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Thickness and power dependence of the spin-pumping effect in $\text{Y}_3\text{Fe}_5\text{O}_{12}/\text{Pt}$ heterostructures measured by the inverse spin Hall effect

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The dependence of the spin-pumping effect on the yttrium iron garnet ($\text{Y}_3\text{Fe}_5\text{O}_{12}$, YIG) thickness detected by the inverse spin Hall effect (ISHE) has been investigated quantitatively. Due to the spin-pumping effect driven by the magnetization precession in the ferrimagnetic insulator $\text{Y}_3\text{Fe}_5\text{O}_{12}$ film a spin-polarized electron current is injected into the Pt layer. This spin current is transformed into electrical charge current by means of the ISHE. An increase of the ISHE voltage with increasing film thickness is observed and compared to the theoretically expected behavior. The effective damping parameter of the YIG/Pt samples is found to be enhanced with decreasing $\text{Y}_3\text{Fe}_5\text{O}_{12}$ film thickness. The investigated samples exhibit a spin mixing conductance of $g_{\text{eff}}^{\uparrow\downarrow} = (3.87 \pm 0.21) \times 10^{18} \text{ m}^{-2}$ and a spin Hall angle between $\theta_{\text{ISHE}} = 0.013 \pm 0.001$ and 0.045 ± 0.004 depending on the used spin-diffusion length. Furthermore, the influence of nonlinear effects on the generated voltage and on the Gilbert damping parameter at high excitation powers is revealed. It is shown that for small YIG film thicknesses a broadening of the linewidth due to nonlinear effects at high excitation powers is suppressed because of a lack of nonlinear multimagnon scattering channels. We have found that the variation of the spin-pumping efficiency for thick YIG samples exhibiting pronounced nonlinear effects is much smaller than the nonlinear enhancement of the damping.

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I. INTRODUCTION

The generation and detection of spin currents have attracted much attention in the field of spintronics [1,2]. An effective method for detecting magnonic spin currents is the combination of spin pumping and the inverse spin Hall effect (ISHE). Spin pumping refers to the generation of spin-polarized electron currents in a normal metal from the magnetization precession in an attached magnetic material [3,4]. These spin-polarized electron currents are transformed into conventional charge currents by the ISHE, which allows for a convenient electric detection of spin-wave spin currents [5–8].

After the discovery of the spin-pumping effect in ferrimagnetic insulator (yttrium iron garnet, $\text{Y}_3\text{Fe}_5\text{O}_{12}$, YIG)/nonmagnetic metal (platinum, Pt) heterosystems by Kajiwara *et al.* [7], there was rapidly emerging interest in the investigation of these structures [6,7,9–23]. Since $\text{Y}_3\text{Fe}_5\text{O}_{12}$ is an insulator with a band gap of 2.85 eV [24] no direct injection of a spin-polarized electron current into the Pt layer is possible. Thus, spin pumping in YIG/Pt structures can only be realized by exchange interaction between conduction electrons in the Pt layer and localized electrons in the YIG film.

Spin pumping into the Pt layer transfers spin angular momentum from the YIG film thus reducing the magnetization in the YIG film. This angular momentum transfer results in turn in an enhancement of the Gilbert damping of the

magnetization precession. The magnitude of the transfer of angular momentum is independent of the ferromagnetic film thickness since spin pumping is an interface effect. However, with decreasing film thickness, the ratio between surface to volume increases and, thus, the interface character of the spin-pumping effect comes into play: the deprivation of spin angular momentum becomes notable with respect to the precession of the entire magnetization in the ferromagnetic layer. Thus, the average damping for the whole film increases with decreasing film thicknesses. It is predicted theoretically [3] and shown experimentally in ferromagnetic metal/normal metal heterostructures ($\text{Ni}_{81}\text{Fe}_{19}/\text{Pt}$) that the damping enhancement due to spin pumping is inversely proportional to the thickness of the ferromagnet [25,26].

Since the direct injection of electrons from the insulator YIG into the Pt layer is not possible and spin pumping is an interface effect, an optimal interface quality is required in order to obtain a high spin- to charge-current conversion efficiency [27,28]. Furthermore, Tashiro *et al.* have experimentally demonstrated that the spin mixing conductance is independent of the YIG thickness in YIG/Pt structures [15]. Recent progress in the growth of YIG films allows for the fabrication of low-damping nanometer-thick YIG films [9–11]. Recently, Castel *et al.* reported on the YIG thickness and frequency dependence of the spin-pumping process [29,30]. In contrast to our investigations, they concentrate on rather thick (>200 nm) YIG films, which are much thicker than the exchange correlation length in YIG [31–34] and thicker than the Pt thickness. Thus, the YIG film thickness dependence in the nanometer range is still not addressed till now.

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In this paper, we report systematic measurements of the spin- to charge-current conversion in YIG/Pt structures as a function of the YIG film thickness from 20 nm to 275 nm. The Pt thickness is kept constant at 9 nm for all samples. We determine the effective damping as well as the ISHE voltage as a function of YIG thickness and find that the thickness plays a key role. From these characteristics the spin mixing conductance and the spin Hall angle are estimated. The second part of this paper addresses microwave power dependent measurements of the ISHE-induced voltage U_{ISHE} and the ferromagnetic resonance linewidth for varying YIG film thicknesses. The occurrence of nonlinear magnon-magnon scattering processes that result in a widening of the linewidth as well as their influence on the spin-pumping efficiency are discussed.

II. SAMPLE FABRICATION AND EXPERIMENTAL DETAILS

In Fig. 1(a) a schematic illustration of the investigated samples is shown. Monocrystalline $\text{Y}_3\text{Fe}_5\text{O}_{12}$ samples of 20, 75, 145, 240, 275 nm thickness were deposited by means of pulsed laser deposition (PLD) from a stoichiometric target using a KrF excimer laser with a fluence of 2.6 J/cm^2 and a repetition rate of 10 Hz [10,35]. In order to ensure epitaxial growth of the films, single-crystalline substrates of gadolinium gallium garnet ($\text{Gd}_3\text{Ga}_5\text{O}_{12}$, GGG) in the (100) orientation were used. We achieved optimal deposition conditions for a substrate temperature of $650^\circ\text{C} \pm 30^\circ\text{C}$ and an oxygen pressure of 6.67×10^{-3} mbar. Afterwards, each film was annealed *ex situ* at $820^\circ\text{C} \pm 30^\circ\text{C}$ by rapid thermal annealing for 300 s under a steady flow of oxygen. We determined the YIG thickness by etching a hole in the YIG film while monitoring the etched elements using a mass spectrometer and subsequent measurement of the height profile by atomic force microscopy (AFM). The crystalline quality was measured

