Surface enhanced Raman spectroscopy on a flat graphene surface

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Surface enhanced Raman spectroscopy (SERS) is an attractive analytical technique, which enables single-molecule sensitive detection and provides its special chemical fingerprints. During the past decades, researchers have made great efforts towards an ideal SERS substrate, mainly including pioneering works on the preparation of uniform metal nanostructure arrays by various nanoassembly and nanotailoring methods, which give better uniformity and reproducibility. Recently, nanoparticles coated with an inert shell were used to make the enhanced Raman signals cleaner. By depositing SERS-active metal nanoislands on an atomically flat graphene layer, here we designed a new kind of SERS substrate referred to as a graphene-mediated SERS (G-SERS) substrate. In the graphene/metal combined structure, the electromagnetic “hot” spots (which is the origin of a huge SERS enhancement) created by the gapped metal nanoislands through the localized surface plasma resonance effect are supposed to pass through the monolayer graphene, resulting in an atomically flat hot surface for Raman enhancement. Signals from a G-SERS substrate were also demonstrated to have interesting advantages over normal SERS, in terms of cleaner vibrational information free from various metal-molecule interactions and being more stable against photo-induced damage, but with a comparable enhancement factor. Furthermore, we demonstrate the use of a freestanding, transparent and flexible “G-SERS tape” (consisting of a polymer-layer-supported monolayer graphene with sandwiched metal nanoislands) to enable direct, real time and reliable detection of trace amounts of analytes in various systems, which imparts high efficiency and universality of analyses with G-SERS substrates.
substrate containing a graphene/metal-combined structure, referred to as a graphene-mediated SERS (G-SERS) substrate.

Results

As illustrated in Fig. 1, the active surface of our G-SERS substrate is a monolayer graphene (1LG), with gold/silver nanoislands tightly adhered on the backside. By collecting electromagnetic hot spots from the metal nanoislands on a graphene surface, a G-SERS substrate is anticipated to offer an atomically smooth surface for controllable molecular arrangements as well as well-defined molecular interactions, yet also has the electromagnetic hot spots created by the localized surface plasmon resonance of gold or silver nanoislands adhered on the backside of graphene.

In a G-SERS substrate, the metal nanoislands focus the incident laser, creating localized electromagnetic hot spots close to the graphene surface. The permeability of the electromagnetic field to pass through the graphene is critical for the final enhancement activity. First, theoretical calculations are carried out. For simplicity, two 60-nm diameter gold hemispheres adhering below a monolayer graphene are used to simulate a G-SERS substrate. From the three dimensional finite difference time domain (3D-FDTD) simulations results as shown in Fig. 1B, the simulated electromagnetic field occurs at the gap of the hemisphere dimer, spreads from the central strongest hot spot to the vicinity. The permeability was also supported by extended Mie theory calculations of a graphene-coated 60-nm gold sphere dimer, (SI Appendix, Fig. S1). By comparing with that of gold hemispheres without graphene, we find that graphene exhibits good permeability for the electromagnetic hot spot to pass through, and interestingly the maximum $|E_{\text{loc}}/E_0|^2$ is even almost one time stronger (SI Appendix, Fig. S2). These results encouraged us that the close vicinity of these hot spots from metal nanostructures can make the monolayer graphene a hot surface (Fig. 1C), and thus allows localized electromagnetic enhancement to be achieved on an atomically flat surface.

Fig. 1D and 1E illustrate possible states of molecules adsorbed on a normal SERS substrate (Fig. 1D) and a G-SERS substrate (Fig. 1E). In Fig. 1D, molecules are in various orientations according to the morphology of the metal nanoislands. The enlarged view in Fig. 1D illustrates various possibilities that may make the enhanced Raman signals complicated. In contrast, by taking the flat and hot graphene surface as the Raman enhancer, as illustrated in Fig. 1E, the strong Raman enhancement from localized electromagnetic hot spots created by metal nanoislands can be realized on an atomically smooth surface. The monolayer graphene is expected to act as a multifunctional “SERS mediator”, including: (i) a flat supporting surface to arrange molecules in a more controllable way, (ii) a spacer to separate the metal-molecule contact, (iii) additional effects like a stabilizer of both the substrate and the molecules under laser exposure.

