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Stable and High Performance AlGa_N Self-Aligned-Gate Field Emitter Arrays

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Abstract—AlGa_N alloys are promising for field emission devices due to their low electron affinities. However, there have been limited demonstrations of AlGa_N vacuum transistors so far. This paper combines a new self-aligned-gate (SAG) process and digital-etching tip sharpening to demonstrate three-terminal AlGa_N SAG field emitter arrays (FEAs). These devices show a turn-on voltage of 19.5 V and an anode current density (J_A) of 100 mA/cm² at an overdrive voltage of 20 V, which are comparable with best Si devices. The AlGa_N SAGFEAs can operate in DC mode at a fixed gate-emitter voltage (V_{GE}) with J_A of 3-5 mA/cm² for at least 5 hours without a significant degradation. The gate leakage does not increase after the long DC operation, suggesting high-performance and stable AlGa_N vacuum transistors.

Index Terms—AlGa_N, field emission, vacuum transistor

I. INTRODUCTION

VACUUM devices are excellent candidates for harsh-environment and high-frequency electronics thanks to their radiation hardness and scattering-free electron transport [1]. Si and metal field emitters (FEs) have been developed into mature technologies in recent years [2]–[5]. III-Nitride semiconductors promise to further improve FEs thanks to their tunable electron affinities [6], [7]. With a lower electron affinity, electrons could tunnel from the semiconductor surface to vacuum more easily, leading to devices with a lower operating voltage. Ga-polar heavily-n-doped GaN self-aligned-gate field emitter arrays (SAGFEAs) have been demonstrated and improved with sharpened tips by wet-based digital etching (DE) recently [8], [9]. The gate-emitter turn-on voltage ($V_{GE,on}$) in these devices is comparable to the state-of-the-art Si devices, while the relatively low current density and device stability are still unsolved issues. Reference [8] observed that the gate insulator under the probing region breaks when V_{GE} approaches 50 V, while no damage was observed in the FEA region. Layout optimization is thus expected to improve device stability.

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In addition to GaN, AlGa_N alloys with high-Al ratio are expected to provide better performance than GaN because of reduced electron affinities [6]. However, most work on these materials has focused on two-terminal field emitters [10]–[13]. Very limited demonstrations of AlGa_N three-terminal FEAs have been reported [14], [15], even though these devices are necessary for accurate benchmarking with other materials' systems. Schottky-diode-type electron emission devices based on the low electron affinities of AlGa_N alloys have also been demonstrated; however, their power consumption is not competitive with the SAGFEAs [16], [17]. In this work, we have fabricated AlGa_N SAGFEAs with a similar SAG geometry and DE recently demonstrated for GaN SAGFEAs [8], [9]. We show a clear performance improvement, compared to simultaneously-fabricated GaN devices. For these AlGa_N SAGFEAs, a turn on voltages ($V_{GE,on}$) of 19-22 V and anode current densities (J_A) approaching 100 mA/cm² at an overdrive voltage ($V_{OV} = V_{GE} - V_{GE,on}$) of 20 V were demonstrated, which are similar to the best Si devices. Additionally, these AlGa_N SAGFEAs show over 5 hours of stable DC operation for anode current density (J_A) of 3-5 mA/cm² with no increase of gate leakage. These results show the great potential for III-Nitride devices to outperform Si devices in the future.

II. DEVICE FABRICATION

Baseline GaN SAGFEAs are fabricated on a GaN-on-Si wafer grown by Enkris Semiconductor, Inc. by metal organic chemical vapor deposition (MOCVD) [8]. The structure consists of a 1.4 μm n⁺⁺-Ga_N (Si: 1×10^{19} cm⁻³) layer on a 1.4 μm buffer layer grown on the Si substrate. AlGa_N SAGFEAs are fabricated on a 1 cm \times 1 cm coupon grown at the Georgia Institute of Technology. This AlGa_N material consists of a 420-nm n⁺ AlGa_N layer and a 200-nm non-intentionally-doped AlN layer, which are grown by molecular beam epitaxy (MBE), on a commercial AlN template [18]–[20]. The AlN template is grown on the sapphire substrate via hydride vapor phase epitaxy (HVPE). The Al composition in this AlGa_N material is 37.6%, as measured by x-ray diffraction (XRD). The Hall electron concentration of the n⁺ AlGa_N layer is about 3.9×10^{19} cm⁻³.