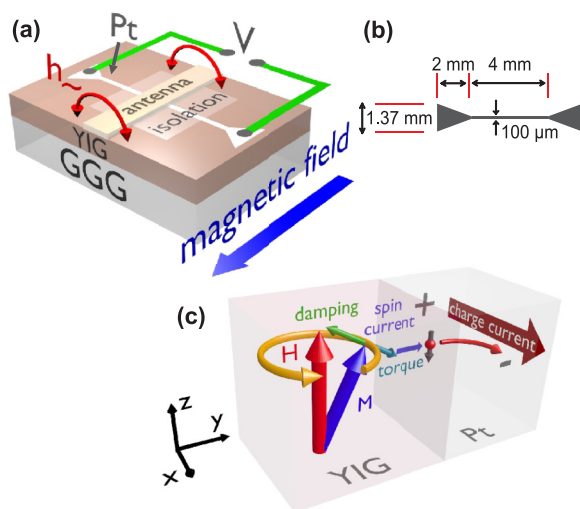


FIG. 1. (Color online) (a) Schematic illustration of the experimental setup. (b) Dimensions of the structured Pt layer on the YIG films. The Pt layer was patterned by means of optical lithography and ion etching. (c) Scheme of combined spin-pumping process and inverse spin Hall effect.

by x-ray diffraction (XRD). In order to deposit Pt onto the samples, they were transferred at atmosphere leading to possible surface adsorbates. Therefore, the YIG film surfaces were cleaned *in situ* by a low-power Ar-ion etching using a plasma source (Gen2, Tectra) for 20 seconds at an angle of 30° at a YIG etch rate $< 0.06 \text{ nm/min}$ before the Pt deposition [27]. We used dc sputtering under an argon pressure of 1×10^{-2} mbar at room temperature to deposit the Pt layers. XRR (x-ray reflectivity) measurements yielded a Pt thickness of 9 nm, which is identical for every sample due to the simultaneously performed Pt deposition. The Pt layer was patterned by means of optical lithography and ion etching. In order to isolate the Pt stripes from the antenna, we deposited a 300 nm thick square of SU-8 photoresist on the top. A sketch of the samples and the experimental setup is shown in Fig. 1(a); the dimensions of the structured Pt stripe are depicted in Fig. 1(b).

In order to corroborate the quality of the fabricated YIG samples, we performed ferromagnetic resonance (FMR) measurements using a vector network analyzer (VNA) [36]. Since the area deposited by Pt is small compared to the entire sample size, we measure the damping α_0 of the bare YIG by VNA (this approach results in a small overestimate of α_0), whereas in the spin-pumping measurement we detect the enhanced damping α_{eff} of the Pt-covered YIG films. The VNA-FMR results are summarized in Table I and in Fig. 2. Apparently, the 20 nm sample features the largest damping of $\alpha_0^{20\text{nm}} = (2.169 \pm 0.069) \times 10^{-3}$. With increasing film thickness α_0 decreases to $\alpha_0^{240\text{nm}} = (0.093 \pm 0.007) \times 10^{-3}$. The inhomogeneous linewidth broadening of our films varies between 1.36×10^{-4} and $3.24 \times 10^{-4} \text{ T}$ [36,38]. The variation of α_0 is attributed to the two-magnon scattering process at the interface which is more pronounced for smaller film thicknesses and gives rise to additional damping. In order to check this assumption, the data illustrated in Fig. 2 were fitted by a $1/d_{\text{YIG}}^2$ dependence. This dependence is expected for the momentum-nonconserving two-magnon scattering process proposed by Arias and Mills and experimentally shown by Azevedo *et al.* [39,40] and we find reasonable agreement. We emphasize that this behavior is due to the magnetization dynamics in the ferromagnetic material and it does not describe a spin-pumping related interface effect which will be discussed in Sec. III B.

The VNA-FMR technique also yields the effective magnetization M_{eff} , whereas the saturation magnetization M_S is determined by a superconducting quantum interference device (SQUID) (see Table I). The saturation magnetization M_S is close to the bulk value [38,41]. From the difference between M_S and M_{eff} , we determine the out-of-plane uniaxial anisotropy K_{out} following the approach presented in Ref. [37] assuming a negligible cubic anisotropy K_{cub} (this assumption is justified since $K_{\text{out}} \gg K_{\text{cub}}$). The results are summarized in Table I.

The spin-pumping measurements for different YIG film thicknesses were performed in the following way. The samples were magnetized in the film plane by an external magnetic field \mathbf{H} , and the magnetization dynamics was excited at a constant frequency of $f = 6.8 \text{ GHz}$ by an Agilent E8257D microwave source. The microwave signals with powers P_{applied} of 1, 10, 20, 50, 100, 250, and 500 mW were applied to a $600 \mu\text{m}$ wide 50 ohm matched Cu microstrip antenna. The stripline

TABLE I. Saturation magnetization M_S measured by SQUID and effective magnetization M_{eff} determined by a VNA-FMR measurement technique for different YIG film thicknesses. The out-of-plane anisotropy constant K_{out} and the corresponding anisotropy field H_{out} are determined according to Ref. [37]. The Gilbert damping parameter α_0 is determined by FMR measurements.

d_{YIG} (nm)	M_S (kA/m)	M_{eff} (kA/m)	K_{out} (J/m ³)	$\mu_0 H_{\text{out}}$ (mT)	α_0 ($\times 10^{-3}$)
20 ± 3	82 ± 12	161.7 ± 0.2	-4110.0 ± 4.2	-100.2 ± 0.1	2.169 ± 0.069
75 ± 10	132 ± 18	176.4 ± 0.1	-3696.6 ± 7.0	-56.0 ± 0.1	0.489 ± 0.007
145 ± 15	140 ± 14	175.1 ± 0.2	-3080.1 ± 5.3	-44.0 ± 0.1	0.430 ± 0.015
240 ± 15	143 ± 19	176.5 ± 0.1	-3000.6 ± 10.6	-42.0 ± 0.2	0.093 ± 0.007
275 ± 20	137 ± 10	176.4 ± 0.1	-3393.0 ± 6.0	-49.5 ± 0.1	0.162 ± 0.008

antenna was fabricated from Cu-laminated duroid substrates and the samples were flipped upside down on the stripline. The bottom side of the duroid was metallized and serves as ground. While the external magnetic field was swept, the ISHE voltage U_{ISHE} was recorded at the edges of the Pt stripe using a lock-in technique with an amplitude modulation at a frequency of 500 Hz, as well as the absorbed microwave power P_{abs} . All measurements were performed at room temperature (292 K).

III. THEORETICAL BACKGROUND

The equations describing the ferromagnetic resonance, the spin pumping, and the inverse spin Hall effect are provided in the following and used in the experimental part of this paper.

A. Ferromagnetic resonance

In equilibrium, the magnetization \mathbf{M} in a ferromagnetic material is aligned along the bias magnetic field \mathbf{H} . Applying an alternating microwave magnetic field \mathbf{h}_\perp perpendicularly to the external field \mathbf{H} results in a torque on \mathbf{M} and causes the magnetic moments in the sample to precess [see also Fig. 1(a)].

