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Detailed Terms
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Electronic Spin Transition in Nano-size Stoichiometric Lithium Cobalt Oxide

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KEYWORDS Spin transition, Lithium cobalt oxide, Intermediate spin, Surface calculation, DFT+U

ABSTRACT: A change in the electronic spin state of the surfaces relevant to Li (de)intercalation of nano-sized stoichiometric lithium cobalt oxide LiCo(III)O$_2$ from low-spin to intermediate and high spin is observed for the first time. These surfaces are the ones that are relevant for Li (de)intercalation. From density functional theory calculations with a Hubbard U correction, the surface energies of the layered lithium cobalt oxide can be significantly lowered as a consequence of the spin change. The crystal field splitting of Co d orbitals is modified at the surface due to missing Co-O bonds. The electronic spin transition also has a significant impact on Co(III)-Co(IV) redox potential, as revealed by the change in the lithium (de)intercalation voltage profile in a lithium half cell.

Lithium cobalt oxide (LiCoO$_2$) is a compound of great importance, as it has been the most widely used positive electrode material for lithium ion batteries for nearly two decades. LiCoO$_2$ adopts the $\alpha$-NaFeO$_2$-type crystal structure with rhombohedral symmetry (space group R-3m) and Li$^+$ and Co$^{3+}$ ions sitting in octahedral sites formed by alternating layers of oxygen. Because of such an ordered layer structure, Li$^+$ can be reversibly deintercalated and re-intercalated from LiCoO$_2$ to Li$_1.2$CoO$_2$, with a high electrochemical potential of up to 4.2 V (vs Li/Li$^+$). In recent years, it has been demonstrated that ultra-fast charge/discharge rate capabilities can be achieved in this compound when nano-scale (< 50nm) particles with morphology optimal for Li intercalation are prepared and tested. Okubo et al. observed several interesting phenomena associated with their nano-sized LiCoO$_2$: First, lattice parameter expansion is observed in particles less than 20nm; second, the magnetic susceptibility increases dramatically when compared with that of bulk-LiCoO$_2$. The authors hypothesized that these phenomena are mainly due to the presence of Co$^{2+}$ on the surface of their nano-sized particles, Co$^{2+}$ being present in the form of Li$_{1.2}$Co$_{1-y}$O$_2$. They attributed this to a consequence of their hydrothermal synthetic process. However, the interpretation of the magnetic data is ambiguous since no direct evidence for the presence of Co$^{2+}$ is obtained. Moreover, Levassuer et al. pointed out that in bulk lithium over-stoichiometric (“Li-excess”) samples, the charge is compensated by oxygen vacancies. This leads to cobalt ions being in a square based pyramidal site with an intermediate spin (IS) configuration. That in this communication we show that, for stoichiometric nano-sized LiCoO$_2$, the anomaly in magnetic susceptibility can similarly be explained by the presence of IS or HS Co$^{3+}$, a phenomenon which alters the lithium (de)intercalation voltage significantly.

The surface energies of LiCoO$_2$ were calculated from first principles with the Hubbard U correction on the generalized gradient approximation (GGA+U) to the density functional theory (DFT). The results are listed in Table 1 and the specific details of the models are presented in the supporting information S1. We find that the surface energies are minimized when the surface Co$^{3+}$ ions are either in the intermediate (IS) or high spin (HS) state depending on the crystallographic orientation. This change in the electronic spin state on the surface can be rationalized by a change in the surface Co crystal field due to the missing Co-O bonds. LiCoO$_2$ particles are typically dominated by the [001] surface, however this surface is not active for Li (de)intercalation. Therefore, we focus on studying the [104] and [110] surfaces, which are the two lowest energy non-polar surfaces identified by previous work. Figure 1a shows the [104] surface of LiCoO$_2$, which represents a major, low energy surface for LiCoO$_2$; this surface slices through the Co, O and Li planes and is expected to be involved in the (de)intercalation process. This represents the [100] surface of the NaCl lattice from which the ordered rocksalt LiCoO$_2$ is derived. In the bulk, and on the [001] surface, octahedrally coordinated Co$^{3+}$ ions are in the low spin (LS) state and do not have unpaired electrons. On the [104] surface, however, the Co$^{3+}$ ions are coordinated by five oxygen ions, resulting in a square pyramidal configuration. The surface energy is then lowered significantly when going from the LS configuration (1188mJ/m$^2$) to the IS (312mJ/m$^2$). Figure 1b depicts the [110] surface of LiCoO$_2$. In this orientation, the Co$^{3+}$ ions on the surface are coordinated by four oxygen ions with a pseudo-tetrahedral configuration. The surface energy is minimized from 2277mJ/m$^2$ (LS) to 1241mJ/m$^2$ (HS). The details of the crystal field splitting of 3d orbitals and the corresponding energy levels of bulk and surface cobalt ions are shown in Figure 1c. A square pyramidal crystal field breaks the degeneracy of both the t$_{2g}$ and e$_g$ orbitals, observed for octahedral symmetry, since the missing O ion along the z-direction reduces the repulsion between 2p electrons and 3d electrons in orbitals pointing towards or closer to the z-axis. A pseudo-tetrahedral crystal field with two missing bonds in the xy plane leads to lower energies for the 3d$_{xy}$ and 3d$_{yz,zx}$ orbitals. A charge density plot, which represents the difference in up and down spins, of the [104] surface clearly show the unpaired electrons as compared to the case of Co$^{3+}$ (LS) in the bulk of LiCoO$_2$. (See Figure 2a and 2b) The first principles calculations reveal that both the [104] and [110] surfaces with optimized Co coordination environments and electronic states expand normal to the surface, the displace-
ment being on the order of 0.1-0.2Å. It is important to point out that such changes in electronic spin states are also seen in first principles simulations of surfaces of CoO and Co₂O₃, as well as LiNiO₂.

