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Phase change materials for nano-polaritonics: a case study of hBN/VO₂ heterostructures

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Abstract:

Polaritonic excitation and control in van der Waals (vdW) materials exhibit superior merits than conventional materials and thus hold new promise for exploring light matter interactions. In this work, we created vdW heterostructures combining hexagonal boron nitride (hBN) and a representative phase change material – vanadium dioxide (VO_2). Using infrared nano-spectroscopy and nano-imaging, we demonstrated the dynamic tunability of hyperbolic phonon polaritons in hBN/ VO_2 heterostructures by temperature control in a precise and reversible fashion. The dynamic tuning of the polaritons stems from the change of local dielectric properties of the VO_2 sublayer through insulator to metal transition by the temperature control. The high susceptibility of polaritons to electronic phase transitions opens possibilities for applications of vdW materials in combination with correlated phase change materials.

Van der Waals (vdW) heterostructures^{1, 2} assembled together by vdW forces offer a versatile novel platform to control physical properties and quantum phenomena³. Specifically, as building blocks for the heterostructures, vdW materials have shown innovative electronic, optical and magnetic properties. They harbor Dirac⁴ and valley⁵ electrons, support superconductivity^{6, 7} and ferromagnetism^{8, 9}, emit light^{10, 11} and exhibit topologically protected states^{12, 13}, as well as many other effects. Stacking various vdW materials into heterostructures can intertwine these properties in a hybrid and controllable fashion for advanced functionalities^{1, 2}. In addition, phase change behaviors as important basics for energy storage^{14, 15}, reconfigurable electronics and optics¹⁶, have just been explored in vdW systems recently^{17, 18}. Therefore, in addition to some preliminary approaches¹⁹⁻²¹, the exploration of heterostructures composed of vdW crystals and phase change materials is imperative.

In this work, we build a novel vdW heterostructure (Figure 1a) by combining hexagonal boron nitride (hBN) and a canonical phase change material vanadium dioxide (VO₂) for the control of polaritons^{22, 23}. Polaritons are hybrid light-matter modes involving collective oscillations of electromagnetic dipoles in materials. One representative polaritonic vdW crystal is hBN where hyperbolic phonon polaritons (HP²s) are supported²⁴⁻³¹. HP²s in hBN originate from anisotropic phonon resonances which naturally facilitate Type I ($\epsilon_z < 0$, $\epsilon_t > 0$) and Type II ($\epsilon_z > 0$, $\epsilon_t < 0$) hyperbolicity. This natural hyperbolicity offer advantages over previous efforts using metamaterials^{32, 33} due to the high momentum cutoff of light set by the interatomic spacing in natural lattices^{34, 35}. Therefore, the hyperbolicity in hBN holds promising applications in subdiffractional focusing^{34, 35}, emission engineering³⁶ and light steering^{26, 30, 33}. Despite the above merits, HP²s in hBN are insensitive to external stimuli due to the insulating nature of the crystals. Dynamic tuning of HP²s in hBN remains limited to electrostatic gating in graphene/hBN

heterostructures³⁷ based primarily on the electronic and plasmonic properties of graphene. Here we report on nano-spectroscopy and nano-imaging results revealing the dynamic tuning of phonon resonances and HP²s in hBN/VO₂ heterostructures via the control of temperature. We remark that optical switching of surface phonon polaritons with phase change materials has been previously achieved in Ge₃Sb₂Te₆ on a quartz substrate³⁸. However, dynamic tuning of polaritons and especially the hyperbolicity haven't been reported therein. We hereby demonstrate the hybridization between polaritons in hBN and insulator-to-metal transition (IMT) in VO₂ via nano-imaging the heterostructures inside the Type II hyperbolic region. We also show the dynamic tuning of spectroscopic phonon resonances in both Type I and Type II region by varying the temperature of the heterostructure.

