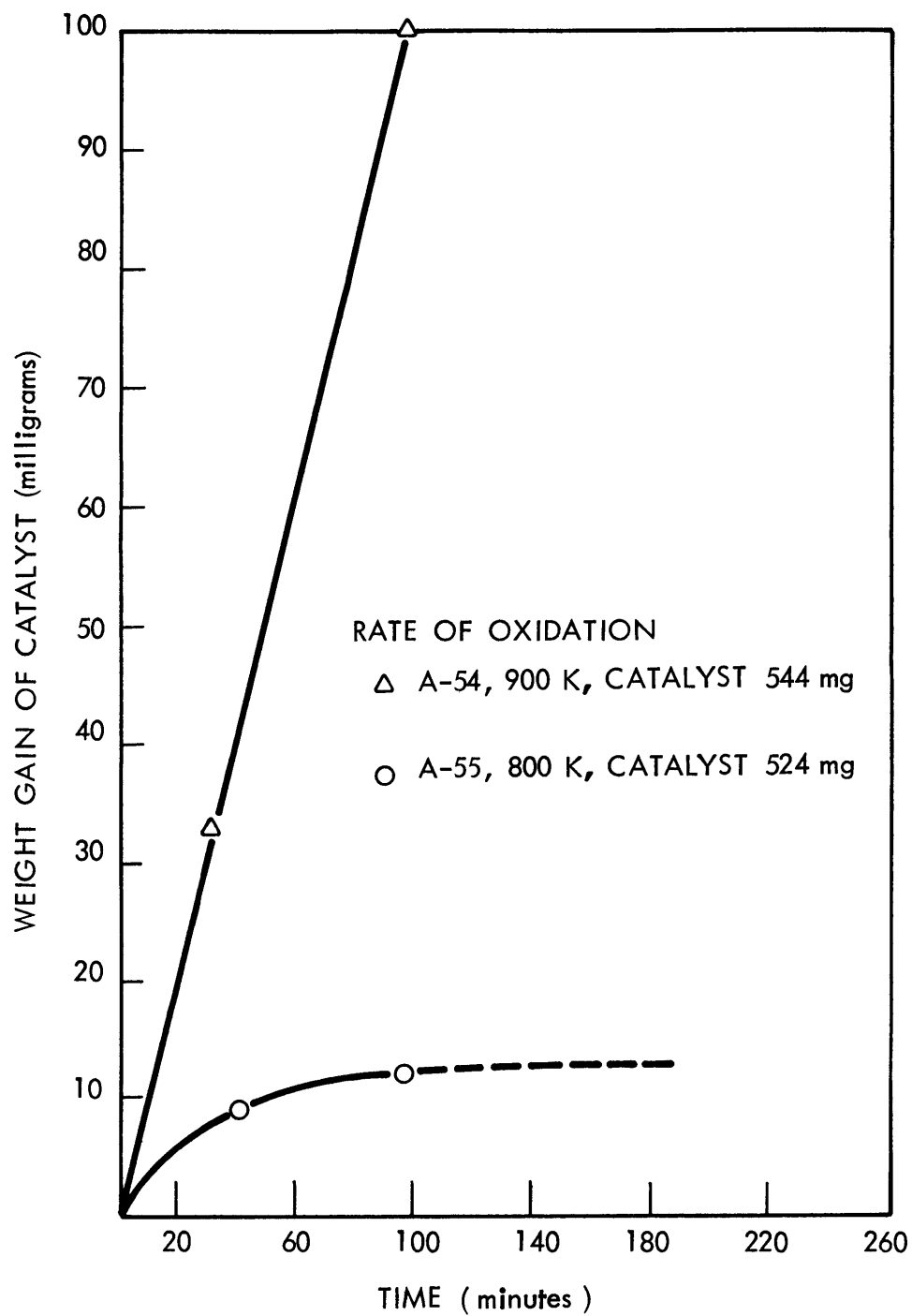
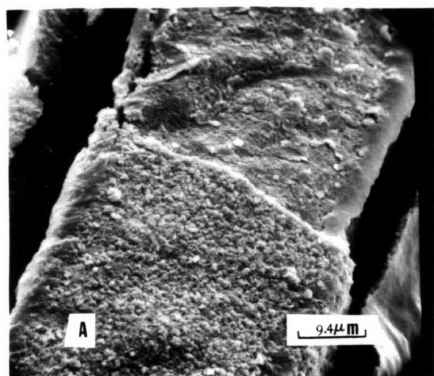
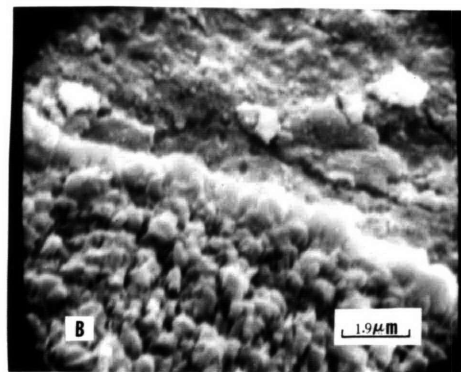


FIGURE 53

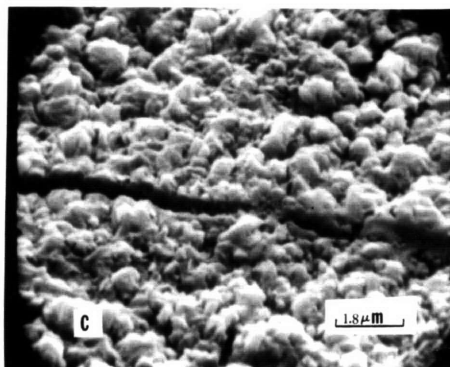
WEIGHT GAIN DURING OXIDATION VERSUS TIME AT 800 K AND 900 K



PRECONDITIONED CATALYST
AT 800 K



SHELL AND CORE STRUCTURE



OUTER SHELL

FIGURE 54 PRECONDITIONED CATALYST AT 800 K

The micrographs in Figure 54 and the data shown in Figure 53 indicate that different processes are controlling oxidation and reduction at 800 and 900 K. At 900 K the reduction process is seen to be controlled by the mechanisms involved with gas-solid reactions. At 800 K, however, the overall reduction process appears to be controlled by solid state diffusion.

4.2.9 Carbon Inhibition at 800 K - $\text{Fe}_3\text{O}_4/\alpha\text{-Fe}$ Phase Boundary

The inhibiting effects of Fe_3O_4 on carbon deposition from a five component gas mixture were next investigated. Again, using the procedure outlined in Section 3.2.1, gas compositions were set by adjusting the $P_{\text{H}_2}/P_{\text{H}_2\text{O}}$ ratio at a fixed O/H ratio, temperature and pressure. Figure 55 illustrates the data obtained at three different O/H ratios over a pre-carboned catalyst at 800 K. Two runs were made at each O/H ratio; one approaching the equilibrium phase boundary from the oxide side, the other from the reduced side. As shown in Figure 55, the data are in excellent agreement with the theoretically predicted $\alpha\text{-Fe}/\text{Fe}_3\text{O}_4$ phase boundary. The effluent concentrations of all runs indicated that methane remained constant; the weight gain observed resulted from carbon monoxide conversion. However, again no definitive statement as to the carbon deposition mechanism could be made.

Considering the scatter observed in the data at 900 K, the excellent agreement between the predicted and experimentally determined phase boundary at 800 K may at first seem

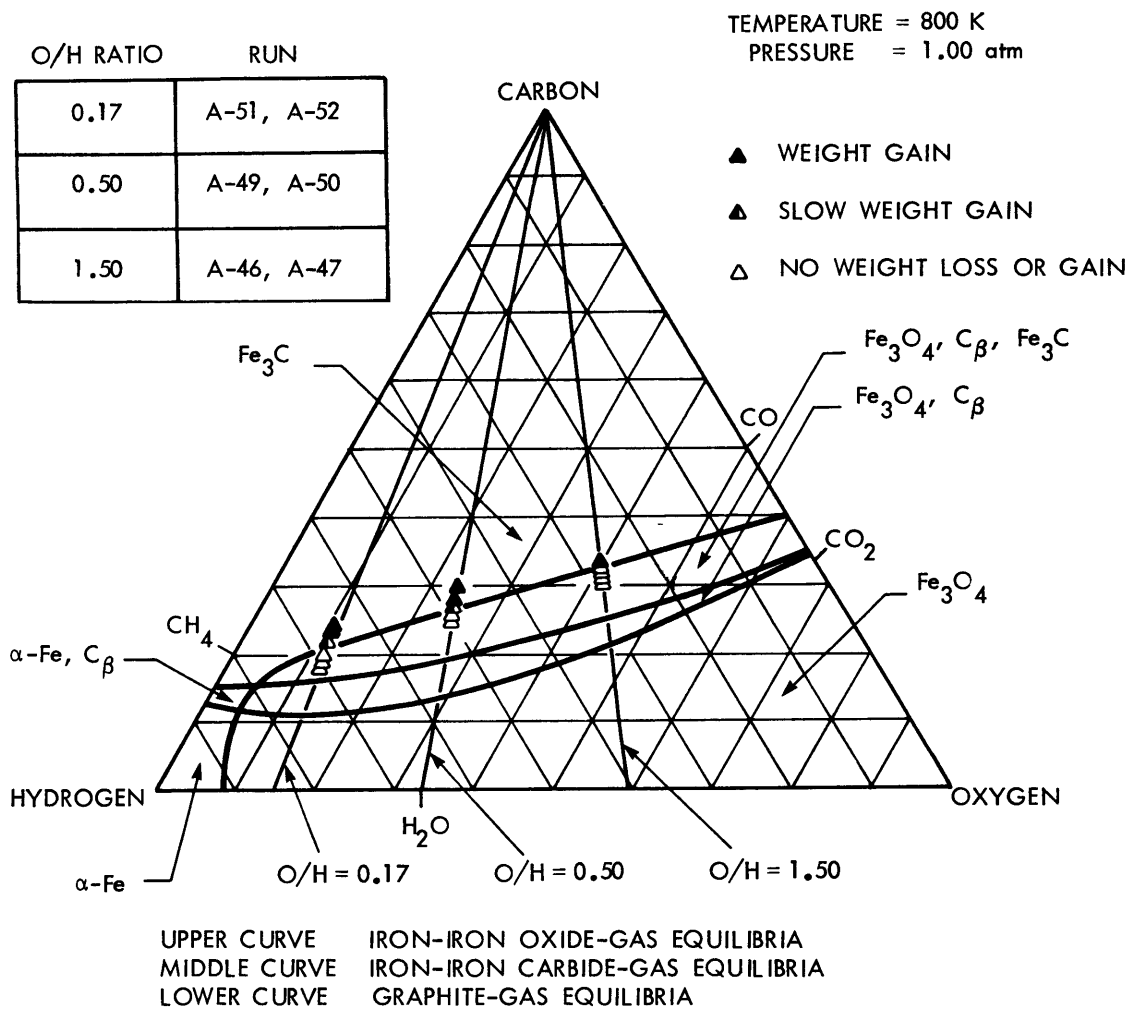


FIGURE 55 INHIBITION OF CARBON DEPOSITION BY Fe_3O_4 AT VARIOUS O/H RATIOS

unusual. However, if one looks at the iron-iron oxide phase diagram in Figure 16, Section 2.2.1, the answer becomes apparent. At temperatures below 833-843 K, the amount of water the system can hold (i.e., oxygen activity) before a second oxide phase forms far exceeds that at temperatures above 833-843 K. Thus, the very serious problem of multiple oxide formation with its concomitant kinetics problems is nonexistent. In fact, in general, at temperatures below 833 K, serious structural change during reaction would not be expected for a steel wool catalyst. That is, structural changes other than those from fiber formation (i.e., sintering, creep).

5. Application of Results

Section 4 presented data which indicated that iron oxides inhibit carbon deposition, one of the important reactions in the Bosch reaction sequence. Also, carbide formation was shown to be slow relative to the rates of oxidation and carbon deposition. Carbide formation, being slow, should not present a problem in an optimal Bosch design. The oxide limitation, however, must be carefully avoided and/or utilized to maximize Bosch efficiency.

The maximum water concentration which can be achieved in the Bosch process, at a specified temperature, is given at the intersection of the graphite-gas/iron-iron oxide phase boundaries.

The reason for this is a complex one. However, one can verify this statement by looking at Figure 56 and visualizing the following.

Envision a series of lines radiating out from the position representing water on the triangular phase diagram. From the properties of equilibrium phase diagrams, any gas mixture having a composition lying closer to the point representing water (for example, point 5) will have a higher water content than another point (point 3) which is further away. Strictly speaking, these points must be colinear with the position representing water.

To the right of the intersection (point 5), the highest water concentration will be given by a gas mixture in

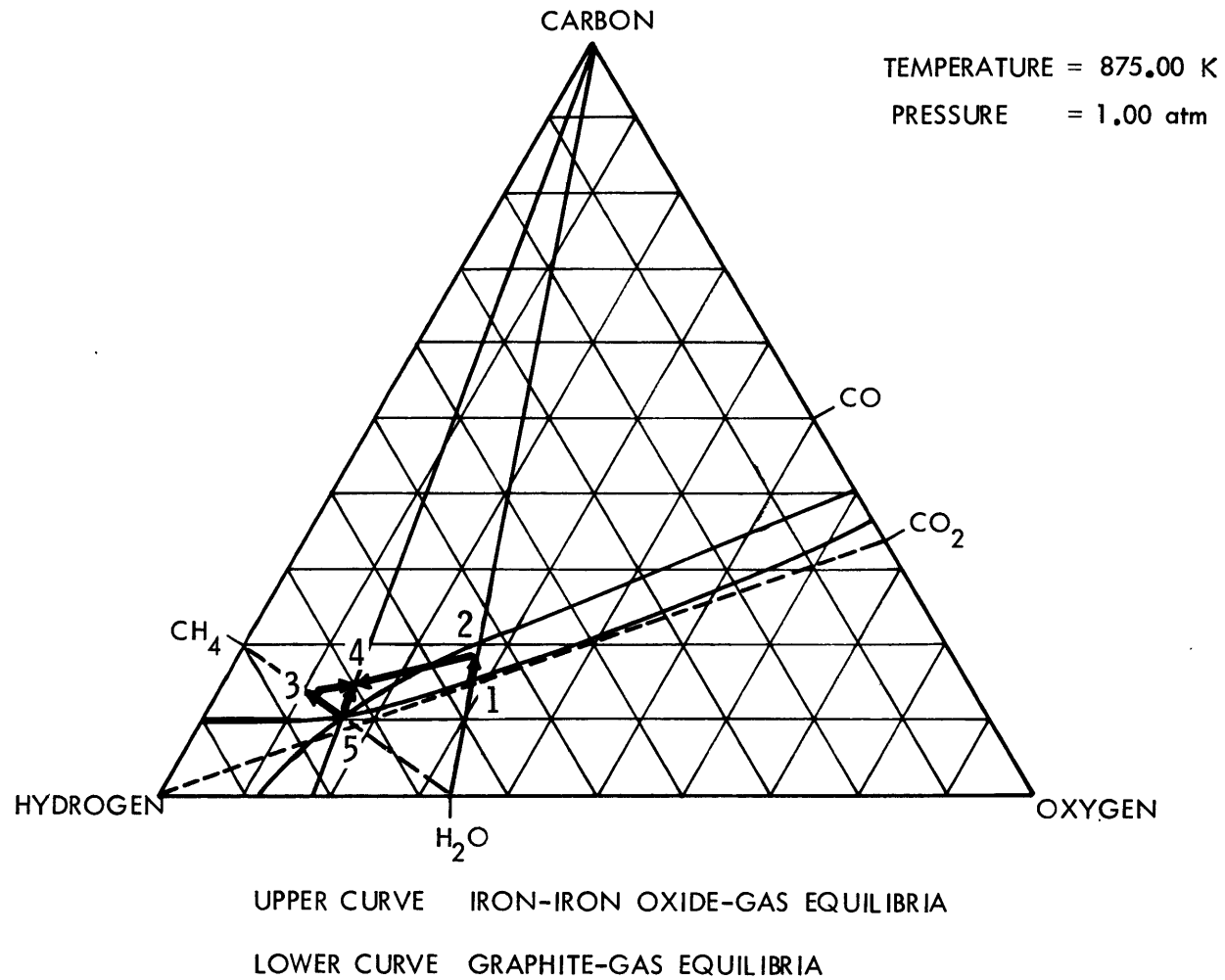


FIGURE 56 OPTIMAL REACTION FLOWPATH FOR THE BOSCH PROCESS

equilibrium with graphite. That is, the graphite-gas phase boundary is closer to the point representing water than is the iron-iron oxide boundary. Unfortunately, the graphite-gas phase boundary lies below the iron-iron oxide phase boundary in the region where iron oxide is the stable iron phase. Therefore, if a gas mixture were to lie in this region, carbon deposition would stop and the gas composition would approach that at equilibrium with iron oxide. Remember, the gas composition in equilibrium with the iron-iron oxide system has a lower water concentration.

The author believes that this is the factor which limited the production of water in previous Bosch prototype reactors.

To the left of point 5, the iron-iron oxide phase boundary is closer to the position representing water than the graphite-gas phase boundary. Thus, to the left of point 5 the graphite-gas phase equilibrium is limiting the maximum water production.

This simple graphical analysis shows that the one point where the maximum water concentration will exist and where there are no limitations imposed on any reaction in the Bosch sequence is point 5. At point 5 iron, iron oxide, graphite, and a gas phase can all exist at equilibrium.

Manning (1976), assuming iron oxides may not be catalysts for carbon deposition, performed some preliminary optimization studies to determine optimal operating conditions. He assumed that the O/H operating ratio was constrained by mass balances

at 0.5 (i.e., the inlet H_2/CO_2 ratio set by system stoichiometry).

Manning investigated a simple recycle Bosch reactor and a variety of different Bosch reactor configurations with prereactors. These prereactors took advantage of the results of Kusner (1962) and Barkley et al. (1952) who determined that the reverse water-gas shift reaction would go to equilibrium in a packed bed; the idea of the prereactor being to take out some of the water prior to entering the Bosch reactor. He determined the prereactor-Bosch reactor configuration to be the most efficient, efficiency being defined as the total moles recycled/mole of CO_2 processed.

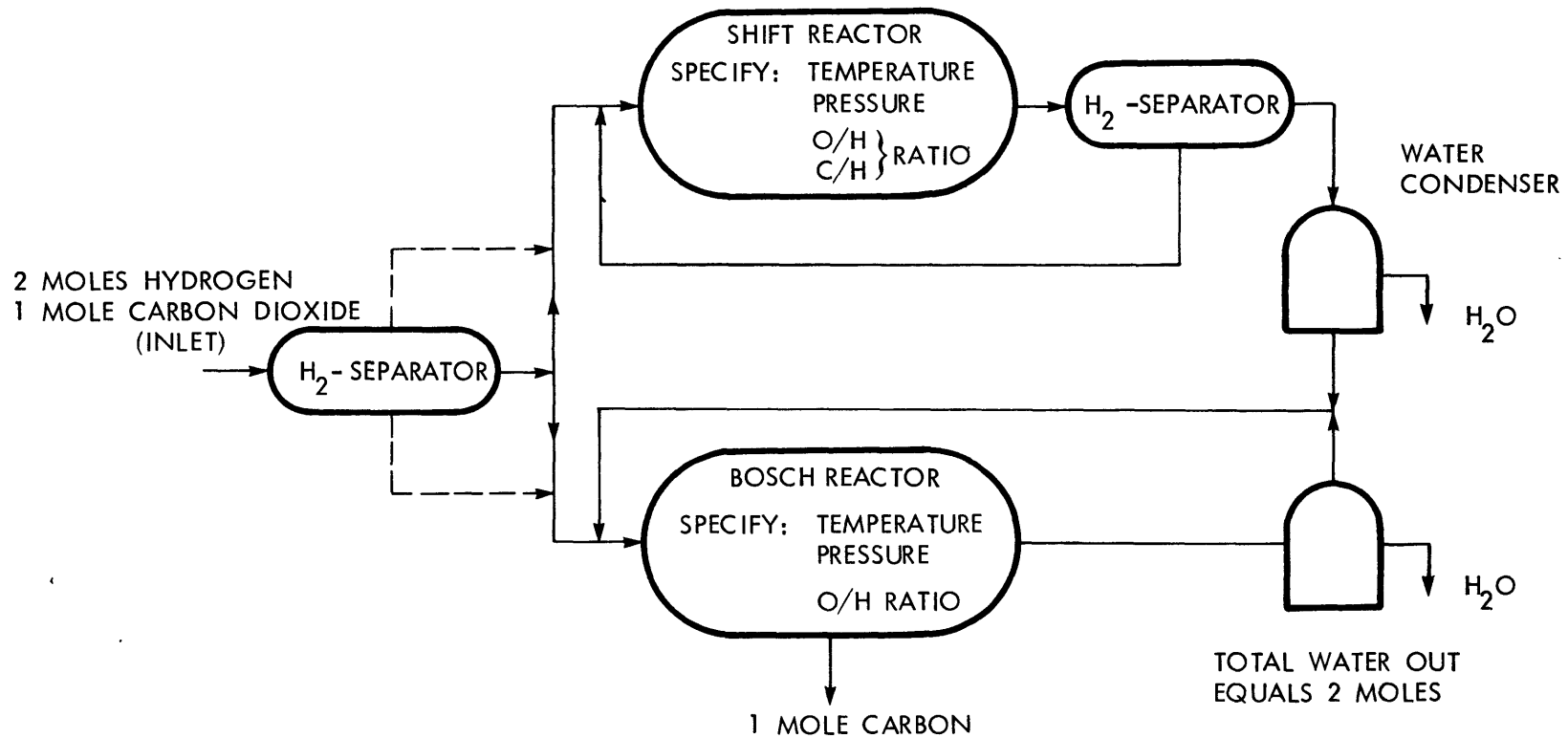
The optimal operating conditions in this case would be determined by the intersection of the optimal O/H ratio (i.e., the interaction between the graphite-gas/iron-iron oxide equilibria, where the O/H ratio equals 0.5) with the operating line for the process where the O/H ratio equals 0.5, the operating line being determined by a line drawn from the carbon apex to the point representing water on a triangular diagram. The optimal conditions were determined to be 10.0 total moles recycled/mole of CO_2 processed at approximately 915 K.

As mentioned previously, the optimal O/H ratio (intersection point) is the point, at any given temperature, where one would ideally like to operate. That is, the point of highest water concentration. This optimal O/H ratio will

increase with temperature. Manning limited his analysis to the case where the O/H ratio throughout the system was fixed by the inlet H_2/CO_2 ratio to a value of 0.5.

This limitation can be overcome, however, if one allows for the addition or removal of hydrogen to the system on start-up. A balance can then be made which equates the product of the recycle rate from the Bosch reactor times the water concentration (at the optimum O/H ratio desired) to the rate at which oxygen, hydrogen, and carbon are being fed to the reactor. This type of design is optimal using a shift reactor to initially take out some of the water. A conceptual reactor design is shown in Figure 57. An actual reaction gas flow path is shown in Figure 56 as envisioned on a triangular phase diagram at 875 K.

Initially, 2 moles of hydrogen are mixed with 1 mole of CO_2 and fed to the shift reactor. Point 1 represents the position of the mixture which must fall on the intersection between the O/H operating line for the shift reactor (i.e., 0.5) and a line drawn from the position of carbon dioxide to the position of hydrogen as represented on the phase diagram. Removing the water formed in the shift reactor, the gas mixture moves along the O/H operating line to point 2. The gas mixture at point 2 is mixed with a gas mixture at point 3 to give point 4. The mixture at point 3 was composed of the Bosch reactor effluent at an O/H ratio of 0.204 minus the water formed. At this point, it should be again pointed out



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FIGURE 57 DESIGN FOR OPTIMAL REACTOR SYSTEM

hydrogen was added on start-up to initially get this optimum O/H ratio. The gas composition at 4 having an O/H ratio of 0.204 proceeds down the Bosch reactor operating line to the intersection point, at point 5.

Evaluation of this process indicates a minimum recycle ratio of 9.2 at reactor temperature of 875 K for both the shift and Bosch reactors.

The question of the effect of increased hydrogen content in the shift reactor was next addressed; the idea being that perhaps the moles recycled could be substantially decreased by "pushing" reaction F to the right in the prereactor. This hopefully would allow a larger fraction of water to be produced and consequently removed prior to processing in a Bosch recycle reactor. This could be done by feeding the inlet gases entirely to the shift reactor. Then, using a palladium-silver membrane separator, hydrogen would be recycled to the shift reactor increasing the hydrogen content. Figure 58 is a computer generated plot of the total moles recycled (hydrogen and Bosch recycle) per mole of CO_2 processed in the system versus temperature. The optimal O/H ratio (intersection point) was used in calculating the minimum recycle at each temperature. As shown, the hydrogen recycle increased the total moles recycled per mole of CO_2 processed. The recycle in the Bosch reactor did indeed go down but only insignificantly relative to the increased hydrogen recycle rate.

If one maintains no recycle in the shift reactor, the

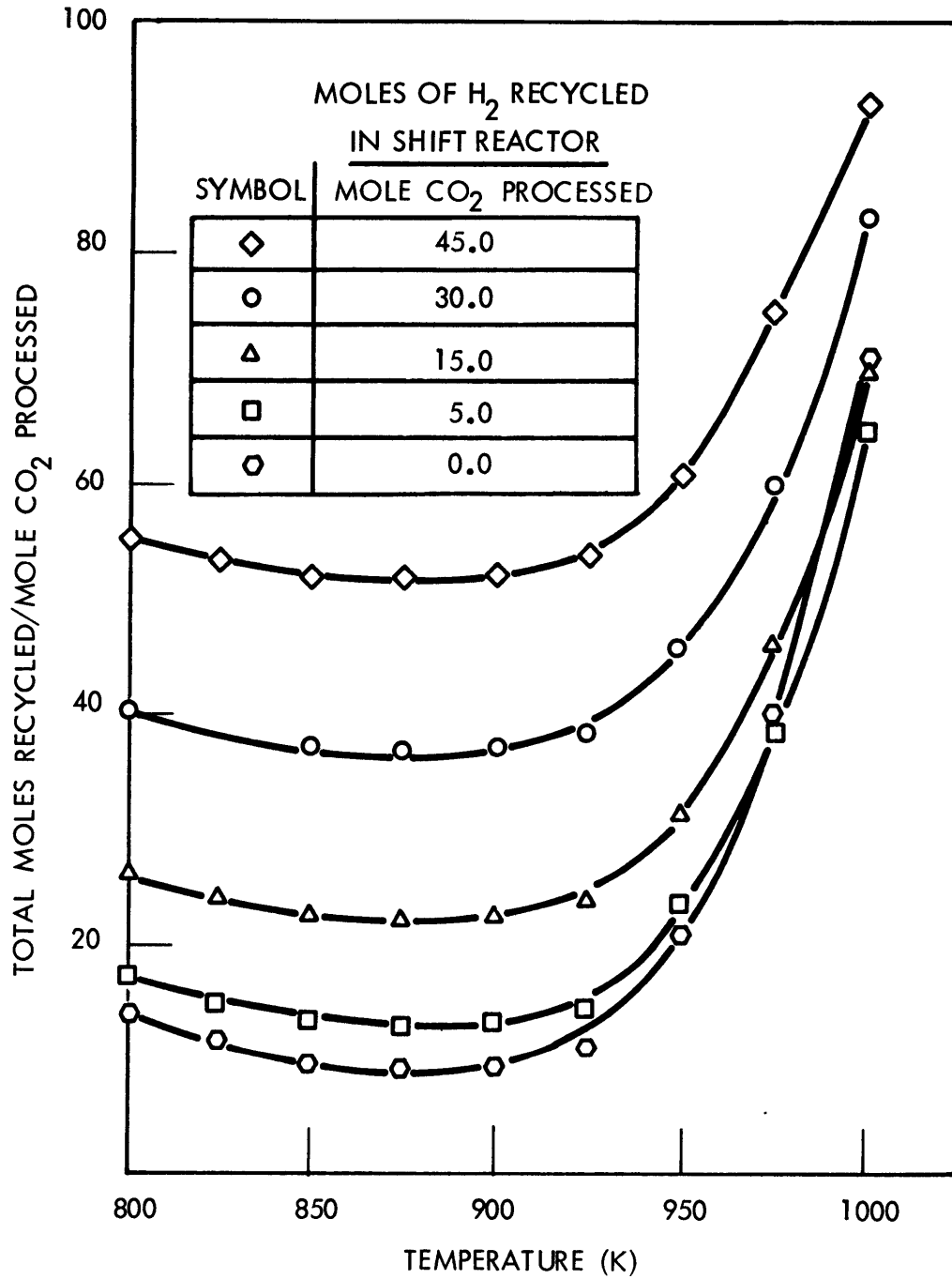


FIGURE 58

EFFECT OF HYDROGEN RECYCLE IN THE SHIFT REACTOR ON TOTAL MOLES RECYCLED

effect of decreased H_2/CO_2 feed ratio to the shift reactor can be investigated by using a palladium-silver membrane separator prior to entering the reactor system, Figure 57. Figure 59 indicates by decreasing the inlet H_2/CO_2 ratio (i.e., increasing the O/H ratio), the minimum moles recycled, again increased.