To seek the possible superiorities of a G-SERS substrate, a rational way was used to fabricate substrates with both G-SERS regions and normal SERS regions for comparison. As shown in Fig. 24, a uniform layer of molecules, such as rhodamine 6G (R6G) or copper phthalocyanine (CuPc), were first deposited on a SiO$_2$/Si substrate by vacuum thermal evaporation. Mechanically exfoliated graphene pieces were then transferred to the top of the molecules, creating regions with and without graphene (for regions without graphene, our Raman inspections were carried out only on those regions that are not touched by the adhesive scotch tape used in the mechanical exfoliation, to make sure no damage was done to the Raman probes). Finally, gold or silver nanoislands adhering to the graphene were constructed by vacuum thermal evaporation of an 8-nm metal film (a preferred thickness for gold as illustrated in SI Appendix, Figs. S3 and S4) on the top of both regions (1LG and the bare substrate). It should be pointed out that gold and silver are the most popular electromagnetic enhancers, where gold is more stable but the enhancement is usually less than that for silver. Silver is more likely to be affected by the environment, such as oxidation processes which will decrease the enhancement over time. Here we show that both of gold and silver can be used effectively for the fabrication of a G-SERS substrate. The method of vacuum thermal evaporation can avoid external contaminations such as surfactants which would be introduced by using nanostructures prepared by colloidal methods (20). Thus, G-SERS and normal SERS regions (defined as G-SERS(M) and SERS(M), where M is the metal used as the electromagnetic enhancer) were constructed corresponding to the regions with and without graphene, respectively.

Typical Raman spectra of R6G with G-SERS(Au), G-SERS(Ag), SERS(Au), and SERS(Ag) are shown in Fig. 2B from bottom to top, respectively. First, it can be seen that the G-SERS regions give more reproducible and cleaner signals as compared to normal SERS. For G-SERS(Au) and G-SERS(Ag), the spectral features are the same as one another, with clear peaks at 612, 635, 660, 771, 1181, 1361, 1511, 1648 cm$^{-1}$ (±1 cm$^{-1}$) (see SI Appendix, Table S1). The peaks at around 1570 and 2625 cm$^{-1}$ (marked by *) are the G-band and G’-band of monolayer graphene, respectively. The relative intensity of the graphene peak as compared to R6G is different in G-SERS(Au) and G-SERS(Ag) (this variability is reproducible from sample to sample in our parallel measurements); this may be because of a different chemical contribution of gold and silver. In contrast, Raman spectra enhanced by SERS(Au) and SERS(Ag) show many additional
peaks (as marked by red arrows) compared to the G-SERS results. Among these additional peaks, such as at 597, 1014, 1411, 1468, 1540 cm$^{-1}$ in SERS(Au) and 554, 590, 1012, 1336, 1403, 1471, 1531 cm$^{-1}$ in SERS(Ag), some of them may be from contamination of surface carbons (as will be described below), some are new emerging bands possibly due to the modified surface selection rules (because of alteration of the symmetry of the vibrational modes under changed configurations), while others remain difficult to assign precisely. Parallel measurements are demonstrated and we found that these additional signals are irreproducible (see SI Appendix, Fig. S6). When looking at those bands, which are also observed in the G-SERS spectra, many bands are shifted (probably due to a charge-transfer effect of the metal-R6G), and the amount of shifts are different for SERS(Au) and SERS(Ag). Thus, the SERS results will depend on various factors, which may result in the appearance and disappearance of various features, mixed with peaks having up or down shifted frequencies. As a result, it would be almost impossible to assign each vibration precisely, since the spectral features are no longer clearly dependent on the primary structure of the sample molecules.

