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Tunable Raman spectroscopy study of CVD and peapod-derived bundled and individual double-wall carbon nanotubes

F. Villalpando-Paez,1 L. G. Moura,2 C. Fantini,2 H. Muramatsu,3 T. Hayashi,3 Y. A. Kim,3 M. Endo,3 M. Terrones,4 M. A. Pimenta,2 and M. S. Dresselhaus5

1Department of Materials Science and Engineering, Massachusetts Institute of Technology, Cambridge, Massachusetts 02139-4307, USA
2Departamento de Física, Universidade Federal de Minas Gerais, Belo Horizonte, Minas Gerais, Brazil
3Faculty of Engineering, Shinshu University, 4-17-1 Wakasato, Nagano-shi 380-8553, Japan
4Department of Materials Science and Engineering and Chemical Engineering, Carlos III University of Madrid, Avenida Universidad 30, 28911 Legans, Madrid, Spain
and Exotic Nanocarbon Research Center, Shinshu University, 4-17-1 Wakasato, Nagano city 380-8553, Japan
5Department of Physics and Department of Electrical Engineering and Computer Science, Massachusetts Institute of Technology, Cambridge, Massachusetts 02139-4307, USA

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We use 40 laser excitation energies to analyze the differences in the Raman spectra from chemical vapor deposition-derived double-wall carbon nanotube (CVD-DWNT) bundles, fullerene-derived DWNT bundles (C60-DWNTs), and individual fullerene-derived DWNTs with inner type-I and type-II semiconducting tubes paired with outer metallic tubes. For the radial breathing mode (RBM) of SWNTs, an experimental $\omega_{RBM}$ vs $d_t$ relationship of the form $\omega_{RBM}=A/d_t+B$ is obtained for the inner tubes of DWNT bundles where the $A$ and $B$ constants are found to be close to those obtained by elasticity theory when modeling the elastic properties of graphite. A similar change in $\omega_{RBM}$ is observed for an inner type-II semiconducting (6,5) tube and a type-I (9,1) tube when inside various metallic outer tubes. The G-band frequency is observed to upshift when switching the laser resonance from DWNTs with semiconducting inner tubes to DWNTs with metallic inner tubes. Finally, we measure the $G'$ feature from C60-DWNTs and note a downshift in frequency with respect to that of CVD-DWNTs.

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I. INTRODUCTION

Double-wall carbon nanotubes (DWNTs) are of scientific interest because they provide the simplest system in which to study the interaction between two concentric carbon nanotubes. Furthermore, the inner and outer tubes of a DWNT can be either metallic (M) or semiconducting (S) so that four different configurations of DWNTs are possible (M@M, S@S, M@S, and S@M). The electronic and optical properties of a DWNT are dependent on the individual properties of its constituent inner and outer tubes and on intertube interactions. Since different DWNT synthesis processes yield different inner-outter tube pairs, the electronic and optical properties of current DWNT samples are highly dependent on the fabrication procedure. The two main fabrication procedures used to synthesize DWNTs include the following: (a) chemical vapor deposition (CVD) (DWNTs)3 or (b) the filling of SWNTs with fullerenes where a second inner tube is created by heat-induced coalescence of the fullerenes (C60-DWNTs).2,3 The CVD-DWNTs and C60-DWNTs have been reported to have some differences in their electronic and optical properties due to differences in the crystallinity of the resulting inner tubes and in the characteristic distance between the inner and outer tubes.4 For instance, C60-DWNTs have been observed to have smaller average wall to wall distances (defined as the distance between the inner and outer walls of the DWNT) than CVD-DWNTs for DWNTs with similar inner diameters. Also, the photoluminescence (PL) emission from the inner tubes of DWNTs appears to be highly dependent on sample fabrication and purity because both high and low intensity PL signals from the inner tubes of DWNTs have been reported for CVD-DWNTs5 and C60-DWNTs,6 respectively. The fact that current DWNT synthesis procedures yield samples that contain a mixture of DWNTs with all four of the configurations listed above further complicates Raman data analysis because the Raman spectra from DWNT bundles contain contributions from most of the four possible DWNT configurations. Thus, it is difficult to quantitatively determine the actual $(n,m)$ indices of the constitutive inner-outer tube pairs present in a DWNT bundle. In an effort to study DWNTs with a known configuration, we have performed tunable Raman experiments on DWNT bundles and complemented these studies with careful studies at the individual DWNT nanotube level capable of finding Raman resonances with both the inner and outer tubes of the same DWNT. Herein, we analyze the differences between the Raman spectra obtained from CVD-derived and C60-derived DWNT bundles with 40 laser excitation energies. We also compare the radial breathing mode (RBM) features from individual DWNTs with the RBM features from DWNT bundles.