The reverse water-gas shift reaction is endothermic. The possibility therefore exists for a decrease in minimum recycle with increase in shift reactor temperature relative to Bosch reactor temperature. In Figure 59 the dotted lines shown indicate conditions under which the Bosch reactor is maintained at the temperature indicated by the intersection of the curve for H_2/CO_2 equal to two and the dotted line, while the minimum recycle is obtained by following dotted lines to the desired shift temperature. The total moles recycled is then read off the abscissa. For example, at a Bosch reactor temperature of 875 K if we increase the shift reactor temperature to 950 K (following the dotted line), the total moles recycled is seen to decrease by 3%.

The improvement in Bosch efficiency with increased shift reactor temperature is small. Also, the penalty for increased hydrogen recycle in the shift reactor was shown to be restrictive. Thus, the suggested mode of operation, based on equilibrium consideration alone, is to run both a shift and Bosch reactor at 875 K with a recycle ratio, at the optimum O/H value, of 9.2 total moles recycled per mole of CO_2 processed.

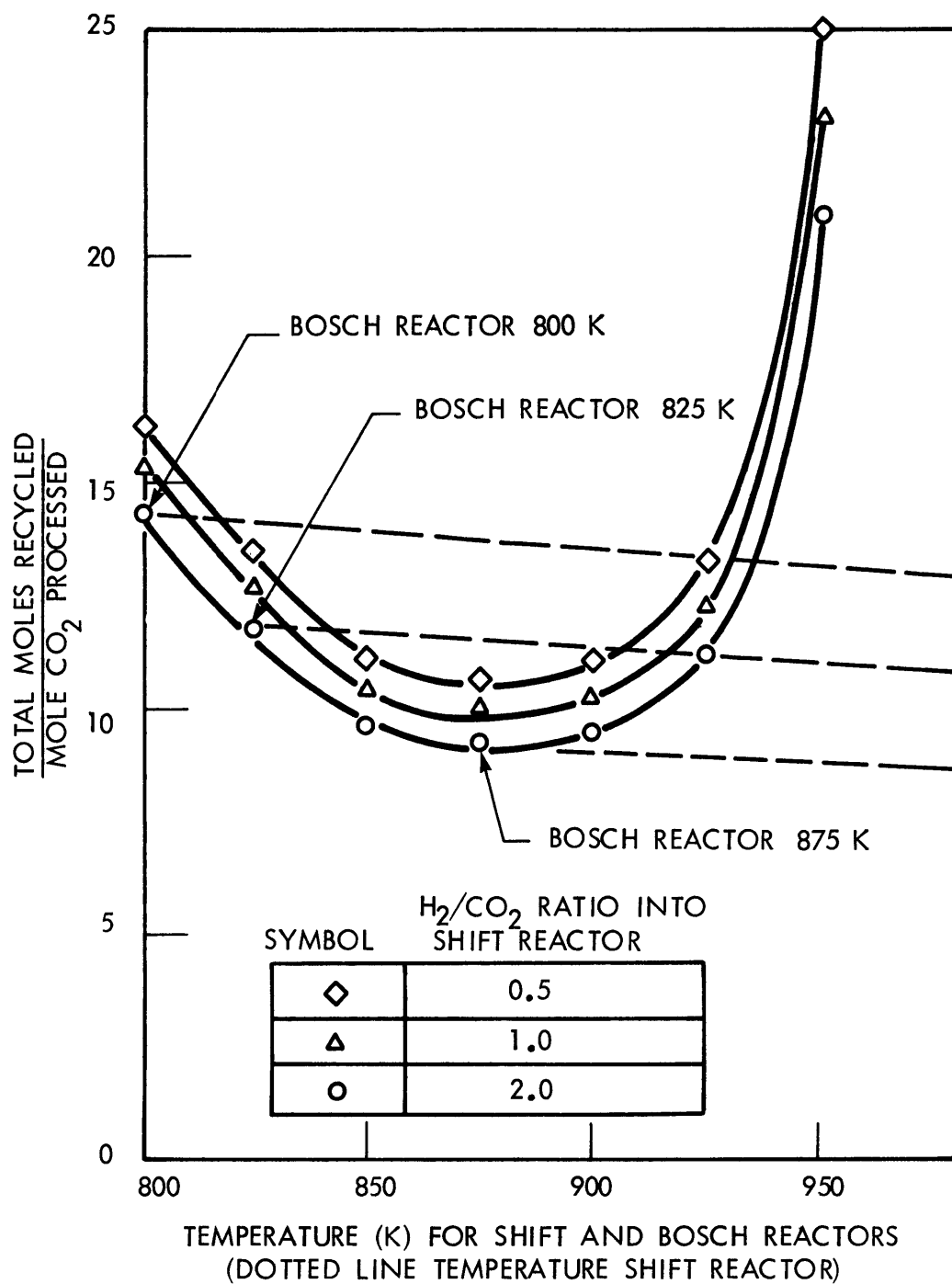


FIGURE 59
EFFECT OF INLET $\text{H}_2/\text{CO}_2 \leq 2.0$ IN THE SHIFT REACTOR ON TOTAL MOLES RECYCLED

6. Conclusions and Recommendations

6.1 Conclusions

The following conclusions were drawn from the data presented in the Results Section and represent to the best of the author's ability, the essence of these data.

6.1.1 Carbon Deposition and Methane Formation

The results of the preliminary experiments indicate that at high temperature (i.e., >930 K) and high carbon monoxide conversion (i.e., $>9\%$), the Boudouard reaction is the primary carbon formation reaction. At low conversion, no definitive statements can be made. The inability to determine the precise mechanism for carbon deposition at low conversion can be attributed to the inherent errors involved in experimentation and data analysis.

Methane formation has been shown to be kinetically favorable at high temperature (>930 K). Once formed, methane concentration appears to remain constant.

6.1.2 Preconditioning at 800 K and 900 K

Preconditioning a steel wool catalyst by first oxidizing in carbon dioxide and water and then reducing in hydrogen has been shown to produce two distinct surface structures depending on temperature.

At 800 K, a dense irregular surface is formed around a smooth central core. This type of morphology is representa-

tive of a reduction process which is controlled by solid state diffusion.

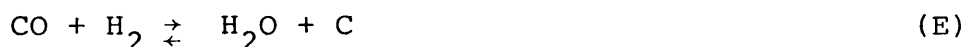
When preconditioning is performed at 900 K, the resulting shell and core type morphology is highly porous. This type of reduction can be best represented by a model assuming a porous catalyst structure under diffusion (gas) or mixed diffusion and interfacial reaction control. The outer porous shell has been shown to consist of two distinct layers, suggesting multiple oxide formation. The solid central core appears to be regular and rather smooth in appearance.

6.1.3 Oxidation Effects on Carbon Deposition

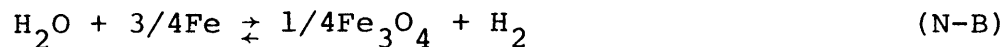
Oxide(s) formed at 800 and 900 K were investigated to determine their effect on carbon deposition. The following results were obtained.

6.1.3.1 Effect of Magnetite (Fe_3O_4) Formation

Data have been presented which indicate Fe_3O_4 is not a catalyst for carbon deposition from reaction D or E



The theoretically predicted alpha-iron/magnetite phase boundary has been experimentally verified. The location was determined by observing the point at which carbon deposition starts or stops. The solid phase favored during reaction was found to be controlled by adjusting the $P_{\text{H}_2}/P_{\text{H}_2\text{O}}$ ratio (reaction N-B).



6.1.3.2 Effect of Wustite (Fe_{1-y}O) Formation

Wustite has been shown to inhibit carbon deposition. The theoretically predicted alpha-iron/wustite phase boundary was shown to be accurately predicted by the experimental data. Scatter observed around the phase boundary was attributed to error involved in experimental procedures and data acquisition and reduction. Multiple oxide formation was also shown to be important in predicting the system response at temperatures above 833-843 K.

Although oxide formation was shown to be critical in determining if carbon would deposit, the data indicate a critical driving force for reactions D and E is also of importance.

6.1.4 Carbide Formation and Carbon Inhibition

Carbide formation and/or carbon deposition from methane was shown to be slow at 823 K. Data indicate that methane formation does not proceed primarily through a carbide intermediate. The main methane formation reaction in a carbon, hydrogen, methane system is reaction G, catalyzed by metallic iron.



At 900 K, in a 5 component gas mixture, carbon deposition from carbon monoxide has been found to be fast relative to carbide formation. The effect of substantial carbiding on

carbon deposition has not been established. However, due to the slow rate of carbide formation from both carbon monoxide and methane, it is believed that oxide formation presents the limiting factor in efficient Bosch operation.

6.1.5 Structural Effects

A series of transmission and scanning electron micrographs have shown a rather complex series of structural changes can occur during normal Bosch operations. Micrographs have been presented which clearly show the formation of multiple oxides. Depending on temperature, the oxidation-reduction history of the catalyst will greatly effect the structure and associated porosity.

Following standard preconditioning, carbon deposition resulted in the shell or surface layer forming localized fiber bundles. These bundles or nodules were suspended from the main body by hollow, tubular shaped carbon fibers and were determined by electron diffraction analysis to be composed of iron and iron compounds. Carbon fibers were seen to use these nodules as growth centers. The fibers themselves were hollow, tubular, or circular in shape, having electron dense material along their length as well as at their tip.

In combination these various structural effects add a high degree of complexity to any analysis of the Bosch process.

6.2 Recommendations

The conclusions drawn from this investigation lead to the

following recommendations: first, if it is desirable to use steel wool as a catalyst, there is an optimum system design. This optimum design utilizes the knowledge that carbide formation is slow and will not influence process operation. However, care must be taken to avoid and/or utilize oxide formation to maximum advantage. Second, if other transition metal catalysts are investigated, the amount of water or carbon dioxide that the system can maintain before oxidation must be carefully evaluated. This is, hopefully, to avoid the oxide limitation found in this investigation. Also, the rate and conditions under which carbides formed should be carefully evaluated.

6.2.1 Optimal Design

A computer simulation of various reactor systems indicates a reverse water-gas shift prereactor in series with a recycle Bosch reactor is the optimal system design. The optimal operating conditions are 9.2 total moles recycled per mole of CO_2 processed at a temperature of 875 K in both reactors.

6.2.2 Other Catalyst Systems

The iron system has been shown to be restrictive due to oxide formation. Other transition metals such as nickel and cobalt have been known to catalyze all the reaction systems involved in the Bosch sequence but, no determination as to the behavior of their oxides is known. Garmirian and Reid (1977) have shown that oxide formation may not be a problem for these systems. That is, the nickel-nickel oxide and cobalt/cobalt

oxide systems have equilibrium water concentrations well above that expected for the Bosch system (i.e., the graphite-gas equilibrium). Thus, these two metals are possible candidates for the Bosch process.

In all metallic catalyst systems, the various phases which form during reaction should be closely evaluated. A tractable way to accomplish this is the phase diagram type of analysis used in this investigation.

7.0 Appendix

7.1 Mathematical Model of Iron Oxide Reduction

A general treatment of the methodology used in deriving mathematical models to describe the kinetics of gas-solid reaction is presented below. The models assume, in all cases, that the combined transport and chemical reaction steps can be represented by electrical circuits. These circuits are analyzed using conventional circuit analysis, resulting in the desired mathematical representation.

The rate of removal of oxygen (gram-atoms per second) from a single iron oxide particle is equal to the molar oxygen density of the core times the rate of consumption of the core volume.

$$\dot{N}_O = C_O \frac{dV}{d\theta} = C_O 4\pi x_i^2 \frac{dx_i}{d\theta} \quad (11)$$

From the stoichiometry of iron-oxide reduction:

$$\dot{N}_O = \dot{N}_A = -\dot{N}_B \quad (12)$$

where A is the reactant (i.e., hydrogen or carbon monoxide) and B is the product (i.e., water or carbon dioxide). The following are the "resistances" of importance.

7.1.1 Gas-Film Resistance

The molar flux between the bulk stream and the solid surface is given by:

$$\dot{N}_{j,\text{film}} = - \left(\frac{k_{m,j}}{R_g T} 4\pi x_o^2 \right) \cdot (P_j^{(b)} - P_j^{(o)}), \quad j=A,B \quad (13_a, 13_b)$$

k_m was determined by:

$$\frac{k_m 2x_o}{D_{AB}} = 2.0 + 0.60 \left(\frac{2x_o G_\infty'}{\mu} \right)^{1/2} \left(\frac{\mu}{\rho D_{AB}} \right)^{1/3} \quad (14)$$

The physical properties of the gas phase are a function of composition. Thus, the Reynolds and Schmidt numbers were evaluated at the film composition, taken as the arithmetic mean of the bulk-phase and the surface composition.

7.1.2 Shell-Layer Resistance

The molar current through a porous layer is given by:

$$\dot{N}_j(t) = - \left(\frac{4\pi x_i x_o}{x_o - x_i} \right) \frac{D_j^{\text{eff}}}{R_g T} \cdot (P_j^{(o)} - P_j^{(i)}), \quad j=a,b \quad (15_a, 15_b)$$

$$D_j^{\text{eff}} = D_j^{(P)} \frac{\epsilon_{Fe}}{\tau} \quad (16)$$

$D_j^{(P)}$ is the diffusion coefficient in a single pore, ϵ_{Fe} is the void fraction of the porous solid layer, and τ is a tortuosity factor depending on structure of the reduced oxide layer.

7.1.3 Interface Resistance

The rate of consumption of reactant (A) and formation of product gas (B) due to chemical reaction is proportional to

the area of the receding interface.

$$-N_O = r_O 4\pi x_i^2 \quad (17)$$

assuming a first order reversible reaction:

$$r_O = \frac{k_r(t)}{R_g T} \left(P_a^{(i)} - \frac{P_b^{(i)}}{K_e} \right) \quad (18)$$

As an example we can take the case of the dense pellet model developed by Spitzer et al. (1966). Equations 13, 15, and 18 can be put in terms of oxygen removal and summed to eliminate the surface and interface partial pressures.

$$\left(\frac{R_g T}{k_{m, (A)}} \cdot \frac{1}{4\pi x_O^2} \right) \cdot (-\dot{N}_O) = P_A^{(b)} - P_A^{(o)}$$

$$\left(\frac{R_g T}{D_A^{eff}} \cdot \frac{\{x_O - x_i\}}{4\pi x_i x_O} \right) \cdot (-\dot{N}_O) = P_A^{(o)} - P_A^{(i)}$$

$$\left(\frac{R_g T}{k_r} \cdot \frac{1}{4\pi x_i^2} \right) \cdot (-\dot{N}_O) = P_A^{(i)} - P_B^{(i)} / K_e$$

$$\left(\frac{R_g T}{K_e D_B^{eff}} \cdot \frac{\{x_O - x_i\}}{4\pi x_i x_O} \right) \cdot (-\dot{N}_O) = \frac{P_B^{(i)}}{K_e} - \frac{P_B^{(o)}}{K_e}$$

$$\left(\frac{R_g T}{K_e k_{m, (B)}} \cdot \frac{1}{4\pi x_O^2} \right) \cdot (-\dot{N}_O) = \frac{P_B^{(o)}}{K_e} - \frac{P_B^{(b)}}{K_e}$$

$$\left(\frac{1}{\alpha 4\pi x_o^2} + \frac{x_o - x_i}{\beta 4\pi x_i x_o} + \frac{1}{k_r 4\pi x_i^2} \right) \cdot \{-\dot{N}_o\} = \frac{1}{R_g T} \left\{ P_A^{(b)} - \frac{P_B^{(b)}}{K_e} \right\} \quad (19)$$

where

$$\alpha = \frac{K_e k_{m,(A)} k_{m,(B)}}{K_e k_{m,(B)} + k_{m,(A)}}$$

and

$$\beta = \frac{K_e D_A^{eff} D_B^{eff}}{K_e D_B^{eff} D_A^{eff}}$$

substituting equation 11 into equation 19 results in equation 20

$$\frac{dx_i}{d\theta} = - \frac{k_{ov}}{R_g T C_o} \left(P_A^{(b)} - \frac{P_B^{(b)}}{K_e} \right) \quad (20)$$

The more complex electrical resistance analogies formulated by Spitzer et al. (1966) and Szekely et al. (1976), used the same basic approach. For additional details, the reader is encouraged to consult the appropriate articles.

7.2 Determination of Triangular Phase Diagrams

In this investigation the catalytic activity of iron oxides and iron carbides for carbon deposition were to be determined. This determination was to be made under actual Bosch operating conditions; the gas phase being composed of CH_4 , H_2 , H_2O , CO , and CO_2 , while the solid phase would consist of carbon deposited on a steel wool catalyst.

A convenient way to approach this problem is to construct

an equilibrium phase diagram including all pertinent solid phases. By simultaneously adjusting the gas phase composition to thermodynamically favor carbon deposition and also to be in the phase region of interest, the catalytic activity of this particular phase could be determined. The solid phase being investigated is determined to be catalytic if a weight increase in the catalyst assembly is noticed and if the reaction effluent indicates carbon monoxide conversion. If no weight increase is observed, the particular phase of interest is assumed to be noncatalytic towards carbon deposition.

There are many different forms of equilibrium phase diagrams. In this investigation the approach developed by Cairns and Tevebaugh (1964) was adopted. Cairns and Tevebaugh derived the equilibrium phase boundary for the graphite-gas system and plotted it on a triangular diagram. These triangular phase diagrams are convenient in that they graphically indicate the gas phase composition in percentage of O, C, and H in equilibrium with the solid phase of interest.

7.2.1 Graphite-Gas Phase Boundary

To derive the graphite-gas phase boundary one first must determine the intensive variables necessary to completely specify the system. This is accomplished through the application of Gibb's phase rule.

For the graphite-gas system:

$N =$ number of components ($\text{CH}_4, \text{H}_2, \text{H}_2\text{O}, \text{CO}, \text{CO}_2, \text{C}$) = 6

$\pi =$ " " phases (gas, solid) = 2

$f = 6 + 2 - 2 - R = 6 - R$

R , the number of independent reactions, can be shown to equal 3; thus, the number of intensive variables, $f = 3$. By specifying the pressure and O/H ratio (in the gas phase) at a given temperature, the system is completely fixed.

$$P_{\text{Total}} = P_{\text{H}_2} + P_{\text{CH}_4} + P_{\text{H}_2\text{O}} + P_{\text{CO}} + P_{\text{CO}_2} \quad (21)$$

$$\text{O/H} = \frac{P_{\text{CO}} + P_{\text{H}_2\text{O}} + 2P_{\text{CO}_2}}{2P_{\text{H}_2} + 2P_{\text{H}_2\text{O}} + 4P_{\text{CH}_4}} \quad (22)$$

$$2\text{H}_2 + \text{C} \rightleftharpoons \text{CH}_4 \quad K_a = \frac{P_{\text{CH}_4}}{P_{\text{H}_2}^2 \cancel{a_c^1}} \quad (23)$$

$$\text{H}_2 + \text{CO}_2 \rightleftharpoons \text{CO} + \text{H}_2\text{O}, \quad K_b = \frac{P_{\text{CO}} P_{\text{H}_2\text{O}}}{P_{\text{H}_2} P_{\text{CO}_2}} \quad (24)$$

$$3\text{H}_2 + \text{CO}_2 \rightleftharpoons \text{CH}_4 + \text{H}_2\text{O}, \quad K_c = \frac{P_{\text{CH}_4} P_{\text{H}_2\text{O}}}{P_{\text{CO}} P_{\text{H}_2}^3} \quad (25)$$

Equations 21, 22, 23, 24, and 25 were solved by successive substitution giving a single polynomial of the form.

$$W_6 P_{\text{H}_2}^6 + W_5 P_{\text{H}_2}^5 + W_4 P_{\text{H}_2}^4 + W_3 P_{\text{H}_2}^3 + W_2 P_{\text{H}_2}^2 + W_1 P_{\text{H}_2} + W_0 = 0 \quad (26)$$

Where $W_0 - W_6$ are expressed as functions of P_{Total} , $\sum_{i=a}^Z K_i$, and O/H ratio.

Using conventional iterative search techniques, such as the Newton-Raphson method, P_{H_2} was numerically determined and the procedure was reversed to obtain the remaining partial pressures.

7.2.2 Iron-Iron Oxide-Gas Phase Boundary

Manning (1976) extended the method developed by Cairns and Tevebaugh to the iron-iron oxide equilibrium. Here $N = 7$, $\pi = 3$, and $R = 3$; thus, $f = 3$ and again temperature, pressure, and, this time, C/H ratios were specified. The following set of equations was determined to specify completely the system at a given temperature.

$$P_{\text{Total}} = P_{\text{H}_2} + P_{\text{CH}_4} + P_{\text{H}_2\text{O}} + P_{\text{CO}} + P_{\text{CO}_2} \quad (21)$$

$$a\text{H}_2\text{O} + b\text{Fe} \rightleftharpoons \text{Fe}_b\text{O}_a + a\text{H}_2, \quad K_d = \frac{P_{\text{H}_2}^a P_{\text{Fe}_b\text{O}_a}^1}{P_{\text{H}_2\text{O}}^a P_{\text{Fe}}^b} \quad (27)$$

$$a\text{CO}_2 + b\text{Fe} \rightleftharpoons \text{Fe}_b\text{O}_a + a\text{CO}, \quad K_e = \frac{P_{\text{CO}}^a P_{\text{Fe}_b\text{O}_a}^1}{P_{\text{CO}_2}^a P_{\text{Fe}}^b} \quad (28)$$

$$3\text{H}_2 + \text{CO} \rightleftharpoons \text{CH}_4 + \text{H}_2\text{O}, \quad K_c = \frac{P_{\text{CH}_4} P_{\text{H}_2\text{O}}}{P_{\text{CO}} P_{\text{H}_2}^3} \quad (25)$$

$$\text{C/H} = \frac{P_{\text{CO}} + P_{\text{CO}_2} + P_{\text{CH}_4}}{2P_{\text{H}_2} + 2P_{\text{H}_2\text{O}} + 4P_{\text{CH}_4}} \quad (29)$$

Again, the method of successive substitutions was used to obtain a single polynomial which was solved using the Newton-Raphson interactive search technique.

7.2.3 Iron-Iron Carbide-Gas Boundary

The final solid phases of interest were carbides. Following the same approach as discussed previously, the following equations at fixed temperature were found to completely determine the system.

$$P_{\text{Total}} = P_{\text{H}_2} + P_{\text{H}_2\text{O}} + P_{\text{CH}_4} + P_{\text{CO}} + P_{\text{CO}_2} \quad (21)$$

$$O/H = \frac{P_{\text{CO}} + 2P_{\text{CO}_2} + P_{\text{H}_2\text{O}}}{2P_{\text{H}_2} + 2P_{\text{H}_2\text{O}} + 4P_{\text{CH}_4}} \quad (22)$$

$$3\text{Fe} + \text{CH}_4 \rightleftharpoons 2\text{H}_2 + \text{Fe}_3\text{C} , \quad K_g = \frac{a_{\text{Fe}}^3 a_{\text{C}}^1 P_{\text{H}_2}^2}{P_{\text{CH}_4} a_{\text{Fe}}^3 a_{\text{C}}^1} \quad (30)$$

$$\text{CH}_4 + \text{CO}_2 \rightleftharpoons 2\text{H}_2 + 2\text{CO} , \quad K_h = \frac{P_{\text{H}_2}^2 P_{\text{CO}}^2}{P_{\text{CO}_2} P_{\text{CH}_4}} \quad (31)$$

$$4\text{H}_2 + \text{CO}_2 \rightleftharpoons \text{CH}_4 + 2\text{H}_2\text{O} , \quad K_I = \frac{P_{\text{CH}_4} P_{\text{H}_2\text{O}}^2}{P_{\text{CO}_2} P_{\text{H}_2}^4} \quad (32)$$

Following the same mathematical procedure as before, the solution of these equations determined the appropriate gas phase composition in equilibrium with iron and iron carbide.

The numerical solution for each of the individual phase boundaries was done by subroutine computer programs which were loaded on call to a main plotting routine as desired. The main plotting routine and support subroutines are shown at the end of this section. Figure 60 is a sample of a typical triangular phase diagram showing the various solid phase regions.

7.2.4 Determination of Reactant Gas Compositions

The inlet gas compositions were determined in a similar fashion. In order to be able to move from one phase field to the next, it was necessary to be able to fix the P_{H_2}/P_{H_2O} at a desired value as well as the temperature, pressure, and O/H or C/H ratios. Thus, the number of intensive variables (f) is seen to be:

$$f = N + 2 - \pi - R - 4$$

Assuming initially that all solid phases will catalyze all reactions which make up the Bosch reaction sequence, $\pi = 1$ (gas), $R = 2$ (reactions 24 and 25), fixing temperature, pressure, C/H or O/H and the P_{H_2}/P_{H_2O} ratio for the 5 component gas mixture will completely specify the system. A computer program is shown at the end of the section which calculates the gas composition, gives the appropriate flow settings for the feed gas delivery section, and sets the saturator bath temperature.

This approach presents a straight-forward way to obtain the desired gas compositions over the phase field of interest.