The ferromagnetic resonance for a field applied in the film plane occurs at the frequency [39,40]

$$f = \frac{\mu_0 \gamma}{2\pi} \sqrt{H_{\text{FMR}}(H_{\text{FMR}} + M_{\text{eff}})}, \quad (1)$$

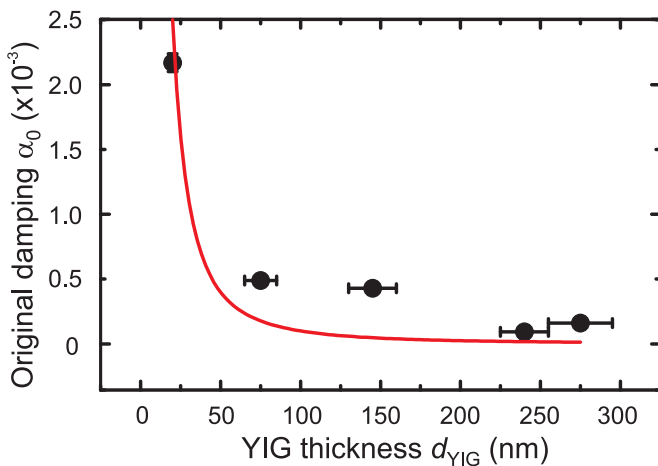


FIG. 2. (Color online) Original Gilbert damping parameter α_0 measured by VNA-FMR technique. The fit shown as a red solid line represents a $1/d_{\text{YIG}}^2$ dependence due to momentum-nonconserving two-magnon scattering at the interface [39,40].

where μ_0 is the vacuum permeability, γ is the gyromagnetic ratio, H_{FMR} is the ferromagnetic resonance field, and M_{eff} is the effective magnetization.

The FMR linewidth ΔH (full width at half maximum) is related to the Gilbert damping parameter α as [26,28,38]

$$\mu_0 \Delta H = \mu_0 \Delta H_0 + 4\pi f \alpha / \gamma. \quad (2)$$

Here, ΔH_0 describes the inhomogeneous linewidth broadening. Equation (2) is a general expression and describes the relation of the frequency dependence of the linewidth and the magnetic damping. Later it will be used to determine the original damping of the bare YIG films α_0 as well as the enhanced damping due to spin pumping α_{eff} .

B. Spin pumping

By attaching a thin Pt layer to a ferromagnet, the resonance linewidth is enhanced [3], which accounts for an injection of a spin current from the ferromagnet into the normal metal due to the spin-pumping effect [see illustration in Fig. 1(c)]. In this process the magnetization precession loses spin angular momentum, which gives rise to additional damping and, thus, to an enhanced linewidth. The effective Gilbert damping parameter α_{eff} for the YIG/Pt film is described as [26]

$$\alpha_{\text{eff}} = \alpha_0 + \Delta\alpha = \alpha_0 + \frac{g\mu_B}{4\pi M_S d_{\text{YIG}}} g_{\text{eff}}^{\uparrow\downarrow}, \quad (3)$$

where $\alpha_0(d_{\text{YIG}})$ is the intrinsic damping of the bare YIG film (described also in Sec. II), g is the g factor, μ_B is the Bohr magneton, d_{YIG} is the YIG film thickness, and $g_{\text{eff}}^{\uparrow\downarrow}$ is the real part of the effective spin mixing conductance. The effective Gilbert damping parameter α_{eff} is inversely proportional to the YIG film thickness d_{YIG} : with decreasing YIG thickness the linewidth and, thus, the effective damping parameter increases.

When the system is resonantly driven in the FMR condition, a spin-polarized electron current is injected from the magnetic material (YIG) into the normal metal (Pt). In a phenomenological spin-pumping model, the dc component of the spin-current density j_s at the interface, injected in the y direction into the Pt layer [Fig. 1(c)], can be described as [25,26,42]

$$j_s = f \int_0^{1/f} \frac{\hbar}{4\pi} g_{\text{eff}}^{\uparrow\downarrow} \frac{1}{M_S^2} \left\{ \mathbf{M}(t) \times \frac{d\mathbf{M}(t)}{dt} \right\}_z dt, \quad (4)$$

where $\mathbf{M}(t)$ is the magnetization. $\{\mathbf{M}(t) \times \frac{d\mathbf{M}(t)}{dt}\}_z$ is the z component of $\{\mathbf{M}(t) \times \frac{d\mathbf{M}(t)}{dt}\}$, which is directed along the equilibrium axis of the magnetization [see Fig. 1(c)].

Due to spin relaxation in the normal metal (Pt) the injected spin current j_s decays along the Pt thickness [y direction in

Fig. 1(c)] as [25,26]

$$j_s(y) = \frac{\sinh \frac{d_{\text{Pt}} - y}{\lambda}}{\sinh \frac{d_{\text{Pt}}}{\lambda}} j_s^0, \quad (5)$$

where λ is the spin-diffusion length in the Pt layer. From Eq. (4) one can deduce the spin-current density at the interface ($y = 0$) [25]:

$$j_s^0 = \frac{g_{\text{eff}}^{\uparrow\downarrow} \gamma^2 (\mu_0 h \sim)^2 \hbar (\mu_0 M_S \gamma + \sqrt{(\mu_0 M_S \gamma)^2 + 16(\pi f)^2})}{8\pi \alpha_{\text{eff}}^2 [(\mu_0 M_S \gamma)^2 + 16(\pi f)^2]}. \quad (6)$$

Since j_s^0 is inversely proportional to α_{eff}^2 and α_{eff} depends inversely on d_{YIG} [Eq. (3)], the spin-current density at the interface j_s^0 increases with increasing YIG film thickness d_{YIG} .

C. Inverse spin Hall effect

The Pt layer acts as a spin-current detector and transforms the spin-polarized electron current injected due to the spin-pumping effect into an electrical charge current by means of the ISHE [see Fig. 1(c)] as [6,7,16,25,26]

$$\mathbf{j}_c = \theta_{\text{ISHE}} \frac{2e}{\hbar} \mathbf{j}_s \times \boldsymbol{\sigma}, \quad (7)$$

where θ_{ISHE} , e , $\boldsymbol{\sigma}$ denote the spin Hall angle, the electron's elementary charge, and the spin-polarization vector, respectively. Averaging the charge-current density across the Pt thickness and taking into account Eqs. (4)–(7) yields

$$\bar{j}_c = \frac{1}{d_{\text{Pt}}} \int_0^{d_{\text{Pt}}} j_c(y) dy = \theta_{\text{ISHE}} \frac{\lambda}{d_{\text{Pt}}} \frac{2e}{\hbar} \tanh\left(\frac{d_{\text{Pt}}}{2\lambda}\right) j_s^0. \quad (8)$$

Taking into account Eqs. (3), (6), and (8) we calculate the theoretically expected behavior of $I_{\text{ISHE}} = A \bar{j}_c$, where A is the cross section of the Pt layer. Ohm's law connects the ISHE voltage U_{ISHE} with the ISHE current I_{ISHE} via $U_{\text{ISHE}} = I_{\text{ISHE}} R$, where R is the electric resistance of the Pt layer. R varies between 1450 Ω and 1850 Ω for the different samples; the resistivities vary between 3.3×10^{-7} and 4.2×10^{-7} Ωm , which is in agreement with values reported by Wang *et al.* for Pt on YIG [23]. Since the samples were grown using the same technique in the same growth chamber, it is not very likely that this variation of the resistance can be attributed to the surface roughness. Atomic force microscopy imaging reveals a rms surface roughness between 0.2 to 0.33 nm for all YIG film thicknesses measured at several points on the sample. It is not expected that this slight change in the surface roughness would lead to a 30% change in the resistance. A possible explanation for the difference in resistance might be a Pt redeposition at the edges occurring during the ion-etching process.