II. EXPERIMENTAL DETAILS

We used a CVD method to synthesize the CVD-DWNT bundles used in this study.1 In order to remove unwanted catalyst residues, the sample was heat treated in an HCl solution at 100 °C for 10 h. Also, since DWNTs can withstand higher heat treatment temperatures than SWNTs, oxidation in air at 500 °C for 30 min was used to remove amorphous

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carbon and impurities in the form of SWNTs. The high purity of our CVD-DWNT bundles relative to residual catalyst particles was confirmed by diamagnetic susceptibility measurements, and the purity with regard to the absence of SWNTs was confirmed by transmission electron microscopy studies. The \( \text{C}_{60}\)-DWNT bundles used herein were fabricated by reacting \( \text{C}_{60}\) and SWNTs under vacuum conditions at 600 °C for 24 h. The starting SWNT material was synthesized by the arc method and purified by Hanwha Corporation (Korea). The as-grown peapods (SWNTs containing encapsulated \( \text{C}_{60}\) fullerenes) were washed with toluene to remove residual \( \text{C}_{60}\) and heat treated at 1700 °C in Ar (1 atm) to transform them into \( \text{C}_{60}\)-DWNT bundles. The individual \( \text{C}_{60}\)-DWNTs were obtained by dispersing 1 mg of \( \text{C}_{60}\)-DWNTs in \( \text{D}_{2}\text{O} \) with 0.5 wt % sodium dodecylbenzenesulfonate and sonicating the dispersion (600 W cm\(^{-2}\)) for 15–30 min. The solution was then ultracentrifuged (320 000 g) for 30 min, and 70% of the supernatant solution was collected. The supernatant solution containing the isolated \( \text{C}_{60}\)-DWNTs was spin coated on a Si chip with gold markers, and a Raman mapping technique described elsewhere was used to obtain Raman resonance spectra from the inner and outer layers of the same individual DWNT. For bundles, the laser power levels were kept below 0.5 mW to avoid excessive heating of the DWNTs. The scattered light was collected through a 100× objective using a backscattering geometry. A Nd:YAG laser was used to generate \( E_{\text{laser}} = 2.33 \text{ eV} \) light and a Kr+ ion laser generated \( E_{\text{laser}} = 1.92 \text{ eV} \). An argon laser working in the multimode configuration (6 W) was used to pump a dye laser containing 4-dicyanomethylene-2-methyl-6-p-dimethylamino-styryl-4H-pyran, rhodamine 6G, and R560 dyes to generate a range of laser energies \( E_{\text{laser}} = 1.89–1.99 \text{ eV} \), \( E_{\text{laser}} = 2.05–2.09 \text{ eV} \), and \( E_{\text{laser}} = 2.24–2.30 \text{ eV} \), respectively. To generate \( E_{\text{laser}} = 1.58–1.66 \text{ eV} \), a Ti:sapphire laser was pumped by the argon laser. A thermoelectrically cooled Si charge coupled device detector operating at \(-75^\circ\text{C}\) was used to collect the Raman spectra.

