A ONE-DIMENSIONAL FOURIER **ANALOGUE** COMPUTER **AND** ITS APPLICATION TO THE REFINMiENT OF **THE STRUCTURE** OF CUBANITE

by

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A 3E-DIMENSIONAL FOURIER **ANALOGUE** COMPUTER

AND ITS APPLICATION TO THE REFINEMENT OF THE

STRUCTURE OF CUBANITE

LEONID V. AZAROFF

Submitted to the Department of Geology on October 1, *1953* in partial fulfillment of the requirements for the degree of Doctor of Philosophy.

A computer has been constructed to sum Fourier series having up to **30** terms. Although this is a onedimensional computer, it can **be** used for double and triple summations **by** using standard trigonometric expansions. Secondarily, it can be used for computing trial structure factors.

> $\texttt{0.D.F.A.C. sums} \sum_{i=1}^{n} \frac{1}{n} \text{ only all electrically.}$ The n --

trigonometric function is produced **by** a variable-angle transformer known as a resolver. Each amplitude is set **by** a variao which regulates the input to a particular resolver. The frequencies 2mnx for 31 values of n are solver. The irequencies z in the *ji* values of in the contract of the resolvers in ratios 0, 1, 2, . . . 30. The resultant individual currents are added in parallel and the value at point \underline{x} (in intervals of 1, 1, or 1 of a cell edge) is read on a voltmeter 6~ **1Zo** 220

and the phase is read on an oscilloscope.

The relative speed of a computation is five to
fester than the standard strip method. The ten times faster than the standard strip method. average error in a computation compares favorably with the rounding off error in conventional two-place strips.

This computer has been utilized in the refinement **of** the structure of cubanite. The original structure determined **by** M. **J.** Buerger has been confirmed. The space group is Pomn with $\underline{a} = 6.46A$, $\underline{b} = 11.117$ A, $\underline{0} = 6.233$ A.

The intensities used in the refinement procedure were corrected for Lorentz and polarization factors and for absorption **by** the crystal. The refinement was carried

out **by** successive Fourier approximations. First, the a and q axis projections were refined, followed **by** the refinement of two plane sections. The final stage of refinement was carried out **by** passing three, mutually-perpendioular line sections through each atom. Difference electron densities were also computed for these line sections yielding the final atomic coordinates. The final atomic coordinates are listed and the bond lengths and bond angles are discussed.

Thesis Supervisor: Martin J. Buerger

Title: **.** Professor of Minerology and Crystallography

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INTRODUCTION

The work described in this thesis comprises two distinct problems. The first part deals with the design and construction of a one-dimensional analogue computer for performing Fourier series summations. Fourier series are continuously utilized in such crystallographic problems as structure determination, structure refinement and others. Although this computer was designed for express use in the crystallography laboratory, its application is more universal, since Fourier series are used in solving different problems in all fields of science.

The second part of this thesis describes the procedure of refining the structure of oubanite. Since the computer described in Part I was used extensively in this refinement procedure, Part II forms a logical sequel. $\mathbf{1}$

Part I. **A** One-Dimensional Fourier Analogue Computer

Introduction

Earlier computers. Fourier series are important in many branches of science. One-dimensional, two-dimensional, and three-dimensional Fourier series are especially important in x-ray crystallography. Since the computation of these functions is tedious, a number of devices have been developed to perform the computation, only a few of which have come into common use. Among these are the electrical digital devices of Beevers^{1,2}, the electrical analogue machines of **H**Egg and Laurent³, Ramsay, et al⁴. the mechanical analogue devices of McLachlan and Champaygne⁵, Rose6, Vand7 and Beevers and Robertson8. **All** these devices sum one-dimensional series. Robertson⁹, Pepinsky^{10,11}. and McLachlan et al 12 have devised electrical analogue machines for two-dimensional Fourier summations.

Basis for desigg. When the number of Fourier syntheses to be computed in the Crystallographic Laboratory of M.I.T. became large enough to warrant using a special computing device, several of these machines were closely investigated. Some were found to have obvious defects, such as contact trouble when multiple telephone switches were used. Most of them were found to handle too limited a number of Fourier terms. Guided **by** this survey, a decision was reached to build a one-dimensional electrical analogue computer which would sum a comparatively

large number of Fourier terms. This was specifically set at **30,** since this is about as high as ever required in the analysis of ordinary non-protein crystal structures. McLachlan's computer 12 served as a guide and as a point of departure. His machine compounds phases electrically **by** using selsyns. It performs a two-dimensional synthesis but is limited to **8** x **8** terms.

The transformation of an angle, φ , into a trigonometric function can be performed electrically in many different ways. The two devices considered in the early stages of designing **ODFAC** were sine potentiometers and resolvers. **A** sine potentiometer consists of a continuous resistor winding tapped to produce a voltage which varies as the sine of the angle of rotation of the main shaft. **A** resolver is a transformer using as its primary the rotor member and as its secondary two separate stator windings placed **90** degrees apart. When a voltage is applied to the primary winding the voltages of the two secondary windings vary respectively as the sine and cosine of the angle of rotation, Resolvers were chosen for the construction **of ODFAO** because the isolating property of a transformer makes the adding circuit independent of variations in the input stages.

Principle of operation. **Fig. 1** shows, in outline form, the operation **of ODFAC.** The input to each resolver, R , is fed from a variac, Y . The variac thus controls the

amplitude or Fourier coefficient, \underline{A} , of the Fourier component of a particular resolver. The two outlet leads of the resolver then deliver voltages proportional to **A** cosY and **A** sinY. The shafts of **30** such resolvers are geared so that the shaft displacement of a particular resolver, n, is an integral multiple, n, of the shaft displacement of a fundamental shaft. The outputs of the individual resolvers are therefore a set of voltages proportional to

if the outputs are appropriately coupled, the machine produces voltages proportional to

$$
\sum_{n=0}^{30} A_n \cos \varphi_n \quad \text{and} \quad \sum_{n=0}^{30} A_n \sin \varphi_n
$$

which can be read on a voltmeter, **VM**. In crystallographic problems γ_n = $2\pi nx$ where <u>n</u> is the number of the harmonic and x is the sampling interval expressed as a fraction of one complete period.

