Chapter 3. Step Structures and Epitaxy on Semiconductor Surfaces

3.1 Anisotropic Coarsening of Self-Assembling Periodic Grooves

Sponsors

Joint Services Electronics Program
Grant DAAH04-95-1-0038

National Science Foundation
Grant DMR 94-23641

Faceting refers to the phase transformation at which an initially uniform surface decomposes into coexisting domains of different orientation. Studies of faceting to date have established its close analogy to the phase separation of a binary mixture. Generally, following a quench from a uniform phase into a coexistence region, domains of the low temperature phases form and subsequently coarsen in time. For binary mixtures, the mechanism of coarsening (ripening) is well-known: larger domains with smaller domain wall curvature accrete material at the expense of smaller domains with larger domain wall curvature. Application of this principle results in the prediction that the average domain size grows as a power law versus time with an exponent of 1/3. In addition, the domain morphology exhibits dynamic scaling behavior so that the distribution of domain sizes or the scattering function, when scaled to the average domain size, appears self-similar at different times. These predictions have been realized experimentally in many systems.

For faceted surfaces, however, the domain walls that must disappear so that coarsening may proceed are linear edges, requiring a different mechanism. Although faceting has been of considerable interest for many decades, pioneering studies of faceting kinetics have focused mainly on the early stage nucleation and growth. Only recently has the subject of late stage coarsening begun to receive attention.

Of particular relevance to the present paper, it was found that when a miscut Si(113) surface was quenched from above a faceting temperature to below, quasi-periodic grooves formed and subsequently coarsened in time. The width of the grooves was observed to grow following approximately a 1/6 power law versus time, and the scattering function from the grooves revealed a dynamic scaling behavior. Another interesting observation was made on a grooved alumina surface, where a rapid increase in the aspect ratio of the grooves was noted as a function of annealing time.

Subsequently, Milner proposed a theory that reproduces the 1/6 power law and in addition predicts that the length of grooves would increase as the third...
power of their width, \(^{6}\) i.e., as the \(1/2\) power of time. Our motivation was to directly test the theory by quantifying the time-dependent aspect ratio of coarsening grooves on Si(113), by means of time-resolved x-ray scattering. Preliminary work by atomic force microscopy (AFM) on a quenched surface indicated an aspect ratio of approximately 50. Simultaneous measurement of the evolution of two length scales that differ by nearly two orders of magnitude posed a technical challenge. This was met by utilizing an area detector\(^{10}\) in a glancing-angle reflection geometry, which nicely matches the instrumental resolution to the requisite length scales. Our results generally endorse the theory.\(^{11}\) The simplicity of the theory and its apparent success lead us to suspect that it will be applicable to other faceting transitions and, more generally, to different types of uniaxial systems. Despite the prevalence of uniaxial phases in nature, the kinetics of domain formation in such systems remains largely unexplored.\(^{12}\)

We begin with a brief review: At non-zero temperature, steps on a crystal surface are subject to long-wavelength fluctuations. The relation between amplitude (\(h\)) and wavelength (\(M\)) of the fluctuations is determined in a statistical manner by the stiffness of the steps\(^{13}\): \(\langle h^2 \rangle \equiv M k_B T / \Sigma\). The theory asserts that a bunch of \(N\) steps on a faceted surface may itself be treated as a (macro) step possessing a stiffness that is \(N\) times larger than the stiffness of a single step \((\Sigma_{\text{step}} = \Sigma / N)\) and a mobility that is \(N\) times smaller than that of a single step \((\Gamma_{\text{step}} = \Gamma / N)\). Faceting proceeds via coalescence of adjacent step bunches as they thermally fluctuate and collide with each other. The rate of coalescence is controlled by the time to develop fluctuations of sufficiently large amplitude, i.e., \(h \equiv L\) where \(L\) is the mean separation between step bunches. It follows that \(L^2 \equiv M k_B T / \Sigma_{\text{step}}\). Because \(N\) and \(\Sigma_{\text{step}}\) increase linearly with \(L\), this relationship between amplitude and wavelength of the colliding step bunches yields \(L^2 \equiv M k_B T / \Sigma\), where \(\Sigma\) is the mean step separation. The theory furthermore takes the characteristic time for a collision between two step bunches to be \(\tau \equiv k_B T / \Gamma_{\text{step}} \Sigma_{\text{step}} / k_B T / \Gamma_{\text{step}} \Sigma_{\text{step}}\), assuming that step fluctuations are determined by adatom attachment and detachment.\(^{15}\) As a consequence, \(L\) will increase as a \(1/6\) power of time in the coarsening process of the step bunches. \(M\) increases more rapidly, with a coarsening exponent of \(1/2\). The step-free facets bounded by step bunches should become increasingly elongated along the step edge direction with increasing time.