**III. RESULTS**

**A. RMB from CVD-DWNT and \( \text{C}_{60}\)-DWNT bundles**

We scanned both our CVD-DWNT and \( \text{C}_{60}\)-DWNT bundled samples with 40 different laser energies \( E_{\text{laser}} \) ranging from 1.58 to 3.23 eV and we measured the \( \omega_{\text{RBM}} \) values of each identifiable \((n,m)\) tube. The RMB contour maps (two-dimensional color plots of the RMB intensity for \( \omega_{\text{RBM}} \) vs \( E_{\text{laser}} \)) shown in Fig. 1 correspond to (a) CVD-DWNTs and (b) \( \text{C}_{60}\)-DWNTs, and both are superimposed on a SWNT-based Kataura plot of optical transition energy \( E_{\text{ii}} \) vs the frequency of the radial breathing mode. The Kataura plot was calculated within the extended tight binding framework including many-body corrections, and fitted to the resonance Raman scattering data from sodium dodecyl sulfate wrapped high pressure CO conversion SWNTs. A SWNT-based Kataura plot was found to be accurate enough for a qualitative identification of the \((n,m)\) indices of the inner and outer tubes of the DWNTs. In principle, each of the four possible DWNT configurations is expected to have unique electronic and optical properties and hence to produce a unique Kataura plot. By comparing the Raman maps from CVD-DWNTs and \( \text{C}_{60}\)-DWNTs in Fig. 1, we find, in agreement with previous work, a marked difference between the diameter distributions of the tubes contained in each type of sample. On one hand, the CVD-DWNT Kataura map reveals the presence of inner and outer tubes whose RMBs span the entire range from 140 to 350 cm\(^{-1}\). On the other hand, the \( \text{C}_{60}\)-DWNT Raman map does not show any resonant tubes in the 190–250 cm\(^{-1}\) range but only shows outer tubes with \( \omega_{\text{RBM}} \leq 190 \text{ cm}^{-1} \) and inner tubes with \( \omega_{\text{RBM}} > 250 \text{ cm}^{-1} \) [see Fig. 1(b)]. This apparent difference in diameter distributions between CVD-DWNT and \( \text{C}_{60}\)-DWNT bundles arises from the fact that, unlike the CVD-DWNTs, where both inner and outer tubes grow simultaneously, \( \text{C}_{60}\)-DWNTs experience different diameter constraints during growth. For the case of the \( \text{C}_{60}\)-DWNTs the possible diameters of the inner tubes are constrained by the diameters of the starting SWNT material (\( \approx 1.3–1.4 \text{ nm} \)) which forms the outer tubes of the DWNTs.

![Figure 1](http://example.com/figure1.png)
The relationship between \( \omega_{\text{RBM}} \) and \( d_i \) for the inner tubes and outer tubes of DWNTs is not expected to be the same as CVD-DWNTs. A list of the small diameter tubes \( (d_i < 0.9 \text{ nm}) \) whose \( (n,m) \) indices were unambiguously identified in the CVD-DWNT and C\(_{60}\)-DWNT samples is presented in Table I.

Further inspection of our DWNT bundle spectra reveals a splitting of the RBM from the inner tubes. The RBM splitting in DWNTs has been reported to occur when various inner tubes with a specific \( (n,m) \) are contained inside different outer tubes with various diameters and chiralities. When, for a given \( (n,m) \) inner tube, the diameter of the outer tube is decreased, the wall to wall distance decreases and the \( \omega_{\text{RBM}} \) of the inner tubes upshifts. Thus, a group of DWNTs, with the same \( (n,m) \) inner tube and different outer tubes, generates a cluster of RBM peaks and we refer to this effect above as a splitting of its \( \omega_{\text{RBM}} \).

In accordance with previous reports, we observe this RBM splitting effect to be more pronounced in the C\(_{60}\)-DWNTs than in the CVD-DWNTs. For instance, the cluster of RBM peaks generated by the (7,5) inner tubes (see Fig. 3) in the C\(_{60}\)-DWNT (CVD-DWNT) bundles contains eight (five) distinguishable RBM peaks spread over a 29 (22) cm\(^{-1} \) range. However, we also find that the RBM splitting effect is not necessarily always present for the inner tubes of DWNTs. For instance, the RBMs from the (6,4) inner tubes in C\(_{60}\)-DWNT bundles have been reported to span over a 30 cm\(^{-1} \) range, whereas in our DWNTs the RBMs from the (6,4) inner tubes for both CVD-DWNTs and C\(_{60}\)-DWNTs only split within a \(< 11 \text{ cm}^{-1} \) range [see Figs. 3(a) and 3(b)]. The small diameter (7,2) inner tube [see Fig. 3(b)] observed in the C\(_{60}\)-DWNTs also does not show any significant splitting. Thus, even when one would expect smaller diameter inner tubes to show increased splitting because they can fit inside a larger variety of DWNT outer tubes, energy minimizing processes that occur during sample preparation are key to defining the final \( (n,m) \) indices and wall to wall distances of the inner and outer tubes that constitute the DWNTs contained in a given sample.