Fig. **1 ODFAC** Outline

Design and construction

Mechanical features. The production of thirty harmonics in the Fourier series requires that one complete rotation of the rotor in the first resolver corresponds to two complete rotations of the rotor in the second resolver, etc., up to thirty complete rotations of the rotor in the thirtieth resolver. This is accomplished in **ODFAC by** means of a gear train represented **by** Fig. 2.

The main shaft leading from the motor is geared **by** worms and worm gears to three horizontal shafts whose relative angles of rotation are given **by** the ratios of the worms and gears. Each horizontal shaft carries a series of spur gears which engage with the spur gears mounted on the rotors of the individual resolvers. The angle of rotation of each rotor shaft is, therefore, a function of the ratio of its spur gears modified **by** the angular rotation of the horizontal shafts. The actual gear train is shown in Fig. **3.** This particular gear train has several advantages. One is that most gears used are commercially available, stock sizes. Only **10** of the **82** gears are nonstandard size and had to be obtained **by** special order. Another advantage is that high gear ratios are avoided, the largest step-down ratio being 5:4. This decreases inaccuracies in the positioning of the resolver shafts.

Because of low frictional losses, all **30** resolvers can be driven at the desired speed **by** a small, **1** h.p. motor. *1500*

Fig. **3** Gear Train

Electrical circuit. The wiring diagram **of** the electrical components of ODFAC is shown in Fig. 4. The input voltage is fed from the variacs, V, through a *DPiT* phase-selector switch, S , to the rotors of the resolvers, R, whioh are linked **by** the gear train illustrated in Fig. **3.** Each output line of the stator winding of the resolvers contains a series resistance, r_, of 500,000 ohms whose purpose is to make any stray **losses** in the secondary circuit negligible. **All** the cosine (and sine) windings of even harmonics are connected in parallel. Similarly the cosine (and sine) windings of the odd harmonics are connected in parallel. These lines lead to a gang switch which permits various combinations of the lines to **be** made.

The use of a parallel, rather than a series adding circuit is preferred for the following reasons: In a series circuit all the stator windings would be directly connected causing a cumulation of errors due to mutual inductance between the individual primary and secondary windings and the secondary windings of adjacent resolvers. The addition of voltages in a series circuit would also have the disadvantage of building up very high voltages and currents with possible damage to the windings. **All** of these disadvantages are overcome **by** a parallel circuit in which the isolation of the individual resolvers limits the current in each secondary winding to that induced **by** the primary.

This isolation permits the insertion of a large resistance to minimize losses in the line. The maximum output voltage obtained from such a circuit can never exceed the input line voltage because the individual voltages are averaged over all **31** branohes.

The output voltage lines pass through a 4-gang switch which selects combinations of odd and even harmonics of the sine or cosine lines for transmission to the voltmeter or oscilloscope. In practice, there are four such lines, viz. the lines for cosine even, cosine odd, sine even and sine odd. Each line is connected to appropriate terminals of the gangs. Thus, when the switch is rotated to a given position, it makes contact with only those leads that are connected at that position. The combinations presently available are: \cos_{θ} , \cos_{θ} , \sin_{θ} , \sin_{θ} , \cos_{θ} + cos_o, $\sin_{0}t$ $\sin_{0}t$, $\cos_{0}t$ $\sin_{0}t$, and $\cos_{0}t$ $\sin_{0}t$. The separation of the odd and even harmonics into separate lines permits the utilization of the symmetry inherent in sine and cosine functions. This is standard practice in crystallographic applications¹³.

The meter has a 50μ a movement and three ranges which correspond to the voltages expected from the number of variacs supplying input voltages. **A** fourth range, actuated **by** a push-button, permits a clear reading of low voltages. The actual meter circuit is illustrated in Fig. 5.

The cathode-ray tube is part of a standard oscilloscope circuit not having a sweep generator. The circuit is illustrated in Fig. **6.** The output voltage is placed across two deflecting plates and a reference voltage across the other two. Since both sets of plates have the samae frequency applied to them, the resulting Lissajous figure on the tube face is a straight line whose angular inclination depends on the magnitude and relative phase of the output voltage.

Construction details. The rear view of the assembled machine is shown in Fig. **7.** The resolvers are held in place **by** aluminum clamps (see also Fig. **3)** which, in turn, are fastened to two aluminum angles forming a tack-like shelf. Each shelf holds five resolvers and supports its horizontal drive shaft mounted in ballbearing supports. The clamps have built-in means for adjusting the angular position of the resolvers, consisting of a slot to accept a gear-bearing key which engages with the gear on the resolver body, permitting minor variations in angular alignement after the resolver has been positioned. The horizontal drive shafts are driven by the main drive shaft which runs vertically, in the center of Fig. **7,** from the motor (hidden **by** the bottom shelf) to the control mechanism.

The control mechanism is connected directly to the drive shaft **by** a set of change gears. It consists of

Fig. 5 Meter Circuit

Fig. 6 Oscilloscope Circuit

Fig. 6 Oscilloscope Circuit

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a shaft which carries a change gear and a cam. The cam actuates a micro-switch which controls the number of revolutions made **by** the motor for the sampling interval. The motor is a dynamic-braking motor which can be stopped instantaneously **by** applying a reverse field to its rotor. The actual control circuit is illustrated in Fig. **8. A** horizontal shaft, permanently geared to the main drive shaft, rotates a dial on the front panel indicating the angular position of the main shaft. The change gears in the control mechanism permit the selection of intervals of $\frac{1}{60}$, $\frac{1}{120}$, or $\frac{1}{240}$ of one complete period.