Scattering-type scanning near-field optical microscopy (s-SNOM) is used to study the hybrid polariton and phase transition in hBN/VO₂ heterostructures. Briefly, we illuminate the tip of an atomic force microscope (AFM) with infrared (IR) quantum cascade lasers (QCL, solid red arrow, Figure 1a) and collect the back-scattered IR signal (dashed red arrow). The scattered signal is demodulated at higher harmonics of the tip tapping frequency to yield essential near-field signal. The s-SNOM can record optical signals inaccessible by conventional optics with a resolution down to 10 nm (Methods) and has been extensively used to investigate nano-optical phenomena, including polaritons in vdW materials^{22, 24, 39, 40} and nanoscale phase transitions^{41, 42}. The VO₂ film studied in this work has a thickness of 50 nm and was deposited on a [110]_R TiO₂ substrate to ensure a large conductivity jump (five orders of magnitude) during the IMT⁴¹. This VO₂/TiO₂ system also renders a unidirectional phase separation which permits an anisotropic electronic phase modulation⁴³.

The $[110]_R$ VO_2 undergoes anisotropic IMT over the temperature range from 320 K to 345 K⁴¹. The evolution of this IMT was first characterized with s-SNOM at the surface of VO_2 films: from the homogenous insulating state to metallic nucleation clusters, and finally unidirectional metallic stripes. These different stages of IMT with increasing temperature are plotted at the hBN/ VO_2 interfaces in Figure 1a to Figure 1d. We note that the s-SNOM recorded IMT is nondispersive in the mid-IR as identical images were obtained regardless of the IR frequency (Figures 1d-f, hBN/ VO_2 interfaces).

In the heterostructures of hBN on VO_2 , nano-optical features of the phase transition were revealed at the hBN top surface via the hybridization between HP^2 s and the IMT. Once illuminated with the IR light (Figure 1a, red arrow), the metalized AFM tip launches propagating HP^2 s by bridging the momentum from photons to polaritons^{22, 34, 35, 44}. The phase boundaries between the insulating and metallic state in hBN/ VO_2 interfaces (Figures 1d-f) act as effective reflectors for HP^2 s. Ring-shape polariton fringes appear on the hBN top surface (Figure 1b) due to standing wave interference between reflected and launched HP^2 s. These fringes can connect with each other as the density of metallic nucleation sites increases at a higher temperature in the IMT of VO_2 (335 K, Figure 1c). As the temperature further increases, metallic stripes emerge on the surface of VO_2 and the corresponding HP^2 s fringes align into a similar stripe shape on the hBN top surface (Figure 1d). Our interpretation of hybrid polariton-phase-transition for the observed results in Figures 1a-d is supported by the s-SNOM image at a shorter IR wavelength ($\lambda_{\text{IR}} = 6.4 \mu\text{m}$, Figure 1e). At this IR frequency, the polariton fringes become narrower, revealing a shorter HP^2 s wavelength, in accord with the frequency – momentum dispersion of HP^2 s^{24, 27}. The hybrid polariton-phase-transition is further verified in a control experiment: no evident nano-optical features can be observed at a much shorter IR wavelength ($\lambda_{\text{IR}} = 5.7 \mu\text{m}$, Figure 1f) out-side-of the hyperbolicity

regime. In comparison to previous works relying on topographic reflectors/launchers for nanopolaritons^{24, 39, 40, 45}, the conducting stripes in VO₂ here are predominantly electronic reflectors and do not reveal any evident topographic features in AFM measurements. The reflection of HP²s can be attributed to local permittivity variations that are *electronically* formed at the phase boundaries in VO₂.

Having established the hybridization between polaritons in hBN and the IMT in VO₂, we now demonstrate the dynamic tuning of the phonon resonances by performing Fourier Transform Infrared Nano-spectroscopy (nano-FTIR) on the hBN/VO₂ heterostructure. The spectroscopic nano-FTIR response is averaged over the sample surface to avoid areal-dependent polaritonic effects (Figure 1). hBN supports z axis and xy plane phonon resonances in the mid-IR that are recorded as spectral peaks in the low (Figure 2a, Type I) and high (Figure 2b, Type II) frequency regions, respectively. At room temperature, the spectral features are similar to reported results in hBN/SiO₂ structures³⁷. As the temperature increases, interesting spectral evolutions in the heterostructure of hBN/VO₂ can be observed in both Type I and Type II regions. In the Type I region (Figure 2a), the spectral intensity of the z axis phonon increases as VO₂ undergoes the IMT at the increasing temperature. In the Type II region (Figure 2b), on the contrary, the spectral intensity of the xy plane phonon decreases at the increasing temperature. While the intensity of the nano-FTIR spectra shows opposite trends in the two spectral regions, both the Type I and Type II resonances red-shift (black dotted arrows in Figures 2a and b) during IMT with increasing temperature. Therefore, both the intensity and the peak frequency of the hBN phonon resonance can be tuned by the control of temperature-dependent IMT in VO₂.