To validate the hypothesis that the Co³⁺ on the surface is in an intermediate spin state by experimental spectroscopic techniques, stoichiometric LiCoO₂ samples with extremely small particle sizes (thus very large surface areas) provide the best opportunities. However, it is difficult to make stoichiometric nanoparticles of LiCoO₂ by conventional solid state or hydrothermal methods. Okubo et al. reported samples with small particle sizes (8 to 32 nm), but the samples are most likely over-stoichiometric, i.e., Li₁+xCoO₂. By contrast, the molten salt method reported earlier represents a much better approach to prepare stoichiometric LiCoO₂ nanoparticles. In this work, stoichiometric LiCoO₂ with very small particle sizes (10nm, 16nm, 20nm, 30nm and 40nm) were synthesized by using a modified molten salt method based on the previously reported method (see supporting information S2). We observed expansion of the lattice parameters from Rietveld analysis of the XRD results, similar to that observed by Okubo et al., along with magnetic susceptibility data that indicates the presence of unpaired electrons. X-ray diffraction (XRD) data indicates the formation of pure single-phase LiCoO₂ for the 20, 30 and 40 nm samples, all the peaks corresponding to the layered α-NaFeO₂ structure (Figure 3a). Lattice parameters are extracted and the c/a ratio is approximately 4.99, indicating a well-formed layered structure. Detailed information regarding the lattice parameters is given in the supporting information, S3. The reflections in the XRD pattern for the 10nm particles are broad and a quantitative refinement could not be achieved.

X-ray photoelectron spectroscopy (XPS) was carried out with a Thermo-Scientific K-Alpha spectrometer using a focused monochromatic Al Kα anode source (see supporting information S4). As shown in Figure 3b, all compounds show a Co 2p₁/₂ main peak at 779.5eV with a satellite peak at 789.5eV and a Co 2p₃/₂ main peak at 794.5eV with a satellite peak at 804.5eV. This observation confirms that the oxidation state is Co³⁺. This is strong evidence that our nano-sized LiCoO₂ samples are stoichiometric, Co³⁺ coordinated by oxygen is characterized by a strong broadening of the main line and a very intense satellite peak at 785.5 eV (Co 2p₁/₂) and 802.5 eV (Co 2p₃/₂), which are both absent in the spectra.

Figure 1. (a) {104} and (b) {110} surfaces of LiCoO₂ (c) Octahedrally, square pyramid and pseudo-tetrahedrally coordinated Co ions. (red- oxygen, green – lithium, blue – Co)

Figure 2. Spin density plot of the {104} plane of the bulk (a) and the surface (b). (Notice the scale differences in the spin density)

⁷Li magic-angle-spinning (MAS) NMR spectroscopy was performed at a magnetic field strength of 4.7 T with a spinning speed of 35 kHz to explore the Li nonstoichiometry in nano-sized LiCoO₂. Figure 4a shows that the spectra of the 10 to 30 nm particles are dominated by a single resonance at 0 ppm, as expected for a stoichiometric, diamagnetic LiCoO₂ sample. The larger NMR linewidth observed for the 10 nm sample compared to the others may be due to the presence of a trace amount (<1%) of Co₃O₄ impurity, which results in faster transverse relaxation leading to larger line width. A weak hyperfine shifted resonance is observed at +185 ppm for the 30 nm sample. This shift, along with peaks at 3, -6, -16, -40 ppm (not observed in our system here, presumably because they are too weak) were previously suggested to be associated with excess Li ions that replace Co³⁺ sites in the bulk, resulting in a defect structure discussed above, with two adjacent square-based pyramids containing two intermediate-spin Co³⁺ ions per oxygen vacancy. Even in the 30 nm sample, the percentage of excess Li is only approximately 0.6%. Two new hyperfine-shifted resonances were observed at -115 and -260 ppm (Fig. 4b), which are attributed to the presence of intermediate and/or high spin Co³⁺ (IS, HS). To the best of our knowledge, these shifts have not been reported in the literature. An analysis of the spin density on the {104} surface, as shown in Figure 2(b), indicates that the Li¹⁺ ions on this surface contain negative spin density and will thus give rise to a negative shift. Integration of the spin density around the Li nuclei to 0.8 Å, using the approach developed by Carlier et al., confirms this observation and indicates that the Li nuclei in the subsequent {104} surface is negative, while the spin density of the Li in the layer below is smaller, but positive (and thus it may be difficult to resolve from the intense bulk LiCoO₂ resonance/and or be buried under the spinning sidebands). Further calculations will focus on the direct calculations of hyperfine shifts on this and a wider range of surfaces. Importantly, quantitative fitting of the NMR spectra to extract the concentration of Li¹⁺ ions nearby paramagnetic Co ions, confirms that the percentage of paramagnetic ions increases with decreasing particle size (Supporting information S5).
Magnetic measurements were performed, using a superconducting quantum interference device (SQUID), at a magnetic field of 1.0 T in the temperature range of 5-300K. The molar magnetic susceptibility of the various sized nano-LiCoO$_2$ particles are plotted as a function of temperature in Figure 5a. The magnetic susceptibility of bulk LiCoO$_2$ is low and practically temperature independent, which is attributed to a Van Vleck-type of paramagnetism associated with diamagnetic Co$^{3+}$(LS) in the layered structure.$^{12}$ In contrast, our nano-size LiCoO$_2$ exhibits a typical Curie-Weiss behavior for T>100K. The Curie constant was determined in the temperature range of 200-300K. The Curie constant increases with decreasing particle size and fitting of the Curie constant indicates that the molar Curie constant is as high as 0.20 in 10nm-sized LiCoO$_2$. (Details on the Curie constant fitting can be found in the supporting information S6).