The motor-generator set at the bottom of Fig. **7** provides the input voltage (120 v., 400; **)** to the variaos only. The driving mechanism and service components operate on the regular line voltage. The purpose of the motorgenerator set is to provide a constant current source with an undistorted wave shape, making the adding circuits in **ODFAC** independent of line fluctuations.

Fig. **8** Control Cirouit

Performance

Operation. **A** front view of **ODFAC** in operation is shown in Fig. 9. The variacs are numbered and arranged on removable panels in sets of five. The amplitudes are set by means of a friction-drive dial permitting an acourate, rapid setting from 0-100. The phases are set **by** means of a plus-minus toggle switch placed directly above each dial. Each bank of five resolvers can be switched in or out of the input curouit **by** means of an additional toggle switch on the left of each panel. The numbering and location of the dials is such that the operator sits facing the first sixteen dials, the next fifteen dials being within arm's-length to his right. The dials need be set only once **for** each one-dimensional summation.

A panel to the left of the operator contains the meter on which the value of the Fourier series is read, the CRO tube on which the positive and negative quadrants are marked, the sine-cosine selector switch, and a pushbutton that advances the computer mechanism to the next sample setting. **A** pilot light indicates that the motor has advanced the computer to the next sample setting and the meter may be read. Once the amplitudes (variao dials) and the phases (plus-minus switches) have been set, and

Fig. **9 ODFAC,** Front View

the type series *(sine* or cosine) has been selected, all the operator does is read the meter, note the phase, and push the button to advance to the next reading. **A** dropleaf table is attached to the front of **ODFAC** to provide a convenient surface for recording readings.

The speed of a computation on **ODFAC** depends primarily on the type of series desired, i.e., onedimensional, two-dimensional, etc., and only secondarily on the number of terms or frequency of interval of sampling desired. The machine time for a complete cycle from **0** to 217 has been selected to take **7** minutes. If the time for setting the variaos and recording the values obtained is added to this, the total computing time amounts to approximately **10-15** minutes, depending on the sampling interval selected. The relative time of computing an average twodimensional series on **ODFAC** ranges from **1** to **1** of the time consumed using standard strip methods¹⁴, ⁵₄ ¹⁵ The more terms there are in the series and the finer the interval of sampling desired, the more efficient **ODFAC** becomes when compared with strip methods.

Accuracy. The accuracy of the components used in OIfAC is the highest attainable at a reasonable cost. The resolvers are accurate to within $1\frac{1}{2}$ mechanical degrees. The variacs and resistors are accurate to within **1% of** maximum ratings. The meter is accurate to better than

1% of full seale deflection. The backlash in the gears is almost non-existent and the angular accuracy of the gear settings is within a fraction of one degree.

Table **1** shows an actual comparison of the meter readings of **ODFAC** with those computed using three-place trigonometric tables. The amplitudes for this onedimensional series were supplied **by** Professor M. **I.** Buerger from a Harker line synthesis for realgar. In the same table are listed corresponding values as computed with the aid of Patterson-Tunell strips¹⁴ and Beevers-Lipson strips¹⁵. If the deviations from the true values are examined it is evident that the errors in the values given **by ODFAC** are, on the average, as small as, if not smaller than, those due to the rounding-off errors inherent in the strip methods.

Finally, **by** making all operations other than recording the numerical values automatic, **ODFAC** eliminates the ever-present source of error **-** the human error.

Conclusions. The chief advantages of **ODFAC** are its ability to handle up to 31 coefficients in a Fourier series and to perform the summation rapidly with an accuracy adequate for most purposes. The simplicity of the design of the machine virtually eliminates almost all possibilities of electrical or mechanical failure. **In** the event such failure should occur, all components are

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readily accessible and removable for repair.

The slowest part of the operation of **ODFAC** lies in reoording the results. It is proposed to add an automatic recorder for this purpose.

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Extension to two dimensions

Use of two units. One **ODFAC** unit performs a two-dimensional synthesis **by** means of a Beevers-Lipson expansion. **A** combination of two such units performs this summation more rapidly. If these units are equipped with chart-type recorders, the most complicated twodimensional series having **30 x 30** Fourier coefficients can be performed in an afternoon. The utilization of two such units follows the plan indicated below.

The series to **be** summed has a general expression: φ (xy) = \sum_{h} $\left[\sum_{k} \mathbf{F}_{h,k}$ cos 2 $\hat{\pi}$ (hx+ky)

$$
=\sum_{h}\left[\sum_{k} F_{hk} \cos 2\pi k y\right] \cos 2\pi h x - \sum_{h}\left[\sum_{k} F_{hk} \sin 2\pi k y\right] \sin 2\pi h x
$$

The summations over **k** are first performed at the same time by the two units working independently. The results of these first summations are then used as ooefficients for the second summation over **h.** In the second operation the two units are connected in parallel to only one recorder, which thus records the results **of** the final two-dimensional summation.

Two-dimensional computer. **A** more elaborate extension to two-dimensions can **be** built **by** combining two

Fig. **10** Two-Dimensional Computer

resolvers for every term in the series. The construction entails building **k ODFAC** units and one "master" **ODFAC** having h resolvers (without variacs) for each harmonic **k.** Such a plan is illustrated in Fig. 10 for $k = h = 4$.