Fig. 1(a) shows the fabrication process flow, and the details of different processing steps can be found in prior work [8]. The III-N tips are firstly formed by dry etching and are then sharpened by DE to obtain sub-20 nm tip width [9].

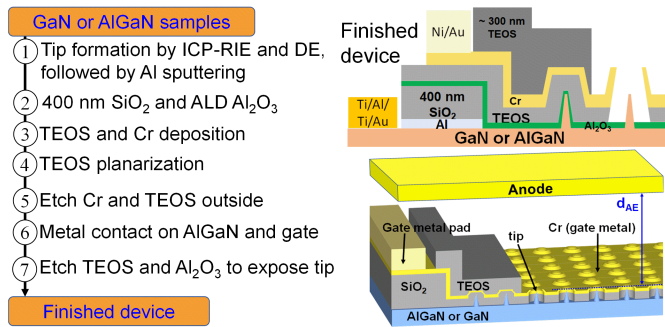


Fig. 1. Process flow, geometry of the finished device, and 3D image of the measurement setup featuring an anode electrode suspended on top of the SAGFEA. The anode-emitter distance (d_{AE}) is fixed at about 2 mm in all measurements.

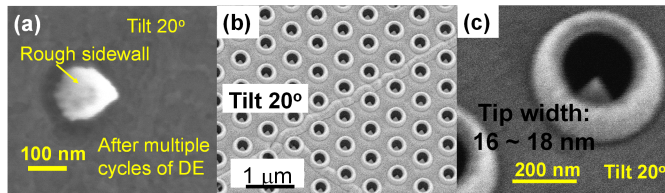


Fig. 2. Tilted SEM images of (a) an AlGaIn tip after DE, (b) large-area and (c) zoom-in SEM images of a finished AlGaIn FEA. Compared to GaN tips, the sidewall of AlGaIn tips after DE is rough [9].

Contrary to GaN, it is observed that the AlGaIn sidewalls become rough after DE (Fig. 2(a)). The rough sidewalls can potentially provide additional electron emission sites and increase gate leakage. After tip formation, an additional 400-nm plasma enhanced chemical vapor deposition (PECVD) SiO₂ is added with respect to our prior work in [8] to prevent early breakdown happening under the gate pad. A 10-nm Al₂O₃ is then deposited by atomic layer deposition (ALD) to protect III-N tips from subsequent plasma dry etching. In the last step, tips are exposed by a two-step etching. The tetraethyl orthosilicate (TEOS) is firstly etched by timed dry etching, and the Al₂O₃ and residual TEOS are then etched by shortly dipping the sample in buffer oxide etchant (BOE). The scanning electron microscopy (SEM) images of a finished AlGaIn FEA with sub-20-nm tip width are shown in Fig. 2(b) and (c).

III. RESULTS AND DISCUSSION

After SEM inspection, samples are loaded into a ultra-high vacuum (UHV) system for measurement at a base pressure of $1\text{-}2 \times 10^{-9}$ Torr. The anode is a suspended 0.5-mm-diameter tungsten ball which can be moved to position on top of the measured FEA. The anode-emitter distance (d_{AE}) is kept around 2 mm in all measurements in this work.

