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Controlling the spontaneous emission rate of monolayer MoS$_2$ in a photonic crystal nanocavity

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We report on controlling the spontaneous emission (SE) rate of a molybdenum disulfide (MoS$_2$) monolayer coupled with a planar photonic crystal (PPC) nanocavity. Spatially resolved photoluminescence (PL) mapping shows strong variations of emission when the MoS$_2$ monolayer is on the PPC cavity, on the PPC lattice, on the air gap, and on the unpatterned gallium phosphide substrate. Polarization dependences of the cavity-coupled MoS$_2$ emission show a more than 5 times stronger extracted PL intensity than the un-coupled emission, which indicates an underlying cavity mode Purcell enhancement of the MoS$_2$ SE rate exceeding a factor of 70. © 2013 Author(s). All article content, except where otherwise noted, is licensed under a Creative Commons Attribution 3.0 Unported License. [http://dx.doi.org/10.1063/1.4826679]

The recent finding that a single atomic layer of transition metal dichalcogenides can exhibit a large, direct bandgap$^{1-4}$ opens the possibility of a new range of atomically thin materials for electronic and electro-optic devices. Monolayer molybdenum disulfide (MoS$_2$) has been used to fabricate field-effect transistors (FETs) with a carrier-mobility of 200 cm$^2$ V$^{-1}$s$^{-1}$ and On/Off ratios exceeding $10^8$ at room temperature, comparable to those obtained in graphene nanoribbon-based FETs.$^5$ Optical studies have shown that monolayer MoS$_2$ exhibits a photoluminescence (PL) quantum yield that is enhanced by a factor more than $10^7$ compared with the bulk crystal.$^2,6$ However, the PL efficiency of monolayer MoS$_2$ is still very low at $\sim 10^{-2}$ because the nonradiative recombination rate $1/\tau_{nr}$ far exceeds the spontaneous emission (SE) rate $1/\tau_s$. For MoS$_2$ monolayers on SiO$_2$ substrates, values of $\tau_{nr} \sim 100$ ps and $\tau_s \sim 10$ ns were estimated at room temperature.$^2,7$ Here, we show that the SE efficiency of an MoS$_2$ monolayer can be greatly enhanced by exploiting the strong Purcell effect in photonic crystal nanocavities to shorten the radiative recombination time. After depositing an MoS$_2$ monolayer onto a planar photonic crystal (PPC) nanocavity, we observe an enhancement of the external extracted PL intensity by a factor of 5.4 above the background. This strong enhancement exists even though the collection is from both the sub-wavelength cavity mode and the surrounding focal spot region. Taking into account this spatial averaging, we deduce that the SE rate enhancement into the cavity mode corresponds to nearly a factor of 70, in close agreement with theory. These results indicate that by exploiting the strong Purcell effect in optical cavities with wavelength-scale mode volume and high quality ($Q$) factor, it is possible to achieve roughly two orders of magnitude improvement in the MoS$_2$ PL efficiency. This gain opens the door to efficient light emissions from, and strong light-matter interactions with, materials of atomic thickness.

The experiment employs PPC nanocavities fabricated in a 138 nm thick gallium phosphide (GaP) membrane using electron-beam lithography, dry etching, and wet chemical undercutting of an AlGaP sacrificial layer.$^8$ The cavity design is a linear three-missing hole (L3) defect$^9$ with a lattice spacing $a = 165$ nm and an air-hole radius $r = 0.3a$, yielding resonant modes in the wavelength range of 600 nm-700 nm to overlap the PL spectrum of the monolayer MoS$_2$. Figure 1(a) shows a scanning electron microscope (SEM) image of the PPC nanocavity before the deposition of MoS$_2$. Trenches around the PPC lattice aid in the removal of the sacrificial layer in a hydrofluoric acid bath. The monolayer MoS$_2$ is prepared by mechanical exfoliation onto a