The **ODFAC** units, whose resolvers have angular speed ratios proportional to their particular h , are shown in solid lines. The rectangles in the dotted lines represent the resolvers of the "master" unit. **All** of these latter resolvers have the same angular settings proportional to **k.**

The coupling between all units is electrical only. The resolvers of one **ODFAC** unit produce voltages proportional to:

$$
F_{h_1k_1} \cos 2\pi h_1 x
$$

$$
F_{h_2k_1} \cos 2\pi h_2 x
$$

$$
F_{h_3k_1} \cos 2\pi h_3 x
$$

which are fed to the resolvers, R_k , of the master unit. The outputs of these resolvers are thus proportional to:

$$
(\mathbf{F}_{h_1k_1} \cos 2\pi h_1 x) \cos 2\pi k_1 y
$$
$$
(\mathbf{F}_{h_{2}k_{1}} \text{ cos } 2\pi h_{2} \mathbf{x}) \text{ cos } 2\pi k_{1} \mathbf{y} \text{ etc.}
$$

 $\sim 10^{-1}$

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Appropriate electrical coupling of the resolvers in the master unit then produces a voltage proportional to

$$
\sum_{h} \sum_{k} F_{hk} \cos 2\pi (hx + ky)
$$

 \bullet

Part II. The Refinement of the Structure of Cubanite

Introduction

The crystal structure of cubanite was determined by M.J. Buerger¹⁶ in 1947. The ferromagnetic property of cubanite and the possibility of its relationship to the structure of another ferromagnetic mineral, pyrrhotite, made the knowledge of the exact structure of cubanite of more than routine interest. For these reasons this writer decided to use the refinement of the structure of cubanite as a vehicle **for** demonstrating the application of the computer described in the first half of this thesis.

The space group of cubanite is **Pomn.** The cell constants reported by Buerger¹⁶ are $\underline{a} = 6.46$, $\underline{0}$, \underline{b} = **11.117 A,** $\underset{\sim}{\circ}$ **= 6.233 Å and were found to be in agreement** with those determined from the measurements of the precession photographs taken in the course of this investigation. This cell contains $4 \text{ CuFe}_2\text{S}_3$.

In this part of the thesis the refinement procedure will be described, with special emphasis on the role **ODFAC** played in simplifying the attendant computations. The refinement was carried out in three stages. During the first stage successive Fourier refinements were carried out in the **(Okl)** and **(hkO)** zones. In the second stage

the complete three-dimensional intensities were used to refine the \underline{x} and \underline{z} atomic coordinates with the aid of plane sections. In the third stage, three line sections were passed through each atom, and difference maps were used to determine the final atomic positions.

 $\bar{\mathbf{v}}$

 $\sim 10^7$

Experimental Procedure

Photographio Technique. The crystal selected for intensity measurements came from the Frood Mine, Sudbury, Ontario and is the same crystal that was used by M.J. Buerger in the original structure determination¹⁶. The crystal was mounted on its **b** crystallographic axis and placed on the Buerger precession **17** camera. The **(hko)** and **(Okl)** zones were then photographed with molybdenum $K\propto$ radiation and the intensities were measured by the M.I.T. modification of the Dawton¹⁸ method.

The complete set of three-dimensional intensities was determined similarly. The individual levels of the * * reciprocal lattice perpendicular to the a and c axes were first photographed. Due to the blind area arising in the center of upper-level precession photographs, it was necessary to take additional photographs at another setting. It turns out that for crystals having orthorhombic symmetry all the missing intensities can be found on reciprocal lattice planes perpendicular to \underline{d}_{101}^* (see Fig. **11)*** It was therefore necessary to make only one additional setting of the crystal to obtain all the intensities contained in the limiting sphere of reflection.

The symmetry of the reciprocal lattice combined with the repeated appearance of the same spots on more than

33.

 $\sim 10^{11}$ km $^{-1}$

Fig. **11** Reciprocal Lattice

one photograph provided the possibility to make four independent measurements of approximately **70%** of the observable intensities. The remaining intensities were measured independently at least twice. The averages of these measurements were then used as the final observed intensities.

Conversion of Intensities to Structure Factors. The intensities were corrected **by** applying the Lorentz and polarization factors as determined for the zero-level precession photographs **by** iaser **9** and upper levels **by** Burbank²⁰and Wells and Abrahams²¹. The charts prepared **by** the above-mentioned authors were enlarged to a radius of **10** cm. and printed on transparent film. These films were then placed over a reciprocal-lattice net drawn to the same scale. The use of these enlarged charts and reciprocal-lattice nets rather than the original films allows a more accurate and rapid determination of the value of the correction.

The intensities were next corrected for absorption **by** the crystal. Since the crystal was an irregular, jagged fragment, an exact determination of the absorption correction was nearly impossible. It was therefore decided to apply an absorption correction based on the assumption that the crystal was spherical. Although this assumption is not strictly valid, it was felt that such an absorption correction would, nevertheless, appreciably decrease the errors in the observed intensities due to absorption effects.

The absorption corrections made were based on a method suggested **by** Ekstein and Evans 22 . The radius **r'** of the sphere is first determined. In this case the average radius of the crystal $(r = 0.0175 cm.)$ was used.

36.

The value of the radius is then multiplied **by** the linear absorption coefficient of the crystal, in this case μ_1 = 127.5 cm $^{-1}$. The per cent transmission as a function of 8 can then be determined **by** forming the product $\mu_1 \cdot r = 2.23$) and referring to the above-cited tables. The tables list the per cent of x-radiation transmitted at the angles $\theta = 0^{\circ}$, $22\frac{1}{2}^{\circ}$, 45° , $67\frac{1}{2}$ and 90° . These values are then plotted and the best curve drawn through them. In this case the plot of transmissivity was made against sin θ rather than θ as suggested by Evans²², and is shown in Fig. 12. The absorption correction was then applied to the observed intensities **by** reading the per cent transmitted corresponding to the value of sin **0** for the reflection concerned and dividing the observed intensity **by** this value.

Finally, the square root of the corrected intensities was taken and these values were used was the observed structure factors. The observed structure factors were placed on an absolute basis, **by** determining a temperature factor and applying it to the computed structure factors. The procedure used is described in Appendix I and the observed structure factors are listed in Appendix II.

