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# Crystal growth and structure analysis of perovskite chalcogenide $\text{BaZrS}_3$ and Ruddlesden-Popper phase $\text{Ba}_3\text{Zr}_2\text{S}_7$

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## Abstract

$\text{BaZrS}_3$  and  $\text{Ba}_3\text{Zr}_2\text{S}_7$  crystals were successfully grown by flux method. X-ray diffraction analyses showed space group of  $Pnma$  with lattice constants of  $a = 7.056 \text{ \AA}$ ,  $b = 9.962 \text{ \AA}$ ,  $c = 6.996 \text{ \AA}$  for  $\text{BaZrS}_3$  and  $P4_2/mnm$  with  $a = 7.071 \text{ \AA}$ ,  $b = 7.071 \text{ \AA}$ ,  $c =$

25.418 Å for  $\text{Ba}_3\text{Zr}_2\text{S}_7$ . Rocking curves with full-width-at-half-maximum of  $0.011^\circ$  for  $\text{BaZrS}_3$  and  $0.027^\circ$  for  $\text{Ba}_3\text{Zr}_2\text{S}_7$  were observed. Pole figure analysis, scanning transmission electron microscopy images and electron diffraction patterns also establish high quality of grown crystals. The octahedra tilting in the corner-sharing octahedra network are analyzed by extracting the torsion angles.

## Introduction

Perovskite chalcogenides have been gaining increased attention as a class of emerging semiconductors with rich tunability and excellent optoelectronic properties in the visible to infrared spectrum.<sup>1-10</sup> Perovskite chalcogenides can be viewed as the inorganic alternatives to hybrid organic-inorganic halide perovskites, with stable, benign, abundant composition, and ultrahigh absorption coefficients. On the other hand, perovskite chalcogenides can also be viewed as the chalcogenide counterparts of perovskite oxides, with much lower bandgap and improved response to visible and infrared light.<sup>1,11</sup> The combination of ultrahigh absorption coefficient, good carrier mobility, along with tunable band gap, good thermal and aqueous stability, and benign, earth abundant composition creates opportunities for a broad range of photonic, optoelectronic, and energy applications, including solar cells,<sup>1,2,4,6</sup> photodetectors,<sup>5,8,10</sup> lighting devices,<sup>9</sup> and photoelectrochemical devices.<sup>3</sup>

These materials were known to exist for a long time and the synthetic efforts can date back to over half a century ago.<sup>12-15</sup> Synthetic methods of ceramic samples include heating binary sulfide mixture for several weeks,<sup>16</sup> sulfurization of corresponding oxides with  $\text{CS}_2$ <sup>13,17</sup> or  $\text{H}_2\text{S}$ ,<sup>14</sup> and solid state reaction with catalytic addition of iodine.<sup>18</sup> Crystal growth efforts have been reported for perovskite related Ruddlesden-Popper phases<sup>19-21</sup> and hexagonal coordinated phases.<sup>15</sup> However, most reports on perovskite chalcogenides were limited to the structural studies of polycrystalline samples. Recently, these materials were rediscovered and explored as a class of semiconductors for optoelectronic applications. Experimental explorations of relevant physical properties started on the prototypical perovskite chalcogenide,

BaZrS<sub>3</sub>.<sup>18,22–25</sup> Synthesis of high quality samples are of vital importance in such explorations. Meng et al.<sup>22</sup> synthesized BaZrS<sub>3</sub> by conventional solid-state reaction of binary mixtures with repeated annealing. Perera et al.<sup>23</sup> synthesized BaZrS<sub>3</sub> by high temperature sulfurization of oxides with CS<sub>2</sub>. We have demonstrated synthesis of BaZrS<sub>3</sub> with catalytic addition of iodine in solid state reaction to enable one-shot, shorter synthesis.<sup>18</sup> However, growth of single crystals for BaZrS<sub>3</sub> is yet to be demonstrated, which is necessary to study the intrinsic optical properties of this model perovskite chalcogenide material. In this work, we will report the crystal growth of BaZrS<sub>3</sub>, along with its  $n = 2$  Ruddlesden-Popper phase Ba<sub>3</sub>Zr<sub>2</sub>S<sub>7</sub>.

## Experimental Section

The single crystals were grown using salt flux method in sealed quartz ampoules with BaCl<sub>2</sub> flux. Barium Sulfide powder (Sigma-Aldrich 99.9%), Zirconium powder (STREM, 99.5%), and Sulfur pieces (Alfa Aesar 99.999%) were stored and handled in an Argon-filled glove box. Stoichiometric quantities of precursor powders with a total weight of 0.5 g were mixed and loaded into a 3/4 inch diameter quartz tube with 1.5 mm thickness along with around 0.5g BaCl<sub>2</sub> (Alfa Aesar 99.998%) inside the glove box. The tube was capped with ultra-torr fittings and a bonnet needle valve to avoid exposure to the air, until it was evacuated and sealed using a blowtorch. The sealed tubes were about 12 cm in length. The tubes were heated to 1050 °C with a ramping rate of 0.3 °C/min, held at 1050 °C for 40 hours and then cooled to 400 °C with a cooling rate of 1 °C/min, and then allowed to naturally cool down after that by shutting off the furnace. The obtained samples were washed with deionized water repeatedly to remove the access flux, and dried with acetone and isopropyl alcohol. In some cases, residue flux after washing were removed by sonicating the crystals in isopropyl alcohol.