Careful transmission electron microscopy (TEM) examination of the nano-sized LiCoO$_2$ shows that all particles exhibit platelike morphology, plates corresponding to the {001} planes; the edges are dominated by the {104} planes justifying our computational study of this surface. (See TEM images in the supporting information S7). If the planes are terminated by the {001} surface, according to first principles computation of this work and previous work by Kramer and Ceder,$^3$ Co$^{3+}$ remains octahedrally coordinated with a low-spin configuration. The contribution of Co$^{3+}$(IS) on the {104} and the Co$^{3+}$(HS) on the {110} surfaces are most likely attributing to the abnormally high magnetic susceptibility seen in nano-size LiCoO$_2$.

The proposed electronic spin state of the surfaces relevant for (de)intercalation has significant impact on the lithium (de)intercalation voltage profile, as revealed by first principles computation performed to calculate the Li extraction potential from different surfaces. If Co$^{3+}$ remains as LS on the {104} surface, the voltage of lithium extraction is as low as 2.32V; by contrast, when a spin transition from LS to IS occurs, the voltage of lithium extraction is 3.69V, close to the bulk value of 3.65V. The narrowing energy gap between the occupied and unoccupied states in IS and HS state Co ions indicates that the insulating nature of LiCoO$_2$ can be altered due to the change in spin state. (See density of state plots in supporting information S8) This helps in explaining the fact that in the lithium half-cell with nano-sized LiCoO$_2$ as the cathode, a clear first order phase transition, associated with the metal-insulator-transition, is absent as no flat voltage plateau is observed upon the first charge$^{15}$ (also see supporting information S9).

In conclusion, it is proposed in this work that electronic spin state transitions occur on the surfaces of stoichiometric Li-CoO$_2$, where trivalent cobalt ions adjacent to the surface adopt an intermediate spin state if they are square pyramid coordinated and a high spin state if they are pseudo-tetrahedrally coordinated. This phenomenon is quantified in nano-sized stoichiometric LiCoO$_2$. We also observed in first principles calculations that both {104} and {110} surfaces with optimized electronic spin states expand normal to the surface and the displacement is of the order of 0.1-0.2Å. This work suggests that changes in electronic spin state could be a common phenomenon in transition metal oxides. The low coordinated geometries on the surface of the oxides result in spin states that are distinct from the bulk. Consequently, unique magnetic and electronic properties arise and alter the materials performance in devices. We show that in this case, the voltage profile
of (de)intercalation is dramatically changed. It is therefore promising to control the surfaces and interfaces of nano-sized materials to alter the electronic and magnetic properties, significantly different from the bulk behaviors.

![Graph of magnetic susceptibility vs temperature] Figure 5. (a) Molar magnetic susceptibilities of particle size 10nm, 16nm, 20nm, 30nm and 40nm samples as a function of temperature. (b) The Curie constants of these samples determined from the $1/\chi$ vs T. (Both are measured with sample holder corrected)

**TABLE 1.**

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<th>Surface</th>
<th>Coordination # of oxygen</th>
<th>$\gamma$ (mJ/m$^2$) with LS</th>
<th>$\gamma$ (mJ/m$^2$) This work with LS</th>
<th>$\gamma$ (mJ/m$^2$) this work spin trans.</th>
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<td>1118</td>
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**ASSOCIATED CONTENT**

Supporting Information: Computation and experiment details, fitting for Curie Constant, morphology, electrochemical measurement, density of states (DOS) plots. This material is available free of charge via the Internet at http://pubs.acs.org.

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**Author Contributions**

The manuscript was written through contributions of all authors. All authors have given approval to the final version of the manuscript.

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**ABBREVIATIONS**

XRD, XPS, SQUID, MAS-NMR, TEM

**REFERENCES**

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