Fig. 12 Absorbtion Correction Curve

Refinament of Coordinates

Refinement in Projection. The atomic coordinates determined by Buerger¹⁶ (see Table II, pp. 56) were used as the starting point. The phases determined by these coordinates were used to compute the electron density $\varphi(\underline{yz})$, $P1g. 13.$

The equation for this projection is:

for
$$
\underline{1} = 2\underline{n}
$$

$$
\rho(yz) = \frac{8}{A} \left\{ \sum_{k=1}^{N} \mathbf{F}_{0k1} \cos 2 \pi k \mathbf{y} \cos 2 \pi i z - \sum_{k=2n}^{N} \mathbf{F}_{0k1} \sin 2 \pi k \mathbf{y} \sin 2 \pi i z \right\}
$$

for $\underline{1} = 2\underline{n+1}$

 $\sqrt{(yz)} = 0$

*In this equation and all those that follow \underline{x} , \underline{y} , \underline{z} are fractional coordinates representing dimensioneless ratios between the true distance from the origin in Angstroms and the parallel translation distances also expressed in Angstroms.

 $b/2$

Fig.13 $\rho(yz)$

The appearance of only even values of **1** permits the actual synthesis to **be** carried out with a dummy index $\underline{1}^{\dagger} = \frac{1}{2}$. This has the effect of halving the true <u>c</u> axis 2^{1} and makes the sampling interval $(\underline{z} = \frac{1}{60})$ equal to $\frac{1}{120}$ of the true o axis.

As expected, the atoms are well resolved in this projection. The new atomic parameters (see Table II) were then used to compute a new set of structure factors. This new set gave a residual coefficient of **19%** and showed no changes in phase for any reflection. It was therefore concluded that no further refinement was possible with this projection.

The new coordinates were then used to compute the structure factors for the **(hkO)** reflections. These computations yielded a residual ooefficient of **9.7%** which vas so low that it was felt that nothing could be gained **by** attempting a refinement in this projection, particularly since the c axis projection contains an overlap of each atom **by** another.

Refinement by Plane Sections. The possibilities for refinement in projection thus being limited, it was decided to continue the refinement in sections taken through the crystal. Since the symmetry requires that two of the four atoms in the asymmetric unit, *viz.,* copper and sulphur,, **lie** on the mirror plane perpendicular to the **b** axis at $\underline{y} = \frac{1}{4}$, this plane was chosen as one of the sections to be computed. The other two atoms, in the asymmetrio unit, iron and sulphur₂, lie on a plane perpendicular to **b** at $y =$ $\frac{1}{12}$ (see Fig. 13) and so this plane was chosen for the other section.

The equation for ρ (xyz) for the space group Pnma **is:23**

h+1=2n, k=2n **(xyz) = 0** F **cos** 2Rhx cos **21Tky** cos 2lz $h+1=2n^{8}$ k=2n+1 F_{hkl} cos 2Thx sin 2Tky sin 2Tlz h **k 1 0** h+1=2n+1 k=2n
 \sum_{h} \sum_{k} \sum_{i} F_{hkl} sin 2Thx cos 2Tky sin 2T1z -**0** $h+1=2n+1$ $k=2n+1$ **F_{hkl}** sin 2Thx sin 2Tky cos 2Tlz **k 1 0**

When $\underline{y} = \frac{1}{4}$ this can be written in the form: φ ($\frac{1}{4}z$) = $\frac{8}{7} \left(\sum_{1}^{h+1=2n} \left(\sum_{k}^{k=2n} \frac{F_{hkl}(-1)^{\frac{k}{2}}}{k} \right) \cos 2\pi nx - \frac{F_{hkl}(1-z)^{\frac{k}{2}}}{2} \right)$

$$
\sum_{h}^{h+1=2n-1}, \quad k=2n+1
$$
\n
$$
\sum_{h} \left(\sum_{k}^{F} F_{hkl} (-1) \frac{k-1}{2} \right) \sin 2 \pi h x \cos 2 \pi 1 z -
$$

$$
h+1=2n, k=2n+1
$$

\n
$$
\sum_{1}^{\infty} \left(\sum_{k}^{n} F_{nki} (-1) \right)^{k-1} \cos 2 \pi h x +
$$

$$
\sum_{h=0}^{\infty} \left(\sum_{k=0}^{\infty} F_{hkl} (-1)^{\frac{k}{2}} \right) \sin 2\pi h x \quad \text{sin } 2\pi 12
$$

When $\underline{y} = \frac{1}{12}$ the form of the equation is the same except that the summation over k takes the form:

when $k = 2n$,

$$
\sum_{k} \quad F_{hkl} \quad \cos \quad \frac{2 \pi k}{12}
$$

when $\underline{k} = 2n+1$,

 \mathcal{A}^{\pm}

$$
\sum_{k} F_{hkl} \sin \frac{2\pi k}{12}
$$

As can be seen from the above equations the first summation to be performed is that over **k.** This can be done most expediently with the help of some computational aid like the Patterson-Tunell strips¹⁴. Since only one value of **y** is required the time consumed setting the dials of **8DFAO** is greater in comparison.

The next step of the summation, on the other hand, can be carried out most efficiently **by ODFAC.** Since for any given value of **1,** the terms having h **odd** or even are combined separately and then added, it is possible to put all the terms *into* the machine and **by** putting the selector switch on \cos_{θ} + \sin_{θ} or \cos_{θ} + \sin_{θ} , as the case may be, perform both summations simultaneously and obtain their combined sum directly. The final summations over **1** are then performed in the usual manner.

The two sections $(\underline{x}_L^1 \underline{z})$ and $(\underline{x}_L^1 \underline{z}^2)$ are shown in Figs. 14 and *15.* As can be seen, the atoms are clearly resolved and are spherical in shape, indicating that the phases are correct. The weak background may be due to stray electrons or just to a cumulation of series termination errors.

The changes in the atomic parameters obtained from these two sections were very slight (see Table II). Only the weak intensities were recomputed at this stage,

45

and, since no phases changed, it was decided to proceed with the refinement by means of line sections through the individual atoms. The residual coefficient for the entire three-dimensional set **cf** intensities at this state *was* 14%.