Interestingly, the (7,2) inner tube does not appear in our CVD-DWNT sample [see Fig. 3(a)]. Figure 3 also shows that the inner tubes of the CVD-DWNTs have RBM peaks with larger full width at half maxima (FWHMs) than the C\(_{60}\)-DWNTs. This observation suggests that the phonon lifetimes in the inner tubes are longer in C\(_{60}\)-DWNTs than in the CVD-DWNTs. However, the intensity of the D band in C\(_{60}\)-DWNTs (not shown here) is greater than that in the CVD-DWNTs, indicating that the former have a larger defect concentration. In a conventional SWNT sample an increased defect concentration would be expected to give rise to more phonon relaxation pathways, decreased phonon lifetimes, and increased RBM FWHM values. The fact that C\(_{60}\)-DWNTs show both sharper RBM peaks and higher D-band intensities than their CVD counterparts suggests that either (a) the C\(_{60}\)-DWNT sample contains traces of unwanted amorphous carbon or (b) the majority of the defects in the
horizontal axes mark the expected possibility but since our sample is heat treated at a high temperature (1700 °C), the so-called corrugation effect is weak and the inner-outer tube interaction is the dominant factor affecting the number and frequency of the observed inner-tube RBMs.

B. RBM from individual C_{60}-DWNTs

In order to investigate the DWNT configurations contained in our DWNT bundle samples, we isolated DWNTs on a Si substrate and used a Raman mapping technique to obtain Raman resonance spectra from both the inner and outer tubes of the same individual DWNT. The individual DWNT specimens that we specially identified for this study are C_{60}-DWNTs with an S@M configuration, where the inner tubes are either (6,5), in resonance with $E_{\text{last}} = 2.10$ eV, or (9,1), in resonance with $E_{\text{last}} = 1.662$ eV. Although the (6,5) and (9,1) tubes are both contained inside M tubes and have almost the same diameter ($d_t = 0.74$ nm), the (6,5) tube is a type-II semiconductor and the (9,1) tube is a type-I semiconductor. Since these semiconducting type-I and type-II tubes have essentially the same diameter but a different chirality dependence of $E_{\text{last}}$, we considered that the interaction between each type of inner S tube with its corresponding outer M tube may vary somewhat. Figure 4 compares the RBM regions of CVD-DWNT bundles and C_{60}-DWNT bundles with the spectra from the individual S@M C_{60}-DWNTs whose inner tubes are (9,1) type I and (6,5) type II. As expected, the peaks from the inner tubes of the individual S@M DWNT tubes can be found within the frequency region that is occupied by the broader RBM peak generated by the DWNT bundle samples (see Fig. 4).

In the case of the (6,5) type-II tube, the $w_{\text{RBM}}$ from the individual tubes can be found in the 313–329 cm$^{-1}$ range where the highest RBM frequencies correspond to tubes that are paired with the smallest diameter outer tubes.

C_{60}^-DWNTs lie in the outer walls which themselves contain crystalline inner tubes that allow for long phonon lifetimes and produce the observed sharp RBM features.

Besides the above-mentioned RBM splitting caused by inner-outter tube interactions in DWNTs, a second mechanism independent of inner-outer tube interactions, known as corrugation of the inner tube, may be capable of generating extra peaks in the RBM region corresponding to the inner tubes of DWNTs. According to Muramatsu et al., the fullerenes inside a SWNT are annealed at relatively low temperatures that do not provide the required thermal energy to enable complete fullerene coalescence, the resulting inner tubes will contain nonhexagonal carbon rings and have a corrugated structure. Inner tubes with a corrugated structure may show complex RBM-like vibrational modes. The presence of corrugated inner tubes in our DWNT samples is a possibility but since our sample is heat treated at a high temperature (1700 °C), the so-called corrugation effect is weak and the inner-outer tube interaction is the dominant factor affecting the number and frequency of the observed inner-tube RBMs.