Additionally, we also found that graphene makes the SERS system more stable. Photo-induced damage is a common characteristic of SERS, and is popularly known as photocarbonization and photobleaching. Raman spectra of the as prepared G-SERS and SERS substrates (without molecules) are shown in Fig. 2C, top panel. First, one can see that the G-SERS substrate provides a cleaner baseline than that of normal SERS: The photocarbonization background in the 1100 – 1700 cm$^{-1}$ range (marked in blue line) for the SERS spectra is not observed in the G-SERS spectra, together with the emergence of a characteristic signal of monolayer graphene. Because no Raman probes were introduced here, this carbonization background is normally considered to be due to surface carbon-based adsorbates from the atmosphere. In fact, the photocarbonization of adsorbates was observed and studied soon after the SERS effect was discovered (21), but so far this signal contamination problem has not yet been resolved. This photocarbonization effect will be even more pronounced in an actual SERS measurement when probe molecules are present. As shown in Fig. 2C (bottom), even with a much lower laser power (the SERS spectrum is acquired under a 10% laser power compared to G-SERS), the SERS spectrum of R6G shows an obvious photocarbonization background (marked in blue line) while this background is absent in the G-SERS spectrum. The presence of graphene cover layer may suppress the catalytic activity of the gold nanosheets and prevent the photo-induced carbonization, as well as keep R6G molecules from direct contact with the gold film, thus preventing adsorbates (from the atmosphere) and R6G molecules from photocarbonization. The high thermal conductivity of graphene may also partly contribute to the enhanced stability, and actually we have found that the morphology of the metal film changes after a relatively large laser power exposure if there is no graphene layer present, while it keeps stable for the G-SERS regions. An additional effect, the lower photoluminescence background in a G-SERS substrate than that of normal SERS, is consistent with the reported results that graphene is an effective quencher for both the fluorescence of dyes (22, 23) and the photoluminescence of a gold film, through a resonance energy transfer process (24).

On the other hand, the photobleaching (or photodegradation) of the Raman probes induced by the laser is also a well-known feature in normal SERS experiments, especially for dyes. During a normal SERS measurement, the intensity of the Raman signals of molecules may decrease with increased acquisition time (especially under a high laser power), and this effect will lead to a variation of the Raman spectra with acquisition time and laser power. Fig. 2D shows a comparison of the stability of the Raman spectra of CuPc of SERS(Au) (top) and G-SERS(Au) (bottom) regions. It is seen here that the signal of CuPc in the normal SERS region decreases quickly in intensity during a 600 s measurement while in the G-SERS region it remains totally stable. It is speculated that both the separation of CuPc from the rough gold film and the formation of a graphene/CuPc complex through π-π interactions may contribute to the stability of CuPc.

The above results in Fig. 2 suggest that, in a G-SERS substrate (when the rough metal substrate is replaced by a hot and atom-
ically flat graphene surface), Raman enhancement turns out to be cleaner (with characteristic and reproducible vibrational modes in which fluctuating information is removed) and more stable (against laser-induced damage such as photocarbonization and photobleaching). It should be emphasized that the above characteristics of a normal SERS substrate do not mean shortcomings of the SERS technique itself, and actually it oppositely reflects the fascinating capability of SERS (even capable of monitoring the actions of a single molecule). However, currently the state of the art of SERS is still mostly limited to the study of a mixture of molecules in unclear states with poor controllability. Thus, the G-SERS substrate is anticipated to act as an effective tool to implement SERS, both for general analyses (with better controllability and reproducibility among parallel measurements) and for many advanced applications.

Further, we checked the enhancement factor of G-SERS substrates. For comparison, the Raman spectra of CuPc (pristine), 1LG/CuPc [graphene-enhanced Raman scattering (25), “GERS”], Au/CuPc [SERS(Au)], Au/1LG/CuPc [G-SERS(Au)], Ag/CuPc [SERS(Ag)], and Ag/1LG/CuPc [G-SERS(Ag)] are shown in Fig. 3(A–F). All the spectra are acquired under the same conditions and all the spectra are shown with the same intensity scale, where the Raman signal data is multiplied by 30 in the case of the pristine and GERS and by 10 in the case of G-SERS(Au) and SERS (Au). Here, the relative intensity of the graphene signal seems much lower (the G-band of graphene is almost invisible but the G'-band can still be clearly distinguished) as compared to measurement of R6G in Fig. 2B, and this is because of a resonance Raman enhancement effect of CuPc under our measurement condition (632.8 nm). Through the integrated area of the 1530 cm$^{-1}$ peak, taking the pristine spectra as a reference, it is enhanced by a factor of 14 for GERS, 61 for SERS(Au), 85 for G-SERS(Au), 580 for SERS(Ag), and 755 for G-SERS(Ag). It should be noted that here the enhancement factors for the SERS region and the G-SERS region are both seriously undervalued since the absorption of the metal film makes only molecules at the gaps between the metal nanoislands (of course, this is quite a minority) contribute to their final signal intensity. However, here it is fair to compare the relative sensitivity between a normal SERS substrate and a G-SERS substrate. The weak signal of pristine CuPc is magnified, and under the measurement conditions (632.8 nm excitation), silver is a better electromagnetic enhancer than gold. Most importantly, for both the case of gold and silver, the enhancement factors of the G-SERS regions are even slightly higher than those of normal SERS for most of the bands. This means that the presence of the graphene layer will not cause the localized electromagnetic field to decay observably in its vicinity, which is consistent with our theoretical results as mentioned before. The small enhancement of the graphene substrate is consistent with our earlier results (25, 26), possibly due to a chemical enhancement. By a careful comparison of their spectral features, it can be seen that the spectral feature from a G-SERS substrate remains the same as that from a graphene substrate while the enhancement was greatly improved by the introduction of the electromagnetic enhancement. Consistent results can be observed from the signals of R6G from a G-SERS substrate and a graphene substrate as shown in Table S1 (SI Appendix), regardless of the material of the metal enhancer. Thus, by collecting hot spots of metal nanoislands on a flat graphene surface, G-SERS substrate provides comparable sensitivity with that of normal SERS. The better-defined molecule-substrate interactions in a G-SERS substrate is also intriguing, as an atomically compact and chemically inert graphene surface seems to cut down signal fluctuations among different metal enhancers.