C	TDIA CONSTRUCTS A TRIANGLER PHASE DIAGRAM AND LABELS IT	AL 001
C	EQKS CALCULATES THE APROPRIATE EQUILBRIUM CONSTANTS, ASSUMING IDEA	AL 002
C	L GAS CONDITIONS AND UNIT FUGASITY FOR CONDENSED PHASES	AL 003
C	GRAPH CONSTRUCTS THE GRAPHITE GAS PHASE BOUNDARY	AL 004
C	BFE3C CALCULATES THE CARBON -OXYGEN HYDROGEN-OXYGEN BOUNDARY CONDI	AL 005
C	TION FOR THE FE3C-FE-GAS PHASE BOUNDARY	AL 006
C	FE3C CONSTRUCTS THE FE3C- FE-GAS PHASE BOUNDARY	AL 007
C	BIHO CALCULATES THE HYDROGEN-OXYGEN BOUNDARY CONDITION FOR THE IR	AL 008
C	ON-IRON OXIDE PHASE BOUNDARY	AL 009
C	BICO CALCULATES THE CARBON-OXYGEN BOUNDARY CONDITION FOR THE IRON-	AL 010
C	IRON OXIDE PHASE BOUNDARY	AL 011
C	IRON CONSTRUCTS THE IRON-IRON OXIDE PHASE BOUNDARY	AL 012
C	BHC CALCULATES THE HYDROGEI-CARBON BOUNDARY CONDITION FOR THE GRAP	AL 013
C	HITE-GAS PHASE BOUNDARY	AL 014
C	BGCO CALCULATES THE CARBON-OXYGEN BOUNDARY FOR THE GRAPHITE-GAS EQ	AL 015
C	UILIBRIUM	AL 016
	EXTERNAL IEOEQ,TCT2,DA	AL 017
	COMMON CO,C1,C2,C3,C4,C5,C6,EPS,IEND,XST	AL 018
	PRES=1.	AL 019
	PI=3.14159	AL 020
	K=1	AL 021
	DO 10 I=1,K,1	AL 022
	CALL SCALF(1.,1.,0.0,2.)	AL 023
	READ(2,999)TEMP	AL 024
999	FORMAT(F5.1)	AL 025
	CALL EQKS(TEMP,EQCH4,EQCO2,EQH2O,EQH2,EQCO,EQC,EQFE3,EQFE2,E	AL 026
	1QCO1,EQH21)	AL 027
	CALL DATSW(7,J)	AL 028
	IF(J-1)22,98,99	AL 029
22	WRITE(3,269)	AL 030
269	FORMAT('TROUBLE IN DATSW')	AL 031
	GO TO 99	AL 032
98	WRITE(3,97)	AL 033
97	FORMAT(4X,'KCH4',4X,'KCO2',4X,'KH2O',4X,'KH2',5X,'KCO',5X,'KC',5X,	AL 034
	1'KFE3C',3X,'KFE2C',4X,'KCO1',4X,'KH21')	AL 035

WRITE(3,900)EQCH4,EQCO2,EQH20,EQH2,EQCO,EQC ,EQFE3,EQFE2,EQCO1,EQH	AL 036
121	AL 037
900 FORMAT(1X,10F8.3)	AL 038
99 CONTINUE	AL 039
CALL TDIA(TEMP,PRES)	AL 040
CALL BIHO(PRES,EQH2,PIH2,PIH20)	AL 041
CALL BICO(PRES, EQCO,PICO,PICO2)	AL 042
CALL IRON(EQH2,EQH20,EQCO,IEOEQ,TCT2,PIH2,PIH20,PICO,PICO2,TEMP)	AL 043
CALL BHC (PRES,EQCH4,PGH2,PGCH4)	AL 044
CALL BGCO(PRES,EQC,PGCO,PGCO2)	AL 045
CALL GRAPH(EQCO2,EQH20,EQCH4,IEOEQ,TCT2,PGH2,PGCH4,PGCO,PGCO2,TEMP	AL 046
1)	AL 047
CALL BFE3C(PRES,EQFE2,EQFE3,PFCO,PFCO2,PFCH4,PFH2)	AL 048
CALL FE3C(EQFE3,EQCO1,EQH21,IEOEQ,TCT2,PFH2,PFCH4,PFCO,PFCO2,DA,TE	AL 049
1MP)	AL 050
10 CONTINUE	AL 051
CALL EXIT	AL 052
END	AL 053

SUBROUTINE TDIA(TEMP,PRES)	TDIA 001
C TDIA USED TO CONSTRUCT A TRIANGULAR PHASE DIAGRAM FOR THE IRON, IRON-O	TDIA 002
C XIDE, IRON-CARBIDE, CARBON, NICKEL-NICKEL OXIDE, COBALT-COBALT OXIDE A	TDIA 003
C ND GAS SYSTEMS	TDIA 004
1 FORMAT('HYDROGEN')	TDIA 005
2 FORMAT('OXYGEN')	TDIA 006
3 FORMAT('CARBON')	TDIA 007
4 FORMAT('UPPER CURVE*')	TDIA 008
5 FORMAT('IRON-IRON OXIDE-GAS EQUILIBRIA')	TDIA 009
6 FORMAT('LOWER CURVE*')	TDIA 010
7 FORMAT('GRAPHITE-GAS EQUILIBRIA')	TDIA 011
8 FORMAT('TEMP.=', F6.2, 'K')	TDIA 012
9 FORMAT('PRES.=', F4.2, 'ATM')	TDIA 013
11 FORMAT('CO')	TDIA 014
12 FORMAT('CO2')	TDIA 015
13 FORMAT('H2O')	TDIA 016
14 FORMAT('CH4')	TDIA 017
15 FORMAT('MIDDLE CURVE ')	TDIA 018
16 FORMAT('IRON-IRON CARBIDE-GAS EQUILIBRIA')	TDIA 019
CALL FPLOT(3,12.,2.)	TDIA 020
CALL SCALF(.85,.85,0.,0.)	TDIA 021
X=5.	TDIA 022
Y=5.*SQRT(3.)	TDIA 023
CALL FPLOT(2,10.,0.0)	TDIA 024
CALL FPLOT(0,X,Y)	TDIA 025
CALL FPLOT(0,0.,0.)	TDIA 026
PI=3.14159	TDIA 027
D=PI/6.	TDIA 028
X=1.	TDIA 029
DO 28 I=1,9,1	TDIA 030
CALL FPLOT(0,X,0.)	TDIA 031
Z=X-(X*SIN(D))	TDIA 032
W=X*COS(D)	TDIA 033
CALL FPLOT(-3,Z,W)	TDIA 034
X=X+1.	TDIA 035

```

      CALL FPLOT(-2,X,0.)
28  CONTINUE
      Q=1.
      C=X
      DO 29 I=1,9,1
      Z=C-(Q*SIN(D))
      W=Q*COS(D)
      CALL FPLOT(0,Z,W)
      X=X-1.
      CALL FPLOT(0,X,0.)
      CALL FPLOT(0,C,0.)
      Q=Q+1.
29  CONTINUE
      Q=1.
      DO 30 I=1,5,1
      Z=C-(Q*SIN(D))
      W=Q*COS(D)
      CALL FPLOT(3,Z,W)
      T=Q-(Q*SIN(D))
      G=Q*COS(D)
      CALL FPLOT(2,T,G)
      Q=Q+1.
      IF(9.-Q)32,34,34
34  T=Q-(Q*SIN(D))
      G=Q*COS(D)
      CALL FPLOT(3,T,G)
      Z=C-(Q*SIN(D))
      W=Q*COS(D)
      CALL FPLOT(2,Z,W)
      Q=Q+1.
30  CONTINUE
32  CALL FCHAR(-.5,-.3,.1,.1,0.)
      WRITE(7,1)
      CALL FCHAR(9.62,-.3,.1,.1,0.)
      WRITE(7,2)

```

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TDIA 036
TDIA 037
TDIA 038
TDIA 039
TDIA 040
TDIA 041
TDIA 042
TDIA 043
TDIA 044
TDIA 045
TDIA 046
TDIA 047
TDIA 048
TDIA 049
TDIA 050
TDIA 051
TDIA 052
TDIA 053
TDIA 054
TDIA 055
TDIA 056
TDIA 057
TDIA 058
TDIA 059
TDIA 060
TDIA 061
TDIA 062
TDIA 063
TDIA 064
TDIA 065
TDIA 066
TDIA 067
TDIA 068
TDIA 069
TDIA 070

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CALL FCHAR(4.6,8.96 ,.1,.1,0.)
WRITE(7,3)
CALL FCHAR(.6,-.55,.1,.1,.0)
WRITE(7,4)
CALL FCHAR(2.,-.55,.2,.2,.0)
WRITE(7,5)
CALL FCHAR(.6,-1.1,.1,.1,.0)
WRITE(7,15)
CALL FCHAR(2., -1.1,.2,.2,.0)
WRITE(7,16)
CALL FCHAR(.6,-1.65,.1,.1,.0)
WRITE(7,6)
CALL FCHAR(2.,-1.65,.2,.2,.0)
WRITE(7,7)
CALL FCHAR(8.,8.,.2,.2,.0)
WRITE(7,8) TEMP
CALL FCHAR(8.,7.,.2,.2,.0)
WRITE(7,9) PRES
CALL FPLOT(3,7.5,4.33)
CALL FPLOT(2,7.5,4.33)
CALL POINT(1)
CALL FCHAR(7.55,4.33,.1,.1,.0)
WRITE(7,11)
CALL FPLOT(3,8.33,2.89)
CALL FPLOT(2,8.33,2.89)
CALL POINT(1)
CALL FCHAR(8.38,2.89,.1,.1,.0)
WRITE(7,12)
CALL FPLOT(3,3.33,.0)
CALL FPLOT(2,3.33,.0)
CALL POINT(1)
CALL FCHAR(3.13,-.15,.1,.1,.0)
WRITE(7,13)
CALL FPLOT(3,1.,1.73)
CALL FPLOT(2,1.,1.73)

TDIA 071
TDIA 072
TDIA 073
TDIA 074
TDIA 075
TDIA 076
TDIA 077
TDIA 078
TDIA 079
TDIA 080
TDIA 081
TDIA 082
TDIA 083
TDIA 084
TDIA 085
TDIA 086
TDIA 087
TDIA 088
TDIA 089
TDIA 090
TDIA 091
TDIA 092
TDIA 093
TDIA 094
TDIA 095
TDIA 096
TDIA 097
TDIA 098
TDIA 099
TDIA 100
TDIA 101
TDIA 102
TDIA 103
TDIA 104
TDIA 105


```
CALL POINT(1)
CALL FCHAR(.61,1.73,.1,.1,.0)
WRITE(7,14)
RETURN
END
```

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TDIA 106
TDIA 107
TDIA 108
TDIA 109
TDIA 110
```

<pre> SUBROUTINE BFE3C(PRES,EQFE2,EQFE3,PFCO,PFCO2,PFCH4,PFH2) C CALCULATES THE BOUNDARY CONDITIONS FOR *FE3C* PCO=(-1.+SQRT(1.+4.*EQFE2*PRES))/(2.*EQFE2) IF(PCO)3,2,2 2 PCO2=PRES-PCO IF(PCO2)3,5,5 3 PCO=(-1.-SQRT(1.+4.*EQFE2*PRES))/(2.*EQFE2) IF(PCO)4,6,6 6 PCO2=PRES-PCO IF(PCO2)4,5,5 5 PFCO2=PCO2 PFCO=PCO PH2=(-EQFE3+SQRT((EQFE3**2.)+4.*EQFE3*PRES))/2. IF(PH2)7,8,8 8 PCH4=PRES-PH2 IF(PCH4)7,12,12 7 PH2=(-EQFE3-SQRT((EQFE3**2.)+4.*EQFE3*PRES))/2. IF(PH2)10,11,11 11 PCH4=PRES-PH2 IF(PCH4)10,12,12 4 WRITE(3,13) 13 FORMAT(1X,'PROBLEM IN BFE3C, PFCO*PFCO2 SECTION') GO TO 5 10 WRITE(3,14) 14 FORMAT(1X,'PROBLEM IN BFE3C, PFH2*PFCH4 SECTION') 12 PFH2=PH2 PFCH4=PCH4 RETURN END </pre>	<pre> BFE3C 001 BFE3C 002 BFE3C 003 BFE3C 004 BFE3C 005 BFE3C 006 BFE3C 007 BFE3C 008 BFE3C 009 BFE3C 010 BFE3C 011 BFE3C 012 BFE3C 013 BFE3C 014 BFE3C 015 BFE3C 016 BFE3C 017 BFE3C 018 BFE3C 019 BFE3C 020 BFE3C 021 BFE3C 022 BFE3C 023 BFE3C 024 BFE3C 025 BFE3C 026 BFE3C 027 BFE3C 028 BFE3C 029 </pre>
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SUBROUTINE FE3C(EQFE3,EQCO1,EQH21,IEOEQ,TCT2,PFH2,PFCH4,PFCO,PFCO2	FE3C 001
1,DA,TEMP)	FE3C 002
C CONSTRUCTS THE IRON-IRON CARBIDE PHASE BOUNDARY	FE3C 003
EXTERNAL TCT2	FE3C 004
COMMON CO,C1,C2,C3,C4,C5,C6,EPS,IEND,XST	FE3C 005
PI=3.14159	FE3C 006
PRES=1.	FE3C 007
N=0	FE3C 008
Z=PI/6.	FE3C 009
PH2O=0.0	FE3C 010
PCO=0.0	FE3C 011
PCO2=0.0	FE3C 012
ATOMH=2.*PH2O+2.*PFH2+4.*PFCH4	FE3C 013
ATOMC=PCO+PCO2+PFCH4	FE3C 014
ATOMO=0.0	FE3C 015
ATOMT=ATOMC+ATOMH+ATOMO	FE3C 016
ULH=10.-(ATOMH/(ATOMT*.1))	FE3C 017
ULO=ATOMO/(ATOMT*.1)	FE3C 018
ULC=ATOMC/(ATOMT*.1)	FE3C 019
X1=ULH-(ULH-ULO)*SIN(Z)	FE3C 020
Y1=(ULH-ULO)*COS(Z)	FE3C 021
XST=PFH2*PRES	FE3C 022
PTEST=PFH2	FE3C 023
IEND=1000	FE3C 024
EPS=.001	FE3C 025
CALL FPLOT(3,X1,Y1)	FE3C 026
CALL FPLOT(2,X1,Y1)	FE3C 027
ROH=.01	FE3C 028
84 D=2.*ROH	FE3C 029
CALL DA(PRES,D,EQFE3,EQCO1,EQH21,A,B,C,E,AA,BB,CC,DD,EE,G,Z,	FE3C 030
1W,T,X,TL,FF,GG,HH)	FE3C 031
CO=T*HH+TL*EE	FE3C 032
C1=HH*X+T*GG+TL*DD-A*EE	FE3C 033
C2=HH*W+GG*X+FF*T+TL*CC-A*DD	FE3C 034
C3=HH*Z+W*GG+X*FF+TL*BB-A*CC	FE3C 035

C4=GG*Z+W*FF+TL*AA-A*BB	FE3C 036
C5=FF*Z-A*AA	FE3C 037
C6=0.0	FE3C 038
45 CALL IEQEQ (X,F,DERF,TCT2,XST,EPS,IEND,IER)	FE3C 039
IF(IER-1)42,43,44	FE3C 040
43 IEND=10000	FE3C 041
EPS=.001	FE3C 042
GO TO 45	FE3C 043
44 XST=XST+.01	FE3C 044
GO TO 45	FE3C 045
42 PH2=X	FE3C 046
PCH4=(PH2**2.)/EQFE3	FE3C 047
S=AA*(PH2**4.)+BB*(PH2**3.)+CC*(PH2**2.)+DD*(PH2)+EE	FE3C 048
Y=FF*(PH2**2.)+GG*(PH2)+HH	FE3C 049
PCO2=S/Y	FE3C 050
PH20=(PRES/D)-A*PH2-(B/EQFE3)*(PH2**2.)+(1./D)*(S/Y)	FE3C 051
PCO=PH20/(G*PH2)	FE3C 052
PRES= PH2+PH20+PCO+PCO2+PCH4	FE3C 053
RCH=(PCO+PCO2+PCH4)/(2.*(PH2+PH20)+4.*PCH4)	FE3C 054
CRHM=PH2/PCH4	FE3C 055
CRCC2=PCO/PCO2	FE3C 056
CRHW=PH2/PH20	FE3C 057
CALL DATSW(8,J)	FE3C 058
IF(J-1)22,98,99	FE3C 059
22 WRITE(3,269)	FE3C 060
269 FORMAT('TROUBLE IN DATSW')	FE3C 061
GO TO 99	FE3C 062
98 IF(N)695,695,694	FE3C 063
695 N=N+1	FE3C 064
WRITE(3,691)	FE3C 065
691 FORMAT(///,44X,'IRON=IRON (ARBIDE-GAS EQUILIBRIUM',/)	FE3C 066
WRITE(3,692) TEMP,PRES	FE3C 067
692 FORMAT(38X,'TEMPERATURE=',F5.1,'K',3X,'TOTAL PRESSURE=',F2.0,'ATM'	FE3C 068
1)	FE3C 069
694 WRITE(3,97)	FE3C 070

97	FORMAT(5X,'PH2',8X,'PH2O',7X,'PCO',8X,'PCO2',7X,'PCH4',4X,'PRESSUR	FE3C 071
	1E',5X,'O/H',9X,'C/H',4X,'PH2/PH2O',4X,'PCO/PCO2',4X,'PH2/PCH4')	FE3C 072
	WRITE(3,100)PH2,PH2O,PCO,PCO2,PCH4,PRES,ROH,RCH,CRHW,CRCC2,CRHM	FE3C 073
100	FORMAT(1X,2F10.6,1X,F10.6,;X,F10.6,F10.6,1X,F10.6,1X,F10.6,1X,F10.	FE3C 074
	16,F6.2,6X,F6.2,6X,F6.2,//)	FE3C 075
99	CONTINUE	FE3C 076
	IF(ABS(PRES-1.)-.01)82,82,(1	FE3C 077
81	ROH=ROH+.01	FE3C 078
	PRES=1.	FE3C 079
	GO TO 84	FE3C 080
82	IF(PH2=PTEST)87,81,81	FE3C 081
87	ROH=ROH+.05	FE3C 082
	ATOMH=2.*PH2O+2.*PH2+4.*PCH4	FE3C 083
	Z=PI/6.	FE3C 084
	ATOMC=PCO+PCO2+PCH4	FE3C 085
	ATOMO=PCO+2.*PCO2+PH2O	FE3C 086
	ATOMT=ATOMH+ATOMO+ATOMC	FE3C 087
	ULH=10.-(ATOMH/(ATOMT*.1))	FE3C 088
	ULO=ATOMO/(ATOMT*.1)	FE3C 089
	ULC=ATOMC/(ATOMT*.1)	FE3C 090
	X2=ULH-(ULH-ULO)*SIN(Z)	FE3C 091
	Y2=(ULH-ULO)*COS(Z)	FE3C 092
	CALL FPLLOT(2,X2,Y2)	FE3C 093
	PTEST=PH2	FE3C 094
	PRES=1.	FE3C 095
	XST=PH2*PRES	FE3C 096
	IF(ROH= 6.)85,86,86	FE3C 097
85	GO TO 84	FE3C 098
86	CONTINUE	FE3C 099
	PCH4=0.0	FE3C 100
	PH2=0.0	FE3C 101
	PH2O=0.0	FE3C 102
	ATOMH=0.0	FE3C 103
	ATOMC=PFCO+PFCO2+PCH4	FE3C 104
	ATOMO=PFCO+2.*PFCO2+PH2O	FE3C 105

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ATOMT=ATOMC+ATOMH+ATOMO
ULH=10.-(ATOMH/(ATOMT*.1))
ULO=ATOMO/(ATOMT*.1)
ULC=ATOMC/(ATOMT*.1)
Y3=(ULH-ULO)*COS(Z)
X3=ULH-(ULH-ULO)*SIN(Z)
CALL FPLOTT(0,X3,Y3)
CALL FPLOTT(3,.0,.0)
RETURN
END
```