IV. YIG FILM THICKNESS DEPENDENCE OF THE SPIN-PUMPING EFFECT DETECTED BY THE ISHE

In Fig. 3 the magnetic field dependence of the generated ISHE voltage U_{ISHE} as a function of the YIG film thickness is shown. Clearly, the maximum voltage U_{ISHE} at the resonance field H_{FMR} and the FMR linewidth ΔH vary with the YIG film thickness. [Please note that the polarity of the ISHE

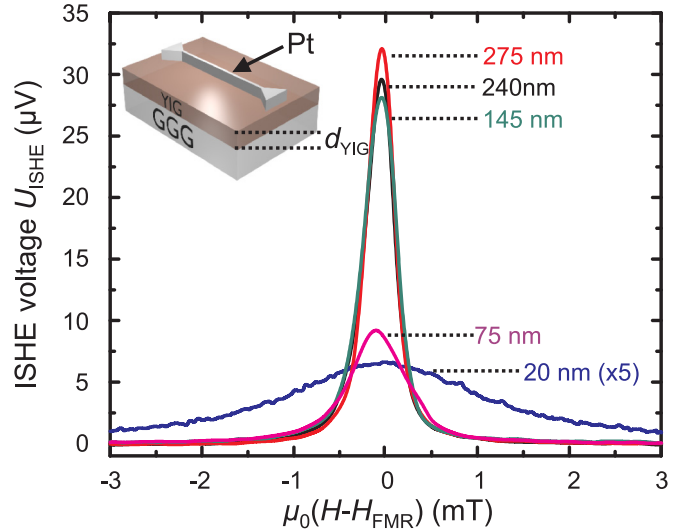


FIG. 3. (Color online) ISHE-induced voltage U_{ISHE} as a function of the magnetic field H for different YIG film thicknesses d_{YIG} . Applied microwave power $P_{\text{applied}} = 10$ mW; ISHE voltage for the 20 nm thick sample is multiplied by a factor of 5.

voltage changes sign upon field reversal in accordance with Eq. (7).] The microwave absorption data follow this behavior. The general trend shows that the thinner the sample the smaller is the magnitude of the observed voltage U_{ISHE} . At the same time the FMR linewidth increases with decreasing YIG film thickness.

In the following the ISHE voltage generated by spin pumping is investigated as a function of the YIG film thickness. For these investigations we have chosen a rather small exciting microwave power of 1 mW. Thus, nonlinear effects such as the FMR linewidth broadening due to nonlinear multimagnon processes can be excluded (such processes will be discussed in Sec. V). Section IV A covers the YIG thickness dependent variation of the enhanced damping parameter α_{eff} . From these measurements the spin mixing conductance $g_{\text{eff}}^{\uparrow\downarrow}$ is deduced. In Sec. IV B we focus on the maximum ISHE voltage driven by spin pumping as a function of the YIG film thickness. Finally, the spin Hall angle θ_{ISHE} is determined.

A. YIG film parameters as a function of the YIG film thickness

As described in Sec. III B, the damping parameter is enhanced when a Pt layer is deposited onto the YIG film. This enhancement is investigated as a function of the YIG film thickness: the effective Gilbert damping parameter α_{eff} [see Eq. (3)] is obtained from a Lorentzian fit to the experimental data depicted in Fig. 3 and Eq. (2). The result is shown in Fig. 4. With decreasing YIG film thickness the linewidth and, thus, the damping enhancement $\Delta\alpha$ increases. This behavior is theoretically expected: according to Eq. (3) $\Delta\alpha$ is inversely proportional to d_{YIG} . Even if the thinnest sample with the largest contribution of the two-magnon scattering to the damping (resulting in a larger damping α_0 ; see also fit in Fig. 2) had a lower quality, the analysis of the spin mixing would not be altered since only the enhancement of the Gilbert damping parameter $\Delta\alpha$ due to the presence of Pt is

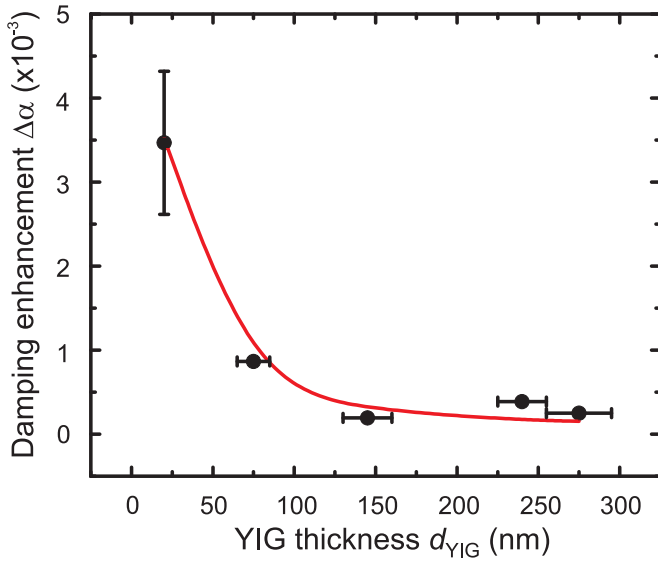


FIG. 4. (Color online) Damping enhancement $\Delta\alpha$ of the YIG/Pt samples obtained by spin-pumping measurements. The red solid curve shows a fit to Eq. (3) taking the measured values for M_S and a constant value for $g_{\text{eff}}^{\uparrow\downarrow}$ into account. $P_{\text{applied}} = 1$ mW. The error bars for the measurement points at higher sample thicknesses are not visible at this scale.

important [see Eq. (3)]. Since the Pt film is grown onto all YIG samples simultaneously, the spin mixing conductance $g_{\text{eff}}^{\uparrow\downarrow}$ at the interface is considered to be constant for all samples [15]. Assuming $g_{\text{eff}}^{\uparrow\downarrow}$ as constant and taking the saturation magnetization M_S obtained by SQUID measurements (Table I) and the original damping parameters $\alpha_0(d_{\text{YIG}})$ (Table I) into account, a fit to Eq. (3) yields $g_{\text{eff}}^{\uparrow\downarrow} = (3.87 \pm 0.21) \times 10^{18} \text{m}^{-2}$. The fit is depicted as a red solid line in Fig. 4. Apparently, the spin-pumping induced damping enhancement shown in Fig. 4 decreases with increasing YIG-film thickness and saturates. This behavior is in qualitative agreement with predictions by Rezende *et al.* [43].

B. YIG thickness dependence of the ISHE voltage driven by spin pumping

Figure 5(a) shows the maximum voltage U_{ISHE} at the resonance field H_{FMR} as a function of the YIG film thickness. U_{ISHE} increases up to a YIG film thickness of around 240 nm when it starts to saturate (in the case of an applied microwave power of $P_{\text{applied}} = 1$ mW). The corresponding charge current I_{ISHE} is shown in Fig. 5(b). The observed thickness-dependent behavior is in agreement with the one reported for $\text{Ni}_{81}\text{Fe}_{19}/\text{Pt}$ [26] and for $\text{Y}_3\text{Fe}_5\text{O}_{12}/\text{Pt}$ [15]. With increasing YIG film thickness the generated ISHE-current increases and tends to saturate at thicknesses near 240 nm [Fig. 5(b)]. According to Eqs. (3), (6), and (8) it is $I_{\text{ISHE}} \propto j_s^0 \propto 1/\alpha_{\text{eff}}^2 \propto (\alpha_0 + c/d_{\text{YIG}})^{-2}$, where c is a constant. Therefore, the ISHE current I_{ISHE} increases with increasing YIG film thickness d_{YIG} and goes into saturation at a certain YIG thickness.