The tuning of phonon resonances stems from the modification of hyperbolic polaritons in hBN/VO₂ heterostructures by varying the temperature. We account for this effect by performing

the simulation of frequency (ω) – momentum (k_t) dispersion (Figures 2c and d) in hBN/VO₂ heterostructures. Our nano-FTIR probes spectroscopic response of materials with in-plane momenta (k_t) that can be effectively coupled to the s-SNOM. The nano-FTIR spectra are therefore related to optical modes within a finite momentum region scoped by the time-averaged near-field coupling weight function⁴⁶ $G = \langle k_{xy}^2 e^{-2k_{xy}z_d} \rangle_t$ (white dashed curves in Figures 2c, d). Here z_d is the distance between sample surface and the simulated point dipole⁴⁶. In hBN, multiple branches of HP²s (shown as false color in Figures 2c, d) in Type I and Type II regions contribute to the z -axis and xy -plane phonon resonance observed in the nano-FTIR spectra, respectively (300 K, black curves in Figures 2a, b). The phase transition in VO₂ is modeled as a variable dielectric ground plane under the hBN crystal. The complex dielectric constants of VO₂ at the fully insulating and metallic phases are extracted from previous experiments⁴⁷. As VO₂ undergoes the IMT, Type I HP²s shift into the strong coupling region (white dashed curves) whereas in the Type II region, HP²s shift outside of the region. This is evident by comparing the $\omega - k_t$ dispersion of the hBN on insulating (false color in Figures 2c and 2d) and on metallic (red curves in Figures 2c and 2d) VO₂. Therefore, during the IMT, the coupling between the s-SNOM with more polaritonic modes can account for the increasing spectral intensity in the Type I region. In contrast, coupling with fewer polaritonic modes in the Type II region will cause the decrease of the spectral intensity. Moreover, with increasing metallicity, the hybrid HP²s-IMT modes red-shift in both Type I and Type II region, in accordance with our experimental observations at the increasing temperature (Figures 2a-b).

The nano-imaging and nano-spectroscopy results augmented with the frequency – momentum dispersion simulations in Figures 1 and 2 demonstrate dynamic tuning of polaritons in heterostructures of hBN/VO₂ made possible by the phase transition. Effective redirection of HP²s in hBN has also been achieved using metallic nucleation and stripes in VO₂ as an electronic

reflector. The dynamic tuning of nano-polaritons and hyperbolicity in hBN can be attributed to the delicate control of the dielectric environment of the polaritonic medium using IMT in VO₂. Future studies will be directed towards ultrafast pump-probe control⁴⁸ of the tuning of nano-polaritons with phase change materials. We also envision the possibility of engineering local patterns³⁸ in phase change materials for memory polaritonics, polaritonic circuits, and other advanced nanophotonic functionalities⁴⁹. Furthermore, the methodology we utilized in hBN/VO₂ can be extended for the tuning of nano-polaritons in heterostructures combining phase change materials with other vdW materials, including graphene⁵⁰, transition metal dichalcogenides¹⁷, topological insulators⁵¹ and black phosphorus⁵² as well as metamaterials and metasurfaces where the polaritonic modes are also sensitive to the dielectric surroundings.

During the preparation of this manuscript, we became aware of another experimental work on phonon polaritons in hBN/VO₂ heterostructures⁵³.