FIG. 1. (a) SEM image of the L3 PPC nanocavity before the transfer of MoS$_2$ monolayer. (b) Optical microscope image of the exfoliated MoS$_2$ film on a polymeric sacrificial substrate. The monolayer is shown in the purple region indicated by the dashed black line. (c) Optical microscope image of a finished device. The single-layer MoS$_2$ is not visible, but its overlap with the PPC cavity is verified by the above multi-layer MoS$_2$ flake and by the fluorescence mapping image shown in Fig. 2(a).
polymeric sacrificial substrate, as shown in the optical microscope image in Fig. 1(b). Due to the optical interference, MoS$_2$ monolayer is clearly visible in the purple region indicated by the dashed black line, which is also confirmed by a micro-Raman spectroscopy. The MoS$_2$ sheet is then transferred onto PPC nanocavities through a precision transfer technique with the help of the polymeric sacrificial substrate, which is removed from the final device by high-temperature annealing. Figure 1(c) shows the finished device. An PPC nanocavity is covered uniformly by the MoS$_2$ monolayer, which is clearly distinguished by correlating the above multi-layer MoS$_2$ flake.

We characterize the device on a micro-PL confocal microscope with a 532 nm continuous-wave excitation laser, focused to a beam diameter of $\sim$400 nm and with a power of $\sim$50 $\mu$W. To study the modifications on the MoS$_2$ SE, we spatially scan the device in 50 nm steps on a piezo stage and detect the MoS$_2$ PL using an avalanche photodiode. Figure 2(a) shows the spatially resolved PL. By correlating it with the SEM image shown in Fig. 1(a), we observe four individual emission profiles of MoS$_2$ due to different substrates, as marked in Fig. 2(a). The PL spectra of the four regions are shown in Fig. 2(b). The result reveals that the PL collected from region 3, where the MoS$_2$ sheet is suspended over a 300 nm wide trench, is significantly brighter than that obtained from region 4 on the bulk GaP membrane. This is expected due to the suppression of the PL quantum yield by the substrate and the total internal reflection of the high-index GaP slab, which sharply reduces the PL collection efficiency. On both regions, the monolayer MoS$_2$ emits the same fluorescence spectrum centered around 660 nm due to the direct electronic bandgap.

On the PPC, we observe both an enhancement and a suppression of the MoS$_2$ PL emission. In region 2, due to the coupling between the periodic air-holes of the PPC lattice and the MoS$_2$ sheet, the in-plane emission channel is inhibited by the in-plane photonic bandgap, which overlaps with the emission band of the monolayer MoS$_2$. Therefore, the SE should be re-directed into near-vertical $k$-vectors within the PPC light cone. This SE redistribution and the higher collection efficiency from the PPC lattice enhance the collection of emission into the vertical direction via the suppression of emission into in-plane PPC modes. Hence, the collected photon flux from region 2 is brighter than that from the bulk GaP membrane, as confirmed from the PL spectra. However, the PL collected from the L3 defect (region 1) shows even brighter emission than that from region 2. Comparing the spectra acquired from region 1 and region 2, it is clear that this enhancement mainly results from a greatly amplified photon flux of the two peaks centered at the wavelengths of 655.4 nm and 656.9 nm. The polarization dependences of the two peaks from region 1 are then resolved by rotating a polarizer in the PL collection path of the microscope setup. The obtained spectra are shown in Fig. 3(a), where $\phi$ denotes the angle between the cavity y-axis and the polarization direction of the polarizer. These spectra indicate the two peaks at 655.4 nm and 656.9 nm are resonant modes of the L3 cavity with expected polarization and wavelength dependences given by three-dimensional finite difference time domain simulations, which also confirm other resonant modes at longer wavelength. Therefore, over the

FIG. 2. (a) Micro-PL spatial mapping of the device, showing four individual emission profiles. (b) PL spectra collected from the four different locations on the sample.
cavity defect region, the simultaneous suppression of SE into in-plane PPC modes together with the cavity mode Purcell enhancement of SE rate results in a dramatic reshaping of the MoS\textsubscript{2} SE, as was previously shown for single emitters\textsuperscript{15-17} and quantum wells\textsuperscript{12} in PPC cavities.