A “universal” substrate should be sufficiently economical, sufficiently convenient, and sufficiently compatible with various samples in a different state. As described in the following, we developed a general type of G-SERS substrate (which we call a “G-SERS tape”), which is transparent, freestanding, and flexible (by virtue of the high light transmittance and good flexibility of both graphene and the selected polymer). As a schematic representation in Fig. 1A, a G-SERS tape consists of three parts: monolayer graphene, a polymer supporting layer, and sandwiched metal nanoislands adhering to the backside of the active face of graphene. Fabrication of G-SERS tapes and their applications in various situations, including surfaces with arbitrary morphology, or even directly on a solution surface are demonstrated. Such a G-SERS process does not require special sample preparation, is a noninvasive process, and is very convenient for in situ characterization or real time monitoring.

As shown in Fig. 4A, taking the G-SERS(Au) tape as example, an 8-nm gold film was first deposited on a large area monolayer graphene grown on a 8 × 8 cm$^2$ copper foil [prepared by the chemical vapor deposition method (27)], which is uniform with high quality (see SI Appendix, Fig. S7). A poly(methyl methacrylate) (PMMA) film, which is optically transparent, inert, and Raman inactive, was then fabricated by spin-coating and dried at 170 °C for 30 min (27, 28). Finally, a transparent, freestanding, and flexible G-SERS tape with a PMMA/metal/graphene structure was obtained by etching the copper foil in a FeCl$_3$/H$_2$O solution. This technique is suitable for scaled production of G-SERS tapes (Fig. 4C). The transmission electron microscopy (TEM) image in Fig. 4B shows the morphology of an 8-nm gold film, in which most gaps between the nanoislands are 2 ~ 3 nm, thereby allowing for a considerable electromagnetic enhancement. As shown in Fig. 4D, the atomic force microscopy characterization shows that the active side (graphene side) of the G-SERS tape is flat, with a fluctuation (surface roughness) along the film length of less than ±2 nm.

![Fig. 3. Comparison of the Raman spectra of CuPc with different enhancement methods.](image-url)
Mechanically exfoliated graphene is also suitable for use in fabricating G-SERS tapes (see SI Appendix, Fig. S8). A G-SERS tape with silver as an electromagnetic enhancer [G-SERS(Ag) tape] are prepared through a similar procedure (with a few modifications) and this tape is also described in the SI Appendix. As the active metal is buried inside the graphene and the PMMA layer (in the case of Ag, there is an additional protection layer) without exposure to the atmosphere, it is anticipated that these G-SERS tapes could have a long shelf life. Because of the flexibility of a G-SERS tape, the contact between the tape and the target material will be good enough to achieve the G-SERS enhancement.