FIG. 3. (Color online) $I_{\text{RBM}}$ vs $w_{\text{RBM}}$ for (a) CVD-DWNT and (b) C_{60}^-DWNT bundles for the 1.89–1.999 eV $E_{\text{last}}$ range. The top horizontal axes mark the expected $d_t$ (nm) of the tubes according to our experimentally obtained $w_{\text{RBM}}$ vs $d_t$ relationship ($w_{\text{RBM}} = 228.8/d_t + 2.4$ cm$^{-1}$). The regions marked with an M (S) correspond to metallic (semiconducting) tubes. In the cases where the exact $(n,m)$ of the tube cannot be unambiguously identified, the $(2n+m)$ family of the corresponding group of tubes is marked.
TUNABLE RAMAN SPECTROSCOPY STUDY OF CVD AND C_{60}-DWNT BUNDLES

The G band is generated by in-plane C-C bond stretching modes that occur in the form of optical phonons along the axial [longitudinal optical (LO) mode] and circumferential [transverse optical (TO) mode] directions of a carbon nanotube. In metallic (semiconducting) tubes, the G* and G features originate from the LO (TO) and TO (LO) phonons, respectively. Since the curvature in a nanotube weakens the bonds in the circumferential direction, the \( \omega_{G,TO} \) downshifts with decreasing nanotube diameter.

By measuring the Raman spectra from our DWNTs in the \( E_{\text{\text{Laser}}} = 1.58-2.33 \) eV range (see Fig. 5), we scanned across regions of the resonance windows of the inner and outer tubes that belong to DWNTs with the four possible configurations mentioned above. The inner and outer tubes of the same DWNT are not necessarily in resonance with the same \( E_{\text{\text{Laser}}} \). However, if the RBM and G-band resonance windows of a certain inner tube are centered at roughly the same energy, we can assume, for a certain \( E_{\text{\text{Laser}}} \), that the observed line shape of the G band is predominantly generated by those tubes that show a resonant RBM signal. We can thus analyze the RBM region of a spectrum, relate it to its corresponding G-band region, and assume that the tubes with RBMs which are in resonance with a given \( E_{\text{\text{Laser}}} \) have a predominant effect on the observed line shape of the G band.

In the \( E_{\text{\text{Laser}}} = 1.58-1.99 \) eV range for CVD-DWNTs, we observe that the inner tubes in resonance are predominantly semiconducting (S@S and S@M) [see Fig. 1(a)]. In this case, the frequencies of the G* band (\( \omega_{G,*} \)) are centered around 1588 cm\(^{-1} \) with a full width at half maximum intensity of \( \text{FWHM}_{G,*} \approx 15 \) cm\(^{-1} \) [see Fig. 5(a)] and \( \omega_{G^{-}} = 1555 \) cm\(^{-1} \) with \( \text{FWHM}_{G^{-}} \approx 42 \) cm\(^{-1} \). The presence of the outer M tubes is likely to be contributing to the large FWHM of \( \omega_{G^{-}} \). On the other hand, for \( E_{\text{\text{Laser}}} \geq 2.07 \) eV, besides being in resonance with S@S and S@M CVD-DWNTs, we also find resonance with CVD-DWNTs having inner metallic tubes (M@S and M@M) that belong to \( (2n+m) \) families 24 and 21 [see Fig. 1(a)] and we observe that the line shape of the G band undergoes a series of noticeable changes [see Fig. 5(a)] as \( E_{\text{\text{Laser}}} \) is increased. These changes in the G-band line shape, which coincide with the appearance of the resonance with M inner tubes, are (a) an \( \omega_{G^{-}} \) upshift of \( \approx 7 \) cm\(^{-1} \) (from \( \omega_{G^{-}} \approx 1588 \) cm\(^{-1} \) at 1.581 eV to \( \omega_{G^{-}} \approx 1595 \) cm\(^{-1} \) at 2.33 eV), (b) an \( \approx 16 \) cm\(^{-1} \) splitting of the G* band and G band (\( \omega_{G^{-}} \approx 1580 \) cm\(^{-1} \) and \( \omega_{G^{-}} \approx 1595 \) cm\(^{-1} \) at 2.33 eV), and (c) an \( \approx 6 \) cm\(^{-1} \) increase in \( \text{FWHM}_{G^{-}} \) (from \( \text{FWHM}_{G^{-}} \approx 15 \) cm\(^{-1} \) at 1.581 eV to \( \text{FWHM}_{G^{-}} \approx 21 \) cm\(^{-1} \) at 2.33 eV). The above-mentioned changes can be appreciated in Fig. 5(c), where spectra for the two extrema in \( E_{\text{\text{Laser}}} \) are shown (predominant resonance with S@S and S@M DWNTs at \( E_{\text{\text{Laser}}} = 1.581 \) eV to be compared with the resonance with M@S and M@M DWNTs at \( E_{\text{\text{Laser}}} = 2.33 \) eV).

