

MIT Open Access Articles

Forbidden Reflection Moiré Patterns in metal-2D Material Interfaces

The MIT Faculty has made this article openly available. **Please share** how this access benefits you. Your story matters.

Citation: Reidy, Kate, Varnavides, Georgios, Thomsen, Joachim Dahl, Blackburn, Arthur, Pham, Thang et al. 2020. "Forbidden Reflection Moiré Patterns in metal-2D Material Interfaces." *Microscopy and Microanalysis*, 26 (S2).

Published Version: 10.1017/S1431927620016104

Publisher: Cambridge University Press (CUP)

Permanent Link: <https://hdl.handle.net/1721.1/144374>

Version: Author's final manuscript: final author's manuscript post peer review, without publisher's formatting or copy editing

Terms of use: <http://creativecommons.org/licenses/by-nc-sa/4.0/>



Title: Forbidden reflection moiré patterns in metal-2D material interfaces

Kate Reidy¹, Georgios Varnavides¹, Joachim Dahl Thomsen¹, Arthur Blackburn², Thang Pham¹, Abinash Kumar¹, James LeBeau¹ and Frances Ross¹

1. Massachusetts Institute of Technology, Cambridge, Massachusetts, United States

2. University of Victoria, Victoria, British Columbia, Canada

Moiré patterns are a classical interference phenomenon arising from the superposition of two lattices with slightly different periodicities or rotations, and can be observed by local variations in transmitted electron intensity in HR-TEM and STEM.¹ While the foundations of rotational epitaxy and moiré patterns have been well studied in the fields of thin-film growth and surface science,² there has recently been a resurgence of interest in these topics and their application to two-dimensional (2D) materials.^{3,4} In particular, specific rotations between stacked 2D layers can enhance interlayer coupling, leading to unexpected properties such as Mott insulator and superconducting phases in twisted 2D bilayers. In such 2D/2D systems, the rotation angle can be tuned during fabrication,⁵ resulting in a variety of possible moiré superlattices.

The use of 3D metallic layers on 2D substrates holds equal promise, with evidence that 2D/3D interlayer coupling and charge transfer can strongly influence the contact resistance, photo-response and high frequency performance of 2D/3D heterostructures.^{6,7} Several metals are known to grow epitaxially on 2D materials, creating well-defined rotation angles at the 2D/3D interface. When epitaxial alignment is obtained, studying these 2D/3D moiré patterns with HRTEM presents new opportunities to quantify the crystal rotation and examine defects in the metallic layer and their relationship to structural features on the 2D substrate.

In this work we explore, theoretically and experimentally, moiré patterns arising from epitaxially aligned {111} triangular Au nanoislands on 2D substrates. The Au is deposited in an ultra-high vacuum (UHV) chamber at $T/T_m \sim 0.2$, which results in equilibrium shape Au nanoislands that are flat, triangular, and epitaxially aligned with respect to the underlying 2D substrate. An example is shown in Figure 1(a) for a hBN substrate. We develop a geometric convolution technique to analytically predict the possible moiré periods, which can be mapped to the wave-vector connecting any pair of spots on the selected area diffraction pattern (SAEDP).⁴ It is interesting to note that the 11Å-period moiré pattern visible in Figure 1(a) cannot be explained using the zero order Laue zone (ZOLZ) diffraction spots. Instead, note that the FFT of the sample (Fig. 1(a) inset) suggests the presence of strong $1/3\{422\}$ reflections. These “forbidden” spots are present in the Au nanoislands in the [111] zone axis, due to both their finite size in the beam direction and the non-integer number of unit cells. The spots can be indexed to Ewald sphere intersections with the first order Laue zone (FOLZ), and their interference with the ZOLZ spots has been shown to result in HRTEM image contrast, akin to the contrast seen at stepped surfaces.⁹ If we include these $1/3\{422\}$ spots in the geometric convolution analysis we can explain the observed 11 Å moiré period by a 20° rotation with respect to the substrate. Further, at this rotation we predict another weaker 19Å-period moiré pattern, consistent with the FFT and geometric convolution analysis (Fig. 1(b)).