The thin film X-ray diffraction and rocking curve measurements were carried out in a Bruker D8 Advance X-ray diffractometer in parallel beam configuration, using a germanium

(004) two-bounce monochromator for Cu  $K_{\alpha 1}$  radiation. X-ray pole figure analysis was carried out using a Bruker D8 Discover GADDS with a Co  $K_{\alpha}$  source, 1/4 Eulerian cradle and Vantec-2000 area detector. The pole figure data was collected with  $1^{\circ}$  resolution in rotation at tilts of  $25^{\circ}$ ,  $50^{\circ}$ ,  $75^{\circ}$  to ensure coverage over the entire pole sphere. The two-dimensional diffraction data was reduced with Bruker Multex area 2 software and the pole figures were analyzed in MTEX texture analysis software.

The single crystal X-ray diffraction data were collected on a Bruker SMART APEX DUO 3-circle platform diffractometer and using Mo  $K_{\alpha}$  radiation ( $\lambda = 0.71073 \text{ \AA}$ ) monochromatized by a TRIUMPH curved-crystal monochromator. The diffractometer was equipped with an APEX II CCD detector and an Oxford Cryosystems Cryostream 700 apparatus for low-temperature data collection. The crystals were mounted in a Cryo-Loop using Paratone oil. The measurement was performed at 100 K. A complete hemisphere of data was scanned on  $\omega$  ( $0.5^{\circ}$ ) at a detector distance of 50 mm and a resolution of  $512 \times 512$  pixels. A total of 2520 frames were collected. The frames were integrated with the Bruker SAINT software package using a SAINT V8.38A (Bruker AXS, 2013) algorithm. The structure was solved and refined using the Bruker SHELXTL Software Package.<sup>26</sup>

Scanning electron microscopy (SEM) images were obtained in a JEOL JEM-7001F analytical field-emission scanning electron microscope equipped with EDAX Apollo X  $10 \text{ mm}^2$  EDS. The SEM images and energy dispersive X-ray spectroscopy (EDS) spectra were acquired with 15 kV accelerating voltage and a working distance of 15 mm. A thin layer of Pt was sputter coated on the crystals to mitigate the charging effects. Magnifications ranging from  $100\times$  to  $10000\times$  were used to check the consistency of chemical composition ratios across the crystals. The spectra shown were recorded at  $400\times$ . We used the commercially pre-analyzed BaS as a standard, and measured its EDS side by side with the crystals to serve as calibration for Ba:S ratios. The BaS powders were sealed under argon until entering SEM chamber to minimize potential oxidation. The Ba:S ratio obtained in BaS across multiple magnifications was consistently to be 1.14:1, justifying the need for such calibration.

The scanning transmission electron microscopy (STEM) images were acquired using an aberration-corrected FEI Titan (Thermo Fisher Scientific Inc.) transmission electron microscope operated at an accelerating voltage of 300 kV. Samples for TEM observations were prepared by focused ion beam (FIB) milling using a Ga-ion beam at an accelerating voltage of 30 kV, followed by a cleaning/polishing procedure by Ar-ion milling at 1.5 kV and 700 V to remove FIB induced residual Ga and surface amorphization in the sample.

## Results and Discussion

The single crystals were grown using salt flux method with  $\text{BaCl}_2$  flux. We have also tried growing these crystals with other salts, such as KI and  $\text{BaCl}_2/\text{MgCl}_2$  eutectic. The results were similar, the salt removing process was notably easier, but the obtained crystals were not as large. In the case of  $\text{BaZrS}_3$ , the predominant morphology of the obtained crystals was cube-like with sharp edges and well-defined surfaces that correspond to crystal facets, as shown in the images in Fig.1(b),(e). The obtained  $\text{BaZrS}_3$  crystals were up to 500  $\mu\text{m}$  in size, and showed shiny/reflective metallic luster when looked at normal incidence. In the case of RP phases, the attempt for single phase crystal growth by varying the stoichiometry of precursors was not successful. Salt flux growth is a slow process, and the energy barriers to get across RP phase boundaries are presumably shallow as they possess similar chemistry and structure. As a result, obtained crystallites were a mixture of RP phase, with the relative yield affected by the starting stoichiometry. For  $\text{Ba}_3\text{Zr}_2\text{S}_7$ , the predominant crystal morphology was square platelets, with similar shiny metallic luster on the surface. However, when looked under microscope, the top platelet crystal facets are not as well defined, and show layered like terraces on the surfaces, as shown in Fig.1(c),(f), presumably due to the stacking of perovskite slabs and rock-salt layer along the  $c$  axis. The isolated  $\text{Ba}_3\text{Zr}_2\text{S}_7$  crystal pieces were up to 300  $\mu\text{m}$  in size. A small amount of  $\text{Ba}_4\text{Zr}_3\text{S}_{10}$  crystallites were also found in certain cases. However, the effort to consistently grow large  $\text{Ba}_4\text{Zr}_3\text{S}_{10}$  crystals was

not successful.