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FE3C 106
FE3C 107
FE3C 108
FE3C 109
FE3C 110
FE3C 111
FE3C 112
FE3C 113
FE3C 114
FE3C 115
```

SUBROUTINE BIHO(PRES,EQH2,PIH2,PIH2O)	BIHO 001
C CALCULATES THE BOUNDARY CONDITIONS FOR THE HYDROGEN-OXYGEN SYSTEM FOR	BIHO 002
C IRON	BIHO 003
C IF TEMP. LESS THAN 860 K $3/4\text{FE}+\text{H}_2\text{O}=\text{1/4FE}_3\text{O}_4+\text{H}_2$	BIHO 004
C IF TEMP. GREATER THAN OR = TO 860 K $\text{FE}+\text{H}_2\text{O}=\text{FEO}+\text{H}_2$	BIHO 005
PH2O=PRES/(EQH2+1.)	BIHO 006
IF(PH2O)400,401,401	BIHO 007
401 PH2=PRES=PH2O	BIHO 008
IF(PH2)400,402,402	BIHO 009
400 WRITE(3,403)	BIHO 010
403 FORMAT(1X,'ERROR PRES OR EQH2 IN SUBROUTINE BIHO')	BIHO 011
402 PIH2O=PH2O	BIHO 012
PIH2=PH2	BIHO 013
RETURN	BIHO 014
END	BIHO 015

SUBROUTINE BICO(PRES,EQCO,PICO,PICO2)	BICO 001
C CALCULATES THE BOUNDARY CONDITIONS FOR THE CARBON-OXYGEN SYSTEM FOR	BICO 002
C IRON	BICO 003
C IF TEMP. IS GREATER THAN 860 K $3/4\text{FE}+\text{CO}_2=1/4\text{FE}_3\text{O}_4+\text{CO}$	BICO 004
C IF TEMP IS LESS THAN OR EQUAL TO 860 K $\text{FE}+\text{CO}_2=\text{FEO}+\text{CO}$	BICO 005
PCO2=PRES/(EQCO+1.)	BICO 006
IF(PCO2)500,501,501	BICO 007
501 PCO= PRES-PCO2	BICO 008
IF(PCO)500,502,502	BICO 009
500 WRITE(3,503)	BICO 010
503 FORMAT(1X,'ERROR IN PRES OR EQCO IN SUBROUTINE BICO')	BICO 011
502 PICO=PCO	BICO 012
PICO2=PCO2	BICO 013
RETURN	BICO 014
END	BICO 015

SUBROUTINE IRON(EQH2,EQH20,EQCO,IEOEQ,TCT2,PIH2,PIH20,PICO,PICO2,T	IRON 001
TEMP)	IRON 002
C CONSTRUCTS THE IRON-IRON OXIDE PHASE BOUNDARY	IRON 003
EXTERNAL TCT2	IRON 004
COMMON CO,C1,C2,C3,C4,C5,C6,EPS,IEND,XST	IRON 005
PI=3.14159	IRON 006
N=0	IRON 007
PTEST=.0	IRON 008
Z=PI/6.	IRON 009
PRES=1.	IRON 010
IEND=10000	IRON 011
EPS=.001	IRON 012
RO=.001	IRON 013
PCH4=0.0	IRON 014
PCO=0.0	IRON 015
PCO2=0.0	IRON 016
ATOMH=2.*PIH20+2.*PIH2+4.*PCH4	IRON 017
ATOMO=PCO+2.*PCO2+PIH20	IRON 018
ATOMC=PCO+PCO2+PCH4	IRON 019
ATOMT=ATOMH+ATOMO+ATOMC	IRON 020
ULH=10.-(ATOMH/(ATOMT*.1))	IRON 021
ULO=ATOMO/(ATOMT*.1)	IRON 022
ULC=ATOMC/(ATOMT*.1)	IRON 023
X3=ULH-(ULH-ULO)*SIN(Z)	IRON 024
Y3=(ULH-ULO)*COS(Z)	IRON 025
CALL FPLOTT(-2,X3,Y3)	IRON 026
XST=PIH2*PRES	IRON 027
93 CO=-EQH2*PRES*(EQCO+1.)	IRON 028
C1=(2.*RO+1.)*(EQH2+1.)*(EQCO+1.)	IRON 029
C2=(4.*RO-1.)*EQH2*EQCO*EQH20*(EQH2*PRES)	IRON 030
C3=(1.-2.*RO)*(EQH2+1.)*EQH2*EQCO*EQH20	IRON 031
C4=0.0	IRON 032
C5=0.0	IRON 033
C6=0.0	IRON 034
CALL IEOEQ (X,F,DERF,TCT2,XST,EPS,IEND,IER)	IRON 035

PH2=X	IRON 036
PH2O=PH2/EQH2	IRON 037
ABC=(2.*RO*(PH2O+PH2))	IRON 038
DEF=(1./(2.*RO)-1.)	IRON 039
GHI=(PRES*(4.*RO-1.))	IRON 040
PQR=(4.*RO*(1./(EQCO)+1.))	IRON 041
PCO=(ABC*DEF+GHI)/PQR	IRON 042
PCO2=PCO/EQCO	IRON 043
PCH4=EQH2O*PCO*(PH2**3.)/(PH2O)	IRON 044
P=PH2+PH2O+PCO+PCO2+PCH4	IRON 045
ROH=(PCO+PH2O+2.*PCO2)/(2.*(PH2+PH2O)+4.*PCH4)	IRON 046
CRHM=PH2/PCH4	IRON 047
CRCC2=PCO/PCO2	IRON 048
CRHW=PH2/PH2O	IRON 049
CALL DATSW(8,J)	IRON 050
IF(J=1)22,98,99	IRON 051
22 WRITE(3,269)	IRON 052
269 FORMAT('TROUBLE IN DATSW')	IRON 053
GO TO 99	IRON 054
98 IF(N)695,695,694	IRON 055
695 N=N+1	IRON 056
WRITE(3,691)	IRON 057
691 FORMAT(///,45X,'IRON-IRON OXIDE-GAS EQUILIBRIUM',/)	IRON 058
WRITE(3,692) TEMP,PRES	IRON 059
692 FORMAT(38X,'TEMPERATURE=',F6.1,'K',3X,'TOTAL PRESSURE=',F2.0,'ATM'	IRON 060
1)	IRON 061
694 WRITE(3,97)	IRON 062
97 FORMAT(5X,'PH2',8X,'PH2O',7X,'PCO',8X,'PCO2',7X,'PCH4',4X,'PRESSUR	IRON 063
1E',5X,'O/H',9X,'C/H',4X,'P2/PH2O',4X,'PCO/PCO2',4X,'PH2/PCH4')	IRON 064
WRITE(3,10)PH2,PH2O,PCO,PCO2,PCH4,P,ROH,RO,CRHW,CRCC2,CRHM	IRON 065
10 FORMAT(1X,2F10.6,1X,F10.6,2X,F10.6,F10.6,1X,F10.6,1X,F10.6,1X,F10.	IRON 066
16,F6.2,6X,F6.2,6X,F6.2,/))	IRON 067
99 CONTINUE	IRON 068
IF (ABS(P-1.)-.01)92, 92, 91	IRON 069
91 RO=RO+.01	IRON 070

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GO TO 93
92 ATOMH=2.*PH20+2.*PH2+4.*PC1 4
   ATOMO=PCO+2.*PCO2+PH20
   ATOMC=PCO+PCO2+PCH4
   ATOMT=ATOMH+ATOMO+ATOMC
   ULH=10.-(ATOMH/(ATOMT*.1))
   ULO=ATOMO/(ATOMT*.1)
   ULC=ATOMC/(ATOMT*.1)
   Y2=(ULH-ULO)*COS(Z)
   X2=ULH-(ULH-ULO)*SIN(Z)
   CALL FPLOTT(2,X2,Y2)
   PTEST=PH2
   PRES=1.
   XST=PH2*PRES
   IF(RO= 6.)94,95,95
94 RO=RO+.05
   GO TO 93
95 CONTINUE
   PH2=0.0
   PH20=0.0
   PCH4=0.0
   ATOMH=2.*PH20+2.*PH2+4.*PCH4
   ATOMC=PICO+PICO2+PCH4
   ATOMO=PICO+2.*PICO2+PH20
   ATOMT=ATOMH+ATOMO+ATOMC
   ULH=10.-(ATOMH/(ATOMT*.1))
   ULO=ATOMO/(ATOMT*.1)
   ULC=ATOMC/(ATOMT*.1)
   Y1=(ULH-ULO)*COS(Z)
   X1=ULH-(ULH-ULO)*SIN(Z)
   CALL FPLOTT(0,X1,Y1)
   CALL FPLOTT(3,0.0,0.0)
RETURN
END

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IRON 071
IRON 072
IRON 073
IRON 074
IRON 075
IRON 076
IRON 077
IRON 078
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IRON 092
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IRON 094
IRON 095
IRON 096
IRON 097
IRON 098
IRON 099
IRON 100
IRON 101
IRON 102
IRON 103
IRON 104

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SUBROUTINE BGCO(PRES,EQC,PGCO,PGCO2)	BGCO 001
C CALCULATES THE BOUNDARY CONDITIONS FOR THE CARBON-OXYGEN SYSTEM FOR *G	BGCO 002
C RAPH*	BGCO 003
PCO=(-1.+SQRT(1.+4.*EQC*PRES))/(2.*EQC)	BGCO 004
IF(PCO)300,301,301	BGCO 005
301 PCO2=PRES-PCO	BGCO 006
IF(PCO2)300,303,303	BGCO 007
300 PCO=(-1.-SQRT(1.+4.*EQC*PRES))/(2.*EQC)	BGCO 008
IF(PCO)304,305,305	BGCO 009
305 PCO2=PRES-PCO	BGCO 010
IF(PCO2)304,303,303	BGCO 011
304 WRITE(3,306)	BGCO 012
306 FORMAT(1X,'WE HAVE A PROBLIM IN SUBROUTINE BGCO')	BGCO 013
303 PGCO=PCO	BGCO 014
PGCO2=PCO2	BGCO 015
RETURN	BGCO 016
END	BGCO 017

SUBROUTINE BHC (PRES, EQCH4, PGH2, PGCH4)	BHC 001
C CALCULATES THE BOUNDARY CONDITIONS FOR THE CARBON-HYDROGEN SYSTEM FOR	BHC 002
C GRAPH	BHC 003
PH2=(-1.+SQRT(1.+4.*EQCH4*PRES))/(2.*EQCH4)	BHC 004
IF(PH2)200,201,201	BHC 005
201 PCH4=PRES-PH2	BHC 006
IF(PCH4)200,203,203	BHC 007
200 PH2=(-1.-SQRT(1.+4.*EQCH4*PRES))/(2.*EQCH4)	BHC 008
IF(PH2)204,205,205	BHC 009
205 PCH4=PRES-PH2	BHC 010
IF(PCH4)204,203,203	BHC 011
204 WRITE(3,206)	BHC 012
206 FORMAT(1X,'WE HAVE A PROBLEM IN SUBROUTINE BHC')	BHC 013
203 PGH2=PH2	BHC 014
PGCH4=PCH4	BHC 015
RETURN	BHC 016
END	BHC 017

SUBROUTINE GRAPH(EQCO2,EQH2O,EQCH4,IEOEQ,TCT2,PGH2,PGCH4,PGCO,PGCO	GRAPH 001
12,TEMP)	GRAPH 002
C CONSTRUCTS THE GRAPHITE-GAS PHASE BOUNDARY	GRAPH 003
EXTERNAL TCT2	GRAPH 004
COMMON CO,C1,C2,C3,C4,C5,C6,EPS,IEND,XST	GRAPH 005
PI=3.14159	GRAPH 006
PRES=1.	GRAPH 007
N=0	GRAPH 008
ROH=.01	GRAPH 009
Z=PI/6.	GRAPH 010
PH2O=0.0	GRAPH 011
PCO=0.0	GRAPH 012
PCO2=0.0	GRAPH 013
ATOMH=2.*PH2O+2.*PGH2+4.*PGCH4	GRAPH 014
ATOMC=PCO+PCO2+PGCH4	GRAPH 015
ATOMO=0.0	GRAPH 016
ATOMT=ATOMC+ATOMH+ATOMO	GRAPH 017
ULH=10.-(ATOMH/(ATOMT*.1))	GRAPH 018
ULO=ATOMO/(ATOMT*.1)	GRAPH 019
ULC=ATOMC/(ATOMT*.1)	GRAPH 020
X1=ULH-(ULH-ULO)*SIN(Z)	GRAPH 021
Y1=(ULH-ULO)*COS(Z)	GRAPH 022
XST=PGH2*PRES	GRAPH 023
PTEST=PGH2	GRAPH 024
IEND=1000	GRAPH 025
EPS=.001	GRAPH 026
CALL FPLOT(3,X1,Y1)	GRAPH 027
CALL FPLOT(2,X1,Y1)	GRAPH 028
84 A=(-PRES*EQCH4)	GRAPH 029
B=2.*ROH	GRAPH 030
CC=1.-2.*ROH	GRAPH 031
D=1.+2.*ROH	GRAPH 032
AA=(EQCH4*D-PRES*EQH2O*CC)	GRAPH 033
AB=(EQCH4**2.)*(1.+4.*ROH)+EQH2O	GRAPH 034
AC=EQCH4*EQH2O*D	GRAPH 035

AD=(-EQCO2*(EQCH4**2.))/(4.*EQH20)	GRAPH 036
AE=(((-EQCH4*EQCO2)/4.)*(3.-2.*ROH)-2.*EQCH4*ROH)	GRAPH 037
AF=(((-EQCO2*EQH20/4.)*CC*(3.+2.*ROH)-4.*ROH*(EQCH4**2.)-2.*EQH20*1ROH*D)	GRAPH 038
AG=(((-EQCO2/(4.*EQCH4))*(EQH20**2.)*(CC**2.)*D)-4.*EQCH4*EQH20*1ROH*D)	GRAPH 039
AG=(((-EQCO2/(4.*EQCH4))*(EQH20**2.)*(CC**2.)*D)-4.*EQCH4*EQH20*1ROH*D)	GRAPH 040
CO=(A**2.)+AD*A	GRAPH 041
C1=2.*A*AA+AD*AA+A*AE	GRAPH 042
C2=2.*A*AB+(AA**2.)+AD*AB+A*AE+A*AF+(EQCH4**2.)*(ROH**2.)	GRAPH 043
C3=2.*A*AC+2.*AA*AB+AD*AC+3*AE+AF*AA+A*AG+2.*(ROH**2.)*EQCH4*1EQH20*D+4.*(ROH**2.)*(EQCH4**3.)	GRAPH 044
C4=2.*AA*AC+(AB**2.)+AC*AE+AF*AB+AG*AA+(EQH20**2.)*(ROH**2.)*1(D**2.)+4.*(EQCH4**4.)*(ROH**2.)+8.*(ROH**2.)*(EQCH4**2.)*(EQH201)*D	GRAPH 045
C4=2.*AA*AC+(AB**2.)+AC*AE+AF*AB+AG*AA+(EQH20**2.)*(ROH**2.)*1(D**2.)+4.*(EQCH4**4.)*(ROH**2.)+8.*(ROH**2.)*(EQCH4**2.)*(EQH201)*D	GRAPH 046
C5=2.*AB*AC+AF*AC+AB*AG+8.*(EQCH4**3.)*(EQH20)*(ROH**2.)*D1+4.*(ROH**2.)*EQCH4*(EQH20**2.)*(D**2.)	GRAPH 047
C5=2.*AB*AC+AF*AC+AB*AG+8.*(EQCH4**3.)*(EQH20)*(ROH**2.)*D1+4.*(ROH**2.)*EQCH4*(EQH20**2.)*(D**2.)	GRAPH 048
C6=AC**2.+AG*AC+4.*(ROH**2.)*(EQCH4**2.)*(EQH20**2.)*(D**2.)	GRAPH 049
45 CALL IESEQ (X,F,DERF,TCT2,XST,EPS,IEND,IER)	GRAPH 050
IF(IER=1)42,43,44	GRAPH 051
43 IEND=10000	GRAPH 052
EPS=.001	GRAPH 053
GO TO 45	GRAPH 054
44 XST=XST+.01	GRAPH 055
GO TO 45	GRAPH 056
42 PH2=X	GRAPH 057
PCH4=(PH2**2.)*EQCH4	GRAPH 058
EQ1=EQH20*(PH2**3.)/(PCH4)	GRAPH 059
PH20=(2.-2.*(PH2+EQCH4*(PH2**2.))-2.*ROH*(PH2+2.*EQCH4*(PH2**2.1)))/(2.*ROH+(1./EQ1)+1.)	GRAPH 060
PH20=(2.-2.*(PH2+EQCH4*(PH2**2.))-2.*ROH*(PH2+2.*EQCH4*(PH2**2.1)))/(2.*ROH+(1./EQ1)+1.)	GRAPH 061
PCO=PCH4*PH20/(EQH20*(PH2**3.))	GRAPH 062
PCO2=PH20*PCO/(PH2*EQCO2)	GRAPH 063
PRES=PH2+PH20+PCO+PCO2+PCH4	GRAPH 064
RCH=(PCO+PCO2+PCH4)/(2.*(PH2+PH20)+4.*PCH4)	GRAPH 065
CRHM=PH2/PCH4	GRAPH 066
CRCC2=PCO/PCO2	GRAPH 067
	GRAPH 070

CRHW=PH2/PH2O	GRAPH 071
CALL DATSW(8,J)	GRAPH 072
IF(J-1)22,98,99	GRAPH 073
22 WRITE(3,269)	GRAPH 074
269 FORMAT('TROUBLE IN DATSW')	GRAPH 075
GO TO 99	GRAPH 076
98 IF(N)695,695,694	GRAPH 077
695 N=N+1	GRAPH 078
WRITE(3,691)	GRAPH 079
691 FORMAT(///, 48X,'GRAPHITE-GAS EQUILIBRIUM',/)	GRAPH 080
WRITE(3,692) TEMP,PRES	GRAPH 081
692 FORMAT(38X,'TEMPERATURE=',F6.1,'K',3X,'TOTAL PRESSURE=',F2.0,'ATM'	GRAPH 082
1)	GRAPH 083
694 WRITE(3,97)	GRAPH 084
97 FORMAT(5X,'PH2',8X,'PH2O',7X,'PCO',8X,'PCO2',7X,'PCH4',4X,'PRESSUR	GRAPH 085
1E',5X,'O/H',9X,'C/H',4X,'PH2/PH2O',4X,'PCO/PCO2',4X,'PH2/PCH4')	GRAPH 086
WRITE(3,100)PH2,PH2O,PCO,PCO2,PCH4,PRES,ROH,RCH,CRHW,CRCC2,CRHM	GRAPH 087
100 FORMAT(1X,2F10.6,1X,F10.6,2X,F10.6,F10.6,1X,F10.6,1X,F10.6,1X,F10.	GRAPH 088
16,F6.2,6X,F6.2,6X,F6.2,///)	GRAPH 089
99 CONTINUE	GRAPH 090
IF(ABS(PRES-1.)-.01)82,82,P1	GRAPH 091
81 ROH=POH+.01	GRAPH 092
PRES=1.	GRAPH 093
GO TO 84	GRAPH 094
82 IF(PH2=PTEST)87,81,81	GRAPH 095
87 ROH=ROH+.05	GRAPH 096
ATOMH=2.*PH2O+2.*PH2+4.*PCH4	GRAPH 097
ATOMC=PCO+PCO2+PCH4	GRAPH 098
ATOMO=PCO+2.*PCO2+PH2O	GRAPH 099
ATOMT=ATOMH+ATOMO+ATOMC	GRAPH 100
ULH=10.-(ATOMH/(ATOMT*.1))	GRAPH 101
ULO=ATOMO/(ATOMT*.1)	GRAPH 102
ULC=ATOMC/(ATOMT*.1)	GRAPH 103
X2=ULH-(ULH-ULO)*SIN(Z)	GRAPH 104
Y2=(ULH-ULO)*COS(Z)	GRAPH 105


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CALL FPLOT(2,X2,Y2)
PTEST=PH2
PRES=1.
XST=PH2*PRES
IF(ROH= 6.)85,86,86
85 GO TO 84
86 CONTINUE
PCH4=0.0
PH2=0.0
PH2O=0.0
ATOMH=0.0
ATOMC=PGCO+PGCO2+PCH4
ATOMO=PGCO+2.*PGCO2+PH2O
ATOMT=ATOMC+ATOMH+ATOMO
ULH=10.-(ATOMH/(ATOMT*.1))
ULO=ATOMO/(ATOMT*.1)
ULC=ATOMC/(ATOMT*.1)
Y3=(ULH-ULO)*COS(Z)
X3=ULH-(ULH-ULO)*SIN(Z)
CALL FPLOT(0,X3,Y3)
CALL FPLOT(3,0.0,0.0)
RETURN
END

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GRAPH 106
GRAPH 107
GRAPH 108
GRAPH 109
GRAPH 110
GRAPH 111
GRAPH 112
GRAPH 113
GRAPH 114
GRAPH 115
GRAPH 116
GRAPH 117
GRAPH 118
GRAPH 119
GRAPH 120
GRAPH 121
GRAPH 122
GRAPH 123
GRAPH 124
GRAPH 125
GRAPH 126
GRAPH 127
GRAPH 128

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	SUBROUTINE IE0EQ (X,F,DERF,TCT2,XST,EPS,IEND,IER)	IE0EQ 001
C	NEWTON-RAPHSON ITERATIVE SEARCH TECHNIQUE FOR PH2 (X)	IE0EQ 002
C	PREPARE ITERATION	IE0EQ 003
	IER=0	IE0EQ 004
	X=XST	IE0EQ 005
	TOL=X	IE0EQ 006
	CALL TCT2 (TOL,F,DERF)	IE0EQ 007
	TOLF=100.*EPS	IE0EQ 008
C	START ITERATION LOOP	IE0EQ 009
	DO 6 I=1,IEND	IE0EQ 010
	IF(F)1,7,1	IE0EQ 011
C	EQUATION IS NOT SATISFIED BY X	IE0EQ 012
1	IF(DERF)2,8,2	IE0EQ 013
C	ITERATION IS POSSIBLE	IE0EQ 014
2	DX=F/DERF	IE0EQ 015
	X=X-DX	IE0EQ 016
	TOL=X	IE0EQ 017
	CALL TCT2 (TOL,F,DERF)	IE0EQ 018
C	TEST OF SATISFACTORY ACCURACY	IE0EQ 019
	TOL=EPS	IE0EQ 020
	A=ABS(X)	IE0EQ 021
	IF(A-1.)4,4,3	IE0EQ 022
3	TOL=TOL*A	IE0EQ 023
4	IF(ABS(DX)-TOL)5,5,6	IE0EQ 024
5	IF(ABS(F)-TOLF)7,7,6	IE0EQ 025
6	CONTINUE	IE0EQ 026
C	END OF ITERATION LOOP	IE0EQ 027
C	NO CONVERGENCE AFTER IEND ITERATION STEPS.ERROR RETURN.	IE0EQ 028
	IER=1	IE0EQ 029
7	RETURN	IE0EQ 030
C	ERROR RETURN IN CASE OF ZERO DIVISOR	IE0EQ 031
8	IER=2	IE0EQ 032
	RETURN	IE0EQ 033
	END	IE0EQ 034

SUBROUTINE TCT2 (X,F,DERF)	TCT2 001
C PART OF NEWTON-RAPHSON SEARCH ROUTINE	TCT2 002
COMMON CO,C1,C2,C3,C4,C5,C6,EPS,IEND,XST	TCT2 003
PH2=X	TCT2 004
F=CO+C1*PH2+C2*(PH2**2.)+C3*(PH2**3.)+C4*(PH2**4.)+C5*(PH2**5.)+	TCT2 005
1C6*(PH2**6.)	TCT2 006
DERF=C1+2.*C2*PH2+3.*C3*(PH2**2.)+C4*4.*(PH2**3.)+5.*C5*(PH2**4.)	TCT2 007
1+6.*C6*(PH2**5.)	TCT2 008
RETURN	TCT2 009
END	TCT2 010

C	PROGRAM TO CURVE FIT LNK DATA AND G AND H	CURVE 001
	DIMENSION T(100),Y(100),A(5,6),YC(100),B(5)	CURVE 002
1	FORMAT(I2,1X,I1)	CURVE 003
2	FORMAT(F11.0,5F12.4)	CURVE 004
3	FORMAT(1X,'LN KP=',E12.5,'/T ',E12.5,'LN(T) ',E12.5,'T ',E12.5,	CURVE 005
	1'T**2 ',E12.5,'T**3 ')	CURVE 006
4	FORMAT(6X,'T',11X,'YINPUT',8X,'YCALC')	CURVE 007
5	FORMAT(1X,E12.5,2X,E12.5,2X,E12.5)	CURVE 008
6	FORMAT(1H1)	CURVE 009
7	FORMAT(1H0)	CURVE 010
8	FORMAT(1X,'H= ',E12.5,2X,E12.5,'T ',E12.5,'T**2 ',E12.5,'T**3 '	CURVE 011
	1,E12.5,'T**4 ')	CURVE 012
	DO 20 I=1,5	CURVE 013
	DO 10 J=1,6	CURVE 014
10	A(I,J)=0.	CURVE 015
20	CONTINUE	CURVE 016
	READ(2,1) N,ICODE	CURVE 017
24	IF(ICODE)26,26,25	CURVE 018
25	A(1,3)=N	CURVE 019
	A(3,1)=N	CURVE 020
	READ(2,2) (T(I),Y(I),I=1,N)	CURVE 021
	DO 30 I=1,N	CURVE 022
	A(1,2)=A(1,2)+ALOG(T(I))/T(I)	CURVE 023
	A(1,1)=A(1,1)+1./T(I)**2	CURVE 024
	A(2,1)=A(1,2)	CURVE 025
	A(4,1)=A(4,1)+T(I)	CURVE 026
	A(1,4)=A(4,1)	CURVE 027
	A(1,5)=A(1,5)+T(I)**2	CURVE 028
	A(5,1)=A(1,5)	CURVE 029
	A(3,3)=A(1,5)	CURVE 030
	A(1,6)=A(1,6)+Y(I)/T(I)	CURVE 031
	A(2,2)=A(2,2)+(ALOG(T(I)))**2	CURVE 032
	A(2,3)=A(2,3)+T(I)*ALOG(T(I))	CURVE 033
	A(3,2)=A(2,3)	CURVE 034
	A(2,4)=A(2,4)+ALOG(T(I))*T(I)**2	CURVE 035

	A(4,2)=A(2,4)	CURVE 036
	A(5,2)=A(5,2)+ALOG(T(I))*T(I)**3	CURVE 037
	A(2,5)=A(5,2)	CURVE 038
	A(2,6)=Y(I)*ALOG(T(I))+A(2,5)	CURVE 039
	A(3,4)=A(3,4)+T(I)**3	CURVE 040
	A(4,3)=A(3,4)	CURVE 041
	A(3,5)=A(3,5)+T(I)**4	CURVE 042
	A(5,3)=A(3,5)	CURVE 043
	A(3,6)=A(3,6)+Y(I)*T(I)	CURVE 044
	A(4,4)=A(3,5)	CURVE 045
	A(4,5)=A(4,5)+T(I)**5	CURVE 046
	A(5,4)=A(4,5)	CURVE 047
	A(4,6)=A(4,6)+Y(I)*T(I)**2	CURVE 048
	A(5,5)=A(5,5)+T(I)**6	CURVE 049
30	A(5,6)=A(5,6)+Y(I)*T(I)**3	CURVE 050
	GO TO 40	CURVE 051
26	A(1,1)=N	CURVE 052
	READ(2,2) (T(I),Y(I),I=1,N)	CURVE 053
	DO 27 I=1,N	CURVE 054
	A(1,2)=A(1,2)+T(I)	CURVE 055
	A(2,1)=A(1,2)	CURVE 056
	A(1,3)=A(1,3)+T(I)**2	CURVE 057
	A(2,2)=A(1,3)	CURVE 058
	A(3,1)=A(1,3)	CURVE 059
	A(1,4)=A(1,4)+T(I)**3	CURVE 060
	A(2,3)=A(1,4)	CURVE 061
	A(3,2)=A(1,4)	CURVE 062
	A(4,1)=A(1,4)	CURVE 063
	A(1,5)=A(1,5)+T(I)**4	CURVE 064
	A(2,4)=A(1,5)	CURVE 065
	A(4,2)=A(1,5)	CURVE 066
	A(3,3)=A(1,5)	CURVE 067
	A(5,1)=A(1,5)	CURVE 068
	A(1,6)=A(1,6)+Y(I)	CURVE 069
	A(2,5)=A(2,5)+T(I)**5	CURVE 070

	A(3,4)=A(2,5)	CURVE 071
	A(4,3)=A(2,5)	CURVE 072
	A(5,2)=A(2,5)	CURVE 073
	A(2,6)=A(2,6)+Y(I)*T(I)	CURVE 074
	A(3,5)=A(3,5)+T(I)**6	CURVE 075
	A(4,4)=A(3,5)	CURVE 076
	A(5,3)=A(3,5)	CURVE 077
	A(3,6)=A(3,6)+Y(I)*T(I)**2	CURVE 078
	A(4,5)=A(4,5)+T(I)**7	CURVE 079
	A(5,4)=A(4,5)	CURVE 080
	A(4,6)=A(4,6)+Y(I)*T(I)**3	CURVE 081
	A(5,5)=A(5,5)+T(I)**8	CURVE 082
27	A(5,6)=A(5,6)+Y(I)*T(I)**4	CURVE 083
40	DO 50 K=2,6	CURVE 084
50	A(1,K)=A(1,K)/A(1,1)	CURVE 085
	DO 80 IDEX=2,5	CURVE 086
	JM=IDEX-1	CURVE 087
	DO 60 I=IDEX,5	CURVE 088
	SUM=0.	CURVE 089
	DO 55 K=1,JM	CURVE 090
55	SUM=SUM+A(I,K)*A(K,IDEX)	CURVE 091
60	A(I,IDEX)=A(I,IDEX)-SUM	CURVE 092
	JP=IDEX+1	CURVE 093
	IM=IDEX-1	CURVE 094
	DO 70 J=JP,6	CURVE 095
	SUM=0.	CURVE 096
	DO 65 K=1,IM	CURVE 097
65	SUM=SUM+A(IDEX,K)*A(K,J)	CURVE 098
70	A(IDEX,J)=(A(IDEX,J)-SUM)/A(IDEX,IDEX)	CURVE 099
80	CONTINUE	CURVE 100
	B(5)=A(5,6)	CURVE 101
	DO 100 IDEX=2,5	CURVE 102
	I=6-IDEX	CURVE 103
	IP=I+1	CURVE 104
	SUM=0.	CURVE 105

90	DO 90 K=IP,5	CURVE 106
	SUM=SUM+A(I,K)*B(K)	CURVE 107
100	B(I)=A(I,6)-SUM	CURVE 108
	WRITE(3,6)	CURVE 109
	IF(ICODE)101,101,102	CURVE 110
101	WRITE(3,8) (B(I),I=1,5)	CURVE 111
	GO TO 103	CURVE 112
102	WRITE(3,3) (B(I),I=1,5)	CURVE 113
103	WRITE(3,7)	CURVE 114
	IF(ICODE)104,104,106	CURVE 115
104	DO 105 I=1,N	CURVE 116
105	YC(I)=B(1)+B(2)*T(I)+B(3)*T(I)**2+B(4)*T(I)**3+B(5)*T(I)**4	CURVE 117
	GO TO 112	CURVE 118
106	DO 110 I=1,N	CURVE 119
110	YC(I)=B(1)/T(I)+B(2)*ALOG(T(I))+B(3)*T(I)+B(4)*T(I)**2+B(5)*T(I)	CURVE 120
	1**3	CURVE 121
112	WRITE(3,4)	CURVE 122
	WRITE(3,5) (T(I),Y(I),YC(I),I=1,N)	CURVE 123
	READ(2,1) N,ICODE	CURVE 124
	IF(ICODE=1)115,115,120	CURVE 125
115	DO 117 I=1,5	CURVE 126
	DO 116 J=1,6	CURVE 127
116	A(I,J)=0.	CURVE 128
117	CONTINUE	CURVE 129
	GO TO 24	CURVE 130
120	CALL EXIT	CURVE 131
	END	CURVE 132

```

SUBROUTINE EQKS(TEMP,EQCH4,EQCO2,EQH2O,EQH2,EQCO,EQC,EQFE3,EQFE2,E
1QC01,EQH21)
C CALCULATES THE APPROPRIATE LNK. VALUES TO DELIVER TO THE PHASE BOUNDAR
C Y SUBROUTINES
C REACTION 1 C+2H2=CH4 EQCH4
C REACTION 2 H2+CO2=CO+H2O EQCO2
C REACTION 3 3H2+CO=H2O+CH4 EQH2O
C REACTION 4 3/4FE+H2O=H2+1/4FE3O4 EQH2
C REACTION 5 3/4FE+CO2=CO+1/4FE3O4 EQCO
C REACTION 6 2CO=CO2+C EQC
C REACTION 7* 3FE+CH4=2H2+FE3C EQFE3
C REACTION 8* CO2+CH4=2H2+2CO EQCO1
C REACTION 9* 4H2+CO2=CH4+2H2O EQH21
C RANGE OF CORREL. FOR LNK BETWEEN 298-2000 K
EKCH4= (8372.2/(TEMP)-1.0769*ALOG(TEMP)-(.56435E-2)*TEMP+(.2904
16E-5)*(TEMP**2.)-( .52351E-9)*(TEMP**3.))
EKCO= (13612./(TEMP)+1.8317*(ALOG(TEMP))-(2.7584E-3)*TEMP+
2(.6536E-6)*(TEMP**2.)-( .78772E-10)*(TEMP**3.))
EKCO2= (47280./(TEMP)+(.1322)*(ALOG(TEMP))-(.94025E-3)*(TEMP)+
3(.45112E-6)*(TEMP**2.)-( .91901E-10)*(TEMP**3.))
EKH2O= (28780./(TEMP)-(.69477)*(ALOG(TEMP))-(.14283E-2)*(TEMP)+
4(.74925E-6)*(TEMP**2.)-( .13785E-9)*(TEMP**3.))
WRITE(1,83)
83 FORMAT(5X,'IF SPECIES AT EQUILIBRIUM IS THOUGHT TO BE FEO(CRYSTAL)
1 TYPE IN 1., IF FEO.9470(WUSTITE) TYPE IN 2.,',/, 'IF NIO TYPE IN -1
1. ')
READ(6,84) TEST
84 FORMAT(7X,F4.1)
IF(TEST-1.)88,85,86
85 EKFE0= ((32461./TEMP)-(.15184E+1)*(ALOG(TEMP))+(.44208E-2)*(TEMP
5)-(.17970E-5)*(TEMP**2.)+( .28776E-9)*(TEMP**3.))
GO TO 87
86 EKFE0=(.317327E05/(TEMP))- .150692E01*(ALOG(TEMP))+.383287E-2*TEMP-
1.135694E-5*(TEMP**2.)+.175926E-9*(TEMP**3.)
WRITE(3,90) TEST