A second example, with Au on MoS₂, is shown in Figure 2(a). In this case, the most visible moiré period is 18 Å. Our calculation shows that this arises from interference with the expected ZOLZ {220} Au spots and the {2110} of MoS₂ (Figs. 2(a,b)). However, close examination of the SAEDP (Fig. 2(c)) shows that a second moiré periodicity of 32 Å should be visible, arising from the forbidden reflection and {1010} MoS₂ spots, and is correctly identified by our geometric convolution technique (Fig. 2(b)). This forbidden reflection moiré superlattice is not visible in conventional HR-TEM (Fig. 2(a)) and STEM (Fig. 2(d)) images due to the overlying stronger 18 Å moiré unit cell. However, we can explore it using annular dark field (ADF) 4D STEM datasets (Figures 2(e-f)).

In summary, we find that Au nanoislands on 2D materials produce multiple moiré periodicities with differing levels of contrast, and some of these moirés involve the “forbidden” reflection $\frac{1}{3}\{422\}$. The simultaneous analysis of several moiré periods enables rotation angles to be determined without ambiguity. For the particular samples analysed here, the appearance of the moiré patterns is consistent with epitaxial alignment and no distortion or dislocations in the nanoislands. We suggest that the Au nanoislands are loosely bound to the underlying substrate, a conclusion that is corroborated by island motion that is visible during both STEM imaging at room temperature and under heating in UHV *in-situ* TEM conditions. We suggest that combined use of 4D STEM data and geometric convolution provides a versatile approach to analyze forbidden and bulk reflection moirés, and will help to explore the implications of interference effects on physical properties at the 2D/3D interface, such as enhanced charge transfer and moiré-modulated electrostatic interactions.

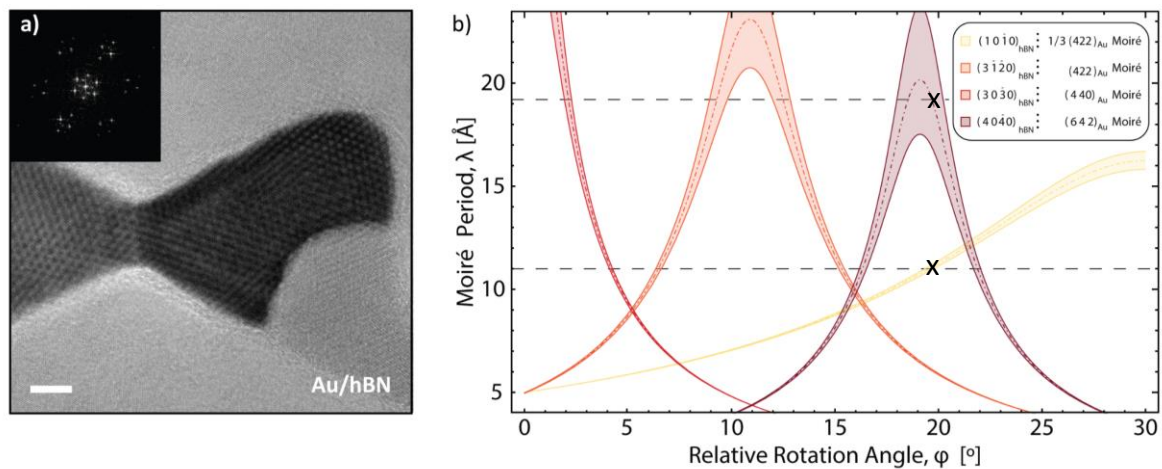


Figure 1: **a)** HRTEM image of two {111} triangular Au nanoislands on hBN, each showing an 11 Å-period moiré pattern. Inset shows the FFT of the image, indicating the $\frac{1}{3}\{422\}$ forbidden reflection spots, moiré satellite spots, as well as suggesting a larger period moiré. The grey region to the lower right is a nanocrystal of Ge, not analyzed further here. Scale bar 5 nm. **b)** Subset of possible moiré supercells for Au on hBN for every possible rotation angle, determined via the geometric convolution technique. Black dotted lines represent the experimentally observed (11 Å) and FFT-suggested (19 Å) moiré periods. Both of these can be explained using a 20° rotation (markers).