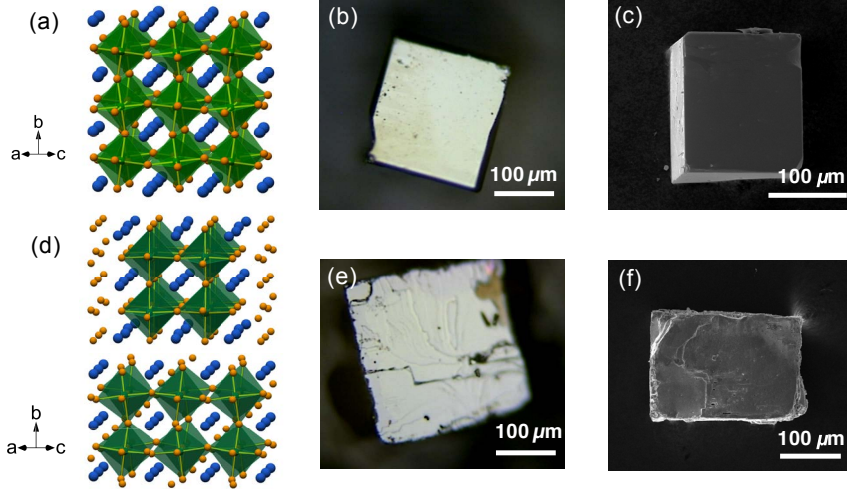


Figure 1: Schematic crystal structure for (a)  $\text{BaZrS}_3$  and (d)  $\text{Ba}_3\text{Zr}_2\text{S}_7$ . The blue, orange, and green spheres represent Ba, S, and Zr, respectively. The  $\text{ZrS}_6$  octahedra are highlighted. Optical pictures of (b)  $\text{BaZrS}_3$ , (e)  $\text{Ba}_3\text{Zr}_2\text{S}_7$  crystals. SEM images of (c)  $\text{BaZrS}_3$ , (f)  $\text{Ba}_3\text{Zr}_2\text{S}_7$  crystals.

We have performed single crystal X-ray diffraction analysis for the crystals. For  $\text{BaZrS}_3$ , we observed twinning in the  $\text{BaZrS}_3$  crystals and had to break the crystals into smaller pieces for a reasonable solution to the diffraction analysis. A lustrous dark red plate-like specimen of  $\text{BaZrS}_3$ , approximate dimensions  $0.005 \text{ mm} \times 0.008 \text{ mm} \times 0.008 \text{ mm}$ , was used for the X-ray crystallographic analysis. The measurement was performed at 100 K. The integration of the data using an orthorhombic unit cell yielded a total of 4218 reflections to a maximum  $\theta$  angle of  $30.54^\circ$  ( $0.70 \text{ \AA}$  resolution), of which 798 were independent (average redundancy 5.286, completeness = 94.8%,  $R_{\text{int}} = 5.33\%$ ,  $R_{\text{sig}} = 4.55\%$ ) and 623 (78.07%) were greater than  $2\sigma(F^2)$ . The final cell constants of  $a = 7.056(3) \text{ \AA}$ ,  $b = 9.962(4) \text{ \AA}$ ,  $c = 6.996(3) \text{ \AA}$ , volume =  $491.8(3) \text{ \AA}^3$ , are based upon the refinement of the XYZ-centroids of 1919 reflections above  $20\sigma(I)$  with  $9.169^\circ < 2\theta < 60.88^\circ$ . Data were corrected for absorption effects using the multi-scan method (SADABS). The ratio of minimum to maximum apparent transmission was 0.764. The calculated minimum and maximum transmission coefficients (based on crystal size) are 0.7310 and 0.8970. The structure was solved and refined using

the space group  $Pnma$ , with  $Z = 4$  for the formula unit,  $BaZrS_3$ . The final anisotropic full-matrix least-squares refinement on  $F^2$  with 29 variables converged at  $R_1 = 3.36\%$ , for the observed data and  $wR_2 = 7.57\%$  for all data. The goodness-of-fit was 1.162. The largest peak in the final difference electron density synthesis was  $1.804 \text{ e}^-/\text{\AA}^3$  and the largest hole was  $-1.788 \text{ e}^-/\text{\AA}^3$  with an root-mean-square deviation of  $0.356 \text{ e}^-/\text{\AA}^3$ . On the basis of the final model, the calculated density was  $4.386 \text{ g/cm}^3$  and  $F(000)$ ,  $576 \text{ e}^-$ .