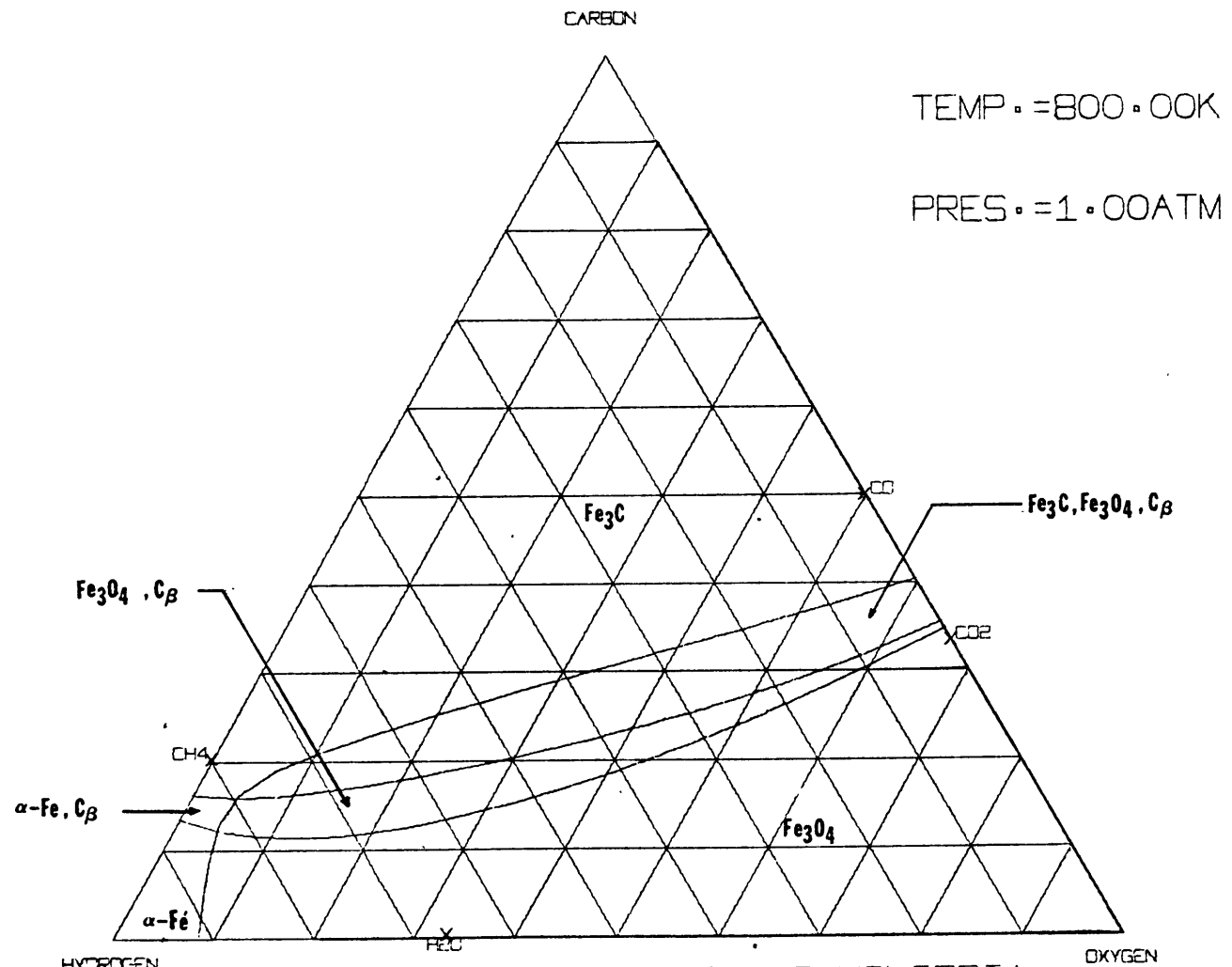
```

```

EQKS 001
EQKS 002
EQKS 003
EQKS 004
EQKS 005
EQKS 006
EQKS 007
EQKS 008
EQKS 009
EQKS 010
EQKS 011
EQKS 012
EQKS 013
EQKS 014
EQKS 015
EQKS 016
EQKS 017
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EQKS 029
EQKS 030
EQKS 031
EQKS 032
EQKS 033
EQKS 034
EQKS 035

```


90	FORMAT(5X,F4.1)	EQKS 036
	GOTO 87	EQKS 037
88	EKFEO=(2.9045E04/TEMP)-2.03977*ALOG(TEMP)+3.7856E-03*TEMP-1.42491E	EQKS 038
	1-06*TEMP**2+2.48586E-10*TEMP**3	EQKS 039
87	EKFE2= ((98359./TEMP)-(.59363E+1))*(ALOG(TEMP))+(.13798E-1)*(TEMP	EQKS 040
	6)-(.39931E-5)*(TEMP**2.)+(.40036E-9)*(TEMP**3.))	EQKS 041
	EKFE3= ((133880./TEMP)-(.77989E+1))*(ALOG(TEMP))+(.20934E-1)*(TEM	EQKS 042
	7P)-(.63194E-5)*(TEMP**2.)+(.63808E-9)*(TEMP**3.))	EQKS 043
	EKF3C=((-.27892E04/TEMP)-(.42754E-01))*(ALOG(TEMP))+(.62806E-02)*(T	EQKS 044
	1EMP)-(.42543E-05)*(TEMP**2.)+(.82417E-09)*(TEMP**3.))	EQKS 045
	EKF2C=((-.26706E04/TEMP)+(.48958)*(ALOG(TEMP)))-(.37848E-02)*(TEMP)	EQKS 046
	1+(.25141E-05)*(TEMP**2.)-(.53351E-09)*(TEMP**3.))	EQKS 047
	EQCH4=EXP(EKCH4)	EQKS 048
	EQCO2=EXP((EKH20+EKCO)-EKC(2))	EQKS 049
	EQH20=EXP((EKH20+EKCH4)-EKCO)	EQKS 050
	EQC=EXP(EKCO2-(2.*EKCO))	EQKS 051
	EQCO1=EXP(2.*EKCO-(EKCH4+EKCO2))	EQKS 052
	EQH21=EXP((2.*EKH20+EKCH4)-(EKCO2))	EQKS 053
	IF(TEST)71,91,91	EQKS 054
91	IF(TEMP=833.)70,71,71	EQKS 055
70	EQH2=EXP(.25*EKFE3-EKH20)	EQKS 056
	EQCO=EXP((.25*EKFE3+EKCO)-EKCO2)	EQKS 057
	GO TO 72	EQKS 058
71	EQH2=EXP(EKFEO-EKH20)	EQKS 059
	EQCO=EXP((EKFE0+EKCO)-(EKCO2))	EQKS 060
72	IF(TEMP=600.)80,80,81	EQKS 061
81	EQFE2=EXP((EKCO2+EKF3C)-2.*(EKCO))	EQKS 062
	EQFE3=EXP(EKF3C-EKCH4)	EQKS 063
	GO TO 82	EQKS 064
80	EQFE2=EXP((EKCO2+EKF2C)-2.*(EKCO))	EQKS 065
	EQFE3=EXP(EKF2C-EKCH4)	EQKS 066
82	CONTINUE	EQKS 067
	RETURN	EQKS 068
	END	EQKS 069



UPPER CURVE IRON-IRON OXIDE-GAS EQUILIBRIA
 MIDDLE CURVE IRON-IRON CARBIDE-GAS EQUILIBRIA
 LOWER CURVE GRAPHITE-GAS EQUILIBRIA

FIGURE 60: TRIANGULAR PHASE DIAGRAM AT 800 K

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7.3 Mass Transfer Limitations in a Five Component Gas Mixture

The following calculations were made to determine if mass transfer limitations were present in Bosch reactor.

Representative Run, A - 25

temperature 900 K, pressure = 1 atmosphere, catalyst charge =
508 mg

<u>COMPONENT</u>	<u>EXPERIMENTAL (%)</u>	<u>THEORETICAL (α-Fe/Fe_{1-y}O, %)</u>
H ₂	0.3729	0.3879
CO	0.2708	0.2175
CO ₂	0.1570	0.1825
CH ₄	0.1010	0.0715
H ₂ O	0.0982	0.1407

Calculating Reynolds Number:

$$N_{Re_d} = d V \rho_{mix} / \mu_{mix} \quad (33)$$

$$V = 20 \text{ cm}^3 / \text{s (STP)} \cdot \frac{900 \text{ K}}{273 \text{ K}} \cdot \frac{4}{\pi (1.9 \text{ cm})^2}$$

$$V = 23.25 \text{ cm/s at 900 K (superficial velocity)}$$

From the ideal-gas law:

$$P V_m = N R_g T$$

$$\frac{M_{wt} N}{V_m} = \rho_{mix} = \frac{M_{wt} P}{R_g T} \quad (34)$$

where

$$M_{wt} = \sum_{i=1}^5 M_i X_i (\text{exp}) \quad (35)$$

$M_{wt} = 18.62 \text{ g/g-mole (molecular weight of reactant gas)}$

$$\rho_{\text{mix}} = \frac{18.62 \text{ g/g-mole} \cdot 1.0 \text{ ATM}}{\frac{82.1 \text{ ATM cm}^3}{\text{g-mole K}} \cdot 900 \text{ K}}$$

$$\rho_{\text{mix}} = 2.52 \times 10^{-4} \text{ g/cm}^3 \quad (\text{gas mixture density})$$

Viscosity of the various gases was obtained at a variety of different temperatures. Equation 36 (Reid and Sherwood, 1950) was used to bring all viscosities to 900 K. The data used and calculated is shown in Table 5.

$$\mu_{900} = \mu_T \left(\frac{T_{900}^*}{T_T^*} \right)^{1/2} \left(\frac{\Omega_{900}}{\Omega_T} \right) \quad (36)$$

Table 5 Viscosity Data

COMPONENT	μ_T (poise) $\times 10^{-4}$	T (K)	$\frac{\epsilon_o}{k}$ ($^{\circ}$ K)	T*	Ω_T
H ₂	1.829	874	59.7	14.64	0.79
CO	2.714	549.9	91.7	5.99	0.90
CO ₂	3.300	763	195.2	3.91	0.97
CH ₄	2.264	772	148.6	5.20	0.93
H ₂ O	1.255	373	809.1	0.46	2.37
<u>Results</u>					
H ₂	1.86	900 K		15.10	0.79
CO	3.76	"		9.80	0.83
CO ₂	3.69	"		4.60	0.94
CH ₄	2.56	"	same	6.10	0.89
H ₂ O	1.19	"		1.11	0.69

Assuming that the mixture viscosity can be reliably approximated by a mole-fraction average:

$$\mu_{\text{mix}} = \sum_{i=1}^5 \mu_i X_i \quad (37)$$

$$\mu_{\text{mix}} = 3.478 \times 10^{-4} \text{ g/cm-s}$$

$$N_{\text{Re}_d} = \frac{0.075 \text{ cm} \cdot 23.25 \text{ cm/s} \cdot 2.52 \times 10^{-4} \text{ g/cm}^3}{3.478 \times 10^{-4} \text{ g/s} - \text{cm}}$$

$$N_{\text{Re}_d} = 1.26$$

Using a correlation for mass transfer to and from tubes in crossflow (Zhukauskas et al., 1968):

$$N_{\text{Sh}} = (0.43 + 0.50 N_{\text{Re}}^{0.5}) \text{Sc}^{0.38} \left(\frac{\text{Sc}_f}{\text{Sc}_w} \right)^{0.25} \quad (38)$$

$$k_{m,j} = \frac{D}{d} \cdot (0.43 + 0.50 N_{\text{Re}}^{0.5}) \text{Sc}^{0.38}$$

Equation 39 was used to calculate the various binary diffusion coefficients. Ω_D (collision integral) and σ (Lennard-Jones force constant) were obtained from equation 40 and 41 in conjunction with Appendix G and Table 11.1 from Reid-Sherwood.

$$D_{AB} = 0.001858 T^{3/2} \left(\frac{M_A + M_B}{M_A M_B} \right)^{1/2} / P \sigma_{AB}^2 \Omega_D \quad (39)$$

$$\epsilon_{\circ AB}/k \left(\left\{ \epsilon_{\circ A}/k \right\} \left\{ \epsilon_{\circ B}/k \right\} \right)^{1/2} \quad (40)$$

$$\sigma_{AB} = 1/2 (\sigma_A + \sigma_B) \quad (41)$$

Table 6 Binary Diffusion Coefficients

	$\underline{D_{H_2}}$ (cm^2/s)	$\underline{D_{CO}}$	$\underline{D_{CO_2}}$	$\underline{D_{CH_4}}$
CO	4.65	-	-	-
CO ₂	4.13	1.04	-	-
CH ₄	4.65	1.45	1.17	-
H ₂ O	5.75	1.18	1.00	1.686

As an approximation the diffusion coefficients of any of the reactant gases into the mixture were represented by the equation

$$D_{Am} = \frac{(1-X_A)}{\frac{N}{\sum_{j=B} (X_j/D_{Aj})}} \quad (42)$$

Using equation 42 and equation 38 along with the data from Table 6, the diffusion coefficients and mass transfer coefficients were calculated. The results are shown in Table 7.

Table 7 Diffusion Coefficients and Mass Transfer Coefficients in a Five Component Gas Mixture

<u>Component</u>	$\underline{D_{i-m}}$ (cm^2/s)	$\underline{k_m}$ (cm/s)
H ₂	4.64	38.67
CO	1.89	21.3
CO ₂	1.58	18.27
CH ₄	1.96	22.67
H ₂ O	1.77	21.28

From run A-25, the maximum rate of carbon deposition was 1.57×10^{-6} g-moles carbon/s, assuming carbon is formed from reaction D or reaction E



$$N = k_{\text{CO}} A \Delta C$$

where

$$C = \frac{N}{V} = \frac{P}{R_g T}$$

The area of a typical preconditioned catalyst is $1912 \text{ cm}^2/\text{gm}$. Therefore, assuming mass transfer limitations do exist, (i.e., the surface concentration equals that of equilibrium)

$$N_{\text{CO}} = (21.30 \text{ cm/s}) \left(\frac{1912 \text{ cm}^2}{\text{gm}} \right) (0.508 \text{ gm}) \left(\frac{1}{\frac{82.1 \text{ atm cm}^3}{\text{g-mole K}} \cdot 900 \text{ K}} \right) \cdot (.2708 - .2175 \text{ atm})$$

$$N_{\text{CO}} = 1.49 \times 10^{-2} \text{ g-moles carbon/s}$$

Mass transfer is seen not to be limiting in the case of carbon monoxide reacting to carbon.

It should be noted that because of the complexities of the Bosch reaction sequence and also the low conversion obtained, it was impossible to calculate whether product concentration was limiting or not. This is due to the fact that the rate of product formation can not be accurately calculated and thus compared to the mass transfer rate.

7.4 Computer Programs Used in Data Analysis

7.4.1 Data Reduction Logic (Computer Analysis)

The inlet and outlet gas phase compositions were determined by a modified computer program originally developed by Manning (1976). Calculations of all gas component concentrations except hydrogen were determined by using the absolute method described by Dal Nogare and Juvet (1962) in conjunction with a precision made external standard. Hydrogen was determined by using an empirical calibration curve as recommended by Purcell and Ettore (1965). A detailed description of the calculation scheme is described by Manning (1976).

The program was modified to increase accuracy and to provide a provision for drawing a triangular phase diagram. The measured inlet and outlet gas compositions were simultaneously plotted on this diagram. The diagram was used to provide a visible check to insure that the solid phase being investigated was the phase of interest.

The main program "Terri" and the associated support programs are shown at the end of this section, including a sample of the output typically obtained. Volume II of this thesis provides a history of all the data obtained both in raw and reduced form.

7.4.2 Propagation of Error Analysis

A propagation of error analysis was performed on the data reduction procedure. This maximum possible error (propagation

error) in each individual gas composition was determined using program "Error". Program Error, along with a sample of its output is presented at the end of this section.

7.4.3 Initial Conditions

As mentioned in Section 3.2.1, the starting conditions were obtained by solving the appropriate equilibrium relationships assuming that all solid phases acted as catalysts for all the reaction systems involved. Then, by setting temperature, pressure, O/H or C/H ratio, and the P_{H_2}/P_{H_2O} ratio, the composition of the gas phase could be made to fix the system in the phase field of interest. This was done by Program "Set", which follows Program "Error".

7.4.4 Optimal Reactor Design

In order to determine the optimal reactor design, it was first necessary to determine the intersection of the iron-iron oxide and graphite-gas phase boundaries (i.e., optimal O/H ratio). This was accomplished by a Newton-Raphson search technique. The Program called "Inter" is presented on conclusion of this section.

Program "Proc" used the results from Program Inter to determine the optimal reactor design. In Program Process the operator specifies Bosch temperature and pressure, shift O/H ratio, total carbon dioxide into the system, and the hydrogen and carbon dioxide directly fed to the shift reactor. Through messages sent out to the key board, the operator can specify

shift recycle and temperature. Program Proc follows Program Inter.

C PROGRAM TERRI USED TO PLOT DATA ON PHASE DIAGRAMS	TERRI 001
EXTERNAL TABLE,DPLOT	TERRI 002
COMMON ITIME(95),DELP(95),IGAS (95),DATA(95,5),CAL(6),	TERRI 003
1CALCO(6),SUMD(6),ISAMP(95),RCC(6),IDATE(5),HSLOP,A,B,CZERO,FEZER,	TERRI 004
2PATM, PH2O,NIT,NSUM1,NSUM2,NSUM3,NSAMP,NRUN	TERRI 005
X=1.	TERRI 006
01 X=X-1.	TERRI 007
CALL ID1	TERRI 008
CALL ID2	TERRI 009
CALL CALC	TERRI 010
CALL DOUT1	TERRI 011
IF(X)91,91,90	TERRI 012
90 GO TO 01	TERRI 013
91 CONTINUE	TERRI 014
CALL EXIT	TERRI 015
END	TERRI 016

SUBROUTINE DA(PRES,D,EQFE3,EQCO1,EQH21,A,B,C,E,AA,BB,CC,DD,EE,G,Z,	DA 001
1W,T,X,TL,FF,GG,HH)	DA 002
C CALCULATES CONSTANTS NEEDED IN PRINT-OUT ROUTINES	DA 003
P=PRES	DA 004
A=(1.+D)/D	DA 005
B=(1.+2.*D)/D	DA 006
C=(2./D)+1.	DA 007
E=(1.-D)/D	DA 008
AA=(1./(EQFE3**2.))*(1.-2.*B-C*((D*B)**2.))	DA 009
BB=(2./(EQFE3))*(1.-A-B-(A*B*C)*(D**2.))	DA 010
CC=(2.*P*(E+B)/EQFE3)+(1.-2.*A)-(C*(D**2.))*((A**2.)-2.*B*P/(EQFE3	DA 011
1*D))	DA 012
DD=2.*P*(E+A*(1.+C*D))	DA 013
EE=2.*(P**2.)*((D-2.)/(2.*D)-C/2.)	DA 014
G=EQFE3*SQRT(EQH21/EQCO1)	DA 015
Z=(1./EQFE3)*(B-1.)	DA 016
W=(EQFE3*G*(A-1.)+B)/(EQFE3*G)	DA 017
T=(-P)/(D*G)	DA 018
X=P*(A/(P*G)-E)	DA 019
TL=-1./(D*G)	DA 020
FF=(2.*B/EQFE3)*(1.-C*D)-(EQFE3*EQH21)*(1.+C*(D**2.))-2.*A/EQFE3	DA 021
GG=-2.*A*C*D	DA 022
HH=2.*P*((D*C-1.)/D+A)+EQCO1/EQFE3	DA 023
RETURN	DA 024
END	DA 025

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SUBROUTINE ID1	ID1 001
C READS IN INITIAL IDENTIFICATION DATA	ID1 002
COMMON ITIME(95),DELP(95),IGAS (95),DATA(95,5),CAL(6),	ID1 003
1CALCO(6),SUMD(6),ISAMP(95),RCC(6),IDATE(5),HSLOP,A,B,CZERO,FEZER,	ID1 004
2PATM, PH20,NIT,NSUM1,NSUM2,NSUM3,NSAMP,NRUN	ID1 005
READ (2,101)NRUN,NSAMP,PATM,PH20,IDATE,CZERO,FEZER	ID1 006
02 WRITE(3,102)NRUN,IDATE	ID1 007
101 FORMAT(5X,I5,5X,I5,5X,F5.2,5X,F6.3,4X,5A2,5X,2F10.3)	ID1 008
102 FORMAT(1H1,33X,21HDATA TAKFN DURING RUN,I4,3H ON,5A2)	ID1 009
WRITE(3,116)	ID1 010
116 FORMAT(1H0, 9HTIME SAMP,5),70H DELP IGAS H2 CO	ID1 011
1 CH4 CO2 H2O C)	ID1 012
PATM=PATM*25.4	ID1 013
CAL(1)=25.18	ID1 014
CAL(2)=24.91	ID1 015
CAL(3)=24.97	ID1 016
CAL(4)=24.94	ID1 017
CAL(5)=PH20*100./PATM	ID1 018
CAL(6)=100.	ID1 019
NIT=0	ID1 020
RETURN	ID1 021
END	ID1 022

SUBROUTINE ID2	ID2 001
C READS IN RAW DATA AND LISTS IT	ID2 002
COMMON ITIME(95),DELP(95),IGAS (95),DATA(95,5),CAL(6),	ID2 003
1CALCO(6),SUMD(6),ISAMP(95),RCC(6),IDATE(5),HSLOP,A,B,CZERO,FEZER,	ID2 004
2PATM, PH20,NIT,NSUM1,NSUM2,NSUM3,NSAMP,NRUN	ID2 005
IH20=PH20	ID2 006
NSUM1=0	ID2 007
NSUM2=0	ID2 008
NSUM3=0	ID2 009
DO 05 N=1,6,1	ID2 010
SUMD(N)=0.	ID2 011
05 CONTINUE	ID2 012
DO 11 I=1,NSAMP,1	ID2 013
11 READ (2,103)ITIME(I),ISAMP(I),DELP(I),IGAS(I),(DATA(I,J),J=1,5)	ID2 014
DO 10 I=1,NSAMP,1	ID2 015
IF(IGAS(I)=6)18,51,51	ID2 016
18 WRITE(3,103)ITIME(I),ISAMP(I),DELP(I),IGAS(I),(DATA(I,J),J=1,5)	ID2 017
103 FORMAT(2I5,5X,F5.2,15,5F10.1)	ID2 018
DO 20 K=1,5,1	ID2 019
DATA(I,K)=DATA(I,K)*PATM/(PATM+DELP(I))	ID2 020
20 CONTINUE	ID2 021
K=IGAS(I)	ID2 022
GO TO (30,40,50,10,10),K	ID2 023
30 DO 70 J=1,4,1	ID2 024
SUMD(J)=SUMD(J)+DATA(I,J)	ID2 025
70 CONTINUE	ID2 026
NSUM1=NSUM1+1	ID2 027
GO TO 10	ID2 028
40 SUMD(6)=SUMD(6)+DATA(I,1)	ID2 029
NSUM2=NSUM2+1	ID2 030
GO TO 10	ID2 031
50 SUMD(5)=SUMD(5)+DATA(I,5)	ID2 032
NSUM3=NSUM3+1	ID2 033
GO TO 10	ID2 034
51 K=IGAS(I)-5	ID2 035

GO TO (52,53,54,56),K	ID2 036
52 WRITE(3,106)ITIME(I),IGAS(I),(DATA(I,J),J=1,5)	ID2 037
106 FORMAT(1H0,I4,15X,I5,5F10.1)	ID2 038
GO TO 10	ID2 039
53 WRITE(3,107)ITIME(I),IGAS(I),DATA(I,1)	ID2 040
107 FORMAT(I5,15X,I5,50X,F10.3)	ID2 041
GO TO 10	ID2 042
54 WRITE(3,106)ITIME(I),IGAS(I),DATA(I,1)	ID2 043
GO TO 10	ID2 044
56 WRITE(3,999)ITIME(I),IGAS(I)	ID2 045
999 FORMAT(1H0,I4,15X,I5,' CHANGE INLET FLOW TO INERT HELIUM AT 20 CC/ 1SEC(STP)')	ID2 046
10 CONTINUE	ID2 047
RETURN	ID2 048
END	ID2 049
	ID2 050

SUBROUTINE CALC	CALC 001
C INITIAL CALCULATIONS TO DETERMINE CHROMATOGRAPH RESPONSE FACTORS	CALC 002
COMMON ITIME(95),DELP(95),IGAS (95),DATA(95,5),CAL(6),	CALC 003
1CALCO(6),SUMD(6),ISAMP(95),RCC(6),IDATE(5),HSLOP,A,B,CZERO,FEZER,	CALC 004
2PATM, PH20,NIT,NSUM1,NSUM2,NSUM3,NSAMP,NRUN	CALC 005
DO 45 L=1,4,1	CALC 006
CALCO(L)=SUMD(L)*100./(NSUM1*CAL(L))	CALC 007
45 CONTINUE	CALC 008
IH20=PH20	CALC 009
IF(IH20)22,46,47	CALC 010
22 WRITE(3,100)	CALC 011
100 FORMAT(1X,'WE HAVE A PROBLEM IN CALC')	CALC 012
46 CALCO(5)=CALCO(4)*0.69	CALC 013
GO TO 48	CALC 014
47 CALCO(5)=SUMD(5)*100./(NSUM3*CAL(5))	CALC 015
48 CALCO(6)=SUMD(6)*100./(NSUM2*CAL(6))	CALC 016
A=SUMD(6)/NSUM2	CALC 017
B=SUMD(1)/NSUM1	CALC 018
HSLOP=(ALOG(100.)-ALOG(25.18))/(ALOG(A)-ALOG(B))	CALC 019
RETURN	CALC 020
END	CALC 021