Fig. 5 shows several typical analyses. As shown in Fig. 5A, G-SERS tapes shown in Fig. 4 can be used directly for real time analysis of samples in an aqueous solution, where we first put such a G-SERS tape floating on water, and a clean baseline was obtained with a clear graphene signal. Next, the same G-SERS tape was placed on a 1 × 10^{-2} M aqueous solution of R6G, and the intrinsic signal of R6G appeared in the Raman spectra (with the same spectral features as the G-SERS results in Fig. 2B). Additionally, the R6G signal disappeared after washing and replacing the G-SERS tape on water. This reversibility suggests that G-SERS tape may be exploited in real time sensing processes, such as online monitoring of water contaminants. Furthermore, the Raman spectra of a self-assembled monolayer of p-aminothiophenol on a flat gold surface (Fig. 5B) and of CuPc molecules adsorbed on a cauliflower surface (Fig. 5C) shows that G-SERS tapes can be used for the analysis of trace amounts of molecules on any surface with arbitrary morphology. Moreover, we extend G-SERS measurements to other solvents like ethanol by making a small modification to the G-SERS tape (see SI Appendix, Fig. S13). Therefore, this kind of G-SERS substrate is widely applicable for direct, noninvasive and ultrasensitive Raman measurements, including real time monitoring on a solution surface and on other surfaces with any arbitrary morphology.

Discussion

SERS is a powerful tool and it can be more powerful when exploited with a carefully designed substrate. Among the wide researches on SERS during the past years, great efforts have been made towards gaining a theoretical understanding of the enhancement mechanism (4, 29, 30) and resulting in extended practical applications (13, 31–33). The basic physical characteristic that a localized surface plasmon resonance (the origin of a large SERS enhancement) occurs only at those closely gapped or highly curved plasmonic metal nanostructures has hindered the SERS technique from being used on a flat surface. In this report, we demonstrated a G-SERS substrate in which a plasmonic metal naisland array is adhered to a graphene surface, creating an atomically smooth surface with strong electromagnetic hot spots that can be used for Raman enhancement. The concept of enhancing the Raman signals on a flat graphene surface simplifies the real time and reversible G-SERS characterization of R6G directly in a 1 × 10^{-2} M aqueous solution. (I, II, III are the Raman spectra with the same spectral features as the G-SERS results in Fig. 2B). Additionally, the R6G signal disappeared after washing and replacing the G-SERS tape on water. This reversibility suggests that G-SERS tape may be exploited in real time sensing processes, such as online monitoring of water contaminants. Furthermore, the Raman spectra of a self-assembled monolayer of p-aminothiophenol on a flat gold surface (Fig. 5B) and of CuPc molecules adsorbed on a cauliflower surface (Fig. 5C) shows that G-SERS tapes can be used for the analysis of trace amounts of molecules on any surface with arbitrary morphology. Moreover, we extend G-SERS measurements to other solvents like ethanol by making a small modification to the G-SERS tape (see SI Appendix, Fig. S13). Therefore, this kind of G-SERS substrate is widely applicable for direct, noninvasive and ultrasensitive Raman measurements, including real time monitoring on a solution surface and on other surfaces with any arbitrary morphology.

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determination of the precise amount of molecules, their detailed position and orientation information, as well as the molecule-substrate interaction. The characteristic Raman signal and atomically precise amount of graphene also make it a natural internal standard. G-SERS substrates are anticipated to be exploited in many fields, such as reliable qualitative and label-free quantitative detection, and the characterization of intrinsic structural information from samples without the introduction of spurious spectral features coming from metal contamination.

In conclusion, by introducing plasmonic metal to a graphene layer, an atomically flat substrate for SERS was developed. Cleaner and more reproducible signals were obtained with a comparable enhancement factor. The use of a transparent, freestanding layer, an atomically flat substrate for SERS was developed. Cleaning the substrate interaction. The characteristic Raman signal and atomically precise amount of molecules, their detailed position and orientation information, as well as the molecule-substrate interaction. The characteristic Raman signal and atomically precise amount of graphene also make it a natural internal standard. G-SERS substrates are anticipated to be exploited in many fields, such as reliable qualitative and label-free quantitative detection, and the characterization of intrinsic structural information from samples without the introduction of spurious spectral features coming from metal contamination.

Materials and Methods
Mechanically exfoliated graphene was prepared from Kish graphite (Covellite Materials Corp.) using Scotch tape on a SiO2 substrate. The graphene was prepared by vacuum evaporation of a 8-nm gold or silver film onto a copper grid (carbon covered). Three-dimensional-FDTD simulations were carried out by commercial software FDTD solutions 6.5 (Lumerical). The dielectric functions used to describe metal and graphene-related plasmonic effects were measured with a M-2000 spectrophotometer. The dielectric functions used to describe metal and graphene-related plasmonic effects were measured with a M-2000 spectrophotometer. The dielectric functions used to describe metal and graphene-related plasmonic effects were measured with a M-2000 spectrophotometer.