The detailed structural parameters are shown in Table 1. Based on these parameters, schematic crystal structures were constructed, as shown in Fig.1(a).  $BaZrS_3$  is isostructural with  $GdFeO_3$  and adopts the distorted perovskite structure with a space group of  $Pnma$ , which agrees well with previous polycrystalline studies.<sup>13,14,18</sup> The octahedra share corners to form a three-dimensional network. This structure differs from ideal cubic perovskite in several ways. If we consider a ‘‘pseudo-cubic’’ structure with three axes pointing along the directions the octahedra are connecting corners. While the  $b$  axis of the orthorhombic cell is aligned with one of the directions,  $a$  and  $c$  axes are close to the face diagonals of the other two pseudo-cubic axes. The pseudo-cubic cell constant, or effectively the spacing between octahedra, is  $4.968\text{\AA}$  along  $[101]$  and  $[10\bar{1}]$  directions, and  $4.981\text{\AA}$  spacing along  $[010]$  direction in  $BaZrS_3$ . Thus, to grow high quality epitaxial films of  $BaZrS_3$ , one can use substrates with a square symmetry and lattice constants of around  $4.97\text{\AA}$  for commensurate epitaxy or around  $3.51\text{\AA}$  for incommensurate epitaxy.

One key feature in  $BaZrS_3$  crystal structure is the tilting of  $ZrS_6$  octahedra. The tilting along  $b$  axis is in phase, with adjacent octahedra layers in  $ac$  plane rotating towards the same direction and remain perfectly overlapped when viewed along  $b$  axis, as shown in Fig.2(a). The tilting along  $[101]$  and  $[10\bar{1}]$  directions are out of phase, with adjacent octahedra layers twisted in different directions, as shown in Fig.2(c). The tilting amplitude can be calculated by analysing the torsion angle of the Zr-S bonds. For in-phase tilt, the torsion angle is half of difference between  $180^\circ$  and the Zr-S-S-Zr torsion angle linking octahedra in the same layer, as shown in Fig.2(c). For out-of-phase tilt, the torsion angle is half of the S-Zr-Zr-S



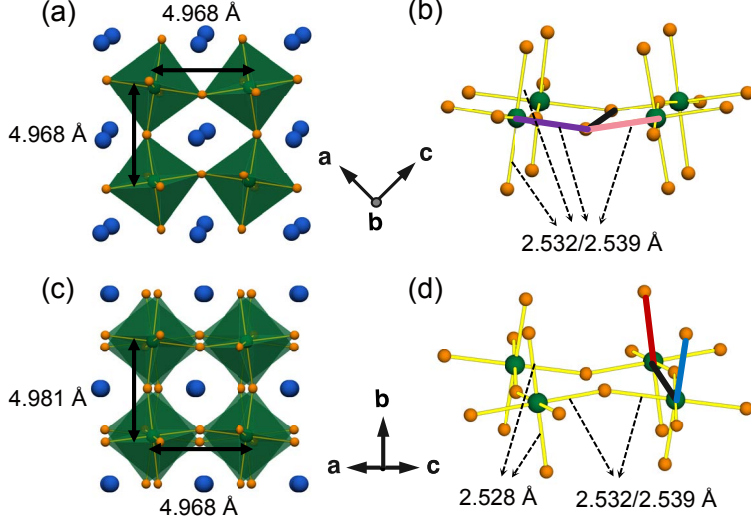


Figure 2: (a) Schematic structure viewed along  $b$  axis in  $\text{BaZrS}_3$  shows the in phase tilt where the tilting of adjacent octahedra layers are overlapping. (b) Ball-stick model in perspective view showing the Zr-S-S-Zr torsion angle (purple-black-pink) to calculate the in-phase tilting angle. (c) Schematic structure viewed along  $(101)$  direction in  $\text{BaZrS}_3$  shows the out of phase tilt where the tilting of adjacent layers are in opposite direction. (d) Ball-stick model in perspective view showing the S-Zr-Zr-S torsion angle (blue-black-red) to calculate out-of-phase tilting angle. The octahedra spacing (Zr-Zr distance) and octahedra shape (Zr-S bond length) along different directions are marked in the schematics.

torsion angle linking octahedra in two adjacent layers, as shown in Fig.2(d). The extracted tilting angle along  $[101]$  and  $[10\bar{1}]$  directions are the same,  $-7.027^\circ$  (negative denoting out-of-phase tilt) and while the tilting angle along  $[010]$  is  $9.0005^\circ$  (positive denoting in-phase tilt). The obtained results show that  $\text{BaZrS}_3$  adopts a Glazer tilting system of  $a^-b^+a^-$  (or standardized  $a^+b^-b^-$ ).<sup>27</sup> It is worth noting that although Zr is sitting in the center of  $\text{ZrS}_6$  octahedra, the octahedra are slightly distorted, with three pairs of Zr-S bonds of bond length  $2.532\text{\AA}$ ,  $2.539\text{\AA}$ , and  $2.528\text{\AA}$ , respectively. The shortest pair of bonds are along  $[010]$ , while octahedra with pairs of  $2.532\text{\AA}$  and  $2.539\text{\AA}$  Zr-S bonds are alternated along  $[101]$  and  $[10\bar{1}]$ . If we compare  $[010]$  to  $[101]/[10\bar{1}]$ , the shorter Zr-S bonds ( $2.528\text{\AA}$  compared to  $2.532\text{\AA}$  and  $2.539\text{\AA}$ ) and larger octahedra (Zr-Zr) spacing ( $4.981\text{\AA}$  compared to  $4.968\text{\AA}$ ) is enabled by a smaller Zr-S-Zr bond angle ( $160.236^\circ$  compared to  $156.905^\circ$ ), due to the larger octahedra tilting amplitude around  $[010]$  ( $9.0005^\circ$  compared to  $7.027^\circ$ ). The in-phase tilting around  $b$

axis also results in net displacement of S atoms, and consequently displacement of Ba atoms in *ac* plane.