```

SUBROUTINE DOUT1
C MAIN CALCULATIONS DONE FOR DATA REDUCTION AND APPROPRIATE DATA PUT IN
C ECT FORM FOR PLOTTING ON THE PHASE DIAGRAM
COMMON ITIME( 95),DELP( 95),IGAS ( 95),DATA( 95,5),CAL(6),
1CALCO(6),SUMD(6),ISAMP( 95),RCC(6),IDATE(5),HSLOP,A,B,CZERO,FEZER,
2PATM, PH2O,NIT,NSUM1,NSUM2,NSUM3,NSAMP,NRUN
ID=5
Y=7.
NCAL =0
NCAL1=0
NCAL2=0
CALL SCALF(.85,.85,0.,0. )
CALL FPLOTT(3.,0.,0)
71 DO 65 I=1,NSAMP,1
IF(NCAL1)22,80,81
22 WRITE(3,100)
100 FORMAT ('WE HAVE A PROBLEM IN DOUT1')
80 WRITE(3,102)NRUN,IDATE
WRITE(3,113)
113 FORMAT(/38H THE CALIBRATION GASES ARE KNOWN TO BE/21X,55HIGAS
1 H2 CO CH4 CO2 H2O )
SUM=100.-CAL(5)
WRITE(3,114)SUM,CAL(5)
114 FORMAT(24X,52H1 25.17 24.94 24.97 24.94 0.00
1 /24X,52H2 100.00 0.00 0.00 0.00 0.00
2 /24X,36H3 0.00 0.00 0.00 ,F6.2,5X,F5.2)
WRITE(3,115)
115 FORMAT(/54H CALIBRATION GAS SAMPLES FOR THIS RUN WERE ANALYSED AS)
WRITE(3,116)
116 FORMAT(1H0, 9HTIME SAMP,5X,70H DELP IGAS H2 CO
1 CH4 CO2 H2O C )
NCAL1=NCAL1+1
81 IF(NCAL)22,82,83
82 IF(IGAS(I)=4)72,65,65
83 IF(NCAL2)22,84,85
DOUT1 001
DOUT1 002
DOUT1 003
DOUT1 004
DOUT1 005
DOUT1 006
DOUT1 007
DOUT1 008
DOUT1 009
DOUT1 010
DOUT1 011
DOUT1 012
DOUT1 013
DOUT1 014
DOUT1 015
DOUT1 016
DOUT1 017
DOUT1 018
DOUT1 019
DOUT1 020
DOUT1 021
DOUT1 022
DOUT1 023
DOUT1 024
DOUT1 025
DOUT1 026
DOUT1 027
DOUT1 028
DOUT1 029
DOUT1 030
DOUT1 031
DOUT1 032
DOUT1 033
DOUT1 034
DOUT1 035

```

84	WRITE(3,102)NRUN,IDATE	DOUT1 036
102	FORMAT(1H1,33X,21HDATA TAKEN DURING RUN,I4,3H ON,5A2)	DOUT1 037
	WRITE(3,116)	DOUT1 038
	NCAL2=NCAL2+1	DOUT1 039
85	IF(IGAS(I)=4)65,86,86	DOUT1 040
86	IF(IGAS(I)=6)72,69,69	DOUT1 041
72	J=1	DOUT1 042
	DATA(I,J)=EXP(ALOG(100.)-(ALOG(A)-ALOG(DATA(I,J)))*HSLOP)	DOUT1 043
	SUM=DATA(I,J)	DOUT1 044
	DO 66 J=2,5,1	DOUT1 045
	DATA(I,J)=DATA(I,J)*100./CALCO(J)	DOUT1 046
	SUM=SUM+DATA(I,J)	DOUT1 047
66	CONTINUE	DOUT1 048
	DO 67 J=1,5	DOUT1 049
	DATA(I,J)=DATA(I,J)*100./SUM	DOUT1 050
67	CONTINUE	DOUT1 051
	INDEX=(SUM-100.)/10.	DOUT1 052
	IF(INDEX)690,691,690	DOUT1 053
690	WRITE(3,108)	DOUT1 054
108	FORMAT(1X,100HTHE SUM OF THE FOLLOWING DATA DIFFERED FROM 100 PERC 1ENT BY MORE THAN 10 PERCENT BEFORE JUSTIFICATION)	DOUT1 055
691	WRITE(3,105)ITIME(I),ISAMP(I),DELP(I),IGAS(I),(DATA(I,J),J=1,5)	DOUT1 056
105	FORMAT(2I5,5X,F5.2,I5,5F10.2)	DOUT1 057
	PH2=DATA(I,1)	DOUT1 058
	PCO=DATA(I,2)	DOUT1 059
	PCH4=DATA(I,3)	DOUT1 060
	PCO2=DATA(I,4)	DOUT1 061
	PH2O=DATA(I,5)	DOUT1 062
	IGASS=IGAS(I)	DOUT1 063
	GO TO (925,925,925,926,927),IGASS	DOUT1 064
926	IF(NIT=2)23,24,25	DOUT1 065
23	NUM=NIT-1	DOUT1 066
	GO TO 928	DOUT1 067
24	NUM=NIT	DOUT1 068
	GO TO 928	DOUT1 069
		DOUT1 070

250

25	NUM=NIT+1.	DOUT1	071
	GO TO 928	DOUT1	072
928	CALL DPLOT(PH2,PCO,PCH4,PCC2,PH2O,NUM)	DOUT1	073
	GO TO 925	DOUT1	074
927	IF(NIT=2)26,27,28	DOUT1	075
26	NUM=NIT	DOUT1	076
	GO TO 930	DOUT1	077
27	NUM=NIT+1	DOUT1	078
	GO TO 930	DOUT1	079
28	NUM=NIT+2	DOUT1	080
	GO TO 930	DOUT1	081
930	CALL DPLOT(PH2,PCO,PCH4,PCO2,PH2O,NUM)	DOUT1	082
925	GO TO 65	DOUT1	083
69	K=IGAS(I)=5	DOUT1	084
	GO TO (152,156,157,998) ,K	DOUT1	085
152	DATA(I,1)=DATA(I,1)/1.6125	DOUT1	086
	DATA(I,2)=DATA(I,2)/3.4250	DOUT1	087
	DATA(I,3)=DATA(I,3)/ 4.3125	DOUT1	088
	DATA(I,4)=DATA(I,4)/3.0000	DOUT1	089
	SUM=0.	DOUT1	090
	DO 153 K=1,4	DOUT1	091
153	SUM=SUM+DATA(I,K)	DOUT1	092
	DATA(I,5)=SUM*DATA(I,5)/PATM	DOUT1	093
	SUM=SUM+DATA(I,5)	DOUT1	094
	DO 154 K=1,5	DOUT1	095
154	DATA(I,K)=100.*DATA(I,K)/SUM	DOUT1	096
	WRITE(3,109)ITIME(I),(DATA(I,K),K=1,5),SUM	DOUT1	097
109	FORMAT(1H0,I4,21H CHANGED INLET GAS TO,F9.2,4F10.2,11X,11HTOTAL FL	DOUT1	098
	LOW ,F4.1,13H CC/SEC (STP))	DOUT1	099
	PH2=DATA(I,1)	DOUT1	100
	PCO=DATA(I,2)	DOUT1	101
	PCH4=DATA(I,3)	DOUT1	102
	PCO2=DATA(I,4)	DOUT1	103
	PH2O=DATA(I,5)	DOUT1	104
	NIT=NIT+1	DOUT1	105

IF(NIT=3)77,77,78	DOUT1 106
78 NIT=1	DOUT1 107
GO TO 921	DOUT1 108
77 GO TO (921,922,923),NIT	DOUT1 109
921 NUM=0	DOUT1 110
CALL TABLE(PH2,PCO,PCH4,PCO2,PH2O,NUM,Y,NRUN,IDATE)	DOUT1 111
GO TO 924	DOUT1 112
922 NUM=2	DOUT1 113
CALL TABLE(PH2,PCO,PCH4,PCO2,PH2O,NUM,Y,NRUN,IDATE)	DOUT1 114
GO TO 924	DOUT1 115
923 NUM=4	DOUT1 116
CALL TABLE(PH2,PCO,PCH4,PCO2,PH2O,NUM,Y,NRUN,IDATE)	DOUT1 117
924 GO TO 65	DOUT1 118
156 C=DATA(I,1)-CZERO	DOUT1 119
WRITE(3,110)ITIME(I),C	DOUT1 120
110 FORMAT(I5,70X,F10.3,23H GR, MS CARBON DEPOSITED)	DOUT1 121
GO TO 65	DOUT1 122
157 WRITE(3,111)ITIME(I),DATA(I,1)	DOUT1 123
111 FORMAT(/I5,21H REACTOR TEMPERATURE ,F4.0,19H DEGREES CENTIGRADE/)	DOUT1 124
GO TO 65	DOUT1 125
998 WRITE(3,999)ITIME(I),IGAS(I)	DOUT1 126
999 FORMAT(1H0,I4,15X,I5,' CHANGE INLET FLOW TO INERT HELIUM AT 20 CC/ 1SEC(STP)')	DOUT1 127
65 CONTINUE	DOUT1 128
NCAL =NCAL +1	DOUT1 129
IF(NCAL2)22,89,90	DOUT1 130
89 DO 55 I=1,6,1	DOUT1 131
RCC(I)=CALCO(I)/CALCO(4)	DOUT1 132
55 CONTINUE	DOUT1 133
WRITE(3,104)(RCC(I),I=1,6)	DOUT1 134
WRITE(3,117)PATM	DOUT1 135
WRITE(3,118)FEZER	DOUT1 136
WRITE(3,119)CZERO	DOUT1 137
117 FORMAT(1H0,26H ATMOSPHERIC PRESSURE WAS ,F7.2,2H .)	DOUT1 138
118 FORMAT(1H0,60H THE WEIGHT OF CATALYST INITIALLY CHARGED TO THE REA	DOUT1 139
	DOUT1 140

1	CTOR IS ,F6.3,7H GRAMS.)	DOUT1 141
119	FORMAT(1H0,70H THE GROSS WEIGHT OF THE CATALYST CARRIER AND SUSPEN	DOUT1 142
	SION MECHANISM IS ,F7.3,7H GRAMS.)	DOUT1 143
104	FORMAT(1H0,24H THE RELATIVE CALIBRATION/25H AREA COEFFICENTS ARE	DOUT1 144
	1 ,6F10.6)	DOUT1 145
	GO TO 71	DOUT1 146
90	RETURN	DOUT1 147
	END	DOUT1 148

```

SUBROUTINE TABLE(PH2,PCO,PCH4,PCO2,PH2O,NUM,Y,NRUM,IDATE)
C LISTS INLET AND OUTLET GAS COMPOSITIONS ON THE PHASE DIAGRAM
  DIMENSION IDATE(5)
  INUM=NUM
  IF(NUM=2)3,4,4
  3 CALL FCHAR(-2.9,9.,.1,.1,0.0)
  WRITE (7,5) NRUM,IDATE
  5 FORMAT('DATE TAKEN DURING RUN',I4,'ON',5A2)
  CALL FCHAR(-2.2,8.,.1,.1,0.0)
  WRITE(7,1)
  1 FORMAT('INLET GAS COMPOSITION  ')
  CALL FCHAR(-2.3,7.5,.1,.1,0.0)
  WRITE(7,2)
  2 FORMAT('H2',5X,'CO',3X,'CH4',3X,'CO2',3X,'H2O')
  4 CALL FPLOT(3,-2.5,Y)
  CALL FPLOT(2,-2.5,Y)
  CALL POINT(INUM)
  CALL FCHAR(-2.4,Y,.1,.1,0.0)
  WRITE(7,200) PH2,PCO,PCH4,PCO2,PH2O
200 FORMAT(F5.1,1X,F5.1,1X,F5.1,1X,F5.1,1X,F5.1)
  INUM=INUM+1
  Y1=Y-.5
  CALL FPLOT(3,-2.5,Y1)
  CALL FPLOT(2,-2.5,Y1)
  CALL POINT(INUM)
  CALL FCHAR(-2.3,Y1,.1,.1,0.0)
  WRITE(7,300)
300 FORMAT('OUTLET GAS COMP. ')
  Y=Y-1.
  RETURN
  END

```

```

TABLE 001
TABLE 002
TABLE 003
TABLE 004
TABLE 005
TABLE 006
TABLE 007
TABLE 008
TABLE 009
TABLE 010
TABLE 011
TABLE 012
TABLE 013
TABLE 014
TABLE 015
TABLE 016
TABLE 017
TABLE 018
TABLE 019
TABLE 020
TABLE 021
TABLE 022
TABLE 023
TABLE 024
TABLE 025
TABLE 026
TABLE 027
TABLE 028
TABLE 029
TABLE 030
TABLE 031

```

```

SUBROUTINE DPLOT(PH2,PCO,PCH4,PCO2,PH2O,NUM)
C PLOTS INLET AND OUTLET GAS COMPOSITIONS ON THE PHASE DIAGRAM
PI=3.14159
Z=PI/6.
ATOMH=2.*PH2O+2.*PH2+4.*PCH4
ATOMC=PCO+PCO2+PCH4
ATOMO=PCO+2.*PCO2+PH2O
ATOMT=ATOMH+ATOMO+ATOMC
ULH=10.-(ATOMH/(ATOMT*.1))
ULO=ATOMO/(ATOMT*.1)
ULC=ATOMC/(ATOMT*.1)
X1=ULH-(ULH-ULO)*SIN(Z)
Y1=(ULH-ULO)*COS(Z)
CALL FPLLOT(3,X1,Y1)
CALL FPLLOT(2,X1,Y1)
CALL POINT(NUM)
RETURN
END

```

```

DPLOT 001
DPLOT 002
DPLOT 003
DPLOT 004
DPLOT 005
DPLOT 006
DPLOT 007
DPLOT 008
DPLOT 009
DPLOT 010
DPLOT 011
DPLOT 012
DPLOT 013
DPLOT 014
DPLOT 015
DPLOT 016
DPLOT 017
DPLOT 018

```

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DATE TAKEN DURING RUN 470N 7 APR 77

CARBON

TEMP. = 800.00K

PRES. = 1.00ATM

INLET GAS COMPOSITION %

	HE	CO	CH4	CO2	H2O
+ INLET GAS COMP.	5.9	39.7	16.1	36.9	1.1
x OUTLET GAS COMP.	8.0	31.0	15.3	42.8	2.7
▽ OUTLET GAS COMP.	6.5	29.4	15.1	43.8	3.1
△ OUTLET GAS COMP.	0.0	0.0	0.0	96.8	3.1
x OUTLET GAS COMP.	5.9	39.7	16.1	36.9	1.1
▽ OUTLET GAS COMP.	100.0	0.0	0.0	0.0	0.0
△ OUTLET GAS COMP.			CH4		

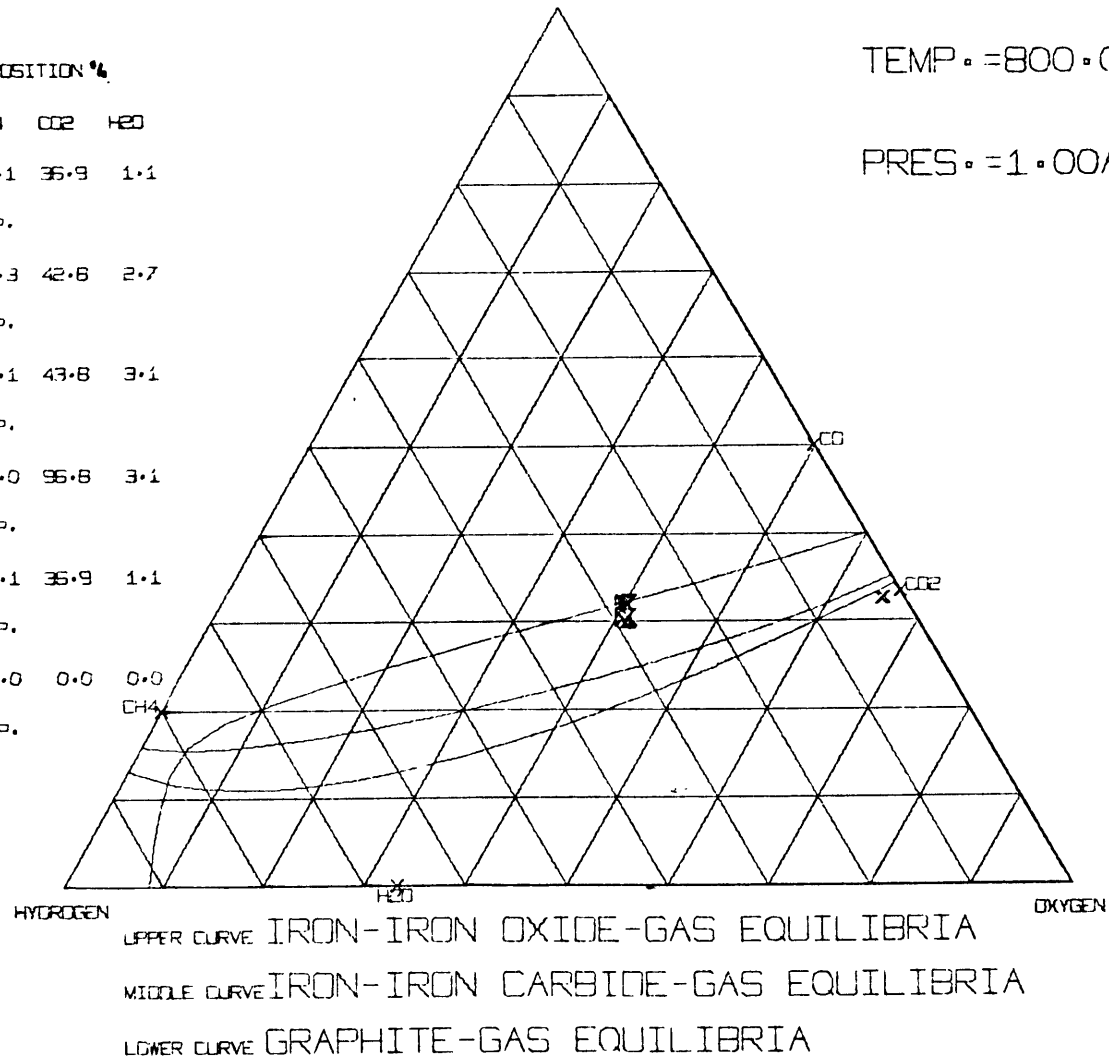


Figure 61 Example of Experimental Data as Plotted on a Triangular Phase Diagram

DATA TAKEN DURING RUN 47 ON 8 APR 77

TIME SAMP	DELTA	IGAS	H2	CO	CH4	CO2	H2O	N2	HE	(MILLIGRAMS)	
800			9 CHANGE INLET FLOW TO INERT HELIUM AT 20 CC/SEC(STP)								
930			8	527.0							
1023	1	0.00	1	3825.0	194597.0	162589.0	214608.0	0.0			
1028	2	0.00	1	3812.0	193370.0	161754.0	212472.0	0.0			
1032	3	0.00	1	3872.0	195833.0	164073.0	214550.0	0.0			
1036	4	0.00	1	3828.0	193928.0	162597.0	212021.0	0.0			
1042			7						33.269		
1045			6	1.8	26.6	13.6	21.7	9.2			
1051	5	0.00	4	779.0	312487.0	104147.0	312281.0	9708.0			
1058	6	0.00	4	789.0	309400.0	103094.0	308819.0	7207.0			
1101			7						33.296		
1116	7	-3.00	5	724.0	298610.0	104679.0	330293.0	8680.0			
1123	8	-3.50	5	729.0	294878.0	103472.0	327619.0	6786.0			
1126			7						33.355		
1143	9	0.00	4	812.0	313404.0	105212.0	315123.0	9290.0			
1148	10	0.00	4	809.0	311321.0	104648.0	311519.0	6763.0			
1152			7						33.445		
1207	11	-3.50	5	748.0	290210.0	105050.0	339968.0	9471.0			
1214	12	-3.50	5	742.0	285455.0	104140.0	337918.0	7538.0			
1218			7						33.536		
1219			9	CHANGE INLET FLOW TO INERT HELIUM AT 20 CC/SEC(STP)							
1308			7						33.524		
1318			6	2.5	20.6	12.9	24.9	21.5			
1330	13	0.00	4	1075.0	244659.0	100105.0	367833.0	17689.0			
1336	14	2.00	4	1061.0	243809.0	99746.0	363794.0	14896.0			
1340			7						33.547		
1401	15	-2.50	5	1030.0	243610.0	98931.0	375179.0	16604.0			
1408	16	-3.50	5	1008.0	240291.0	97751.0	370281.0	14395.0			
1413			7						33.558		
1448	17	0.00	4	1082.0	243834.0	99606.0	369909.0	18214.0			
1454	18	0.00	4	1054.0	242814.0	99296.0	365326.0	15923.0			
1458			7						33.565		
1510	19	-3.50	5	1019.0	242031.0	98035.0	375119.0	16652.0			
1517	20	-4.00	5	1013.0	239012.0	96753.0	370516.0	14129.0			
1521			7						33.570		
1524			9	CHANGE INLET FLOW TO INERT HELIUM AT 20 CC/SEC(STP)							
1600			7						33.573		
1611			6	2.6	19.5	12.6	25.5	24.5			
1627	21	0.00	4	1179.0	235535.0	97634.0	376478.0	19730.0			
1634	22	0.00	4	1134.0	233305.0	96683.0	371146.0	17048.0			
1638			7						33.579		
1655	23	-3.30	5	1101.0	234578.0	97098.0	382903.0	19060.0			
1702	24	-3.80	5	1072.0	231455.0	95673.0	378647.0	16426.0			
1706			7						33.579		
1728	25	0.00	4	1161.0	235762.0	97911.0	375818.0	21260.0			
1734	26	0.50	4	1139.0	234538.0	97400.0	371600.0	17894.0			
1739			7						33.579		
1755	27	-3.70	5	1093.0	234134.0	96441.0	380717.0	20158.0			
1802	28	-4.00	5	1088.0	230797.0	95240.0	375046.0	16371.0			
1806			7						33.579		
1808			6	0.0	0.0	0.0	27.5	24.5			
1829	29	0.00	3	0.0	1134.0	0.0	843407.1	18158.0			
1836	30	0.00	3	0.0	92.0	0.0	832146.1	15922.0			

Figure 62 Example of Raw Data, Run A-47

1840			7								33.584	
1855	31	-4.20	5	0.0	22876.0	0.0	811413.1	18433.0				
1902	32	-5.00	5	0.0	19630.0	0.0	804270.1	16181.0				
1908			7								33.589	
1910			9	CHANGE INLET FLOW TO INERT HELIUM AT 20 CC/SEC(STP)								
2034			7								33.598	
2043			6	1.8	26.6	13.6	21.7	9.2				
2053	33	-0.50	4	761.0	315284.0	103308.0	317878.0	10681.0				
2059	34	0.00	4	750.0	314611.0	103714.0	310405.0	8043.0				
2102			7								33.586	
2120	35	-3.50	5	722.0	314927.0	103068.0	327163.0	8733.0				
2128	36	3.80	5	716.0	298875.0	102769.0	328014.0	6682.0				
2132			7								33.598	
2145	37	0.30	4	760.0	312313.0	104065.0	312637.0	10421.0				
2152	38	-4.00	5	708.0	290266.0	103472.0	333706.0	8137.0				
2156			7								33.644	
2157			6	32.5	0.0	0.0	0.0	0.0				
2209	39	0.00	2	21938.0	0.0	732.0	0.0	0.0				
2212	40	0.00	2	21695.0	0.0	600.0	0.0	0.0				
2215			7								33.589	
2216			9	CHANGE INLET FLOW TO INERT HELIUM AT 20 CC/SEC(STP)								

Figure 62 (continued)

DATA TAKEN DURING RUN 47 ON 7 APR 77

THE CALIBRATION GASES ARE KNOWN TO BE

IGAS	H2	CO	CH4	CO2	H2O
1	25.17	24.94	24.97	24.94	0.00
2	100.00	0.00	0.00	0.00	0.00
3	0.00	0.00	0.00	96.79	3.20

CALIBRATION GAS SAMPLES FOR THIS RUN WERE ANALYSED AS

TIME SAMP	DELP	IGAS	H2	CO	CH4	CO2	H2O	N2	HE	C (MILLIGRAMS)
1023	1	0.00	1	25.10	24.90	24.92	25.05	0.00	0.00	0.00
1028	2	0.00	1	25.19	24.90	24.94	24.95	0.00	0.00	0.00
1032	3	0.00	1	25.19	24.91	24.99	24.89	0.00	0.00	0.00
1036	4	0.00	1	25.21	24.91	25.01	24.84	0.00	0.00	0.00
1829	29	0.00	3	0.00	0.14	0.00	96.51	3.34	0.00	0.00
1836	30	0.00	3	0.00	0.01	0.00	97.00	2.98	0.00	0.00
2209	39	0.00	2	99.88	0.00	0.11	0.00	0.00	0.00	0.00
2212	40	0.00	2	99.90	0.00	0.09	0.00	0.00	0.00	0.00

THE RELATIVE CALIBRATION

AREA COEFFICENTS ARE 0.017795 0.912158 0.761705 1.000000 0.622108 0.025495 0.875000

ATMOSPHERIC PRESSURE WAS 767.33 .

THE WEIGHT OF CATALYST INITIALLY CHARGED TO THE REACTOR IS 0.518 GRAMS.

THE GROSS WEIGHT OF THE CATALYST CARRIER AND SUSPENSION MECHANISM IS 32.962 GRAMS.

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Figure 63 Chromatograph Response Factors for A-47

DATA TAKEN DURING RUN 47 ON 7 APR 77

TIME SAMP	DELP	IGAS	H2	CO	CH4	CO2	H2O	N2	HE	C(MILLIGRAMS)
800	9 CHANGE INLET FLOW TO INERT HELIUM AT 20 CC/SEC(STP)									
930	REACTOR TEMPERATURE 527. DEGREES CENTIGRADE									
1042										306.999
1045	CHANGED INLET GAS TO			5.70	37.35	16.11	39.62	1.20	TOTAL FLOW 19.9 CC/SEC (STP)	
1051	5	0.00	4	7.01	39.46	15.75	35.97	1.79	0.00	0.00
1058	6	0.00	4	7.17	39.60	15.80	36.05	1.35	0.00	0.00
1101										333.999
1116	7	-3.00	5	6.62	37.78	15.86	38.12	1.61	0.00	0.00
1123	8	-3.50	5	6.74	37.79	15.88	38.30	1.27	0.00	0.00
1126										392.997
1143	9	0.00	4	7.19	39.28	15.79	36.02	1.70	0.00	0.00
1148	10	0.00	4	7.26	39.51	15.90	36.06	1.25	0.00	0.00
1152										483.001
1207	11	-3.50	5	6.76	36.56	15.84	39.07	1.74	0.00	0.00
1214	12	-3.50	5	6.81	36.46	15.93	39.37	1.41	0.00	0.00
1218										573.997
1219	9 CHANGE INLET FLOW TO INERT HELIUM AT 20 CC/SEC(STP)									
1308										561.996
1318	CHANGED INLET GAS TO			7.63	28.90	15.20	45.44	2.80	TOTAL FLOW 19.9 CC/SEC (STP)	
1330	13	0.00	4	8.98	30.67	15.02	42.06	3.25	0.00	0.00
1336	14	2.00	4	9.00	30.94	15.16	42.11	2.77	0.00	0.00
1340										584.999
1401	15	-2.50	5	8.67	30.53	14.84	42.89	3.05	0.00	0.00
1408	16	-3.50	5	8.67	30.63	14.92	43.06	2.69	0.00	0.00
1413										596.000
1448	17	0.00	4	9.01	30.50	14.92	42.21	3.34	0.00	0.00
1454	18	0.00	4	8.94	30.77	15.07	42.24	2.95	0.00	0.00
1458										602.996
1510	19	-3.50	5	8.63	30.45	14.77	43.05	3.07	0.00	0.00
1517	20	-4.00	5	8.73	30.56	14.81	43.22	2.64	0.00	0.00
1521										608.001
1524	9 CHANGE INLET FLOW TO INERT HELIUM AT 20 CC/SEC(STP)									
1600										611.000
1611	CHANGED INLET GAS TO			8.09	27.34	14.91	46.44	3.20	TOTAL FLOW 19.9 CC/SEC (STP)	
1627	21	0.00	4	9.61	29.37	14.57	42.82	3.60	0.00	0.00
1634	22	0.00	4	9.49	29.62	14.70	42.99	3.17	0.00	0.00
1638										616.996
1655	23	-3.30	5	9.11	29.28	14.51	43.60	3.48	0.00	0.00
1702	24	-3.80	5	9.07	29.41	14.55	43.88	3.06	0.00	0.00
1706										616.996
1728	25	0.00	4	9.48	29.35	14.59	42.68	3.88	0.00	0.00
1734	26	0.50	4	9.48	29.63	14.73	42.82	3.31	0.00	0.00
1739										616.996
1755	27	-3.70	5	9.08	29.30	14.45	43.46	3.69	0.00	0.00
1802	28	-4.00	5	9.22	29.46	14.56	43.67	3.06	0.00	0.00
1806										616.996
1808	CHANGED INLET GAS TO			0.00	0.00	0.00	96.79	3.20	TOTAL FLOW 10.3 CC/SEC (STP)	
1840										622.001
1855	31	-4.20	5	0.00	2.89	0.00	93.68	3.42	0.00	0.00
1902	32	-5.00	5	0.00	2.52	0.00	94.42	3.05	0.00	0.00
1908										626.999
1910	9 CHANGE INLET FLOW TO INERT HELIUM AT 20 CC/SEC(STP)									

Figure 64 Example of Reduced Data, Run A-47

2034											636.001
2043	CHANGED	INLET GAS TO		5.71	37.33	16.11	39.63	1.20			TOTAL FLOW 19.9 CC/SEC (STP)
2053	33	-0.50	4	6.81	39.45	15.48	36.28	1.95	0.00	0.00	
2059	34	0.00	4	6.84	39.94	15.76	35.94	1.49	0.00	0.00	
2102											624.000
2120	35	-3.50	5	6.59	38.51	15.58	37.68	1.61	0.00	0.00	
2128	36	3.80	5	6.63	38.17	15.71	38.21	1.25	0.00	0.00	
2132											636.001
2145	37	0.30	4	6.87	39.44	15.73	36.01	1.92	0.00	0.00	
2152	38	-4.00	5	6.57	37.12	15.84	38.92	1.52	0.00	0.00	
2156											681.999
2157	CHANGED	INLET GAS TO		100.00	0.00	0.00	0.00	0.00			TOTAL FLOW 19.6 CC/SEC (STP)
2215											626.999
2216											

9 CHANGE INLET FLOW TO INERT HELIUM AT 20 CC/SEC(STP)

Figure 64 (continued)

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***** INTERRUPT REQUEST *****

C PROGRAM ERROR USED TO DETERMINE THE ERRORS THAT OCCUR IN RAW DATA REDU	ERROR 001
C CTION BASED ON A PROPAGATION OF ERROR ANALYSIS ON THE PRECISION GAS MI	ERROR 002
C XTURE	ERROR 003
DIMENSION ITIME(008),DELPH(008),IGAS (008),DATA(008,5),CAL(6),	ERROR 004
1CALCO(6),SUMD(6),ISAMP(008),RCC(6),IDATE(5), TE(6), EDAT(8,5),DSM	ERROR 005
1D(6),DCLC(6),DCAL(6),TEST(6)	ERROR 006
01 READ (2,101)NRUN,NSAMP,PATM,PH2O,IDATE,CZERO,FEZER	ERROR 007
IF(NRUN)68,68,02	ERROR 008
02 WRITE(3,102)NRUN,IDATE	ERROR 009
101 FORMAT(5X,I5,5X,I5,5X,F5.2,5X,F6.3,4X,5A2,5X,2F10.3)	ERROR 010
102 FORMAT(1H1,33X,21HDATA TAKEN DURING RUN,I4,3H ON,5A2)	ERROR 011
WRITE(3,116)	ERROR 012
C MAXIMUM ERROR IN BAROMETRIC PRESSURE (DPATM) AND ERROR IN PRESSURE OF	ERROR 013
C THE SAMPLE CELL (DDELPH) AND ERROR AS SPECIFIED IN THE PRECISION GAS	ERROR 014
C MIXTURE (DCAL)	ERROR 015
DPATM=4.	ERROR 016
DDELPH=1.	ERROR 017
DSUM=0.0	ERROR 018
IH2O=PH2O	ERROR 019
PATM=PATM*25.4	ERROR 020
CAL(1)=25.18	ERROR 021
CAL(2)=24.91	ERROR 022
CAL(3)=24.97	ERROR 023
CAL(4)=24.94	ERROR 024
CAL(5)=PH2O*100./PATM	ERROR 025
CAL(6)=100.	ERROR 026
NSUM1=0	ERROR 027
NSUM2=0	ERROR 028
NSUM3=0	ERROR 029
DO 05 N=1,6,1	ERROR 030
SUMD(N)=0.	ERROR 031
TEST(N)=0.0	ERROR 032
DSMD(N)=0.0	ERROR 033
DCAL(N)=.02	ERROR 034
05 CONTINUE	ERROR 035

C START OF ANALYSIS	ERROR 036
DCAL(5)=(2.0*PH20/PATM)+(100.*PH20*DPATM/(PATM**2.))	ERROR 037
DO 11 I=1,NSAMP,1	ERROR 038
11 READ (2,103)ITIME(I),ISAMP(I),DELP(I),IGAS(I),(DATA(I,J),J=1,5)	ERROR 039
DO 10 I=1,NSAMP,1	ERROR 040
IF(IGAS(I)-6)18,51,51	ERROR 041
18 WRITE(3,103)ITIME(I),ISAMP(I),DELP(I),IGAS(I),(DATA(I,J),J=1,5)	ERROR 042
103 FORMAT(2I5,5X,F5.2,I5,5F10.1)	ERROR 043
DO 20 K=1,5,1	ERROR 044
DATA(I,K)=DATA(I,K)*PATM/(PATM+DELP(I))	ERROR 045
V=1.	ERROR 046
IF(IGAS(I)-4)451,20,20	ERROR 047
451 EDAT(I,K)=DATA(I,K)*V*DPATM*(1./(PATM+DELP(I))+PATM/(PATM+DELP(I))	ERROR 048
1**2.)+DATA(I,K)*V*DDELP*(PATM/(PATM+DELP(I))**2.)	ERROR 049
20 CONTINUE	ERROR 050
K=IGAS(I)	ERROR 051
GO TO (30,40,50,10,10),K	ERROR 052
30 DO 70 J=1,4,1	ERROR 053
SUMD(J)=SUMD(J)+DATA(I,J)	ERROR 054
DSMD(J)=DSMD(J)+EDAT(I,J)	ERROR 055
70 CONTINUE	ERROR 056
NSUM1=NSUM1+1	ERROR 057
GO TO 10	ERROR 058
40 SUMD(6)=SUMD(6)+DATA(I,1)	ERROR 059
DSMD(6)=DSMD(6)+EDAT(I,1)	ERROR 060
NSUM2=NSUM2+1	ERROR 061
GO TO 10	ERROR 062
50 SUMD(5)=SUMD(5)+DATA(I,5)	ERROR 063
DSMD(5)=DSMD(5)+EDAT(I,5)	ERROR 064
NSUM3=NSUM3+1	ERROR 065
GO TO 10	ERROR 066
51 K=IGAS(I)-5	ERROR 067
GO TO (52,53,54,56),K	ERROR 068
52 WRITE(3,106)ITIME(I),IGAS(I),(DATA(I,J),J=1,5)	ERROR 069
106 FORMAT(1H0,I4,15X,I5,5F10.1)	ERROR 070

GO TO 10	ERROR 071
53 WRITE(3,107)ITIME(I),IGAS(I),DATA(I,1)	ERROR 072
107 FORMAT(I5,15X,I5,50X,F10.3)	ERROR 073
GO TO 10	ERROR 074
54 WRITE(3,106)ITIME(I),IGAS(I),DATA(I,1)	ERROR 075
GO TO 10	ERROR 076
56 WRITE(3,999)ITIME(I),IGAS(I)	ERROR 077
10 CONTINUE	ERROR 078
ID=0	ERROR 079
DO 45 L=1,4,1	ERROR 080
CALCO(L)=SUMD(L)*100./(NSUM1*CAL(L))	ERROR 081
DCLC(L)=(100.*DSMD(L)/(NSUM1*CAL(L))+100.*SUMD(L)*DCAL(L)/(NSUM1*	ERROR 082
1CAL(L)**2.)	ERROR 083
PIT=CALCO(4)*.875	ERROR 084
45 CONTINUE	ERROR 085
IF(IH20)68,46,47	ERROR 086
46 CALCO(5)=CALCO(4)*0.69	ERROR 087
DCLC(5)=DCLC(4)*.69	ERROR 088
GO TO 48	ERROR 089
47 CALCO(5)=SUMD(5)*100./(NSUM3*CAL(5))	ERROR 090
DCLC(5)=100.*DSMD(5)/(NSUM3*CAL(5))+100.*SUMD(5)*DCAL(5)/(NSUM3*CA	ERROR 091
1L(5)**2.)	ERROR 092
48 CALCO(6)=SUMD(6)*100./(NSUM2*CAL(6))	ERROR 093
DCLC(6)=100.*DSMD(6)/(NSUM2*CAL(6))+100.*SUMD(6)*DCAL(6)/(NSUM2*CA	ERROR 094
1L(6)**2.)	ERROR 095
A=SUMD(6)/NSUM2	ERROR 096
B=SUMD(1)/NSUM1	ERROR 097
C=ALOG(A)-ALOG(B)	ERROR 098
D=ALOG(100.)-ALOG(25.18)	ERROR 099
HSLOP=(ALOG(100.)-ALOG(25.18))/(ALOG(A)-ALOG(B))	ERROR 100
DELA=DSMD(6)/NSUM2	ERROR 101
DELB=DSMD(1)/NSUM1	ERROR 102
DELC=DELA/(A)+(DELB/B)	ERROR 103
DSLO=-(D*DELC/(C**2.))	ERROR 104
NCAL =0	ERROR 105