The transfer and output characteristics after proper conditioning of one of the best AlGaIn SAGFEAs are shown in Fig. 3. This AlGaIn FEA consists of 150×150 tips, and the FEA area is about $96 \mu\text{m} \times 80 \mu\text{m}$. The transfer characteristics are stable as there is no clear deviation between 3 sequential measurements. The device turns on at a gate-emitter voltage (V_{GE}) of 19.5 V at which point the anode current (I_A) increases to 10 pA from the noise level. The maximum anode

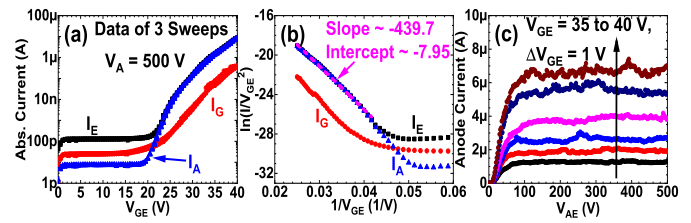


Fig. 3. (a) Transfer characteristics, (b) corresponding Fowler-Nordheim plot, and (c) output characteristics of one of the best AlGaIn FEAs. This FEA consists of 150×150 tips.

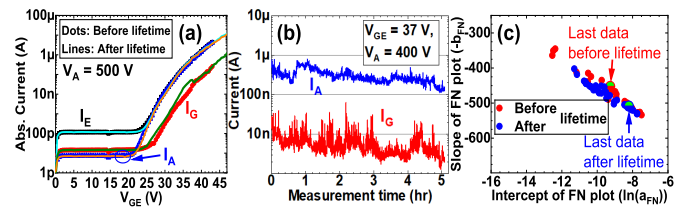


Fig. 4. (a) Transfer characteristics before and after lifetime test, (b) lifetime test, and (c) Seppen-Katamuki (S-K) chart of an AlGaIn FEA. This FEA consists of 150×150 tips. Anode current of $1 \mu\text{A}$ is equal to about 13 mA/cm^2 .

current is $8 \mu\text{A}$, which corresponds to a current density J_A of about 100 mA/cm^2 . Moreover, though the AlGaIn sidewalls are rough after DE (Fig. 2(a)), the gate leakage (I_G) is still at least one-order-of-magnitude lower than the anode current. The Fowler-Nordheim (F-N) plot of I_A shows a clear straight line with a slope ($-b_{FN}$) of -439.7 (V) (Fig. 3(b)). The output characteristics of this AlGaIn device shows a stable anode current for anode-emitter voltages (V_{AE}) above 100 V. This large V_{AE} is due to space-charge limit (Child's law) [21], [22], and it can be reduced by integrating the anode into the device structure and reducing, in that way, the anode-emitter distance, d_{AE} [23]. The device is stable and I_G does not increase after the measurements.

A different AlGaIn SAGFEA with 150×150 tips is measured for both transfer characteristics and DC operation lifetime (Fig. 4). Based on the transfer characteristics (Fig. 4(a)), the device does not show significant degradation after a total 5-hr DC operation at a fixed V_{GE} of 37 V (Fig. 4(b)). At the beginning of the lifetime test, I_A was about 400 nA, which corresponds to a current density J_A of about 5 mA/cm^2 . Both I_A and I_G gradually decrease during the lifetime test, and the measurement stops when the I_A goes below 200 nA. F-N parameters for multiple transfer characteristics before and after lifetime test are plotted in the Seppen-Katamuki (S-K) chart (Fig. 4(c)). As the data points stay in the same region of the S-K chart, it is clear that the device does not degrade significantly [24]. The variability during the lifetime measurement could be related to noise introduced by micro-vibrations happening during the measurement, as the anode is a suspended metal ball in the measurement system, which is not fixed to the device.