Methods

Experimental setup

The infrared (IR) nano-imaging and Fourier transform infrared nano-spectroscopy (nano-FTIR) experiments on hBN/VO₂ heterostructures were performed using a scattering-type scanning near-field optical microscope (s-SNOM). We use a commercial s-SNOM system (www.neaspec.com) based on a tapping-mode atomic force microscope (AFM) and a commercial AFM tip (tip radius ~ 10 nm) with a PtIr₅ coating. In the experiment, we illuminate the AFM tip by monochromatic quantum cascade lasers (QCLs) (www.daylightsolutions.com) and a broadband source from a difference frequency generation system (www.lasnix.com) to cover a frequency

range of $900 - 2300 \text{ cm}^{-1}$. The s-SNOM nano-images were recorded by a pseudo-heterodyne interferometric detection module with an AFM tapping frequency 280 kHz and tapping amplitude around 70 nm. In order to remove the background signal, we demodulate the s-SNOM output signal at the 3rd harmonics of the tapping frequency.

Sample fabrication

The vanadium dioxide (VO_2) substrates studied in this report are deposited on $[110]_R \text{TiO}_2$ substrates ($1 \text{ cm} \times 1 \text{ cm}$) by temperature-optimized sputtering using the reactive bias ion beam deposition in a mixed Ar and O_2 atmosphere. These substrates show consistently anisotropic THz and DC conductivity⁴¹. A 1% ~ 2% tensile strain along c_R axis is present in these substrates due to the lattice mismatch between TiO_2 and VO_2 . Hexagonal boron nitride (hBN) crystals are mechanically exfoliated onto the VO_2/TiO_2 substrates.

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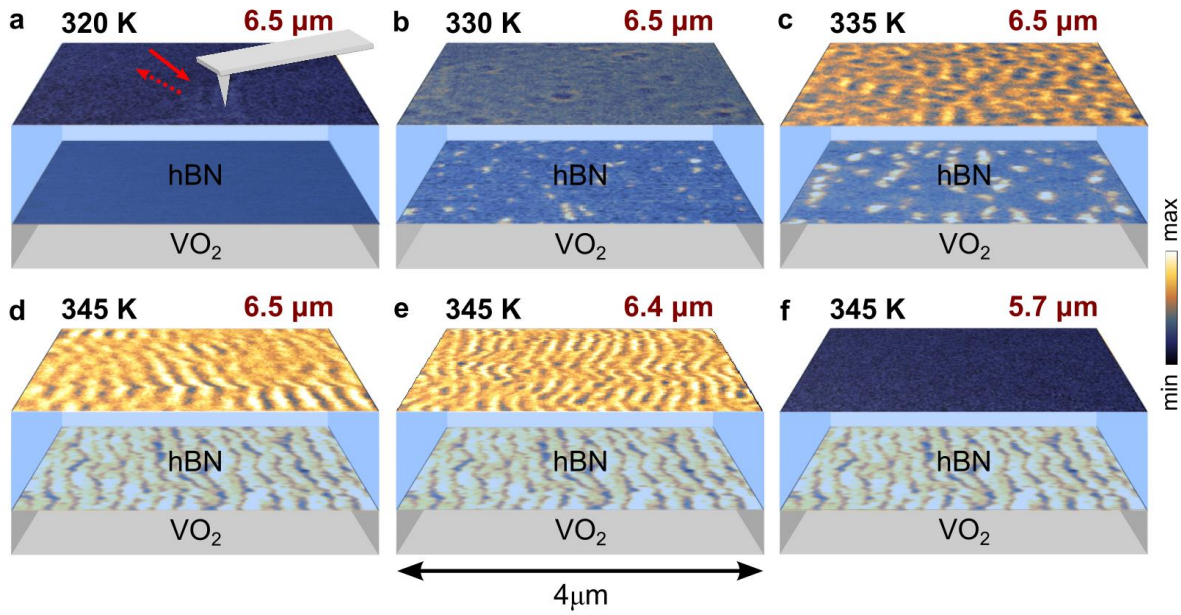


Figure 1. s-SNOM nano-images reveal the hyperbolic phonon polaritons and insulator to metal phase transition in hBN/VO₂ heterostructures. s-SNOM images of hyperbolic phonon polaritons (plotted on hBN top surface) affected by insulator to metal transition (hBN/VO₂ interface) recorded at the temperature and incident IR wavelength of 320K and 6.5 μm (a), 330K and 6.5 μm (b), 335K and 6.5 μm (c), 345K and 6.5 μm (d), 320K and 6.4 μm (e) and 320K and 6.7 μm (f). Thickness of the hBN: 73 nm.