```

    NCAL1=0
    NCAL2=0
71 DO 65 I=1,NSAMP,1
    IF(NCAL1)68,80,81
80 WRITE(3,102)NRUN,IDATE
    WRITE(3,113)
113 FORMAT(/38H THE CALIBRATION GASES ARE KNOWN TO BE/21X,55HIGAS
    1 H2 CO CH4 CO2 H2O )
    SUM=100.-CAL(5)
    WRITE(3,114)SUM,CAL(5)
114 FORMAT(24X,52H1 25.17 24.94 24.97 24.94 0.00
    1 /24X,52H2 100.00 0.00 0.00 0.00 0.00
    2 /24X,36H3 0.00 0.00 0.00 ,F6.2,5X,F5.2)
    WRITE(3,115)
115 FORMAT(/54H CALIBRATION GAS SAMPLES FOR THIS RUN WERE ANALYSED AS)
    WRITE(3,116)
116 FORMAT(1H0, 9HTIME SAMP,5X, ' DELP IGAS H2 CO
    1 CH4 CO2 H2O N2 HE C(MILLIGRAMS)')
    NCAL1=NCAL1+1
81 IF(NCAL)68,82,83
82 IF(IGAS(I)-4)72,65,65
83 IF(NCAL2)68,85,85
85 NCAL2=NCAL2+1
    IF(IGAS(I)-4)65,86,86
86 IF(IGAS(I)-6)72,69,69
72 IF(ID)68,888,887
888 J=1
    IF(IGAS(I)-3)503,512,504
503 EDAT(I,J)=(ALOG(100.)-(ALOG(A)-ALOG(DATA(I,J)))*HSLOP)*(HSLOP*(EDA
    1T(I,J)/DATA(I,J)+DELA/A)+DSLO*(ALOG(DATA(I,J))-ALOG(A)) )
    GO TO 513
512 EDAT(I,1)=0.0
513 DSUM=EDAT(I,J)
504 DATA(I,J)=EXP(ALOG(100.)-(ALOG(A)-ALOG(DATA(I,J)))*HSLOP)
    SUM=DATA(I,J)

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ERROR 106
ERROR 107
ERROR 108
ERROR 109
ERROR 110
ERROR 111
ERROR 112
ERROR 113
ERROR 114
ERROR 115
ERROR 116
ERROR 117
ERROR 118
ERROR 119
ERROR 120
ERROR 121
ERROR 122
ERROR 123
ERROR 124
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ERROR 126
ERROR 127
ERROR 128
ERROR 129
ERROR 130
ERROR 131
ERROR 132
ERROR 133
ERROR 134
ERROR 135
ERROR 136
ERROR 137
ERROR 138
ERROR 139
ERROR 140

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DO 66 J=2,5,1	ERROR 141
IF(IGAS(I)=4)505,510,510	ERROR 142
505 EDAT(I,J)=100.*EDAT(I,J)/(CALCO(J))+(DATA(I,J)*100.*DCLC(J)/(CALCO	ERROR 143
1(J)**2.))	ERROR 144
DSUM=EDAT(I,J)+DSUM	ERROR 145
510 DATA(I,J)=DATA(I,J)*100./CALCO(J)	ERROR 146
SUM=SUM+DATA(I,J)	ERROR 147
66 CONTINUE	ERROR 148
DO 67 J=1,5	ERROR 149
IF(IGAS(I)=4)506,518,518	ERROR 150
506 EDAT(I,J)=(100.*EDAT(I,J)/SUM)+(DATA(I,J)*100.*DSUM/(SUM**2.))	ERROR 151
518 DATA(I,J)=DATA(I,J)*100./SUM	ERROR 152
67 CONTINUE	ERROR 153
PN2=0.0	ERROR 154
HEL=0.0	ERROR 155
DO 777 J=1,5,1	ERROR 156
IF(EDAT(I,J)-TEST(J))777,775,775	ERROR 157
775 TEST(J)=EDAT(I,J)	ERROR 158
TE(J)=(EDAT(I,J)/DATA(I,J))*100.	ERROR 159
777 CONTINUE	ERROR 160
C ERRORS IN THE WEIGHT MEASUREMENT OF THE CATALYST ASSEMBLE	ERROR 161
CB=.003	ERROR 162
DCZ=.003	ERROR 163
DCD=(CB+DCZ)*1000.	ERROR 164
GO TO 884	ERROR 165
887 J=1	ERROR 166
DATA(I,J)=EXP(ALOG(100.)-(ALOG(A)-ALOG(DATA(I,J))))*HSLOP)	ERROR 167
DATA(I,2)=DATA(I,2)*100./CALCO(2)	ERROR 168
DATA(I,3)=DATA(I,3)*100./PIT	ERROR 169
SUM=DATA(I,1)+DATA(I,2)+DATA(I,3)	ERROR 170
DO 886 J=4,5,1	ERROR 171
DATA(I,J)=DATA(I,J)*100./CALCO(J)	ERROR 172
SUM=SUM+DATA(I,J)	ERROR 173
886 CONTINUE	ERROR 174
HEL=100.-SUM	ERROR 175

SUM=SUM+HEL	ERROR 176
PN2=DATA(I,3)	ERROR 177
DATA(I,3)=0.0	ERROR 178
884 ERROR=ABS(SUM-100.)	ERROR 179
C IF THE TOTAL AREA BEFORE NORMALIZATION IS GREATER THAN 5 PERCENT THE E	ERROR 180
C RRROR WILL BE INDICATED ON THE OUTPUT	ERROR 181
IF (ERROR-5.) 691,691,690	ERROR 182
690 WRITE(3,108) ERROR	ERROR 183
108 FORMAT(1X, 'THE SUM OF THE FOLLOWING DATA DIFFERED FROM 100 PERC	ERROR 184
1ENT BY',F6.2,'BEFORE JUSTIFICATION')	ERROR 185
691 WRITE(3,105)ITIME(I),ISAMP(I),DELP(I),IGAS(I),(DATA(I,J),J=1,5),PN	ERROR 186
12,HEL	ERROR 187
105 FORMAT(2I5,5X,F5.2,I5,7F10.2)	ERROR 188
GO TO 65	ERROR 189
69 K=IGAS(I)-5	ERROR 190
GO TO (152,156,157,998),K	ERROR 191
152 DATA(I,1)=DATA(I,1)/1.65	ERROR 192
DATA(I,2)=DATA(I,2)/3.58	ERROR 193
DATA(I,3)=DATA(I,3)/4.25	ERROR 194
DATA(I,4)=DATA(I,4)/2.75	ERROR 195
ID=0	ERROR 196
SUM=0.	ERROR 197
DO 153 K=1,4	ERROR 198
153 SUM=SUM+DATA(I,K)	ERROR 199
PERHC=(1.00-(DATA(I,5)/PATM))	ERROR 200
SUM=SUM/PERHC	ERROR 201
DATA(I,5)=SUM*DATA(I,5)/PATM	ERROR 202
DO 154 K=1,5	ERROR 203
154 DATA(I,K)=100.*DATA(I,K)/SUM	ERROR 204
WRITE(3,109)ITIME(I),(DATA(I,K),K=1,5),SUM	ERROR 205
109 FORMAT(1H0,I4,21H CHANGED INLET GAS TO,F9.2,4F10.2,11X,11HTOTAL FL	ERROR 206
1OW ,F4.1,13H CC/SEC (STP))	ERROR 207
GO TO 65	ERROR 208
998 WRITE(3,999)ITIME(I),IGAS(I)	ERROR 209
ID=1	ERROR 210

GO TO 65	ERROR 211
156 BC=(DATA(I,1)-CZERO)*1000.	ERROR 212
WRITE(3,110)ITIME(I),BC	ERROR 213
110 FORMAT(I5,95X,F10.3)	ERROR 214
111 FORMAT(/I5,21H REACTOR TEMPERATURE ,F4.0,19H DEGREES CENTIGRADE/)	ERROR 215
GO TO 65	ERROR 216
157 WRITE(3,111)ITIME(I),DATA(I,1)	ERROR 217
65 CONTINUE	ERROR 218
NCAL =NCAL +1	ERROR 219
IF(NCAL2)68,89,90	ERROR 220
89 DO 55 I=1,6,1	ERROR 221
PITT=PIT/CALCO(4)	ERROR 222
RCC(I)=CALCO(I)/CALCO(4)	ERROR 223
55 CONTINUE	ERROR 224
WRITE(3,104)(RCC(I),I=1,6),PITT	ERROR 225
WRITE(3,117)PATM	ERROR 226
WRITE(3,118)FEZER	ERROR 227
WRITE(3,119)CZERO	ERROR 228
118 FORMAT(1H0,60H THE WEIGHT OF CATALYST INITIALLY CHARGED TO THE REA	ERROR 229
CTOR IS ,F6.3,7H GRAMS.)	ERROR 230
999 FORMAT(1H0,I4,15X,I5,' CHANGE INLET FLOW TO INERT HELIUM AT 20 CC/	ERROR 231
1SEC(STP) ')	ERROR 232
119 FORMAT(1H0,70H THE GROSS WEIGHT OF THE CATALYST CARRIER AND SUSPEN	ERROR 233
SION MECHANISM IS ,F7.3,7H GRAMS.)	ERROR 234
117 FORMAT(1H0,26H ATMOSPHERIC PRESSURE WAS ,F7.2,2H .)	ERROR 235
GO TO 71	ERROR 236
90 WRITE(3,102) NRUN,IDATE	ERROR 237
WRITE(3,500)	ERROR 238
500 FORMAT(/,16X,'THE FOLLOWING ERROR ANALYSIS WAS PERFORMED USING TH	ERROR 239
1E TECHNIQUE OF PROPAGATION OF ERRORS')	ERROR 240
WRITE(3,501)	ERROR 241
501 FORMAT(/,28X,'H2 CO CH4 CO2 H2O N2	ERROR 242
1 HE C(MILLIGRAMS)')	ERROR 243
WRITE(3,502) (TEST(J),J=1,5),DCD	ERROR 244
502 FORMAT(/,4X,'ABSOLUTE ERROR',4X,5F10.5,23X,F10.5)	ERROR 245

WRITE(3,516) (TE(J),J=1,5)	ERROR 246
516 FORMAT(/,4X,'RELATIVE ERROR',4X,5F10.5)	ERROR 247
WRITE(3,520)	ERROR 248
520 FORMAT(5X,'(PERCENTAGE)')	ERROR 249
GO TO 01	ERROR 250
104 FORMAT(1H0,24HTHE RELATIVE CALIBRATION/25H AREA COEFFICENTS ARE	ERROR 251
1 ,7F10.6)	ERROR 252
68 CALL EXIT	ERROR 253
END	ERROR 254

DATA TAKEN DURING RUN 47 ON 7 APR 77											
TIME	SAMP	DELP	IGAS	H2	CO	CH4	CO2	H2O	N2	HE	C(MILLIGRAMS)
1023	1	0.00	1	3825.0	174597.0	162589.0	214608.0	0.0			
1028	2	0.00	1	3812.0	193370.0	161754.0	212472.0	0.0			
1032	3	0.00	1	3872.0	195833.0	164073.0	214550.0	0.0			
1036	4	0.00	1	3828.0	193928.0	162597.0	212021.0	0.0			
1829	29	0.00	3	0.0	1134.0	0.0	843407.1	18158.0			
1836	30	0.00	3	0.0	92.0	0.0	832146.1	15922.0			
2209	39	0.00	2	21938.0	0.0	732.0	0.0	0.0			
2212	40	0.00	2	21695.0	0.0	600.0	0.0	0.0			

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Figure 65 Raw Data for "Error", Run A-47

DATA TAKEN DURING RUN 47 ON 7 APR 77

THE FOLLOWING ERROR ANALYSIS WAS PERFORMED USING THE TECHNIQUE OF PROPAGATION OF ERRORS

	H2	CO	CH4	CO2	H2O	N2	HE	C(MILLIGRAMS)
ABSOLUTE ERROR	0.60832	1.08676	1.09085	4.77729	0.24632			6.00000
RELATIVE ERROR (PERCENTAGE)	2.41458	4.36286	4.36046	4.92480	7.37461			

Figure 66 Example of Propagation of Error Analysis, A-47

C=(EK1+2.*RC)/(4.*RC*RV)	SET 036
D=(RC*(1.-2.*RV)-2.*RV)/(4.*RV)	SET 037
GO TO 31	SET 038
6 K=5	SET 039
C=(EK1+RC)/(RC*(4.*RO-1.))	SET 040
D=-2.*RO*(RC+1.)/(4.*RO-1.)	SET 041
31 E=(1.+RC+D)	SET 042
F=(1.+(EK1/RC)+C)	SET 043
G=(EK1/RC)**2.	SET 044
IF(RO-.2)800,800,801	SET 045
800 IF(RV-.2)802,802,801	SET 046
802 XCOF(1)=(-D*E)/((F**2.)*G*EK2)	SET 047
XCOF(2)=(G*EK2+(F*D*E)-(E**2.)*C)/((F**2.)*G*EK2)	SET 048
XCOF(3)=-2./F	SET 049
XCOF(4)=1.	SET 050
CALL POLRT(XCOF,COF,M,ROOTR,ROOTI,IER)	SET 051
DO 101 MV=1,3,1	SET 052
PCO2=ROOTR(MV)	SET 053
IF(PCO2)101,102,102	SET 054
102 PCO=PCO2*(EK1/RC)	SET 055
PH2=(1.-F*PCO2)/E	SET 056
IF(PH2)101,810,810	SET 057
810 PCH4=EK2*((EK1/RC)**2.)*(PH2**2.)*PCO2	SET 058
PH20=RC*PH2	SET 059
PT=PCO2+PCO+PCH4+PH20+PH2	SET 060
IF(ABS(PT-1.)-.01)11,11,101	SET 061
101 CONTINUE	SET 062
806 WRITE(3,804)	SET 063
804 FORMAT(50X,'EQUATIONS ARE UNSTABLE')	SET 064
GO TO 100	SET 065
801 XCOF(1)=C/(G*EK2*E)	SET 066
XCOF(2)=(D*F-C*E)/(G*EK2*E)	SET 067
XCOF(3)=-1./E	SET 068
XCOF(4)=1.	SET 069
CALL POLRT(XCOF,COF,M,ROOTR,ROOTI,IER)	SET 070

	DO 1010 NV=1,3,1	SET 071
	PH2=ROOTR(NV)	SET 072
	IF(PH2)1010,103,103	SET 073
103	PH2O=PH2*RC	SET 074
	PCO2=(1.-E*PH2)/F	SET 075
	PCO=PCO2*(EK1/RC)	SET 076
	PCH4=EK2*((EK1/RC)**2.)*(P+2**2.)*PCO2	SET 077
	PT=PCO+PCO2+PCH4+PH2O+PH2	SET 078
	IF(ABS(PT-1.)-.01)11,11,10!0	SET 079
1010	CONTINUE	SET 080
	GO TO 806	SET 081
11	PI=3.14157	SET 082
	PW=760.*PH2O	SET 083
	V=(ALOG(PW))/2.303	SET 084
	IF(TEMP-333.)34,34,35	SET 085
34	A=8.10765	SET 086
	B=1750.2860	SET 087
	C=235.0	SET 088
	GO TO 36	SET 089
35	A=7.96681	SET 090
	B=1668.21	SET 091
	C=228.	SET 092
36	SETT=(-B/(V-A))-C	SET 093
	SET1=PH2*FLOW*1.65	SET 094
	SET2=PCO*FLOW*3.58	SET 095
	SET3=PCH4*FLOW*4.25	SET 096
	SET4=PCO2*FLOW*2.75	SET 097
	Z=PI/6.	SET 098
	ATOMH=2.*(PH2O+PH2)+4.*PCH4	SET 099
	ATOMC=PCO+PCO2+PCH4	SET 100
	ATOMO=PCO+2.*PCO2+PH2O	SET 101
	ATOMT=ATOMC+ATOMH+ATOMO	SET 102
	ULH=10.-(ATOMH/(ATOMT*.1))	SET 103
	ULO=ATOMO/(ATOMT*.1)	SET 104
	ULC=ATOMC/(ATOMT*.1)	SET 105

	Y1=(ULH-ULO)*COS(Z)	SET 141
	CALL FPLOT(-2,X1,Y1)	SET 142
	X=5.	SET 143
	Y=5.*SQRT(3.)	SET 144
	CALL FPLOT(-1,X,Y)	SET 145
	GO TO 100	SET 146
C	CASE WHERE C/H IS CONSTANT	SET 147
809	ATOMO=0.0	SET 148
	ATOMT=ATOMC+ATOMH+ATOMO	SET 149
	ULH=10.-(ATOMH/(ATOMT*.1))	SET 150
	ULC=ATOMC/(ATOMT*.1)	SET 151
	X1=ULH-(ULH-ULO)*SIN(Z)	SET 152
	Y1=(ULH-ULO)*COS(Z)	SET 153
	X=10.	SET 154
	Y=0.0	SET 155
	CALL FPLOT(-2,X,Y)	SET 156
	CALL FPLOT(-1,X1,Y1)	SET 157
100	CONTINUE	SET 158
105	CONTINUE	SET 159
106	CONTINUE	SET 160
	CALL FPLOT(3,0.,0.)	SET 161
	STOP	SET 162
	END	SET 163

C PROGRAM TO OBTAIN INTERSECTION OF GRAPHITE-GAS AND IRON-IRON-OXIDE BO	INTER 001
C UNDARY	INTER 002
EXTERNAL TCT2	INTER 003
COMMON CO,C1,C2,C3,C4,C5,C6, EPS,IEND,XST	INTER 004
IEND=1000	INTER 005
EPS=.001	INTER 006
KK=9	INTER 007
DO 2 K=1,KK	INTER 008
READ(2,4) TEMP,PRES,ROH,PH2 ,NN	INTER 009
CALL EQKS(TEMP,EQCH4,EQCO2,EQH20,EQH2,EQCO,EQC,EQFE3,EQFE2,E	INTER 010
1QCO1,EQH21)	INTER 011
4 FORMAT(4F16.8,I3)	INTER 012
DO 1 N=1,NN	INTER 013
XST=PH2	INTER 014
A=(-PRES*EQCH4)	INTER 015
B=2.*ROH	INTER 016
CC=1.-2.*ROH	INTER 017
D=1.+2.*ROH	INTER 018
AA=(EQCH4*D-PRES*EQH20*CC)	INTER 019
AB=(EQCH4**2.)*(1.+4.*ROH)+EQH20	INTER 020
AC=EQCH4*EQH20*D	INTER 021
AD=(-EQCO2*(EQCH4**2.))/(4.*EQH20)	INTER 022
AE=(((-EQCH4*EQCO2)/4.)*(3.-2.*ROH)-2.*EQCH4*ROH)	INTER 023
AF=((-EQCO2*EQH20/4.)*CC*(3.+2.*ROH)-4.*ROH*(EQCH4**2.)-2.*EQH20*	INTER 024
1ROH*D)	INTER 025
AG=((-EQCO2/(4.*EQCH4))*((EQH20**2.)*(CC**2.)*D)-4.*EQCH4*EQH20*	INTER 026
1ROH*D)	INTER 027
CO=(A**2.)+AD*A	INTER 028
C1=2.*A*AA+AD*AA+A*AE	INTER 029
C2=2.*A*AB+(AA**2.)+AD*AB+AA*AE+A*AF+(EQCH4**2.)*(ROH**2.)	INTER 030
C3=2.*A*AC+2.*AA*AB+AD*AC+AB*AE+AF*AA+A*AG+2.*(ROH**2.)*EQCH4*	INTER 031
1EQH20*D+4.*(ROH**2.)*(EQCH4**3.)	INTER 032
C4=2.*AA*AC+(AB**2.)+AC*AE+AF*AB+AG*AA+(EQH20**2.)*(ROH**2.)*	INTER 033
1(D**2.)+4.*(EQCH4**4.)*(ROH**2.)+8.*(ROH**2.)*(EQCH4**2.)*(EQH20	INTER 034
1)*D	INTER 035