In the case of C_{60}-DWNTs, the distribution of the four possible DWNT configurations present in the sample is different and we do not find a strong resonance with inner M tubes belonging to the \( (2n+m) \) family 24 for \( E_{\text{\text{Laser}}} = 2.242-2.304 \) eV [see Fig. 5(b)]. Instead we only find strong resonance with inner M tubes for \( E_{\text{\text{Laser}}} = 2.33 \) eV which excites (9,3) inner M tubes belonging to family 21.

the data are sufficiently accurate to tell us that the outer tubes of the individual DWNTs with inner (6,5) and (9,1) tubes belong to \( (2n+m) \) families 33 and 36, respectively.
either predominantly semiconducting or metallic. The line shape of the G band for C60-DWNTs is similar for all excitation energies in the 1.581–2.33 eV range. However, when $E_{\text{laser}}=2.33$ eV is reached and DWNTs with inner M tubes are excited [see Fig. 5(c)], the G band upshifts ($\sim 9$ cm$^{-1}$ increase in $\omega_{G1}$), splits ($\sim 12$ cm$^{-1}$ separation between $\omega_{G1}$ and $\omega_{G2}$), and broadens ($\sim 8$ cm$^{-1}$ increase in FWHM). The upshift and broadening of $\omega_{G1}$ when in resonance with M@S and M@M DWNTs is consistently observed in both the CVD-DWNTs and C60-DWNTs. Therefore, in DWNT bundles, the line shape of the G band is not only dependent on the sample’s diameter distribution but also on its DWNT M/S configuration distributions.

**D. G’ band from CVD-DWNT and C60-DWNT bundles**

The G’ band is based on a two-phonon intervalley double resonance process that does not require the presence of defects. In general, the G’ band is generated by four steps where (a) an incoming photon excites an electron with wave vector $k$, (b) the electron is scattered from $k$ to $k+q$ by emitting a phonon with wave vector $q$, (c) the electron is backscattered from $k+q$ to $k$ by emitting a second phonon, and (d) the electron recombines with a hole at $k$. Since the laser excitation energy ($E_{\text{laser}}$) determines the $k$ of the excited electron and the $q$ of the emitted phonons, the G’-band mode is dispersive. For reference, in SWNTs the reported G’-band linewidth is 106 cm$^{-1}/eV$. Also, like other Raman modes, the G’ band has a diameter dependence.