Experiments were performed at beamline X20C at the National Synchrotron Light Source. X-rays with a wavelength of \(\lambda = 1.4\) A were selected by WSi multilayers with a relative bandpass of \(\sim 10^{-5}\). A CCD x-ray area detector\(^{15}\) was mounted 0.92 m downstream of the sample.

After annealing the sample at 1530 K for 60 s, we cooled the surface to 1250 K, where the surface is known to have a uniformly stepped morphology. The scattering from such a surface is a single peak with a resolution-limited width corresponding to specular reflection. We then quenched the surface to 1195 K and collected the scattered intensity every 4 s with an exposure time of 1 s. Examples are displayed in Figure 1 for times 10, 100, 1000, and 10000 seconds after the quench.

In addition to the central peak which corresponds to specular reflection, two vertically displaced peaks are observed. They may be identified as the first order diffraction peaks from the periodic grooves. With increasing time they grow in intensity, narrow in width, and move in towards the center. The vertical position of the peaks is related to the groove periodicity via \(\varphi_y = 2 \pi / L\), so that the growth of \(L\) versus time is evident from these data. Equally evident is the dramatic decrease in the horizontal width of the groove peaks, which arises from the increasing extent of ordering along the grooves. Especially striking is the change in the shape of the peaks. Beginning elongated along \(Q_x\) at early times, the peaks evolve to appear nearly circular in shape at late times. In fact, because of resolution effects, the

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length of the grooves in the x-direction is much larger than the period in the y-direction even at the earliest times.

Our results for the peak position ($\epsilon_y$) and the deconvoluted full widths of the peaks at half maximum ($k_x^+$ and $k_y$) are shown in Figure 2. The width of the specular peak remains resolution-limited at all times. Evolution of the horizontal width proceeds at a much faster rate than that of the vertical width, as shown in Figure 2. Beginning at $4 \times 10^4$ Å at the earliest times measured, the two peaks evolve to $2 \times 10^4$ Å and $4 \times 10^6$ Å, respectively. The behavior of each appears consistent with a power law versus time. The best fit for times less than 2000 s yields $k_x^+ = 0.00077 t^{-0.44}$ Å$^{-1}$ for the peak on the facet side and $k = 0.0066 t^{-0.35}$ Å$^{-1}$ for the peak on the step-bunch side. The exponents 0.44 and 0.35 for $k_x^+$ are unquestionably different from 0.16 for $\epsilon_y$ and $k_y$. The width of the peak on the facet side follows a power law up to 4000 s, at which time it reaches our resolution limit. The measured exponent of 0.44 is close to the predicted exponent of 0.52, given the uncertainties at early times (due to counting statistics) and at very late times (due to instrumental resolution). The dashed line in Figure 2 indicates the expected 1/2 power law. Relating $k_x^+$ to the finite length of grooves via $M = 5.9/k_x^+,$ we find that at 1195 K the groove length increases from 1.4 μm to 16.5 μm for times between 4 s and 1000 s, while $L$ increases from 450 Å to 960 Å. The anisotropy ratio $M/L$ of the shape of the coarsening grooves changes from 32 at 4 s to 172 at 1000 s.