The F-N parameters of 4 different AlGaIn SAGFEAs and a baseline GaN device fabricated through the same process are summarized and compared with our prior work in the S-K chart shown in Fig. 5(a). Low $|b_{FN}|$ and large $\ln(a_{FN})$

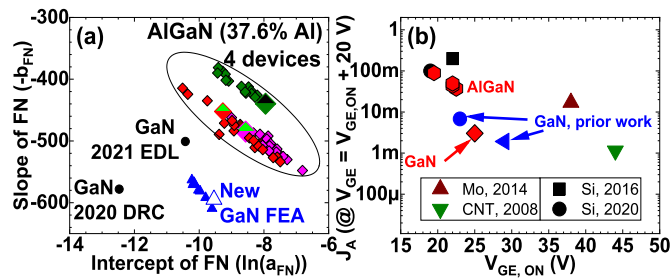


Fig. 5. (a) S-K chart of fabricated III-N self-align-gated FEAs and (b) benchmark of different materials FEAs [3], [4], [8], [26]–[28]. The same-color labels in (a) represent F-N parameters of different transfer characteristics from one FEA after conditioning. The red labels in (b) represent devices in this work.

are desired for lower operating voltage and higher anode current. Although the performance of the baseline GaN device is slightly worse than prior work [8], the changes in device fabrication allow the new GaN device to be more stable than the one in reference [8], and the early breakdown in the insulator layer under the probing region is eliminated. Furthermore, the AlGaIn devices clearly show better performance than GaN devices thanks to their potentially-lower work function. The effective work function is typically lower if the data points are located at a more upper-right region in the S-K chart [24]. Ideally, heavily doped n-type AlGaIn alloy with a higher Al composition ratio should provide better performance because of a lower electron affinity. However, surface states, doping activation energies, and defects, which are generated during growth and device fabrication, are related to the Al composition ratio and will affect the surface effective work function of the fabricated tips [9], [25]. Moreover, high-Al-composition AlGaIn materials are prone to form oxide in air, which will also change the energy barriers for electron field emission. Therefore, there might be an optimum Al composition ratio of the AlGaIn alloy for the field emission devices, while it still requires more detailed study on different aspects mentioned above.

The III-N SAGFEAs demonstrated in this work are also compared with SAGFEAs based on other materials (Fig. 5(b)) [3], [4], [8], [26]–[28]. When setting an overdrive voltage ($V_{OV} = V_{GE} - V_{GE,ON}$) of 20 V, the J_A of our AlGaIn SAGFEAs approaches 100 mA/cm^2 and is comparable with state-of-the-art Si SAGFEAs. Based on the F-N parameter ($b_{FN} = 439.7 \text{ V}$), our AlGaIn SAGFEAs can potentially outperform Si devices. However, the defects introduced in the AlGaIn devices by the dry etching and DE (Fig. 2(a) and (b)) make these devices prone to break at higher V_{GE} . A strong electric field at high V_{GE} in the oxide layer covering the unexposed tips (as shown in the cross-section geometry in Fig. 1) can also cause gate oxide breakdown. These unexposed tips are necessary to extend out the gate metal from the FEA region to the gate pad region, and further optimization is necessary to mitigate or eliminate this failure mechanism. With further optimization on different etching process and device geometry, the device is expected to be more stable and to be able to support a higher gate voltage. More detailed study on device

stability and lifetime at different current densities can then be done for better III-N SAGFEAs in the future.

IV. CONCLUSION

Three-terminal 37.6%-Al AlGaIn SAGFEAs are demonstrated with a turn-on voltage of 19.5 V at $I_A = 10 \text{ pA}$ and a J_A of 100 mA/cm^2 at $V_{GE} = 40 \text{ V}$. Thanks to the lower effective work function, the operation voltage and current density of AlGaIn SAGFEAs are better than GaN devices, and competitive with state-of-the-art Si devices. These AlGaIn SAGFEAs can operate in DC mode with J_A of $3\text{--}5 \text{ mA/cm}^2$ at a fixed V_{GE} for 5 hours without a clear degradation, suggesting stable field emission devices. Though further study on material properties and fabrication optimization is needed, these results show that AlGaIn SAGFEAs have great potential for high-frequency and harsh-environment applications.

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The GaN-on-Si coupon used in this work was obtained from the same 6-inch wafer reported in prior work [8], provided by Enkris, Inc. The AlGaIn materials were grown by MBE on AlN templates and characterized at the Georgia Institute of Technology. The device fabrication and characterization was conducted in MTL, NSL, and MIT.nano at MIT.

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