 $\bar{\mathbf{v}}$

Refinement **by** Line-Sections. The availability of a one-dimensional computer strongly suggested that much of the time spent in computing plane sections could be saved **by** making instead line sections through the atoms. At first thought it may appear that the location of the centers of the atoms must be known rather well before a line section can be expected to accurately locate the center of the atom along that line. It turns out, **how**ever, that, due to the spherical symmetry of the electron distribution about the atom, a line passed through a point several tenths of an Angstrom away from the true center still locates the deasest accumulation of electrons (the center) quite accurately.

A.line section parallel to the a, **b,** o axes was passed through each atom. Since two of the parameters are held fixed, the values of x and z as determined **by** the plane sections were used to first determine the values of \underline{y} for the two atoms (iron and sulphur₂) located in general positions. The newly determined values of y were then used in the computations of the other two line sections. The complete set of twelve line sections is shown in Figs. **16, 17, 18,** and **19.**

The forms of the equations used in computing the line sections can be illustrated **by** the ones used for iron, the others being identical except for the values of the

47.

Fig.18 Line Sections for Cu

Fig. **19** Line Sections for S_I

respective \underline{x} , \underline{y} and \underline{z} coordinates.

For refining y

$$
\rho(\frac{5}{60}, y, \frac{8}{60}) = \frac{8}{90} \left\{ \sum_{k=2n}^{k=2n} \left[\sum_{h=1}^{h+1=2n} \sum_{k=1}^{n} F_{hkl} \cos 2\pi h \frac{5}{60} \cos 2\pi L \frac{8}{60} - \right] \right\}
$$

$$
\sum_{h} \sum_{1}^{F} f_{hkl} \sin 2\pi h \frac{5}{60} \sin 2\pi l \frac{8}{60} \cos 2\pi k y -
$$

k=2n+1 h+1=2n
\n
$$
\sum_{k} \left[\sum_{h} \sum_{i=1}^{F} f_{hk1} \cos 2\pi h \frac{5}{60} \sin 2\pi h \frac{8}{60} + \cdots \right]
$$
\nh+1=2n+1
\n
$$
\sum_{h} \sum_{i=1}^{F} f_{hk1} \sin 2\pi h \frac{5}{60} \cos 2\pi h \frac{8}{60} \sin 2\pi k y
$$

The summations over h and 1 were carried out first with the aid of Patterson-Tunell strips. The summations over k were performed on ODFAC with the selector switch set on \cos_{θ} + \sin_{θ} . The sampling interval used was $\frac{1}{120}$ of a cell edge in all of these computations.

$$
\rho(x, \frac{5}{60}, \frac{8}{60}) = \frac{8}{v_0} \Bigg[\sum_{h} \Bigg(\sum_{k=1}^{F} \sum_{k=1}^{F} \bigg(\sum_{k=1}^{F} \sum_{k=1}^{F} \bigg(\sum_{k=1}^{F} \sum_{k=1}^{F} \bigg) \bigg) \Bigg) \Bigg(\sum_{k=1}^{F} \sum_{k=1}^{F} \bigg(\sum_{k=1}^{F} \sum_{k=1}^{F} \bigg(\sum_{k=1}^{F} \sum_{k=1}^{F} \bigg) \bigg) \Bigg(\sum_{k=1}^{F} \sum_{k=1}^{F} \bigg(\sum_{k=1}^{F} \sum_{k=1}^{F} \bigg) \bigg(\sum_{k=1}^{F} \sum_{k=1}^{F} \sum_{k=1}^{F} \sum_{k=1}^{F} \bigg) \bigg(\sum_{k=1}^{F} \sum_{k=1}^{F} \sum_{k=1}^{F} \
$$

The summations over **k** and **1** were carried out first with the aid of Patterson-Tunell strips. Since the summation over <u>k</u> will be required again for the section along z, it was carried out first. The final summations over \underline{h} were then performed on ODFAC.

$$
\rho(\frac{5}{60},\frac{5}{60},z) = \frac{8}{\pi} \left\{ \sum_{i=1}^{n} \left[\sum_{i=1}^{n} \left(\sum_{k=1}^{n} F_{ik1} \cos 2\pi k \frac{5}{60} \right) \cos 2\pi h \frac{5}{60} - h + 1 = 2n + 1 \right] \right\}
$$

\n
$$
\sum_{i=1}^{n} \left(\sum_{k=1}^{n} F_{ik1} \sin 2\pi k \frac{5}{60} \right) \sin 2\pi h \frac{5}{60} \cos 2\pi 12 - h + 1 = 2n \quad k = 2n + 1
$$

\n
$$
\sum_{i=1}^{n} \left[\sum_{i=1}^{n} \left(\sum_{k=1}^{n} F_{ik1} \sin 2\pi k \frac{5}{60} \right) \cos 2\pi h \frac{5}{60} + h + 1 = 2n \right] \sum_{i=1}^{n} \left(\sum_{k=1}^{n} F_{ik1} \cos 2\pi k \frac{5}{60} \right) \sin 2\pi h \frac{5}{60} \right] \sin 2\pi 12
$$

The summation over **k** having been done before, only the summations over h and 1 remained. The summation over h can be done efficiently on **ODFAC** since for a given **1** the terms for h odd and even are combined separately and then added. The final summations over **1** were then also performed on **ODFAC.**

As can be seen from Figs. 16, **17, 18** and **19,** the shifts in the atomic positions indicated **by** the line sections are very slight. It was therefore decided to use a more precise indication of the amount of movement required. To this end the "difference" electron densities suggested by Cochran²⁴ were used.