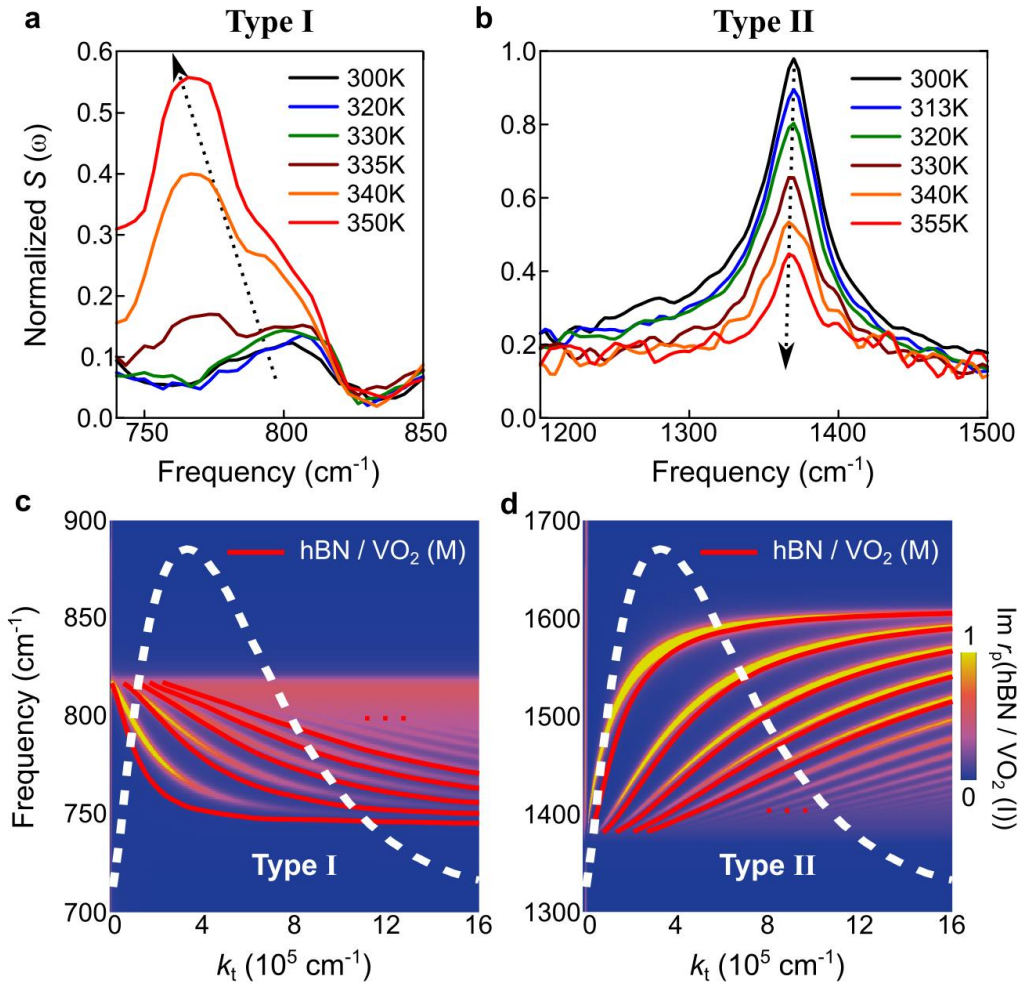


Figure 2. Tuning hyperbolic phonon polaritons in hBN/VO₂ heterostructures by temperature control. **a, b**, nano-FTIR spectra of hBN/VO₂ at various temperature in Type I (**a**) and Type II (**b**) hyperbolic region. **c, d**, Simulation of the hybridization between hyperbolic phonon polaritons and insulator to metal transition in hBN/VO₂ heterostructures in Type I (**c**) and Type II (**d**) hyperbolic region. The polariton dispersion are plotted with red curves at metallic state and with false color at insulating state of VO₂, respectively. The white dashed curves indicate the near-field coupling weight function. hBN thickness: 73 nm.