In DWNT bundles, the diameter distribution is bimodal and we expect a G’ band with two main features that correspond to the inner (G’$^1$) and outer (G’$^2$) tubes. The intensity ($I_{G'}$), the absolute frequency ($\omega_{G'}$), the dispersion ($\partial \omega_{G'}/\partial E_{\text{laser}}$), and the FWHM (FWHM$_{G'}$) linewidth of the two G’-band components in DWNTs depend on nanotube diameter, chirality, metallicity, and other factors, such as doping, strain, and energy of the incoming photons. Figure 6 shows $I_{G'}$ vs $\omega_{G'}$ from (a) CVD-DWNT bundles and (b) C60-DWNT bundles for the 1.881–2.33 eV $E_{\text{laser}}$ range. The CVD-DWNT sample shows two peaked structure where the low frequency peak (G’$^1$), which corresponds to the inner tubes, has a higher intensity than its higher frequency counterpart (G’$^2$). The $I_{G'}$, $\omega_{G'}$, and FWHM$\omega_{G'}$ ratios (not presented here) show a tendency to increase with increasing $E_{\text{laser}}$. Interestingly, the dispersion of the peak corresponding to the inner tubes ($\partial \omega_{G1}/\partial E_{\text{laser}}=122.21$ cm$^{-1}/eV$) is considerably larger than that of the outer tubes ($\partial \omega_{G2}/\partial E_{\text{laser}}=89.44$ cm$^{-1}/eV$), indicating that the inner and outer tubes, which vary in diameter and chirality, possess different sets of $E(k)$ dispersion relations with different slopes and thus enable the selection of different $q$ wave vectors [see Fig. 6(c)]. The inherent difference between the phonon dispersion relations of the inner and outer tubes also contributes to the final $\partial \omega_{G1}/\partial E_{\text{laser}}$, values of the G’1 and G’2 features.

The G’-band spectra from the C60-DWNTs also has a two peaked structure where the intensity of G’$^1$ relative to G’$^2$ tends to increase with increasing $E_{\text{laser}}$. Interestingly, we observe a slight downshift in the frequency of the G’ band of C60-DWNTs with respect to CVD-DWNTs for spectra taken at the same $E_{\text{laser}}$. In particular, the $\omega_{G1}$ of the C60-DWNTs is on average $\sim 6.5$ cm$^{-1}$ below its CVD-derived counterpart.
From previous experiments performed on SWNTs, the $\omega_{G1}$ was found to have a $d_i$ dependence of the form $\omega_{G1} = \omega_0 + \beta/d_i$, where $\omega_0 = 2704$ cm$^{-1}$ and $\beta = -35.4$ cm$^{-1}$ nm for $E_{\text{laser}} = 2.41$ eV. However, the inner tubes from both CVD-DWNT and C$_{60}$-DWNT bundled samples show deviations from the above-mentioned SWNT-based $\omega_{G1}(d_i)$ relation. The SWNT-based $\omega_{G1}(d_i)$ relation predicts that our inner tubes with diameters in the $0.7$–$0.9$ nm range should have $\omega_{G1}$ in the $2653$–$2664$ cm$^{-1}$ range. However, for the CVD-DWNT (C$_{60}$-DWNT) sample at $E_{\text{laser}} = 2.33$ eV, $\omega_{G1} = 2632$ cm$^{-1}$ ($\omega_{G1} = 2634$ cm$^{-1}$) which is in both cases below the SWNT-based prediction.

In the case of the outer tubes, there is reasonably good agreement with the SWNT-based $\omega_{G1}(d_i)$ relation. The SWNT-based relation predicts $\omega_{G1}$ to be in the $2676$–$2681$ cm$^{-1}$ range for tubes with $d_i$ in the $1.3$–$1.6$ nm range and we measure $\omega_{G1} = 2682$ cm$^{-1}$ ($\omega_{G1} = 2671$ cm$^{-1}$) for the CVD-DWNTs (C$_{60}$-DWNTs). In agreement with previous reports, the SWNT-based $\omega_{G1}(d_i)$ relation may predict the $\omega_{G1}$ of the outer tubes quite well but it cannot be applied to the inner tubes because this relation does not take into account the high curvature of tubes with $d_i < 1$ nm and the possible interactions of the inner tubes with the surrounding outer tubes due to strain and electrostatic effects.