Since the line sections based on the observed structure factors had already been computed, it was decided to compute line sections using computed structure factors separately and then to subtract the two resulting electron densities to form the difference synthesis:

$$
D = \bigwedge_{\text{obs}} - \bigwedge_{\text{calc.}}
$$

To this end scattering factors given **by** the Internationale Tabellen²⁵ were used. A temperature factor obtained by $\text{plotting}|\mathbf{F}_{0}| / |\mathbf{F}_{0}|$ as a function of $\sin^{2}\theta$ was applied to the computed structure factors **so** that they would be on the same scale as the observed structure factors. The procedure used is outlined in Appendix **I.**

The line sections based on the computed F's are also shown in Figs. **16, 17, 18,** 19. As can be seen **by** comparing the electron densities based on observed and computed structure factors, the indicated shifts of the atomic centers were not always the same. The difference of the two electron densities was, therefore, plotted, and is shown in Fig. 20.

Cochran²⁴ gives criteria for computing the amount of shift required. Essentially, his equations state that the atom must be moved in the direction of the steepest gradient **by** an amount proportional to the steepness of the

Fig. 10 Differential Syntheses **Fig. 11** Final Differential Syntheses

 $\mathcal{A}(\mathcal{A})$ and $\mathcal{A}(\mathcal{A})$

 \tilde{z}

gradient. In this case an empirical relationship based on the measured gradient and the amount **of** shift indicated **by** the corresponding electron density plot was worked out and applied to all the atoms requiring a shift. Since the criterion for correctness of an atomic position is that the atom must be located on a slope of zero gradient, only those atoms situated on definitely sloping gradients were moved.

The resulting atomic coordinates (see Table II) were then used to compute structure factors for the two zones **(hio)** and **(Ok1).** The residual coefficients of **7.7%** and **11.0%** respectively, indicated that the refinement had been carried as far as experimentally available data would permit.

As a final check, the entire three-dimensional set of structure factors was computed and is tabulated along with the observed structure factors in Appendix II. The residual coefficient index for the entire set of structure factors is **11.9%.** Difference syntheses based on these intensities were also computed, similarly to the ones above, and they are shown in **Fig.** 21. As can be seen from these plots these atomic positions can **be** assumed to be final.

54.

Final Atomic Parameters. The final atomic parameters and the corresponding residual eoefficients for the structure factors computed using these parameters are tabulated in Table II. The table is constructed to indicate the basis of selecting the parameters, the parameters thus determined, and the resulting residual coefficient for each set of parameters.

 $\overline{4}$

56. Table II

Parameter Changes

Discussion of Structure

The structure of cubanite is discussed in detail by Buerger¹⁶. "The structure [is] composed of slabs of the wurtzite arrangement parallel to **(010)** and averaging **b/2** wide. Since the wurtzite arrangement is polar, the metal coordination tetrahedra all point up, or else they all point down. In the cubanite structure, the wurtzitelike slabs are joined to one another **by** inversion centers so that neighboring slabs have their tetrahedral apex directions reversed...Since the inversion center occurs at the midpoint of an edge of the iron tetrahedron, it has the effect of joining all iron tetrahedra in pairs which share this edge."

Since the sharing of a tetrahedral edge is most unorthodox and since this sharing occurs for iron tetrahedra only, it is difficult to avoid the correlation of this curious feature with the unusual characteristic of ferromagnetism. The reason for undertaking this refinement procedure was, primarily, to learn more about the bond distributions in these iron tetrahedra.

The bond lengths and bond angles of the more important atoms are tabulated in Tables III and IV. The designations **of** the atoms refer to the illustration of the structure shown in Fig. 22. The shared edge of the iron tetrahedra is comprised of atoms labeled $S_2(D)$ and $S_2(E)$.

57.

Fig. 22 Structural Unit

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* The subscripts $1,2,3,4$ refer to the coordinates of Fe,Cu,S₁₁,S₁ respectively, as tabulated in Table II.

Table IV

 ~ 10

$$
S_{2}(C)-Cu-S_{2}(C)
$$
\n
$$
S_{2}(C)-Cu-S_{1}(F)
$$
\n
$$
S_{1}(F)-Cu-S_{1}(G)
$$
\n
$$
S_{2}(C)-Cu-S_{1}(G)
$$
\n
$$
S_{2}(C)-Cu-S_{1}(G)
$$
\n
$$
113^{0} 12^{1}
$$

 $\mathcal{A}^{\mathcal{A}}$

The length of this edge, $S_2(D)-S_2(E)=3.61$ Å is shorter than the other $s = S$ distances $(3.84_h, 3.73_h, 3.85₇)$ **3.76₂ A).** The angle $S_2(D)$ -Fe $-S_2(E)$ = 105⁰ 36' is less than the tetrahedral angle of **1090 28'** and is also less than the other S-Fe-S angles **(1090 57',** 1140 44', **111035'). A** somewhat similar distortion occurs for the copper tetrahedron. The copper atoms are located in symmetry planes and are coordinated **by** two sulfur atoms in the same symmetry plane and t wo other sulfurs symetrically displaced from the plane. The two sulfurs in the plane are each shared **by** two copper and two iron atoms, whereas the sulfur atoms not lying in the symmetry plane are shared **by** one copper and three iron atoms. The bond length from the copper to these latter sulfurs is longer $(Cu-S_2(C)=2.33_o\lambda)$ than the Cu-S distances $(2.29₂, 2.27₅²)$ in the symmetry plane and the angle between the bonds to two such sulfurs $(104^{\circ}38!)$ is considerably smaller than the other S-Cu-S angles **(107003', 110054', 1130** 12').

The above, geometrical description of the structure can be used to draw some inferences regarding the relationship between the bond distribution in cubamite and the property of ferromagnetism. The ground state of iron requires that **18** of its **26** electrons comprise the argon core, six be in the **3d** state and two in the 4s state. The magnetic properties of the transition group elements are

attributed to the magnetic properties of the **3d** electrons. According to Slater 26 "... the required condition for ferromagnetism is that there be an inner partly-filled shell of electrons, whose radius is as **small** as possible compared with the interatomic distance." If we consider the **Fe-S** distances, we find that one distance $(Fe-S_2(E)=2.24$ ₅) is shorter than the other three. Since the ionization potentials **of** the **3d** and the 4s electrons **(.60** and **.58)** are nearly the same, it is reasonable to assume that the four tetrahedral bonds of iron are formed **by** three unpaired **3d** eleotrons and one unpaired 4s electron, one of the 4s electrons of the ground state having been raised to the **34** state. This bond distribution is opposed to an earlier explanation of the ferromagnetism of oubanite²⁷ which postulated an Fe-Fe bond leaving one unpaired electron. In addition to the lower potential energy attained **by** pairing all the electrons, the shortening of the shared edge of the iron tetrahedra casts serious doubt on the validity of th1is earlier view.