From Eqs. (3), (6), and (8) we determine the expected behavior of $I_{\text{ISHE}} = A \bar{j}_c$ and compare it with our experimental

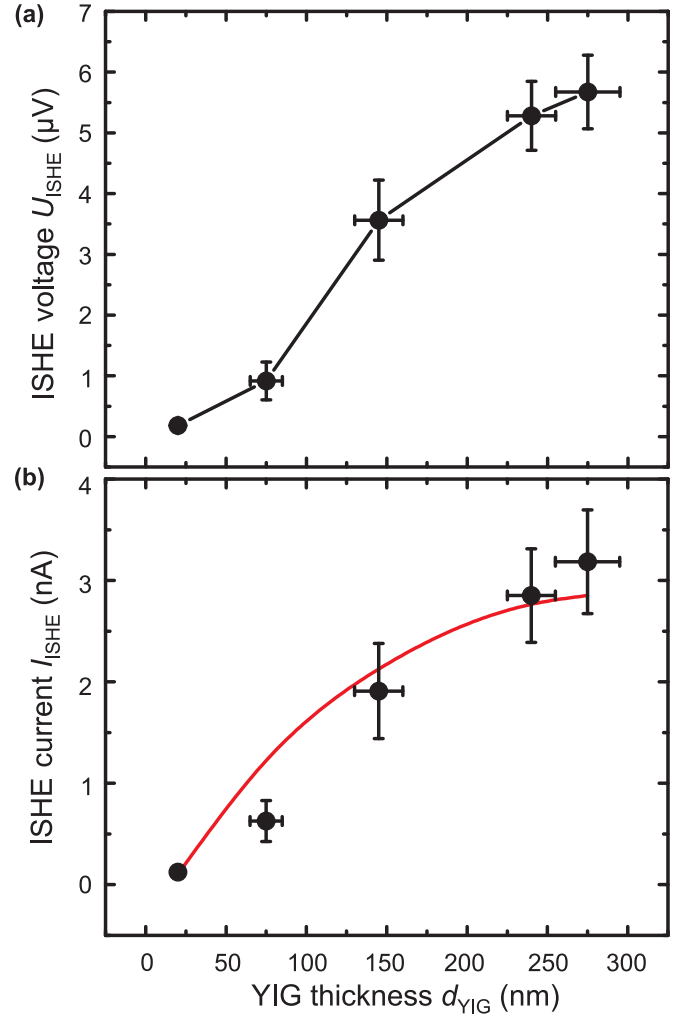


FIG. 5. (Color online) (a) ISHE voltage U_{ISHE} as a function of the YIG film thickness d_{YIG} . (b) Corresponding thickness-dependent charge current I_{ISHE} . The red curve shows a fit to Eqs. (6), (7), (8) with the parameters $g_{\text{eff}}^{\uparrow\downarrow} = (3.87 \pm 0.21) \times 10^{18} \text{m}^{-2}$ and $\theta_{\text{ISHE}} = 0.045 \pm 0.004$. The applied microwave power used is $P_{\text{applied}} = 1$ mW.

data. In order to do so, M_S (see Table I), the original damping parameter α_0 determined by VNA-FMR measurements at 1 mW (see Table I), and the enhanced damping parameter α_{eff} obtained by spin-pumping measurements at a microwave power of 1 mW (see also Fig. 4) are used. The Pt layer thickness is $d_{\text{Pt}} = 9$ nm and the microwave magnetic field is determined to be $h_{\sim} = 3.2$ A/m for an applied microwave power of 1 mW using an analytical expression [44]. The spin-diffusion length in Pt reported in the literature varies roughly between 1 and 10 nm. In order to estimate the spin Hall angle we use a lower and upper value. For a spin-diffusion length of $\lambda = 1.2$ nm [45] we find $\theta_{\text{ISHE}} = 0.045 \pm 0.004$; for a spin-diffusion length of $\lambda = 7.3$ nm [23], the spin Hall angle is $\theta_{\text{ISHE}} = 0.013 \pm 0.001$. Figure 5(b) shows exemplarily the fit to our data using $\lambda = 1.2$ nm. Here, the damping parameter is assumed to be constant as $\alpha_0 = 6.68 \times 10^{-4}$, which is the average of the measured values of α_0 . The spin Hall angle we find here is in agreement with literature values varying in a range of 0.0037–0.086 [45–48]. It is interesting to mention

that although Wang *et al.* [23] investigate similar YIG-film thicknesses and report a comparable spin mixing conductance, they find a larger spin Hall angle. This difference might be attributed to a difference in the Pt quality. Using the fit we estimate the saturation value of the generated current. According to our fit, we find that 90% of the estimated saturation level of 4.5 nA is reached at a sample thickness of $\sim 1.2\mu\text{m}$.

V. INFLUENCE OF NONLINEAR EFFECTS ON THE SPIN-PUMPING PROCESS FOR VARYING YIG FILM THICKNESSES

In order to investigate nonlinear effects on the spin-pumping effect for varying YIG film thicknesses, we performed microwave power dependent measurements of the ISHE voltage U_{ISHE} as function of the film thickness d_{YIG} . For higher microwave powers in the range of 1 mW to 500 mW we observe the same thickness-dependent behavior of the ISHE voltage as in the linear case ($P_{\text{applied}} = 1$ mW; discussed in Sec. IV B): Near 240 nm U_{ISHE} starts to saturate independently of the applied microwave power, as is shown in Fig. 6(a). Furthermore, it is clearly visible from Fig. 6(a) that for a constant film thickness the spin pumping driven ISHE voltage increases with increasing applied microwave power.

At high microwave powers the voltage does not grow linearly and saturates. Figure 6(b) shows the deviation of the ISHE voltage ΔU_{ISHE} from the linear behavior with respect to the measured value of $U_{\text{ISHE}}^{500\text{mW}}$ at the excitation power $P_{\text{applied}} = 500$ mW. In order to obtain the relation between U_{ISHE} and P_{applied} for each YIG film thickness d_{YIG} the low-power regime up to 20 mW is fitted by a linear curve and extrapolated to 500 mW. The inset in Fig. 6(b) shows the corresponding graph for the case of the 20 nm thick sample. As seen from Fig. 6(b), the deviation from the linear behavior is drastically enhanced for larger YIG thicknesses. For the 20 nm and 75 nm samples we observe an almost linear behavior between U_{ISHE} and P_{applied} over the entire microwave power range, whereas for the thicker samples the estimated linear behavior and the observed nonlinear behavior differ approximately by a factor of 2.5 [Fig. 6(b)].

Castel *et al.* reported on the frequency and power dependence of spin pumping from a 200 nm thick YIG film into a 15 nm thick Pt layer [29]. They used rather low microwave powers up to 70 mW and they observed a linear relation between power and ISHE voltage. This result agrees with our observation in the low-power regime (Fig. 6). The frequency dependence was not studied here. We observe an increase of the ISHE voltage as well as a broadening of the FMR linewidth with increasing microwave power.