For  $\text{Ba}_3\text{Zr}_2\text{S}_7$ , there were several possible structural variables from previous structural studies,  $I4/mmm$ ,<sup>20</sup>  $P4_2/mnm$ ,<sup>19</sup> and  $Cccm$ .<sup>28</sup> We performed single-crystal XRD studies on the grown crystals at 100 K and found that the crystals adopted the  $P4_2/mnm$  space group, as shown in Table 2. A dark red prism-like piece of  $\text{Ba}_3\text{Zr}_2\text{S}_7$  crystal, approximate dimensions of 0.056 mm  $\times$  0.086 mm  $\times$  0.118 mm, was used for the X-ray crystallographic analysis in the same setup. The integration of the data using a tetragonal unit cell yielded a total of 29151 reflections to a maximum  $\theta$  angle of 28.32° (0.75 Å resolution), of which 898 were independent (average redundancy 28.762, completeness = 99.7%,  $R_{int}$  = 4.3%,  $R_{sig}$  = 1.14%) and 743 (82.74%) were greater than  $2\sigma(F^2)$ . The structure was solved and refined using the space group  $P4_2/mnm$ , with  $Z = 4$  for the formula unit,  $\text{Ba}_3\text{Zr}_2\text{S}_7$ . The final cell constants of  $a = 7.071(2)$  Å,  $b = 7.071(2)$  Å,  $c = 25.418(5)$  Å,  $\alpha = 90^\circ$ ,  $\beta = 90^\circ$ ,  $\gamma = 90^\circ$ , volume = 1270.9(8) Å<sup>3</sup>, are based upon the refinement of the XYZ-centroids of 6657 reflections above  $20\sigma(I)$  with  $6.412^\circ < 2\theta < 56.35^\circ$ . Data were also corrected for absorption effects. The ratio of minimum to maximum apparent transmission was 0.724. The final anisotropic full-matrix least-squares refinement on  $F^2$  with 40 variables converged at  $R_1 = 2.15\%$ , for the observed data and  $wR_2 = 5.16\%$  for all data. The goodness-of-fit was 1.139. The largest peak in the final difference electron density synthesis was 1.067 e<sup>-</sup>/Å<sup>3</sup> and the largest hole was -0.722 e<sup>-</sup>/Å<sup>3</sup> with an root-mean-square deviation of 0.146 e<sup>-</sup>/Å<sup>3</sup>. On the basis of the final model, the calculated density was 4.28 g/cm<sup>3</sup> and F(000), 1440 e<sup>-</sup>. The obtained crystal structure is shown in Fig.1(d). The stacking of perovskite slabs and rock salt layer is along [001]. Within the perovskite layers, the octahedra are connecting corners along [110] and  $[1\bar{1}0]$ . In the slightly distorted tetragonal  $P4_2/mnm$  structure, the octahedra tilting is only present along one of the in-plane directions. The out-of-phase tilting with a tilting angle of -7.863° keeps alternating from one block of double-layer to the next due to the screw axis  $4_2$  around [001].