For simplicity, we designate the two resonant modes at 655.4 nm and 656.9 nm as mode 1 and mode 2, respectively. Fitting the peaks to Lorentzian line shapes, we find that the \( Q \) factors of the two modes are \( Q_0 = 220 \) and 320, respectively, which degrade from the initial \( Q \) factors of 880 and 800 of the unloaded cavity due to the spectrally overlap with the absorption resonance of the monolayer MoS\textsubscript{2}.\textsuperscript{2,18,19} The simulated cavity fields of modes 1 and 2 are shown in the inset of Fig. 3(a), which have mode volumes (\( V_{\text{mode}} \)) of 0.63 and 0.33 (\( \lambda/n \radian^2 \)), where \( n \) is the refractive index of GaP.

To quantitatively analyze the cavity enhancement of MoS\textsubscript{2} SE rate, we model the coupled MoS\textsubscript{2}-cavity system by considering the MoS\textsubscript{2} monolayer as a collection of excitonic dipole emitters. The exciton recombination rate is given by a sum over radiative and non-radiative recombination rates, \( \Gamma = \Gamma_r + \Gamma_{\text{nr}} \). In our experiments, the PL intensities at the resonant wavelengths show linear dependence on the excitation power, verifying that the SE processes are far below the saturation rate of the MoS\textsubscript{2} sheet. Therefore, the emission power \( P \) is proportional to \( P_{\text{in}} A \Gamma_r / (\Gamma_r + \Gamma_{\text{nr}}) \), where \( P_{\text{in}} \) is the excitation power and \( A \) is the absorbance of monolayer MoS\textsubscript{2} at the excitation wavelength. Because \( \Gamma_{\text{nr}} \gg \Gamma_r \) in MoS\textsubscript{2} and the finite collection angle of optics, we can approximate for all of our experiments that \( P \propto \eta I_{\text{in}} \Gamma_r / \Gamma_{\text{nr}} \), where \( \eta \) is the collection efficiency of the PL emission. Here, we consider the excitons as an ensemble of emitters \( \mu \) in the MoS\textsubscript{2} on a bulk substrate have a natural SE rate \( \Gamma_0(\lambda) d\lambda \) with a transition rate corresponding to the spectral range from \( \lambda \) to \( \lambda + d\lambda \). The modified SE distribution when the MoS\textsubscript{2} sheet is on the PPC nanocavity is given by

\[
\Gamma(\lambda) d\lambda = \Gamma_0(\lambda) d\lambda |F_{\text{c},0} L(\lambda)| \psi(\lambda) [\psi(\lambda)]^* F_{\text{PC}}. \tag{1}
\]

Here, \( L(\lambda) = 1/[1 + 4 Q^2 C^2 (\lambda/\lambda_c - 1)^2] \) denotes the cavity’s Lorentzian spectrum with \( \lambda_c \) as the resonant wavelength, and \( \psi = E \cdot \mu / |E_{\text{max}}| \) denotes the spatial and angular overlaps between the emitter dipole \( \mu \) and the cavity field \( E \). The factor \( F_{\text{c},0} = 3\pi \lambda_c Q_{\text{c},0} (\lambda_c/\lambda)^3 \) is the maximum SE enhancement (Purcell) factor of the cavity mode when the emitter dipole is on resonance with the cavity and spatially aligned with the cavity field. The term \( F_{\text{PC}} \) accounts for the suppression of the SE rate by the PPC lattice and modes other than the cavity mode.\textsuperscript{12,15}