In Fig. 7 the normalized absorbed microwave power P_{abs} (a) and ISHE voltage U_{ISHE} (b) as function of the external magnetic field H are shown for different microwave powers P_{applied} in the range of 1 mW to 500 mW (YIG film thickness $d_{\text{YIG}} = 240$ nm). The linewidths for both signals tend to be asymmetric at higher microwave powers. The shoulder at lower magnetic field is widened in comparison to the shoulder at higher fields. The reason for this asymmetry might be due to the formation of a foldover effect [49,50], due to nonlinear damping or a nonlinear frequency shift [51,52].

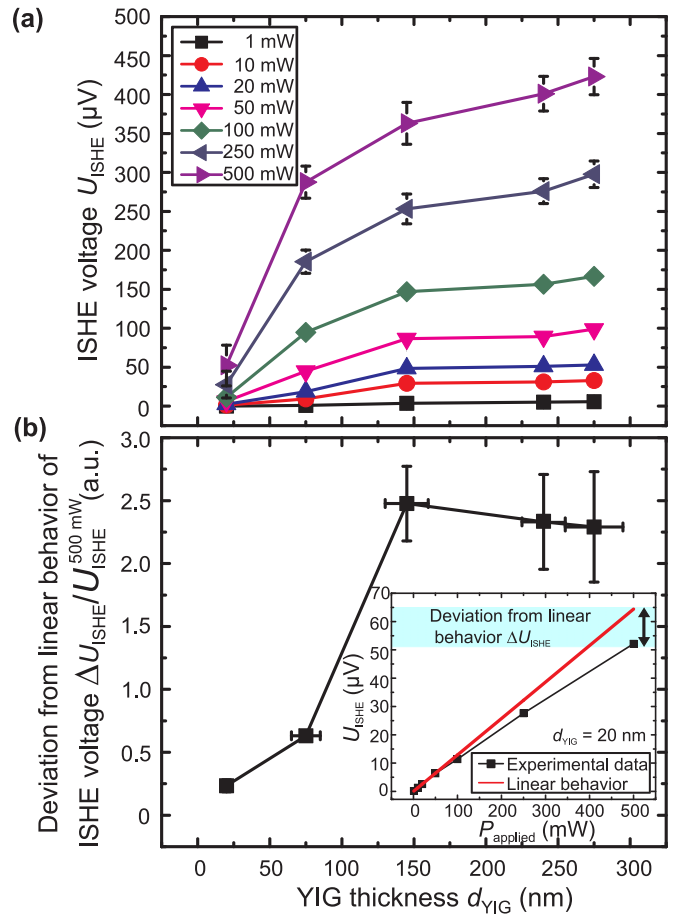


FIG. 6. (Color online) (a) YIG thickness dependence of the ISHE voltage driven by spin pumping for microwave powers in the range between 1 and 500 mW. The general thickness-dependent behavior is independent of the applied microwave power. The error bars for the measurement at lower microwave powers are not visible at this scale. (b) Deviation of the ISHE voltage from the linear behavior with respect to the measured voltage $U_{\text{ISHE}}^{500\text{mW}}$. The inset shows experimental data for a YIG film thickness $d_{\text{YIG}} = 20$ nm and the theoretically expected curve. The error bars of the 20 nm sample are not visible at this scale.

The results of the damping parameter α_{eff} obtained by microwave power dependent spin-pumping measurements are depicted in Fig. 8(a). It can be seen that with increasing excitation power the Gilbert damping for thicker YIG films is drastically increased. To present this result more clearly the nonlinear damping enhancement $(\alpha_{\text{eff}}^{500\text{mW}} - \alpha_{\text{eff}}^{1\text{mW}})/\alpha_{\text{eff}}^{1\text{mW}}$ is shown in Fig. 8(b). The damping parameter at a sample thickness of 20 nm $\alpha_{\text{eff}}^{20\text{nm}}$ is almost unaffected by a nonlinear broadening at high microwave powers. With increasing film thickness the original damping $\alpha_{\text{eff}}^{1\text{mW}}$ at $P_{\text{applied}} = 1$ mW increases by a factor of around 3 at $P_{\text{applied}} = 500$ mW. This factor is very close to the value of the deviation of the ISHE voltage from the linear behavior [Fig. 6(b)].

This behavior can be attributed to the enhanced probability of nonlinear multimagnon processes at larger sample thicknesses: In order to understand this, a fundamental understanding of the restrictions for multimagnon scattering processes can be derived from the energy and momentum

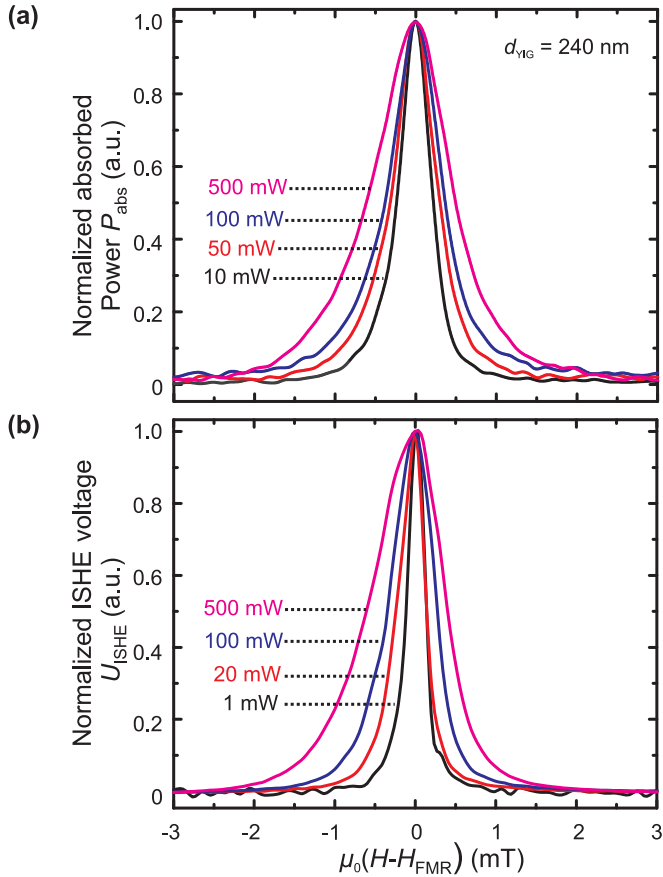


FIG. 7. (Color online) Illustration of the linewidth broadening at higher excitation powers. (a) Normalized absorbed microwave power P_{abs} and (b) normalized ISHE-voltage spectra are shown as a function of the magnetic field H for different excitation powers. Sample thickness: 240 nm.