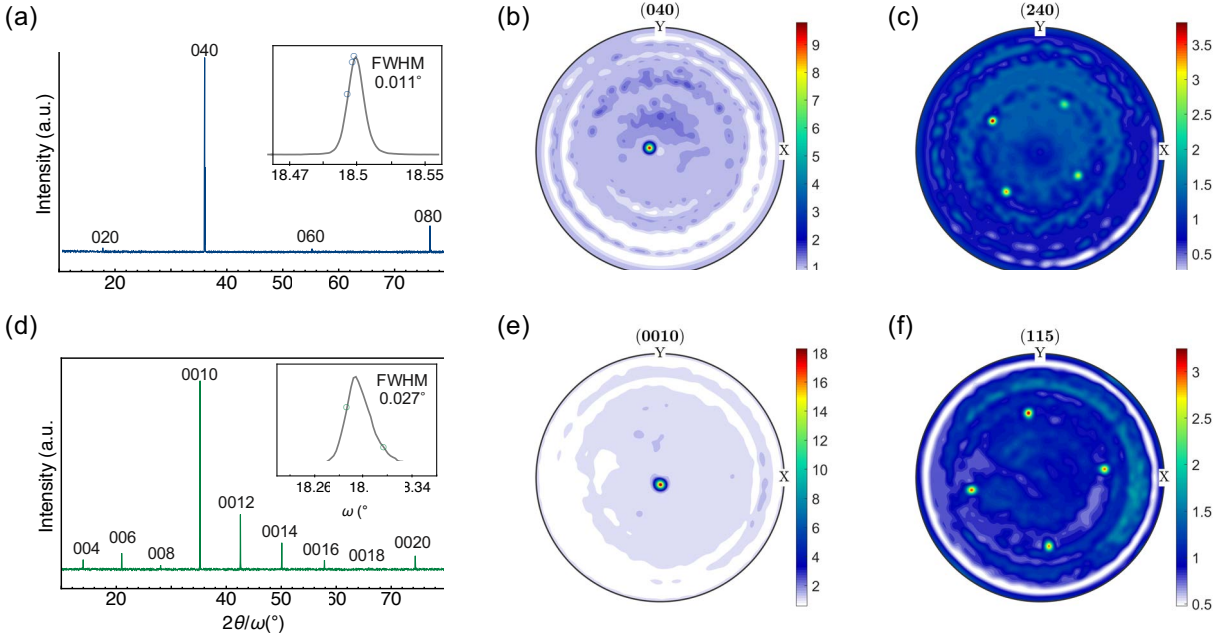


Figure 3: Out-of-plane XRD of individual crystal for (a)  $\text{BaZrS}_3$  and (e)  $\text{Ba}_3\text{Zr}_2\text{S}_7$ . The insets are rocking curves of the most intense peaks. Pole figures for (b)  $\text{BaZrS}_3$  040, (c)  $\text{BaZrS}_3$  240, (e)  $\text{Ba}_3\text{Zr}_2\text{S}_7$  0010, and (f)  $\text{Ba}_3\text{Zr}_2\text{S}_7$  115 plotted as orientation density function. The color bars indicate multiples of distribution function.

For the bigger crystals of  $\text{BaZrS}_3$  and  $\text{Ba}_3\text{Zr}_2\text{S}_7$ , we performed out-of-plane thin film XRD scans on isolated crystal pieces in an X-ray diffractometer with monochromatic radiation. For  $\text{BaZrS}_3$ , a set of narrow  $0k0$  type reflections were observed, as shown in Fig.3(a). The most intense reflections have a full-width-at-half-maximum (FWHM) of less than  $0.04^\circ$ . We also performed high resolution rocking curve (RC) measurements. The inset is the rocking curves of the most intense 040 reflection. Notably, we obtained a RC FWHM of  $0.011^\circ$  for  $\text{BaZrS}_3$  crystal, indicating highly oriented structural quality of the single crystal. We note that cubic looking crystals could also show 101 type termination. As mentioned earlier, the difference in lattice spacing between these two directions is less than 0.3%. out-of-plane XRD scan of  $\text{Ba}_3\text{Zr}_2\text{S}_7$  showed a set of  $00l$  type peaks, as shown in Fig.3(d). This proves that the terminating crystal facets with layered-like features are the  $(00l)$  plane. For  $\text{Ba}_3\text{Zr}_2\text{S}_7$ , the rocking curve width is notably larger, with a FWHM of  $0.027^\circ$  for 0010 reflection, presumably arising from higher degree of mosaicity associated with the layered

nature of the crystal structure. The phase and orientation of the single crystal samples were confirmed with X-ray pole figure analysis. The 040 and 240 pole figures of a  $\text{BaZrS}_3$  crystal are shown in Fig.3(b), (c). Fig.3(b) indicates (040) planes are aligned to the normal of the sample surface, confirming the termination facet. The spot is slightly off the center due to the small misorientation of the sample when mounted on the stage. Fig.3(c) exhibits the 4 fold symmetry of the (042) poles, resulting from very similar lattice parameters for  $a$  and  $c$  lattice parameters. The 0010 and 115 pole figures of a  $\text{Ba}_3\text{Zr}_2\text{S}_7$  crystal were shown in Fig.3(e), (f). Similarly, Fig.3(e) confirms (001) termination and Fig.3(f) exhibits the 4 fold symmetry of the (115) poles in the tetragonal crystal system.