Finally, another factor that affects the lineshape of the $G'$ band is the distribution of the M/S configurations present in a DWNT bundle sample. In SWNTs the $G'$-band intensity ($I_{G'}$) has been reported to be larger for M than for S nanotubes because the electron-phonon matrix elements for the TO phonon at the K point are larger for M tubes. Therefore, it is reasonable to expect that the relative intensities of $G'_1$ and $G'_2$ also depend on whether the inner and outer tubes of a DWNT are M or S. When taking the metallicity factor into account, one must also keep in mind that the resonance window of the $G'$ band is wider than the resonance window of the RBM and therefore other tubes that are not in resonance with the laser excitation energy for the RBM feature can also contribute to the line shape of the $G'$ band. Resonance with scattered photons can also occur. Nevertheless, if we assume that both the $G'$ band and the RBM resonance windows of a carbon nanotube are centered around the same energy, it is reasonable to expect that the major contribution to the line shape of the $G'$ band in DWNTs comes from the inner and outer tubes that are in resonance with the laser excitation energy. In this context, we divide our $G'$-band measurements into two regimes: (I) for all $E_{\text{laser}} > 2.2$ eV that excite M@S DWNTs and the inner M tubes of M@M DWNTs and (II) for all $E_{\text{laser}} < 2.2$ eV that excite S@M DWNTs and the inner S tubes of S@S DWNTs (see Figs. 1 and 6). In both
C_{60}\text{-D}W\text{NTs and CVD-DW}\text{NTs we observe that the } I_{G_1'G_2'}\text{ is greater in regime II than in regime I in part because the laser energy is in resonance with the } E_{11}\text{ transition of the inner M tubes. Thus, when the inner tubes in resonance with } E_{\text{laser}}\text{ are predominantly metallic, we observe an enhancement of the intensity of the lower frequency } G_1'\text{ region with respect to its high frequency } G_2'\text{ counterpart. Experiments at the individual DWNT level may enable quantitative measurements capable of decoupling the individual contributions to the line shape of the } G'\text{ band for each of the above-mentioned factors.}

IV. CONCLUSIONS

By scanning a large number of laser energies we have measured the Raman spectra from CVD-derived and C_{60}\text{-derived double-wall carbon nanotubes with the four possible metallic and semiconducting configurations. For small diameter nanotubes (<0.9 nm), the separation between the } E_{ij}\text{ is large enough to allow for an accurate } (n,m)\text{ identification of the inner tubes. Once the } (n,m)\text{ indices of the observed inner tubes were assigned, we measured their } \omega_{\text{RBM}}\text{ and obtained the } \omega_{\text{RBM}}\text{ to } d_i\text{ relationship of the form } \omega_{\text{RBM}}=A/d_i+B.\text{ The values of the } A\text{ and } B\text{ constants are close to those previously obtained from the elastic properties of graphite,}\text{ indicating that the inner tubes are effectively shielded from the environment by their surrounding outer tubes. By scanning the RBM region from the CVD-DWNTs and C_{60}\text{-DWNTs bundles we find that the splitting effect of the } \omega_{\text{RBM}}\text{ described above is not necessarily always present for all inner tubes. For the same } (n,m)\text{ inner tubes, the RBM splitting behavior is observed to vary between CVD-derived and C_{60}\text{-derived DWNTs, indicating that the energy minimizing processes inherent to each sample’s synthesis method define the final } (n,m)\text{ @ } (n',m')\text{ combinations and wall to wall distances found in DWNTs. We also analyzed the Raman spectra from individual S@M DWNTs with inner (6,5) type-II and (9,1) type-I semiconducting tubes. The upshift in the } \omega_{\text{RBM}}\text{ of the inner tube caused by a decrease in the diameter of the outer tube was observed to follow the same trend in both type-I (6,5) and type-II (9,1) inner semiconducting tubes. The line shape and frequency of the G band from CVD-DWNT and C_{60}\text{-DWNT bundles were analyzed using various laser excitation energies and changes in the G-band line shape were correlated with a switch in resonance for DWNTs whose inner tubes are predominantly semiconducting (S@S and S@M) to resonance with DWNTs with predominantly inner metallic tubes (M@M and M@S). Finally, the } G'\text{-band frequency of C_{60}\text{-DWNT bundles is observed to be downshifted with respect to that of CVD-DWNTs due to differences in the diameter distributions and M/S configuration populations of the samples.}

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