C5=2.*AB*AC+AF*AC+AB*AG+8.*(EQCH4**3.)*(EQH20)*(ROH**2.)*D	INTER 036
1+4.*(ROH**2.)*EQCH4*(EQH20**2.)*(D**2.)	INTER 037
C6=AC**2.+AG*AC+4.*(ROH**2.)*(EQCH4**2.)*(EQH20**2.)*(D**2.)	INTER 038
45 CALL IEOEQ (X,F,DERF,TCT2,XST,EPS,IEND,IER)	INTER 039
IF(IER=1)42,43,44	INTER 040
43 IEND=10000	INTER 041
EPS=.001	INTER 042
GO TO 45	INTER 043
44 XST=XST+.01	INTER 044
GO TO 45	INTER 045
42 PH2=X	INTER 046
PCH4=(PH2**2.)*EQCH4	INTER 047
EQ1=EQH20*(PH2**3.)/(PCH4)	INTER 048
PH20=(2.-2.*(PH2+EQCH4*(PH2**2.))-2.*ROH*(PH2+2.*EQCH4*(PH2**2.	INTER 049
1)))/(2.*ROH+(1./EQ1)+1.)	INTER 050
PCO=PCH4*PH20/(EQH20*(PH2**3.))	INTER 051
PCO2=PH20*PCO/(PH2*EQCO2)	INTER 052
RO=(PCH4+PCO+PCO2)/(2.*(PH2+PH20)+4.*PCH4)	INTER 053
C7=-EQH2*PRES*(EQCO+1.)	INTER 054
C8=(2.*RO+1.)*(EQH2+1.)*(EQCO+1.)	INTER 055
C9=(4.*RO-1.)*EQH2*EQCO*EQH20*(EQH2*PRES)	INTER 056
C10=(1.-2.*RO)*(EQH2+1.)*EQH2*EQCO*EQH20	INTER 057
PH =-(C7/C8+(C9/C8)*PH2**2.+(C10/C8)*PH2**3.)	INTER 058
IF(ABS(PH2-PH)-.004)3,3,1	INTER 059
1 ROH=ROH+.001	INTER 060
3 WRITE(3,5)	INTER 061
5 FORMAT(3X,'TEMP',7X,'PRES',7X,'ROH',7X,'PH2',7X,'PH20',7X,'PCH4	INTER 062
1',7X,'PCO',7X,'PCO2')	INTER 063
WRITE(3,7)TEMP,PRES,ROH,PH,PH20,PCH4,PCO,PCO2	INTER 064
7 FORMAT(2X,F6.1,5X,F3.1,6X,F5.3,5X,F8.6,5X,F8.6,5X,F8.6,6X,F8.6,5	INTER 065
1X,F8.6)	INTER 066
2 CONTINUE	INTER 067
STOP	INTER 068
END	INTER 069

C PROGRAM FOR PROCESS-PARAMETERS STUDY, EQUILIBRIUM ASSUMED	PROC 001
C READING IN FIXED CONDITIONS	PROC 002
NN=9	PROC 003
DO 1 I=1,NN	PROC 004
READ(2,2) TEMP1,PRES,ROH,CAA,CDD,HDD	PROC 005
2 FORMAT(6F10.3)	PROC 006
C CALL DATA SWITCH TO ALTER SHIFT REACTOR TEMPERATURE RELATIVE TO BOSCH	PROC 007
CALL DATSW(7,J)	PROC 008
IF(J-1)16,33,26	PROC 009
33 WRITE(1,24)	PROC 010
24 FORMAT(5X,'PUNCH IN DESIRED SHIFT TEMPERATURE')	PROC 011
READ(6,25) TEMP	PROC 012
25 FORMAT(F6.1)	PROC 013
CALL EQKS(TEMP,EQCH4,EQCO2,EQH2O,EQH2,EQCO,EQC,EQFE3,EQFE2,E	PROC 014
1QCO1,EQH21)	PROC 015
GO TO 30	PROC 016
26 TEMP=TEMP1	PROC 017
CALL EQKS(TEMP,EQCH4,EQCO2,EQH2O,EQH2,EQCO,EQC,EQFE3,EQFE2,E	PROC 018
1QCO1,EQH21)	PROC 019
C DETERMINE EXIT COMPOSITIONS OUT OF THE SHIFT REACTOR ASSUMING T,P,CDD,	PROC 020
C HDD,ROH,CAA	PROC 021
C CALL DATA SWITCH USED TO INITIATE SHIFT HYDROGEN RECYCLE	PROC 022
30 CALL DATSW(6,J)	PROC 023
IF(J-1)16,17,18	PROC 024
16 WRITE(3,20)	PROC 025
20 FORMAT(5X,'PROBLEM IN DATSW(6,J)')	PROC 026
GO TO 23	PROC 027
17 WRITE(1,21)	PROC 028
21 FORMAT(5X,'PUNCH IN HYDROGEN RECYCLE FOR SHIFT REACTOR')	PROC 029
READ(6,22) HS	PROC 030
22 FORMAT(F5.2)	PROC 031
HDD=HS+HDD	PROC 032
GO TO 19	PROC 033
18 HS=0.0	PROC 034
19 A=EQCO2-1.	PROC 035

B=EQCO2*(HDD+CDD) *(-1.)	PROC 036
C=EQCO2*HDD*CDD	PROC 037
WN=(-(B)-SQRT((B**2.)-4.*A*C))/(2.*A)	PROC 038
IF(WN)3,4,4	PROC 039
4 IF(WN-1.)5,5,3	PROC 040
3 WN=(-(B)+SQRT((B**2.)-4.*A*C))/(2.*A)	PROC 041
WRITE(3,6)	PROC 042
6 FORMAT(10X,'PROBLEM IN STATEMENT NUMBER 4')	PROC 043
5 HAA=2.*CAA	PROC 044
CCCP=CAA	PROC 045
HMM=WN	PROC 046
WJJ=HAA-HMM	PROC 047
C READ IN THE GRAPHITE-GAS, IRON-IRON OXIDE PHASE BOUNDARY INTERSECTION	PROC 048
READ(2,7) PH2,PCO2,PCH4,PCO,PH2O	PROC 049
7 FORMAT(5F8.6)	PROC 050
WII=WJJ	PROC 051
TOT=WII/PH2O	PROC 052
HII=PH2*TOT	PROC 053
CII=PCO2*TOT	PROC 054
AMII=PCH4*TOT	PROC 055
COII=PCO*TOT	PROC 056
AMHH=AMII	PROC 057
CON=WN	PROC 058
HHH=HDD-WN+HII	PROC 059
CHH=CDD-CON+CII	PROC 060
COHH=CON+COII	PROC 061
HFF=HAA-HDD	PROC 062
CFF=CAA-CDD	PROC 063
AMHH=AMII	PROC 064
ROHS=CDD/HDD	PROC 065
HNN=HDD-WN	PROC 066
CNN=CDD-WN	PROC 067
AMN=0.0	PROC 068
H=HHH+CHH+COHH+HFF+CFF+AMHH	PROC 069
WRITE(3,8)	PROC 070

8	FORMAT(10X,'SHIFT REACTOR')	PROC 071
	WRITE(3,9)	PROC 072
9	FORMAT(10X,'CO2+H2=CO+H2O',31X,'INLET',29X,'OUTLET')	PROC 073
	WRITE(3,10)	PROC 074
10	FORMAT(8X,'TEMP',6X,'PRES',7X,'O/H',4X,'RECYCLE',6X,'PH2',6X,'PCO2	PROC 075
	1' ,7X,'PH2',6X,'PCO2',7X,'PCO',6X,'PH2O',6X,'PCH4')	PROC 076
	WRITE(3,11) TEMP,PRES,ROHS,HS,HDD,CDD,HNN,CNN,CON,WN,AMN	PROC 077
11	FORMAT(5X,11F10.5,/')	PROC 078
	WRITE(3,12)	PROC 079
12	FORMAT(10X,'BOSCH REACTOR')	PROC 080
	WRITE(3,13)	PROC 081
13	FORMAT(10X,'CO2+2H2=2H2O+C',30X,'INLET',29X,'OUTLET')	PROC 082
	WRITE(3,14)	PROC 083
14	FORMAT(8X,'TEMP',6X,'PRES',7X,'O/H',4X,'RECYCLE',2X,'OUTPUT FROM S	PROC 084
	1HIFT-H2O',4X,'PH2',6X,'PCO2',7X,'PCO',6X,'PH2O',6X,'PCH4')	PROC 085
	WRITE(3,15) TEMP1,PRES,ROH,H,PH2,PCO2,PCO,PH2O,PCH4	PROC 086
15	FORMAT(5X,4F10.5,20X,5F10.5,////)	PROC 087
	1 CONTINUE	PROC 088
23	CONTINUE	PROC 089
	CALL EXIT	PROC 090
	END	PROC 091

7.5 Nomenclature

a_j	activity of species, j
A	area, cm^2
C_j	concentration of species j, moles/ cm^3
C_o	concentration of reducible oxygen in oxide, g-atoms O/ cm^3
d	diameter of steel wool fiber, cm
$D_j^{(eff)}$	effective diffusivity of species j in porous product layer, cm^2/s
$D_j^{(P)}$	diffusivity of species j in a single pore, cm^2/s
$D_{j,k}$	binary gaseous diffusion coefficient, cm^2/s
$D_{j,m}$	gas phase diffusivity of j into multi-component mixture, cm^2/s
G_∞	approach flowrate of reducing gas, moles/ $\text{cm}^2\text{-s}$
k	Bohzmann's constant
$k_{m(j)}$	mass-transfer coefficient for species j, cm/s
k_{ov}	over-all conductance, cm/s
$k_r^{(t)}$	specific rate constant for surface reaction forming Product t, cm/s
$K_a^{(j)}$	adsorption equilibrium constant for species j, cm^3/mole
$K_e^{(t)}$	equilibrium constants for s/t equilibrium $P_{H_2O}/P_{H_2} \Big _{eq} \quad \text{or} \quad P_{CO_2}/P_{CO} \Big _{eq}$
L	thickness of slab
M_j	molecular weight of species j, g/g-mole
M_{wt}	average molecular weight of 5 component reactant

	gas, g/g-mole
\dot{N}_j	molar flow of species j, moles/s
N_{Re_d}	Reynold's number based on catalyst fiber diameter, $d \underline{v} \rho / \mu$
N_{Sc}	Schmidt number, $\mu / \rho D_{j,k}$ {subscript (f) = film {subscript (w) = wall
N_{Sh}	Sherwood number, $d k_{m,j} / D_{j,k}$
P_j	partial pressure of species j, atm (with super- script denoting where located)
P_T	total pressure, atm
r_o	reduction rate, g-atoms O/cm ² s
R	resistance to molar current of gas, (moles/s atm) ⁻¹ (with identifying superscripts)
R_g	gas constant, 82.1 atm cm ³ /g-mole K
t	time, s
T	temperature, Kelvin unless otherwise specified
T^*	$k T / \epsilon_o$
\underline{v}	free stream velocity, cm/s
V	volume of unreacted core, cm ³
V_m	unit volume of gas mixture
x_i	mole fraction of i in 5 component gas mixture
x_o	external radius of oxide sphere, cm
$x_i^{(t)}$	effective interface radius, distance from the origin to inner boundary of the indicated product phase t, cm
\underline{X}	overall conversion

Greek

β_n	defined as $\text{Cos}\beta_n = \beta_n (2Ds/k_{m,j} L)$
ϵ_{Fe}	void fraction in reduced iron layer
ϵ_o	energy-potential parameter
ρ_{mix}	density of mixture, g/cm^3
σ	Lennary-Jones force constant
θ	reaction time, s
μ_i	viscosity of component i, g/cm s
μ_{mix}	viscosity of mixture, g/cm s
Ω_D	collision integral for diffusion
Ω_T	collision integral for viscosity at temperature T

Superscripts

b	Associates principle symbol with the bulk gas phase
Fe	Associates principle symbol with the iron phase
h	Associates principle symbol with the hematite phase
m	Associates principle symbol with the magnetite phase
s	Associates principle symbol with the reactant phase in a given reduction step, s \rightarrow t
t	Associates principle symbol with the product phase in a given reduction step, s \rightarrow t
w	Associates principle symbol with the Wustite phase

Subscripts

A	Designates reactant gas, H_2 or CO
B	Designates product gas, H_2O or CO_2
F	Associates principle symbol with external mass- transfer resistance

- I Associates principle symbol with interface reaction
- j,k Designates chemical species j, k
- S Associates principle symbol with a shell layer reaction product -- Associates principle symbol with solid phase

7.6 Literature References

Baker, R.T.K., and P.S. Harris, "Controlled Atmosphere Electron Microscopy", J. Phys., Sect. E, Scientific Instruments, 5, 793 (1972).

Baker, R.T.K., M.A. Barber, P.S. Harris, F.S. Feates, and R.J. Waite, "Nucleation and Growth of Carbon Deposits from the Nickel Catalyzed Decomposition of Acetylene", J. Catalysis, 26, 51 (1972).

Baker, R.T.K., P.S. Harris, R.B. Thomas, and R.J. Waite, "Formation of Filamentous Carbon from Iron, Cobalt, and Chromium Catalyzed Decomposition of Acetylene", J. Catalysis, 30, 86 (1973).

Baker, R.T.K., and R.J. Waite, "Formation of Carbonaceous Deposits from the Platinum-Iron Catalyzed Decomposition of Acetylene", J. Catalysis, 37, 101 (1975).

Barkley, L.W., T.E. Corrigan, H.W. Wainwright, and A.E. Sands, "Catalytic Reverse Shift Reaction: A Kinetic Study", Ind. Eng. Chem., 44, 1066 (1952).

Boehm, H.P., "Carbon from Carbon Monoxide Disproportionation on Nickel and Iron Catalysts: Morphological Studies and Possible Growth Mechanisms", Carbon, 11, 583 (1973).

Bortolini, P., "Kinetics of the Water-Gas Conversion Reaction", Chem. Eng. Sc., 9, 135 (1958).

Browning, L.C., T.W. DeWitt, and P.H. Emmett, "Equilibria in the Systems $\text{Fe}_2\text{C-Fe-CH}_4\text{-H}_2$ and $\text{Fe}_3\text{C-Fe-CH}_4\text{-H}_2$ ", J. Am. Chem. Soc., 72, 4211 (1950).

Browning, L.C., and P.H. Emmett, "Equilibrium Measurements in the System $\text{C-CH}_4\text{-H}_2$ ", J. Am. Chem. Soc., 73, 581 (1951).

Cahn, R.W., "Physical Metallurgy Principles", D. Van Nostrand Co., New York (1973).

Chatterjee, B., and P.P. Das, "Nature of the Catalyst in the Decomposition of Carbon Monoxide in the Presence of Iron", Nature, 173, 1046 (1954).

Dal Nogare, S., and R.S. Juvet, "Gas-Liquid Chromatography, Theory, and Practice", Wiley, New York (1962), pp. 257-264.

Das, P.P., and B. Chatterjee, "Iron as a Catalyst in the Decomposition of Carbon Monoxide", Trans. Indian Inst. Metals, 6, 279 (1952).

Das, P.P., and B. Chatterjee, "The Nature of the Catalyst in the Decomposition of Carbon Monoxide in the Presence of Iron", Trans. Indian Inst. Metals, 7, 189 (1955).

Dietz, W.A., "Response Factors for Gas Chromatographic Analyses", J. Gas Chromatog., 5, 68 (1967).

Douglas, W.Y.M., and S.W. Churchill, "Recorrelation of Data for Convective Heat Transfer Between Gases and Single

Cylinders with Large Temperature Differences", AIChE Heat Transfer Symp., Nat'l. Meeting, Louisville, Ky., 1955, Reprint 16.

Duggin, M.J., and L.J.E. Hofer, "Nature of χ -Iron Carbide", *Nature*, 212, 248 (1966).

Eckert, E.R.G., and R.M. Drake, Jr., "Analysis of Heat and Mass Transfer", McGraw-Hill Book Co., New York, 1972.

Eckstrom, H.C., and W.A. Adcock, "A New Iron Carbide in Hydro-carbon Synthesis Catalysts", *J. Am. Chem. Soc.*, 72, 1042 (1950).

Edstrom, J.O., and G. Bitsianes, *Trans. Am. Inst. Mining Eng.*, 203, 760 (1955).

Elliott, R.P., "Constitution of Binary Alloys, First Supplement", McGraw-Hill, New York (1965).

Evans, E.L., J.M. Thomas, P.A. Thrower, and P.L. Walker, "Growth of Filamentary Carbon on Metallic Surfaces During the Pyrolysis of Methane and Acetone", *Carbon*, 11, 441 (1973).

Everett, M.R., "The Kinetics of Carbon Deposition Reactions in High-Temperature Gas-Cooled Reactors", D.P. Report 500, Atomic Energy Establishment, Winfrith (January 1967).

Fleureau, B., "Sur la formation de graphite par decomposition catalytique de l'oxyde de carbone", *Compt. rend.*, 237, 330 (1953).

Garmirian, J.E. and R.C. Reid, 10.90 Project Report, Massachusetts Institute of Technology, Massachusetts (1977).

Hansen, M., editor, "Constitution of Binary Alloys", 2nd ed., McGraw-Hill, New York (1958).

Herbstein, F.H., and J.A. Snyman, "Identification of Eckstrom-Adcock Iron Carbide as Fe_7C_3 ", Inorg. Chem., 3, 894 (1964).

Hofer, L.J.E., E.M. Cohn, and W.C. Peebles, "The Modifications of the Carbide, Fe_2C ; Their Properties and Identification", J. Am. Chem. Soc., 71, 189 (1949).

Hollis, O.L., and M.V. Hayes, "Water Analysis by Gas Chromatography Using Porous Polymer Columns", J. Gas Chromatography, 4, 235 (1966).

Holmes, R.F., E.E. Keller, and C.D. King, "A Carbon Dioxide Reduction Unit Using Bosch Reaction and Expendable Catalyst Cartridges", NASA CR-1682, (November 1970).

Jack, K.H., "Iron-Nitrogen, Iron-Carbon, and Iron-Carbon-Nitrogen Interstitial Alloys: Their Occurrence in Tempered Martenstite", Nature, No. 4002, 61 (July 13, 1946).

Jack, K.H., and S. Wild, "Nature of χ -Carbide and Its Possible Occurrence in Steels", Nature, 212, 248 (1966).

Karcher, W., and P. Glaude, "Inhibition of Carbon Deposition on Iron and Steel Surfaces", Carbon, 9, 617 (1971).

Kawasaki, E., J. Sanscraninte, T.J. Walsh, "Kinetics of

Reduction of Iron Oxide with Carbon Monoxide and Hydrogen", A.I.Ch.E. J., 8, 1, 48 (1962).

Kohn, J.A., and D.W. Eckart, "X-Ray Study of Synthetic Diamond and Associated Phases", Am. Mineralogist, 47, 1422 (1962).

Kusner, R.E., "Kinetics of the Iron Catalyzed Reverse Water-Gas Shift Reaction", Ph.D. Thesis, Case Institute of Technology, Ohio (1962).

Landler, P.F.J., and K.L. Komarek, Trans. Met. Soc., AIME, 236, 138 (1970) As referenced by Szekely, #27.

Louw, J.D., J.P. Van Den Berg, L.C. Ferreira, and J.J. Pienaar, "Appearance of FeC in a Hydrocarbon Synthesis Catalyst", J. Am. Chem. Soc., 79, 5899 (1957).

MacIver, D.S., and P.H. Emmett, "Surface Area Measurements on Carbon Black Produced by the Catalytic Decomposition of Carbon Monoxide Over Iron", J. Phys. Chem., 59, 1109 (1955).

Manning, M.P., "An Investigation of the Bosch Process", Massachusetts Institute of Technology, Sc.D. (1976).

Manning, M.P., and R.C. Reid, "Carbon Dioxide Reduction by the Bosch Process", Inter-American Congress of Chemical Engineering, Caracas, Venezuela, July 13-16, 1975.

Mars, P., "Factors Governing the Behavior of the Adiabatic Water-Gas Shift Reactor", Chem. Eng. Sci., 14, 375 (1961).

McCarthy, J.T., L.J.E. Hofer, B. Seligman, J.A. Lecky, W.C. Peebles, and R.B. Anderson, "Electron and X-Ray Diffraction Studies of Iron Fischer-Tropsch Catalysts", J. Phys. Chem., 57, 730 (1953).

McKewan, W.M., Trans. Met. Soc., AIME, 224, 2-5 (1962).

Meissner, H.P., and R.C. Reid, "The Bosch Process", Environmental Control & Life Support Systems Conference, San Francisco, California, August 14-16, 1972.

Muan, A., and E.F. Osborn, "Phase Equilibria Among Oxides in Steel Making", Addison-Wesley Publishing Co., Reading, Massachusetts (1965).

Nabi, G. and W.K. Lu, "Reduction Kinetics of Hematite to Magnetite in Hydrogen-Water Vapor Mixtures", Trans. Met. Soc., AIME, 242, 2471 (1968).

Nelson, H.R., "The Primary Oxide Film on Iron", J. Chem. Phys., 5, 252 (1937).

Podgurski, H.H., J.T. Kummer, T.W. DeWitt, and P.H. Emmett, "Preparation, Stability, and Adsorptive Properties of the Carbides of Iron", J. Am. Chem. Soc., 72, 5382 (1950).

Podolski, W.F., and Y.G. Kim, "Modeling the Water-Gas Shift Reaction", Ind. Eng. Chem. Proc. Des. Develop., 13, 415 (1974).

Purcell, J.E., and L.S. Ettre, "Analysis of Hydrogen with Thermal Conductivity Detectors", J. Gas Chromatog., 3,

69 (1965).

Ratliff, J.T., "Early Stages in the Interaction of Carbon Monoxide with Single Crystal Films", Ph.D. Thesis, Georgia Institute of Technology.

Reed-Hill, R.E., "Physical Metallurgy Principles", D. Van Nostrand Co., 2nd Ed., New York (1973).

Reid, R.C., and T.K. Sherwood, "The Principles of Gases and Liquids", McGraw-Hill, 2nd Ed. (1966).

Renshaw, G.D., C. Roscoe, and P.L. Walker, Jr., "Disproportionation of CO Over Iron and Silicon-Iron Single Crystals", J. Catalysis, 18, 164 (1970).

Robertson, S.D., "Carbon Formation from Methane Pyrolysis Over Some Transition Metal Surfaces I. Nature and Properties of the Carbon Formed", Carbon, 8, 365 (1970).

Rossini, R.D., Ed., "Selected Values of Chemical Thermodynamic Properties", Circular of the National Bureau of Standards 500 (1961).

Ruston, W.R., M. Warzee, J. Hennaut, and J. Waty, "The Solid Reaction Products of the Catalytic Decomposition of Carbon Monoxide on Iron at 550°C", Carbon, 7, (1969).

Ruston, W.R., M. Warzee, J. Hennaut, and J. Waty, "Basic Studies on the Growth of Carbon Deposition from Carbon Monoxide on a Metal Catalyst", D.P. Report 394, Atomic Energy Establishment, Winfrith (1966).

Ruthven, D.M., "A Simple Method of Calculating Mass Transfer Factors for Heterogeneous Catalytic Gas Reactors", Chem. Eng. Sci., 23, 759 (1968).

Sherwood, T.K., and R.L. Pigford, "Absorption and Extraction", McGraw-Hill, New York (1952).

Shunk, F.A., "Constitution of Binary Alloys, Second Supplement", McGraw-Hill, New York (1969).

Spitzer, R.H., F.S. Manning, and W.O. Philbrook, "Generalized Model for the Gaseous Topochemical Reduction of Porous Hematite Spheres", Trans. Met. Soc., AIME, 236, 1715 (1966).

Spitzer, R.H., F.S. Manning, and W.O. Philbrook, "Mixed-Control Reaction Kinetics in the Gaseous Reduction of Hematite", Trans. Met. Soc., AIME, 236, 726 (1966).

Stricoff, R.S., "The Reduction of Carbon Dioxide with Hydrogen Over Iron Catalysts", S.M. Thesis, Massachusetts Institute of Technology (1972).

Stull, D.R., and H. Prophet, "JANAF Thermochemical Tables", 2nd Ed., National Bureau of Standards, Washington (1971).

Szekely, J., J.W. Evans, and H.Y. Sohn, "Gas-Solid Reactions", Academic Press (1976).

Tamar, Y., Y. Nishiyama, and M. Takahashi, "The Effects of Helium Upon Carbon Deposition Onto Metal Surfaces", Carbon,

7, 209 (1969).

Tesner, P.A., E.Y. Rabinovich, I.S. Rafalkes, and E.F. Arefieva, "Formation of Carbon Fibers from Acetylene", Carbon, 8, 435 (1970).

Tevebaugh, A.D., and E.J. Cairns, "CHO Gas Phase Compositions in Equilibrium with Carbon and Carbon Deposition Boundaries at One Atmosphere", J. Chem. Eng. Data, 9, 453 (1964).

Tevebaugh, A.D., and E.J. Cairns, "Carbon Deposition Boundaries in the CHO System at Several Pressures", J. Chem. Eng. Data, 10, 359 (1965).

Todorov, R.P., D.K. Lambieu, and St. M. Mechkova, "Effect of Carbide-Forming Elements and Their Compounds on the Catalytic Action of Iron in Bell's Reaction", Kenetika i Kataliz, 15, 687 (1974).

Tsao, T., "Kinetics of Dissociation of Carbon Monoxide on α -Fe", Carnegie-Mellon University, Ph.D. (1974).

Tsay, Q.T., W.H. Ray, and Julian Szekely, "The Modeling of Hematite Reduction with Hydrogen Plus Carbon Monoxide Mixtures". Part I "The Behavior of Single Pellets", AIChE J., 22, 6, 1064 (1976). Part II, "The Direct Reduction Process in a Shaft Furnace Arrangement", Ibid., 1072 (1976).

Turkdogan, E.T., and J.V. Vinters, "Catalytic Oxidation of Carbon", Carbon, 10, 97111 (1971).

Van Vlack, L.H., "Elements of Materials Science", 2nd Ed., Addison-Wesley Publishing Co., Reading, Massachusetts (1964).

Walker, P.L., Jr., J.F. Rakszawski, and G.R. Imperial, "Carbon Formation from Carbon Monoxide-Hydrogen Mixtures Over Iron Catalysts I. "Properties of Carbon Formed", J. Phys. Chem., 73, 133 (1959), II. "Rates of Carbon Formation", Ibid., 140 (1959).

Warzee, M., C. Sonnen, J. Hennaut, and W.R. Ruston, "Study of Catalytic Decomposition of Carbon Monoxide on Pure Iron, Mild Steels, and Alloyed Steels", D.P. Report 530, Atomic Energy Establishment, Winfrith (1967).

Watase, T., J. Chem. Soc. Japan, 54, 110 (1933).

West, C.A., "Precipitation from Solid Solutions of C and N in α -Iron", J. App. Phys., 20, 943 (1949).

Wert, C.A., "Diffusion Coefficient of C in α -Iron", Phys. Rev., 79, 601 (1950).

Westerman, R.V., "Mechanism and Kinetics of Iron Deterioration in Carbon Monoxide", Ph.D. Thesis, Georgia Institute of Technology (1967).

Wilson, R.B., "Fundamental Investigation of the Bosch Reaction", S.M. Thesis, Massachusetts Institute of Technology (1971).