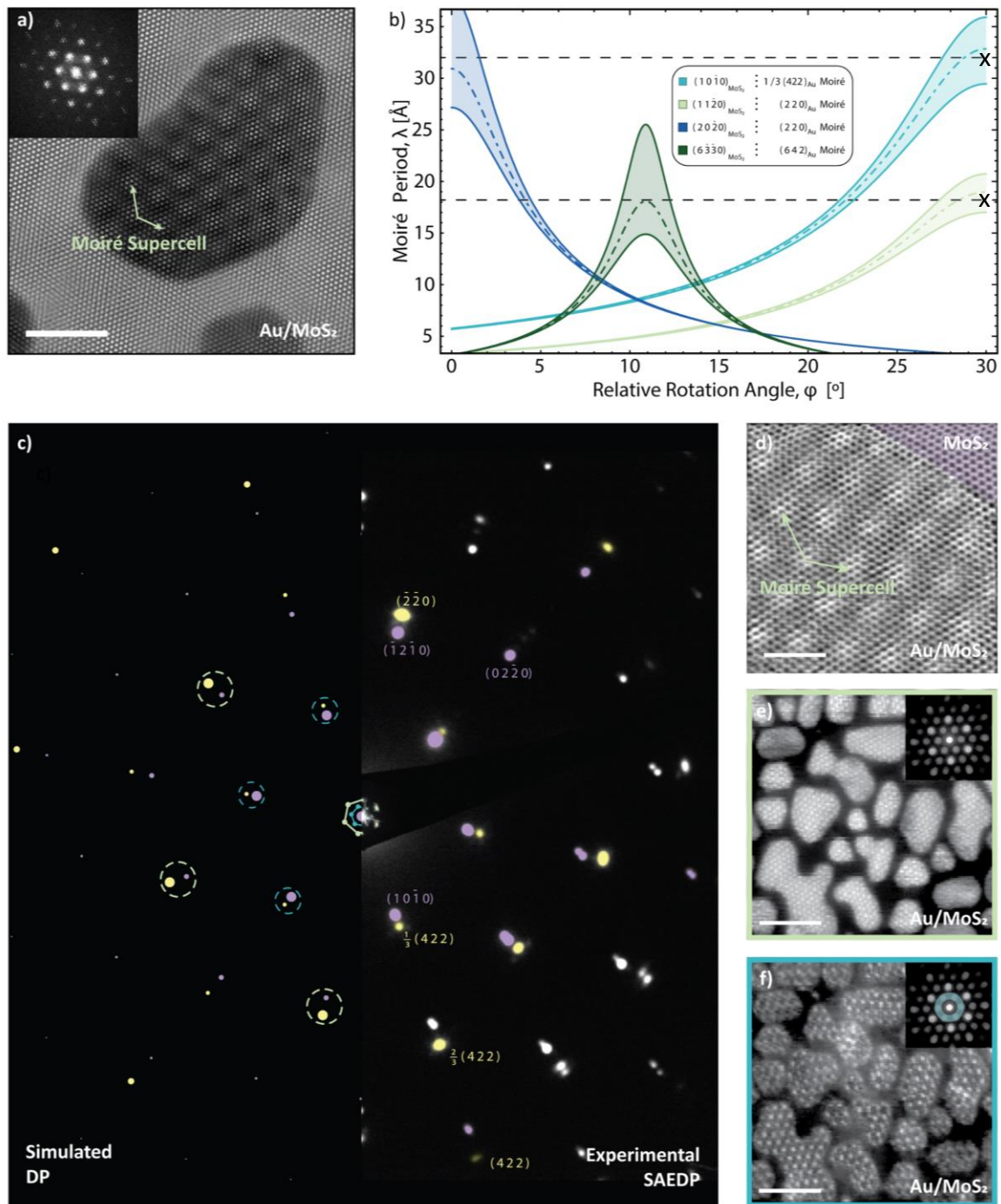


Figure 2: **a)** UHV HR-TEM image of {111} Au nanoislands on MoS₂, showing an 18 Å-period moiré pattern. Inset shows the SAEDP of the sample, highlighting the satellite moiré spots. Scale bar 5nm. **b)** Subset of possible moiré supercells for Au on MoS₂, determined via the geometric convolution technique. Black dotted lines represent the two experimentally observed moiré periods in the HR-TEM and 4D STEM data. At 30° epitaxial orientation, we predict two moiré periods at 18Å and 32Å (markers). **c)** Simulated diffraction pattern for Au and MoS₂ (left), compared with the experimentally obtained SAEDP and the FFT of the HR-TEM moiré (right). False color represents: MoS₂ (purple), Au (yellow), 18 Å moiré (green), and 32 Å moiré (cyan). Green and cyan circled areas represent spots contributing to 18 Å and 32 Å moirés, respectively. **d)** Atomic-resolution STEM image of Au on MoS₂ showing the visible moiré pattern. **e)** Atomic-resolution STEM image of Au on MoS₂ showing the visible moiré pattern. **f)** Atomic-resolution STEM image of Au on MoS₂ showing the visible moiré pattern.

18 Å moiré supercell. Scale bar 2nm and uncorrected scan distortion visible near top of sample caused by charging. **e)** 4D STEM image showing the brightest 18 Å period moiré, and **f)** ADF 4D STEM image using the $1/3 \{422\}$ forbidden reflection spots highlighting weaker 32 Å period moiré. Scale bars 20 nm. Inset shows the 4D STEM DP, alongside the ADF annulus used around $1/3\{422\}$ forbidden reflection spots.