conservation laws:

$$\sum_i^N \hbar\omega_i = \sum_j^M \hbar\omega_j, \quad \sum_i^N \hbar\mathbf{k}_i = \sum_j^M \hbar\mathbf{k}_j, \quad (9)$$

where the left/right sum of the equations runs over the initial/final magnons with indices i/j which exist before/after the scattering process, respectively [53–55]. The most probable scattering mechanism in our case is the four-magnon scattering process with $N = 2$ and $M = 2$ [55]. Furthermore, it was shown theoretically that there is a lower critical thickness for the three-magnon scattering process (with $N = 1$ and $M = 2$ or $N = 2$ and $M = 1$, respectively) [34]. In Eq. (9) the wave vector $\mathbf{k}_{i/j}$ and the frequency $\omega_{i/j}$ are connected by the dispersion relation $2\pi f_{i/j}(\mathbf{k}_{i/j}) = \omega_{i/j}(\mathbf{k}_{i/j})$. The calculated dispersion relations are shown in Fig. 9 [backward volume magnetostatic spin-wave modes with a propagation angle $\angle(\mathbf{H}, \mathbf{k}) = 0^\circ$ as well as magnetostatic surface spin-wave modes $\angle(\mathbf{H}, \mathbf{k}) = 90^\circ$] [56]. In the case of the 20 nm sample thickness, the first perpendicular standing spin-wave mode (thickness mode) lies above 60 GHz, the second above 120 GHz. Thus, the nonlinear scattering probability obeying the energy and momentum conservation is largely reduced. This means magnons cannot find a proper scattering partner

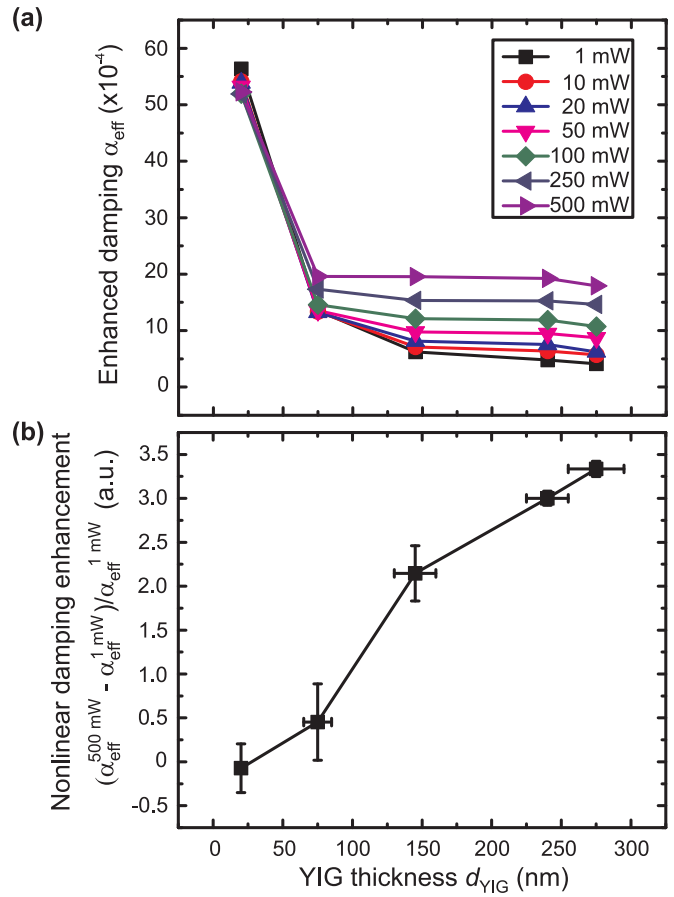


FIG. 8. (Color online) (a) Power-dependent measurement of the damping parameter α_{eff} for different YIG film thicknesses d_{YIG} obtained by a Lorentzian fit to the ISHE-voltage signal. The error bars are omitted in order to provide a better readability of the graph. (b) Nonlinear damping enhancement $(\alpha_{\text{eff}}^{500 \text{ mW}} - \alpha_{\text{eff}}^{1 \text{ mW}}) / \alpha_{\text{eff}}^{1 \text{ mW}}$ as a function of the YIG film thickness d_{YIG} . Due to a reduced number of scattering channels to other spin-wave modes for film thicknesses below 75 nm, the damping is only enhanced for thicker YIG films with increasing applied microwave powers.

and, thus, multimagnon processes are prohibited or at least largely suppressed. With increasing film thickness the number of standing spin-wave modes increases and, thus, the scattering probability grows. As a result, the scattering of spin waves from the initially excited uniform precession (FMR) to other modes is allowed and the relaxation of the original FMR mode is enhanced. Thus, the damping increases and we observe a broadening of the linewidth, which is equivalent to an enhanced Gilbert damping parameter α_{eff} at higher YIG film thicknesses (see Fig. 8). At YIG-film thicknesses above 240 nm the density of states is high enough to satisfy momentum and energy conservation laws [Eq. (9)]. This means the scattering probability is already close to that of a bulk sample and, thus, the enhancement of damping saturates (Fig. 8).

In order to investigate how the spin-pumping efficiency is affected by the applied microwave power, we measure simultaneously the generated ISHE voltage U_{ISHE} and the transmitted (P_{trans}) as well as the reflected (P_{refl}) microwave power, which enables us to determine the absorbed microwave

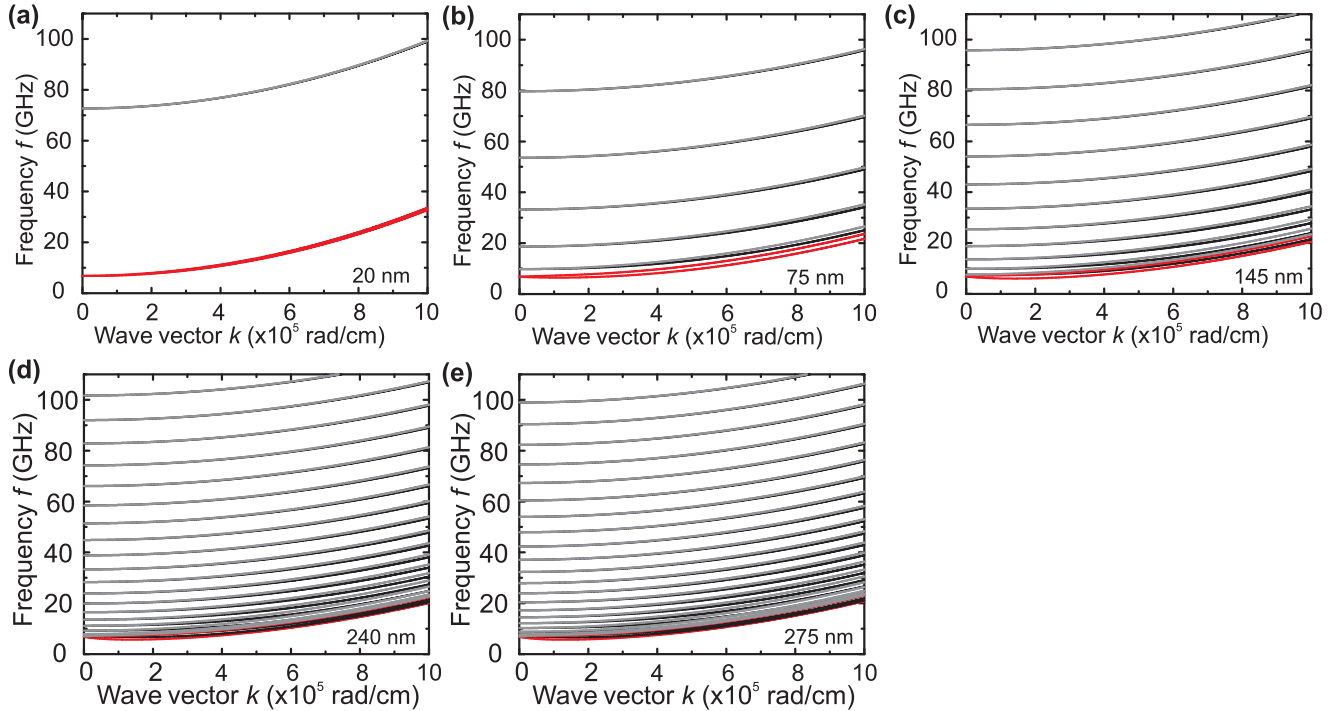


FIG. 9. (Color online) Dispersion relations calculated for each sample thickness taking into account the saturation magnetization measured by SQUID (see Table I). Backward volume magnetostatic spin-wave modes as well as magnetostatic surface spin-wave modes (in red) and the first perpendicular standing thickness spin-wave modes are depicted (in black and gray). (a)–(e) show the dispersion relations for the investigated sample thicknesses of 20 nm to 275 nm.