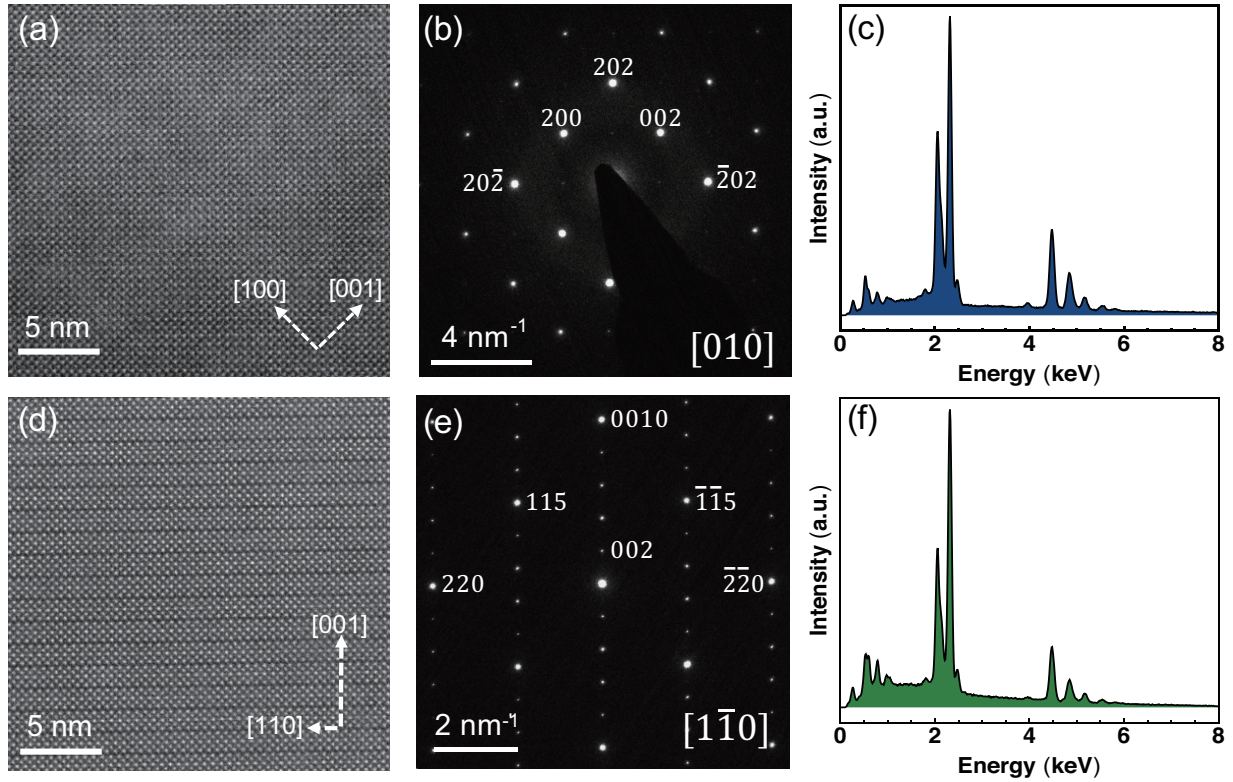


Figure 4: The cross-sectional STEM images of (a)  $\text{BaZrS}_3$  viewed along  $b$  axis and (d)  $\text{Ba}_3\text{Zr}_2\text{S}_7$  viewed along (110) direction. Corresponding SAED patterns of (b)  $\text{BaZrS}_3$  and (e)  $\text{Ba}_3\text{Zr}_2\text{S}_7$ . EDS spectra of (c)  $\text{BaZrS}_3$  and (f)  $\text{Ba}_3\text{Zr}_2\text{S}_7$  samples taken at 400X. The obtained Ba:Zr:S ratios are 1.06:1:3.04 and 1.56:1:3.45, respectively.

We further performed scanning transmission electron microscopy (STEM) studies on the crystals. The high-angle annular dark-field (HAADF) image of a  $\text{BaZrS}_3$  and a  $\text{Ba}_3\text{Zr}_2\text{S}_7$

crystal and corresponding selected area electron diffraction (SAED) patterns are shown in Fig.4. In Fig.4(a), we can see the extended 3D corner sharing octahedra network and a pseudo-cubic pattern in BaZrS<sub>3</sub>. We were not able to resolve the subtle Ba displacements in STEM and the HAADF collection angle excludes scattering from the lighter S atoms. Given the small difference in spacing between the three quasi-cubic directions, (020), (101) and (10 $\bar{1}$ ), the image and diffraction pattern can be indexed with either [010] or [101] being the zone axis. In Ba<sub>3</sub>Zr<sub>2</sub>S<sub>7</sub>, as shown in Fig.4(d), one can see the double-layer perovskite blocks stacked along [001], the blocks are offset by half a unit cell along the face diagonal of the in-plane square lattice. The diffraction pattern in Fig.4(e) can be indexed with a zone axis of [1 $\bar{1}$ 0] (or equivalently [110]). The denser diffraction spots along (001) is due to the much larger lattice constants of *c* axis. The bright 0010 spot and weaker 004, 006, 008 spots resemble very well the intensity of the peaks we see in the out-of-plane scan in Fig.3(d). Such highly symmetric SAED patterns indicate high crystallinity of the grown crystals over the selected region and agree well with the pole figure analysis. Chemical composition analysis by EDS with varying locations and magnifications on the crystals showed only expected elements in a consistent ratio, as shown in Fig.4(c),(f).

Table 1: Space group, lattice constants, atomic coordinates, equivalent isotropic atomic displacement parameters ( $\text{\AA}^2$ ), and anisotropic atomic displacement parameters ( $\text{\AA}^2$ ) from X-ray crystallographic analysis for BaZrS<sub>3</sub>.