The total cavity-coupled MoS\textsubscript{2} emission spectrum \( I_\text{c}(\lambda) \) with different polarizations \( \phi \) can be fitted to a model that considers both the SE rate modifications and the collection efficiencies of the cavity mode and averaged PPC leaky modes. We calculate \( I_\text{c}(\lambda) \) by integrating the SE rate given in Eq. (1) over the spatial and in-plane orientation densities of the emitter dipoles \( \rho(\mathbf{r}, \lambda, \mu) \)

\[
I_\text{c}(\lambda) = \Gamma_0(\lambda) \int d^2 r |\eta_{\text{c},0} F_{\text{c},0} L(\lambda)| \psi_D^2 \sin(\phi) + \eta_{\text{PC}} F_{\text{PC}} \rho(\mathbf{r}, \lambda, \mu). \tag{2}
\]

where \( \eta_{\text{c},0} \) and \( \eta_{\text{PC}} \) are the coupling efficiencies into the objective lens of the PL emissions coupled with the cavity mode and averaged PPC leaky modes. Due to the primarily linear polarization dependence of the cavity modes 1 and 2, the PL spectra shown in Fig. 3(a) with polarizations as \( \phi = 0^\circ \) and \( 90^\circ \) indicate the off- and on-resonance emissions. We calculate the spectrally resolved cavity-enhancement of the collected emission from the two spectra, as shown in Fig. 3(b), which is governed by

\[
I_{\text{c},0}(\lambda) = \eta_{\text{c},0} \int d^2 r |F_{\text{c},0} L(\lambda)| \psi_D^2 \rho(\mathbf{r}, \lambda, \mu) + 1. \tag{3}
\]

By integrating the far-field radiations of a dipole spectrally on- or slightly off-resonance with the cavity mode over the numerical aperture (NA = 0.95) of the objective lens, which locates on the cavity defect, we obtain coupling efficiency ratios \( \frac{\eta_{\text{c}}}{\eta_{\text{PC}}} \) for modes 1 and 2 of 87\% and 73\%, respectively.\textsuperscript{20} The overall integration of \( \langle |E \cdot \mu|/|E_{\text{max}}| \rangle \) \( ^2 \) is 0.169 and 0.079 for modes 1 and 2, respectively, as calculated from their simulated cavity fields. With the calculated \( V_{\text{mode}} \) and the \( Q \) factors derived from the experimental spectra, we calculate the maximum Purcell factor \( F_{\text{c},0} \) for modes 1 and 2 is about 26.5 and 73.8. The suppression factor \( F_{\text{PC}} \) is estimated by simulating the emission power ratio of a dipole on the L3 cavity defect and on the bulk membrane.\textsuperscript{15} The emission frequency of the dipole is chosen in the photonic bandgap of PPC but off-resonance with the cavity mode. The obtained \( F_{\text{PC}} \) is approximately 0.4, which is close to the values found in similar PPC structures.\textsuperscript{12,15,21} Combining the above calculations and the Lorentzian functions \( L(\lambda) \) of modes 1 and 2, the theoretical model described in Eq. (3) shows a good fit to the experimentally obtained enhancement spectrum, as shown in Fig. 3(b).

In conclusion, we have shown that by coupling monolayer MoS\textsubscript{2} to a PPC nanocavity, it is possible to dramatically enhance its internal quantum efficiency for transitions on resonant with the cavity. The experimental results and theoretical calculations reveal that the maximum enhancement of the MoS\textsubscript{2} SE rate by the cavity modes can be higher than 70, with a suppression factor of about 0.4 due to the PPC lattice. In this work, the strong Purcell enhancement was limited to the sub-wavelength size of the cavity; however, a high Purcell enhancement across a larger area could be realized using slow light near the bandedge of photonic crystals or coupled cavity arrays.\textsuperscript{22} The cavity-enhanced light-matter coupling in monolayer MoS\textsubscript{2} indicated by the strong Purcell effect expands the scope of solid state cavity quantum electrodynamics to atomically thin materials with large bandgaps, which has implications for a range of optical devices, including efficient photodetectors\textsuperscript{23} and electroluminescent materials.
systems, cavity-enhanced nonlinearities, and potentially even lasers employing atomically thin gain media.

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6 A. Splendiani, L. Sun, Y. Zhang, T. Li, J. Kim, C. Chim, G. Galli, and F. Wöhrle, Nano Lett. 10, 1271 (2010).