We plot in Figure 3 the ratios $\epsilon_y/k_y = 2.0$ and $\epsilon_y^3/k_x^+$. For times between 4 s and 1000 s, we measure constant ratios $\epsilon_y/k_y = 2.0$ and $= 0.0069$ Å$^3$. Constant signifies dynamic scaling along the $Q_y$ direction. $\epsilon_y^3/k_x^+$ indicates that $M$ is proportional to $L^3$, in agreement with Milner's predictions. We may define the coarsening regime to correspond to times from ~4 s to 1000 s, in which the growth of grooves is governed by simple scaling relations. This is the principal result of this work, namely that the scaling relations, predicted by a model of thermally fluctuating step bunches, are observed to hold for steps on Si(111). To our knowledge, anisotropic dynamic scaling has not previously been observed. In view of the simplicity of the model, we anticipate that this behavior may be generic to different uniaxial systems that undergo phase separation and coarsening. Beyond 1000 s, an ordering regime follows in which the equilibrium grooved morphology is approached, as evidenced by the approach to a limiting groove size and the continually decreasing scattering peak widths.

![Figure 1](image.png) Diffraction patterns from a grooved sample after quench. 10-100-1000-10000 seconds. Intensity is displayed in logarithmic scale. Each frame displays a detector area of 8 x 8 mm$^2$.

Figure 2. Power law evolution of the position and width of the first-order groove peaks. Results for two temperatures are displayed. Resolution corresponds to 30 μm in the x-direction and 4500 Å in the y-direction.

The theoretically constant ratio $M/L^1$ may be estimated as $\Sigma_1 / k_B T_0$. It follows that the single-step stiffness is estimated to be of the order of $\Sigma_1 \approx 0.72 \text{ meV} \text{ Å}^{-1}$ for $k_B T = 100 \text{ meV}$. For comparison, we note that at the same temperature the stiffness of S8 steps on Si(001) is approximately 1 meV Å$^{-1}$, and that of S9 steps is 7 meV Å$^{-1}$. For steps on Si(111), several values of the stiffness have been reported within the range 30-68 meV Å$^{-1}$.\footnote{S. Song \textit{et al.}, \textit{Surf. Sci.} 372: 37 (1997). A. Pimpinnelli \textit{et al.}, \textit{Surf. Sci.} 295: 143 (1993). N.C. Bartelt, R.M. Tromp, and E.D. Williams, \textit{Phys. Rev. Lett.} 73: 1656 (1994).}

Previous to this work, the dynamics of step fluctuations on Si(001) and Si(111) have been studied by electron microscopy,\footnote{N.C. Bartelt \textit{et al.}, \textit{Phys. Rev. B} 48: 15453 (1993); C. Alfonso \textit{et al.}, \textit{Surf. Sci.} 262: 371 (1992); N.C. Bartelt, R.M. Tromp, and E.D. Williams, \textit{Phys. Rev. Lett.} 73: 1656 (1994).} yielding vivid real space images and such quantities as $\Sigma_1$ and $\Gamma_1$. Here, we have used an area detector in glancing-angle reflection geometry to demonstrate that x-ray scattering may provide complementary information concerning the collective behavior of steps undergoing a phase transformation. Specifically, after quenching a miscut Si(113) surface from above to below its faceting point, we characterized \textit{in situ} and in real time the two-dimensional scattering function of the evolving morphology. The width of the scattering along the groove direction evolves as a power law versus time with an exponent close to $1/2$, and the width of the scattering across the grooves and the peak wavevector of the scattering both evolve with an exponent of $1/6$. Such is a clear realization of a simple model of groove coarsening, which focuses on the thermal fluctuations of step bunches and takes the collision between adjacent step bunches as the rate-limiting mechanism.

\begin{figure}[h]
\centering
\includegraphics[width=0.8\textwidth]{fig3}
\caption{Scaling relations are demonstrated in the coarsening regime. (a) Best fit to a constant yields (b) $A^2$. This quantity is proportional to $M/L^3$, from which the single-step stiffness can be deduced.}
\end{figure}
3.2 Publications


3.2.1 Thesis