power $P_{\text{abs}} = P_{\text{applied}} - (P_{\text{trans}} + P_{\text{refl}})$ [27]. Since the 240 nm sample exhibits a strong nonlinearity [large deviation from the linear behavior (Fig. 6) and large nonlinear linewidth enhancement (Fig. 8)], we analyze this sample thickness. In Fig. 10 the normalized absorbed microwave power $P_{\text{norm}} = P_{\text{abs}}/P_{\text{abs}}^{P_{\text{applied}}=1 \text{ mW}}$ and the normalized ISHE voltage in

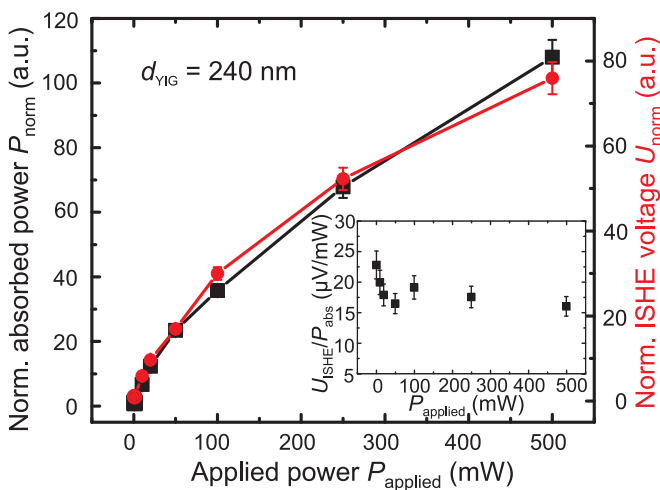


FIG. 10. (Color online) Normalized absorbed power $P_{\text{norm}} = P_{\text{abs}}/P_{\text{abs}}^{P_{\text{applied}}=1 \text{ mW}}$ (black squares) and normalized ISHE voltage $U_{\text{norm}} = U_{\text{ISHE}}/U_{\text{ISHE}}^{P_{\text{applied}}=1 \text{ mW}}$ (red dots) for varying microwave powers P_{applied} . The inset illustrates the independence of the spin-pumping efficiency $U_{\text{ISHE}}/P_{\text{abs}}$ of P_{applied} . YIG thickness illustrated: 240 nm. Error bars of the low-power measurements are not visible at this scale.

resonance $U_{\text{norm}} = U_{\text{ISHE}}/U_{\text{ISHE}}^{P_{\text{applied}}=1 \text{ mW}}$ are shown as a function of the applied power P_{applied} . Both curves tend to saturate at high microwave powers above 100 mW. The absorbed microwave power increases by a factor of 110 for applied microwave powers in the range between 1 and 500 mW, whereas the generated voltage increases by a factor of 80. The spin-pumping efficiency $U_{\text{ISHE}}/P_{\text{abs}}$ (see inset in Fig. 10) varies within a range of 30% for the different microwave powers P_{applied} without clear trend. Since the 240 nm thick film shows a nonlinear deviation of the ISHE voltage by a factor of 2.3 [Fig. 6(b)] and the damping is enhanced by a factor of 3 in the same range of P_{applied} [Fig. 8(c)], we conclude that the spin-pumping process is only weakly dependent on the magnitude of the applied microwave power (see inset in Fig. 10). The reason for this independency is that the energy pumped into the magnetic system is distributed among the entire magnon spectrum via magnon-magnon interactions. In our previous studies reported in Refs. [16,17] we show that secondary magnons generated in a process of multimagnon scattering contribute to the spin-pumping process. Consequently, the spin-pumping efficiency does not depend on the applied microwave power. This also underlines the rich effects and phenomena in magnetization dynamics as the origin of the spin-pumping driven ISHE voltage.

VI. SUMMARY

The $\text{Y}_3\text{Fe}_5\text{O}_{12}$ thickness dependence of the spin-pumping effect detected by the ISHE has been investigated quantitatively. It is shown that the effective Gilbert damping

parameter of the YIG/Pt samples is enhanced for smaller YIG film thicknesses, which is attributed to an increase of the ratio between surface to volume and, thus, to the interface character of the spin-pumping effect. We observe a theoretically expected increase of the ISHE voltage with increasing YIG film thickness with a tendency to saturation above thicknesses of about 240 nm. The spin mixing conductance $g_{\text{eff}}^{\uparrow\downarrow} = (3.87 \pm 0.21) \times 10^{18} \text{ m}^{-2}$ as well as a lower and an upper limit for the spin Hall angle ($\theta_{\text{ISHE}} = 0.013 \pm 0.001$ and $\theta_{\text{ISHE}} = 0.045 \pm 0.004$) are calculated and are found to be in agreement with values reported in the literature for our materials.

The microwave power dependent measurements reveal the occurrence of nonlinear effects for the different YIG film thicknesses: for low powers, the induced voltage grows linearly with the power. At high powers, we observe a saturation of the ISHE voltage U_{ISHE} and a deviation by a factor of 2.5 from the linear behavior. The microwave power dependent investigations of the Gilbert damping parameter by spin pumping show an enhancement by a factor of 3 at high sample thicknesses due to nonlinear effects. This enhancement of the damping is due to nonlinear scattering processes representing an additional damping channel which absorbs energy from the originally excited FMR. We have shown that the smaller the sample thickness, the less dense is the spin-wave spectrum and, thus, the fewer nonlinear scattering channels exist. Hence, the smallest investigated sample thicknesses (20 and 75 nm) exhibit a small deviation of the ISHE voltage from the linear behavior and a largely reduced enhancement of the damping

parameter at high excitation powers. Furthermore, we have found that the variation of the spin-pumping efficiencies for thick YIG samples which show strongly nonlinear effects is much smaller than the nonlinear enhancement of the damping. This is attributed to secondary magnons generated in a process of multimagnon scattering that contribute to the spin pumping. It is shown that even for thick samples (240 nm) the spin-pumping efficiency is only weakly dependent on the applied microwave power and varies only within a range of 30% for the different microwave powers without a clear trend.

Our findings provide a guideline to design and create efficient magnon- to charge-current converters. Furthermore, the results are also substantial for the reversed effects: the excitation of spin waves in thin YIG/Pt bilayers by the direct spin Hall effect and the spin-transfer torque effect [57,58].

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