References

1. Oster, G. & Nishijima, Y. Moiré Patterns. *Scientific American* **208**, 54–63 (1963).
2. Bassett, G. ., Menter, J. . & Pashley, D. . Moiré patterns on electron micrographs, and their application to the study of dislocations in metals. *Proc. R. Soc. London. Ser. A. Math. Phys. Sci.* **246**, 345–368 (1958).
3. Woods, C. R. *et al.* Commensurate-incommensurate transition in graphene on hexagonal boron nitride. *Nat. Phys.* **10**, 451–456 (2014).
4. Zeller, P. *et al.* What are the possible moiré patterns of graphene on hexagonally packed surfaces? Universal solution for hexagonal coincidence lattices, derived by a geometric construction. *New J. Phys.* **16**, (2014).
5. Cao, Y. *et al.* Unconventional superconductivity in magic-angle graphene superlattices. *Nature* **556**, 43–50 (2018).
6. Sundaram, R. S. *et al.* The graphene-gold interface and its implications for nanoelectronics. *Nano Lett.* **11**, 3833–3837 (2011).
7. Wang, Y. *et al.* Van der Waals contacts between three-dimensional metals and two-dimensional semiconductors. *Nature* **568**, 70–74 (2019).
8. Buff, P.-A., Flueli, M., Spycher, R., Stadelmann, P. & Borelb, J.-P. *Crystallographic Structure of Small Gold Particles studied by High-resolution Electron Microscopy. Faraday Discuss* **92**, (1991).
9. Reyes-Gasga, J., Gómez-Rodríguez, A., Gao, X. & José-Yacamán, M. On the interpretation of the forbidden spots observed in the electron diffraction patterns of flat Au triangular nanoparticles. *Ultramicroscopy* **108**, 929–936 (2008).