Space Group	<i>Pnma</i>			Temperature	100 K	
Lattice	<i>a</i> ( $\text{\AA}$ )	<i>b</i> ( $\text{\AA}$ )	<i>c</i> ( $\text{\AA}$ )			
Constants	7.056(3)	9.962(4)	6.996(3)			
Atomic Sites	<i>x/a</i>	<i>y/b</i>	<i>z/c</i>		U(eq)	
Ba1	0.54544(10)	0.25	0.99030(10)		0.00696(17)	
S1	0.4949(4)	0.25	0.4382(4)		0.0070(6)	
S2	0.2896(3)	0.46782(19)	0.7896(3)		0.0060(4)	
Zr1	0.5	0.5	0.5		0.0031(2)	
	U <sub>11</sub>	U <sub>22</sub>	U <sub>33</sub>	U <sub>23</sub>	U <sub>13</sub>	U <sub>12</sub>
Ba1	0.0091(3)	0.0040(3)	0.0078(3)	0	-0.0004(3)	0
S1	0.0101(14)	0.0029(12)	0.0079(13)	0	0.0008(9)	0
S2	0.0075(9)	0.0054(8)	0.0052(9)	0.0012(7)	0.0015(7)	-0.0009(8)
Zr1	0.0038(5)	0.0021(4)	0.0035(5)	0.0001(4)	0.0001(5)	0.0000(3)

Table 2: Space group, lattice constants, atomic coordinates, equivalent isotropic atomic displacement parameters ( $\text{\AA}^2$ ), and anisotropic atomic displacement parameters ( $\text{\AA}^2$ ) from X-ray crystallographic analysis for  $\text{Ba}_3\text{Zr}_2\text{S}_7$ .

Space Group	$P4_2/mnm$			Temperature	100 K	
Lattice	$a(\text{\AA})$	$b(\text{\AA})$	$c(\text{\AA})$			
Constants	7.071(2)	7.071(2)	25.418(5)			
Atomic Sites	$x/a$	$y/b$	$z/c$	U(eq)		
Ba1	0.74627(4)	0.74627(4)	0.5	0.01153(12)		
Ba2	0.26045(3)	0.73955(3)	0.81869(2)	0.01114(10)		
S1	0.2854(2)	0.7146(2)	0.5	0.0161(4)		
S2	0.5	0.5	0.60860(8)	0.0127(4)		
S3	0.22238(15)	0.77762(15)	0.69788(5)	0.0153(3)		
S4	0.0	0.0	0.58414(8)	0.0113 (4)		
S5	0	0.5	0.59528(6)	0.0132(3)		
Zr1	0.25039(5)	0.74961(5)	0.59974(2)	0.00769(12)		
	$U_{11}$	$U_{22}$	$U_{33}$	$U_{23}$	$U_{13}$	$U_{12}$
Ba1	0.01347(15)	0.01347(15)	0.0076(2)	0	0	0.0014(2)
Ba2	0.01371(12)	0.01371(12)	0.00600(16)	-0.00022(8)	0.00022(8)	-0.00017(13)
S1	0.0205(6)	0.0205(6)	0.0072(8)	0	0	0.0030(8)
S2	0.0097(5)	0.0097(5)	0.0188(9)	0	0	0.0027(7)
S3	0.0201(4)	0.0201(4)	0.0057(5)	-0.0001(4)	0.0001(4)	0.0006(6)
S4	0.0098(5)	0.0098(5)	0.0144(8)	0	0	0.0024(7)
S5	0.0094(6)	0.0094(6)	0.0208(6)	0	0	-0.0039(5)
Zr1	0.00801(15)	0.00801(15)	0.0070(2)	0.00023(11)	-0.00023(11)	0.00014(19)

## Conclusion

The growth of high quality  $\text{BaZrS}_3$  crystals was achieved using flux growth method with  $\text{BaCl}_2$  flux. Through X-ray diffraction analysis of the crystal, space group of  $Pnma$  with lattice constants of  $a = 7.056\text{\AA}$ ,  $b = 9.962\text{\AA}$ ,  $c = 6.996\text{\AA}$  was extracted for  $\text{BaZrS}_3$ . The octahedra tilting system was identified to be  $a^+b^-b^-$ , with in-phase tilting of  $9.0005^\circ$  around  $[010]$  and out-of-phase tilting of  $-7.027^\circ$  around  $[101]$  and  $[10\bar{1}]$ . STEM images and electron diffraction pattern were used to probe the local crystallinity of the crystal. High resolution X-ray rocking curve (FWHM of  $0.011^\circ$ ) and pole figure analysis for 040 and 240 revealed overall high crystallinity of the grown crystal.  $\text{Ba}_3\text{Zr}_2\text{S}_7$  crystal was also grown using similar method by varying the starting precursor stoichiometry ratio. X-ray diffraction analysis of  $\text{Ba}_3\text{Zr}_2\text{S}_7$  crystal showed a  $P4_2/mnm$  structure and a rocking curve FWHM of  $0.027^\circ$ . STEM image with  $[110]$  zone axis and corresponding diffraction pattern, along with pole figure analysis for 0010 and 115 were shown to further establish high crystallinity. We expect the crystal growth of perovskite chalcogenides to enable more advanced spectroscopic and transport

measurements to study their optoelectronic properties.

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## Supporting Information Available

The following files are available in supplementary information.

- BaZrS<sub>3</sub>.cif: cif file from X-ray diffraction analysis of BaZrS<sub>3</sub> crystal
- Ba<sub>3</sub>Zr<sub>2</sub>S<sub>7</sub>.cif: cif file from X-ray diffraction analysis of Ba<sub>3</sub>Zr<sub>2</sub>S<sub>7</sub> crystal

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