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APPENDIX I

Temperature Factor Determination

To facilitate the comparison of observed and computed structure factors they must **be** placed on the same scale. If the structure is correct, the ratio between the observed structure factors, measured on an arbitrary scale, and the computed structure factors should be constant, i.e.

$$
\frac{|\mathbf{F}_o|}{|\mathbf{F}_o|} = k
$$

The assumption that the atoms are at rest, made in computing the structure factors, is not valid at room temperatures. **If** a temperature factor is introduced to correct for the thermal motion of the atoms, the ratio between observed and computed structure factors becomest

$$
\frac{|\mathbf{F}_o|}{|\mathbf{F}_o|} = k \quad \text{exp} \quad (-2B \quad \frac{\sin^2 \theta}{\lambda^2})
$$

or
$$
\frac{|\mathbf{F}_o|}{|\mathbf{F}_o|} = \log_k k - 2B \quad \frac{\sin^2 \theta}{\lambda^2}
$$

 $log_a |F_a|$ e

The later equation is an equation of a straight line and is useful in determining the value of B, the

unknown quantity in the above equation.

Since the temperature factor is a function of sin² θ it is convenient, in practice, to first compute the ratio between $\begin{vmatrix} F \\ 0 \end{vmatrix}$ and $\begin{vmatrix} F \\ 0 \end{vmatrix}$ and then to determine the average value of this ratio for limited ranges of $\sin^2\theta$. A plot of these values vs $\sin^2\theta$ is then used to determine the temperature correction.

The actual plot used in this case is represented **by** Fig. **23** which shows the plot of $\log_{e} \frac{|\mathbf{F_0}|}{|\mathbf{F_0}|}$ vs $\sin^2\theta$ for the final set of computed structure factors. The accompanying tables show the details of the computations.

The final computed structure factors, on an absolute basis, are listed along with the observed struoture factors in Appendix II.

Fig. 23 Temperature Correction

Average value of $k \rightarrow 53$

 $.050 \langle \sin^2\!\theta \rangle$.100

 $\label{eq:2} \frac{1}{\sqrt{2}}\int_{0}^{\infty}\frac{1}{\sqrt{2\pi}}\left(\frac{1}{\sqrt{2\pi}}\right)^{2}d\mu\left(\frac{1}{\sqrt{2\pi}}\right) \frac{d\mu}{\sqrt{2\pi}}\,.$

 $\frac{1}{2}$

 $\mathcal{L}(\mathcal{L}(\mathcal{L}))$. The set of $\mathcal{L}(\mathcal{L})$

 $\sim 10^{-10}$

The average value of \underline{k} = .32₉

 \bullet

 $\mathcal{L}^{\text{max}}(\mathcal{L}^{\text{max}})$

Appendix II

Final Structure Factors

This table lists all the observed and computed structure factors placed on an absolute basis as described in Appendix I.

A comparison of the observed and computed structure factors shows an agreement as good as the experimental data warrants. The largest disagreement occurs for the two strongest reflections **(060)** and *(330)* and is probably due to extinction for which no correction has been made.

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 $\mathcal{L}^{\text{max}}_{\text{max}}$

Biographical Note

Leonid Vladimirovitch Azaroff, son of Vladimir I. Azaroff, Ph.D., and Maria Y. Azaroff, B.S., was born in Moscow, Russia on June **19, 1926.** He spent his early **child**hood in Riga, Latvia, where he obtained his elementary education. In **1939,** the Azaroff family emigrated to the United States.

His secondary school education was completed in the United States when he graduated from the Berkeley Preparatory School, Boston, Massachusetts, in August, 1943. In September of that year he was awarded a scholarship at Tuft's College, and enrolled in the Physics Department where he remained until called to enter the service in 1944.

During part of the two years spent in the service, he attended the Virginia Polytechnic Institute, Blacksburg, Va. where he studied civil engineering for three terms. During the last year he was stationed at Fort Belvoir, Va., where he worked as a research assistant in the Infra-Red Section of the Army Engineer Board. He was honorably discharged in 1946.

Upon return to civilian life, he re-entered Tufts College, from which he received a B.S., cum laude, in

88.

Physics, in June, 1948. While attending Tufts College he was elected to $\sum \mathcal{P}\sum$, Physics Honorary Society and became a member of the American Association of Physics Teachers and the American Institute of Physics.

In September of *1948* he was admitted to the Graduate School of the Massachusetts Institute of Technology where he worked towards a Ph.D. in geophysics. In the summer **of** *1950,* he attended a special summer course in x-ray crystallography at Brooklyn Polytechnic Institute, Brooklyn, N.Y., and upon returning to M.I.T. in the fall changed his major field to crystallography. He held an appointment as teaching fellow in **1952** and **1953** during which time he assisted in the teaching of Elementary Crystallography, Theoretical Crystallography and X-Ray Mlineralogy. From September, **1952** to June, **1953** he was appointed a full-time research assistant and conducted research in crystallography. In 1952 he was elected to $\sum X$, Graduate Honor Society. He is also a member of the American Crystallographic Association and the Mineralogical Society of America. He presented a paper titled **"A** One-Dimensional Analogue Computer" at the June, **1953** meeting of the A.C.A. and has published an article titled **"A** Telescoping Direct-Beam Tunnel" in the Review **of** Scientific Instruments. **(1953)**